

A New Composite Restorative Based on a Hydrophobic Matrix

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A hydrophobic restorative composite based on a fluorocarbon analog of an alkyl methacrylate and a bisphenol adduct was formulated into a one-paste system, which polymerized in the presence of blue light. Physical, mechanical, and water-related properties were determined. High contact angles and low water sorption were shown by the experimental composite. Capillary penetration of oral fluids around restorations, therefore, could be prevented in the presence of this highly hydrophobic surface. The physical and mechanical properties of the experimental composite were either comparable to or somewhat less favorable than commercial Bis-GMA composites.

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Introduction.

The majority of current commercial composite restorative systems are based on Bis-GMA and consist of 70-80% by weight of silanated inorganic materials.^{1,2} It has been shown that these composite materials have substantially improved physical and mechanical properties compared to unfilled acrylic materials.³⁻¹⁰ However, a major problem with unfilled acrylics and composites is the penetration of oral fluids via microleakage through capillary spaces between the restoration and the tooth.¹¹⁻¹³ These fluids contain bacteria which can cause decay beneath the filling.

The driving force for the capillary penetration of a liquid between two solids has been shown by O'Brien, Craig, and Peyton¹⁴ to be a function of the free energy changes at the solid-liquid interfaces and can be expressed by the following equation:

$$(F_{S_1V} - F_{S_1L}) + (F_{S_2V} - F_{S_2L}) = -\Delta F^S = \gamma_{LV} (\cos \Theta_1 + \cos \Theta_2)$$

where F is the free energy, S_1 and S_2 are the two solids, V and L are the vapor and liquid, γ_{LV} is the surface energy of the liquid, and Θ_1 and Θ_2 are the contact angles of the liquid against the two solids. If ΔF^S is negative, penetration will be spontaneous, and if positive, it will not be spontaneous. Therefore, the preparation of highly hydrophobic surfaces of teeth and restorations could eliminate this capillary penetration.

Fluorocarbon-containing polymers have a low surface energy and are highly hydrophobic. The wettability of fluorinated acrylates and methacrylates and their polymers has been studied by several workers¹⁵⁻¹⁸ utilizing the concept of critical surface tension which can be determined from contact angle measurements. Their results indicated that fluorine-containing acrylic polymers were difficult to wet with aqueous solution.

The purpose of this study was to formulate a hydrophobic composite restorative system by using a fluorocarbon analog of a methacrylate. The physical, mechanical, and water-related properties of this experimental composite were also evaluated.

Materials and methods.

Formulation of the experimental com-

TABLE I
COMPOSITION OF THE EXPERIMENTAL COMPOSITE

Constituents	% by Weight
Octafluoropentylmethacrylate	18.52
Diacryl 101 ^a	6.17
Silanated quartz ^b	75.00
Camphoro quinone	0.19
N,N-dimethyl-amino-ethyl methacrylate	0.12

^aSupplied by Akzo Chemie bv., Amersfoort, Holland

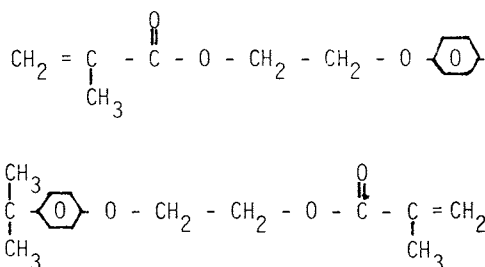
^bSupplied by 3M Company, St. Paul, MN

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posite.—The composition of the experimental composite is shown in Table 1. Octafluoropentylmethacrylate is hydrophobic with an advancing contact angle of 100° on a smooth surface. Diacryl 101 is a difunctional unsaturated methacrylic monomer based on Bisphenol A. It is less hydrophilic and much less viscous than Bis-GMA. The molecular formula of Diacryl 101 is:



The constituents of the parent resin were mutually dissolved to form a clear solution by carefully balancing the ratio between them. Camphoro quinone and N, N-dimethylaminoethyl-methacrylate were used as accelerators. The experimental composite containing silanated quartz consisted of a single paste, which polymerized by initiation in the presence of blue light.

Method of testing.—The methods of testing described by Craig and co-workers⁶⁻⁷ in previous publications were used in this study. In all instances the values reported are means of five replications.

The abrasive wear was determined by the two-body abrasion test used by Powers, Allen, and Craig.¹⁹ The inclined plane method²⁰ was used to determine the contact angles of distilled water on the composite surfaces. A smooth surface of composite prepared against a microscope slide was measured using a cathetometer. A second determination was done after the surface had been finished with a 600 grit silicon carbide paper.

Results.

Results of physical, mechanical, and water-related properties of the experimental composite are summarized in Tables 2, 3, and 4, respectively. A filled (Adaptic) and an unfilled restorative material (Sevriton)^{6,7} are also listed for comparison. Values of each mean and standard deviation in these tables were calculated from five experimental measurements.

In general, the physical properties of the experimental composite shown in Table 2 are comparable to those of Adaptic, except polymerization contraction, and are higher than those of Sevriton. A high filler content of 77% by weight and very short setting time of 35 seconds were recorded for the experimental composite. The value of 3.1% volume contraction is 2.5 times higher than that of Adaptic, but is only 60% of that of Sevriton. The thermal coefficient of expansion measured between 0 and 60°C

TABLE 2
PHYSICAL PROPERTIES OF THE EXPERIMENTAL COMPOSITE COMPARED WITH A
COMMERCIAL COMPOSITE AND AN UNFILLED RESIN

	Adaptic ^a	Exp. Comp.	Sevriton ^b
Inorganic Phase, wt %	77.7 (0.2) ^c	76.2 (0.2)	—
Working Time, min.	3.5 (0.3)	35 sec.	4.0 (0.5)
Polymerization Contraction, Vol. %	1.3 (0.1)	3.1 (0.3)	5.2 (0.5)
Density, g/cc	1.98 (0.01)	2.16 (0.03)	1.12 (0.02)
Thermal Coefficient of Expansion, $\times 10^{-6}/^\circ\text{C}$	39.4 (1.4)	55.7 (5.0) (0-60 $^\circ\text{C}$) 53.4 (5.4) (at 37 $^\circ\text{C}$)	92.0 (2.3)

^aData from Dennison and Craig.⁷

^bData from Macchi and Craig.⁶

^cValues in parenthesis are standard deviations from the mean.

TABLE 3
MECHANICAL PROPERTIES OF THE EXPERIMENTAL COMPOSITE COMPARED WITH A
COMMERCIAL COMPOSITE AND AN UNFILLED RESIN

Property	Adaptic ^a	Exp. Comp.	Sevriton ^b
Indentation depth, mm	0.061 (0.002) ^c	0.065 (0.001)	0.113 (0.003)
Recovery, %	74.9 (1.7)	66.1 (1.5)	74.1 (3.5)
Compressive strength, MPa	237 (21)	127 (2.9)	71 (14)
0.1% Yield strength, MPa	161 (9)	80 (4.8)	52 (7)
Modulus of elasticity, GPa	16.6 (0.9)	15.0 (1.1)	2.3 (0.5)
Tensile strength, MPa	45.5 (2.4)	29.0 (1.2)	22.8 (0.7)
Abrasive wear mm ³ /mm × 10 ⁺⁴	7.2 (0.2)	8.0 (0.2)	13.3 (1.6)

^aData from Dennison and Craig⁷ except abrasive wear.

^bData from Macchi and Craig⁶ except abrasive wear.

^cValues in parentheses are standard deviations from the mean.

TABLE 4
WATER-RELATED PROPERTIES OF THE EXPERIMENTAL COMPOSITE COMPARED WITH A
COMMERCIAL COMPOSITE AND AN UNFILLED RESIN

	Adaptic ^a	Exp. Comp.	Sevriton ^a
Contact Angle			
Advancing (Θ_A) deg.	58.7 (1.7) ^b finished	156 (1.3) finished	61 (2) finished
	68.7 (1.0) smooth	88 (0.6) smooth	
	47.2 (3.4) finished	76 (1.9) finished	38 (3) finished
Receding (Θ_R) deg.			
	54.7 (1.4) smooth	58 (2.5) smooth	
Water Sorption (mg/cm ²)			
24 hours	0.37 (0.07)	0.09 (0.01)	1.58 (0.27)
48 hours	0.42 (0.04)	0.13 (0.01)	1.67 (0.22)
7 days	0.59 (0.02)	0.14 (0.01)	2.03 (0.30)
14 days	0.66 (0.04)	0.15 (0.02)	2.08 (0.29)
Solubility %			
14 days	<0.01	<0.01	0.23 (0.03)

^aData from Dennison and Craig⁷ except contact angles.

^bValues in parentheses are standard deviations from the mean.

by a thermomechanical analyzer* is relatively higher for the experimental composite

than for Adaptic.

Table 3 shows that the mechanical performance of the experimental composite compares with that of Adaptic and exceeds that of Sevriton. The indentation and the

*DuPont Thermal Mechanical Analyzer No. 941, Instrument Division, Wilmington, DE

modulus obtained from the compression test are about the same for both filled composites. The compressive strength, the 0.1% yield strength in compression, and the tensile strength from diametral compression tests of the experimental composite are considerably lower than those of Adaptic, ranging from one-half to two-thirds. The abrasive wear measured from the two-body test against a 600 grit silicon carbide paper is comparable for both filled materials.

The water-related properties, *i.e.*, contact angles, water sorption, and solubility, are shown in Table 4. The advancing contact angle of 88° on a smooth surface is the highest among all. After finishing with a 600 grit paper, the contact angle increased to the very high value of 156° , which is in contrast to a decreasing angle shown in both Adaptic and Sevrton.

The water sorption of the experimental composite measured at one-, two-, seven-, and 14-day periods is much lower than that of Adaptic and Sevrton. It was only 0.15 mg/cm² at 14 days, and was only one-fourth and one-fifteenth of that of Adaptic and Sevrton. In addition, the experimental composite reached equilibrium very quickly in two days, while the amount of water sorption was still increasing for the other two materials. The solubility of the experimental composite was so low that it was difficult to measure.

Discussion.

The high polymerization contraction of the experimental composite can be attributed to two causes. First, a relatively high amount of small molecular weight fluoro-methacrylate monomer was used. The attempt to employ a partially polymerized fluoro-methacrylate resulted in a phase separation from the parent resin. Second, the polymerization does not occur until the material is exposed to light. In chemically initiated systems, however, polymerization begins immediately on mixing, and some shrinkage has already taken place by the time of measurement. Therefore, lower polymerization contractions are usually recorded.

The relatively high thermal coefficient of expansion of the experimental composite can be reduced by increasing the filler

content. This is possible for the experimental composite because of the very low viscosity of the parent resin. The higher dimensional and thermal changes of the experimental composite could possibly be outweighed by the advantages of its highly hydrophobic character, as discussed later.

In regard to mechanical properties, the high hardness and rigidity of the experimental composite, reflected by the indentation and elastic modulus measurements, were as good as those of Adaptic. The compressive and tensile strengths of the experimental composite are considerably lower than those of Adaptic, but still surpass Sevrton. It is important that, in the attempt to enhance hydrophobicity, other properties are not compromised. These lower values of strength are probably a result of extensive use of fluoro-methacrylate monomers with a fairly low molecular weight, since it requires a great deal of reaction before considerable chain length can be built up. This problem, as well as polymerization contraction, could be resolved by using fluorinated molecules of a large aromatic type similar to Bis-GMA. Nevertheless, the exact clinical relevance of a high tensile and compressive strength has not been shown.

The abrasive wear determined from a two-body test is comparable to that of Adaptic. It is believed by Powers *et al.*²¹ that ductile failure is preferable to brittle failure in respect to this property. The experimental composite was comparable in abrasive wear to Adaptic in spite of the lower strength; this result may be caused by a lower coefficient of friction of the fluoro-polymer.

The most promising and interesting qualities of the experimental composite are the water-related properties. The water sorption of the experimental composite is much lower than that of Adaptic and Sevrton. This is believed to be associated with hydrophobic polymers. In addition, the experimental composite had only a negligible solubility after 14 days. This result may suggest that the bonding between the polymer and the inorganic phases is strong.

The advancing contact angle of water on a smooth surface is 88° , which is probably the highest in the literature for this type of composite and compares very

favorably with Adaptic. After finishing with 600 grit paper, the contact angle increased to the very high value of 156° . This result is in contrast to the theory²² which predicts a reduction in the measured contact angle by roughening of surfaces with angles below 90° . This is well shown by Adaptic in Table 4, where the contact angle decreased from 68.7° to 58.7° . It is believed, therefore, that the bulk of the experimental composite has a contact angle higher than that measured for the smooth surface and is hydrophobic.

According to the equation derived by O'Brien *et al.*,¹⁴ which was shown previously, a polymer with a contact angle of greater than 130° would be required in order for ΔF^S to be positive, if the contact angle of saliva on dentin is taken to be 50° . Therefore, the experimental composite, having a contact angle of 156° after finishing, would possibly prevent capillary penetration of oral fluids around restorations. Furthermore, if the walls of the cavity were treated with octafluoropentylmethacrylate, as a varnish, which has a contact angle of 100° , ΔF^S would be more positive, and capillary penetration could be even more difficult. The elimination of fluid penetration should improve the serviceability of the restoration, reduce recurrent caries, and improve dental health. However, the advantages of the experimental composite extend beyond capillary effects to the contoured surface of the restoration. This surface is a low energy surface to which plaque would attach with more difficulty and subsequently be removed with more ease, facilitating oral hygiene regimes. This effect may improve the prognosis for gingival health in the presence of a subgingival composite, which even in competent hands can be associated with a chronic inflammation of the gums.

Conclusions.

It was possible to produce a hydrophobic composite using a fluorocarbon analog of methacrylate, octafluoropentylmethacrylate, and a bisphenol adduct. The experimental composite was designed as a one-paste system which was polymerized in the presence of blue light. The physical and mechanical properties of the experimental composite were comparable in most

instances to a commercial composite, but had higher polymerization contraction and lower compressive and tensile strengths. The profoundly hydrophobic nature of the experimental composite was evident from the high contact angle with water. In addition, the experimental composite had an exceptionally low water sorption value and negligible water solubility.

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