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Electrochemical Oxygen Reduction Reaction on Ag Nanoparticles of Different Shapes

Tim Van Cleve, Emily Gibara, and Suljo Linic*[a]

Silver electrocatalysts are an attractive alternative to platinum for electrochemical oxygen reduction reaction in alkaline fuel cells. Recent advances in the synthesis of metal nanoparticles have enabled the design of silver nanoparticles of different shapes, terminated with different surface facets that exhibit different catalytic properties. In this contribution, we prepared spherical and cubic silver nanoparticle electrocatalysts and

tested their electrocatalytic oxygen reduction reaction activity in $0.1\,\mathrm{m}$ sodium hydroxide. Our work demonstrates that carbon-supported silver nanospheres and nanocubes of similar size exhibit similar ORR activity with the spheres slightly outperforming the cubes. In addition, we suggest possible reasons for the slightly enhanced activity of the nanospheres.

Introduction

Catalyzing the electrochemical oxygen reduction reaction (ORR) is critical for the design of highly efficient fuel cells. Although platinum has been the principal material of interest because of its high catalytic activity and superior stability-Pt and Au are the only stable metals under the conditions of electrocatalytic ORR in acidic electrolytes—the practical implementation of Pt-based electrodes has been hampered by their high cost.[1] While the superior electrocatalytic ORR activity of Pt compared to other metals persists in basic medium (with OHas the charge carrier), the number of materials that are stable under these conditions is significantly larger than in acid. [2] For example, silver (Ag) leaches in the form of oxidized Ag ions in acid under ORR conditions. On the other hand, Ag forms a stable oxide near ORR conditions in basic medium as shown in Figure 1. Ag is also currently approximately seventy times less expensive than Pt.[3] The main problem with Ag electrocatalysts is that in base these materials are at least an order of magnitude less active on a per-surface-site basis toward ORR than platinum.[4]

We recently demonstrated that Ag–Co alloy nanoparticle electrocatalysts perform ORR in basic medium at a rate that is more than 5 times higher than the rate on pure Ag nanoparticles of identical size. We identified these materials based on an analysis of the ORR reaction mechanism on the Ag(111) surface at the elementary step level. This mechanistic analysis suggested that by perturbing the Ag surface atoms in such a way that they bind oxygenated surface intermediates slightly more strongly than the Ag(111) surface yields a more reactive Ag surface. Our studies suggested that the electronic communication between subsurface Co atoms and the surface Ag

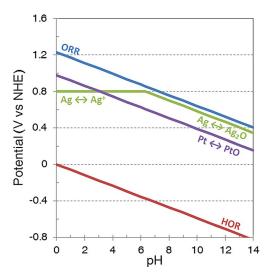


Figure 1. The Pourbaix diagram shows the equilibrium potentials for silver and platinum oxidization reactions as a function of the solution pH and potential with respect to the normal hydrogen electrode potential scale. At high potentials, oxidized species are favored making electrode stability a major concern in acidic electrolytes. Operating in alkaline conditions enables the use of more materials including silver as ORR catalysts.

atoms in the alloy nanoparticles electronically perturbs the Ag surface atoms yielding more reactive ORR materials.^[5]

One way to accomplish the objective of creating Ag surface sites that bind oxygenated intermediates more strongly than the Ag(111) surface is, as mentioned above, to create suitable Ag alloys. [5,7] An alternative approach is to control the distribution of surface facets exposed to the reactants and intermediates involved in ORR by controlling the morphology of Ag nanoparticles. [8] For example, the Ag(100) surface binds oxygenated adsorbates more strongly that the Ag(111) surface. [9] As Ag nanocubes have a significantly larger concentration of the Ag(100) surface sites than the Ag nanospheres, we hypothesized that Ag nanocubes should be more reactive than

[[]a] T. Van Cleve, E. Gibara, Prof. S. Linic Department of Chemical Engineering University of Michigan Ann Arbor, MI 48109-2136 (USA) F-mail·linic@umich.edu



nanospheres. In this contribution we have used colloidal synthetic routes to synthesize Ag nanoparticles of cubic and spherical shapes. We tested these nanostructures in electrochemical ORR in a basic environment. Contrary to our original hypothesis, we found that samples containing Ag spheres and Ag cubes exhibit similar performance with the Ag spheres slightly outperforming the cubes of equal size. We describe our findings below and suggest possible reasons for the superior activity of the nanospheres.

Results and Discussion

Characterization of Ag nanoparticles

The data in Figure 2 show the UV/Vis extinction spectra of various samples of Ag spheres and cubes prior to their deposition onto Vulcan XC 72 carbon. The extinction in the UV/Vis spectra is the consequence of the excitation of localized surface plas-

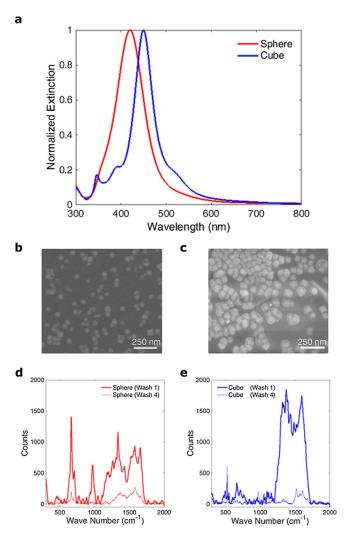


Figure 2. Extinction spectra (a) of representative Ag sphere and cube samples that were tested in our electrochemical experiments. SEM images of silver nanospheres (b) and nanocubes (c) particles deposited on Si substrate. SERS spectra Ag nanospheres (d) and Ag nanocubes (e) following multiple acetone rinses

mon resonance (LSPR) in Ag nanostructures. The wavelength and shape of the LSPR extinction peaks are characteristics of the shape and size of Ag nanoparticles.^[10] The appearance of the single peak in the Ag UV/Vis extinction spectra at around 400 nm is a characteristic of Ag spheres with \approx 40 nm diameters, while the triple extinction peak is a distinguishing feature of nanoscopic Ag cubes.[10] The UV/Vis extinction spectra are consistent with relatively pure samples of Ag cubes and spheres. Following the four acetone washes, the UV/Vis extinction spectra showed no changes compared to the original UV/ Vis spectra, suggesting minimal changes in particle morphology. The size and shape of Ag samples were also verified by SEM of nanoparticles deposited onto a silicon substrate, shown in Figure 2. Analysis of these micrographs revealed that Ag spheres had an average diameter of 43.7 ± 8.5 nm and cubes had an average edge length of 40.8 ± 5.4 nm.

As we used colloidal synthesis methods that employ organic stabilizing agents (mainly polyvinylpyrrolidone, PVP), we attempted to remove these molecules before performing electrochemical tests. The removal of PVP was attempted by washing the sample in acetone multiple times as described above. Data in Figure 2d and e show surface enhanced Raman spectra (SERS) of our Ag samples after the first and fourth washing step. The analysis of the SERS spectra suggests that there is a significant decrease in the PVP signal resulting from the washing process. We note that the Raman peaks in the range 1400–1600 cm⁻¹ correspond to the C–C stretches of PVP.^[11] The data also show that there is some residual PVP that was detected on Ag even after multiple washing steps; however, if comparing to the as-prepared samples and considering that these Ag nanostructures are excellent in enhancing Raman signal, the amount of residual PVP appears to be relatively small.

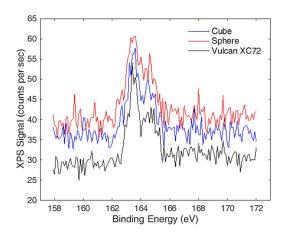
Data in Figure 3 show X-ray photoelectron spectroscopy (XPS) spectra of the Ag nanosphere and nanocube samples on carbon and the Ag-free carbon support. In this XPS analysis, among other elements, we focused on sulfur since it has been shown to affect the chemical activity of metals and it was used in the synthesis of the Ag cubes. [10,12] The S2p XPS spectrum shows that sulfur is present in all the samples in the similar concentrations and chemical states suggesting that the main source of sulfur is the carbon support. Data in Figure 3 also show the Ag3d XPS spectra. It is found that Ag is present in the metallic form.

Electrochemical performance

Linear sweep voltammetry experiments were performed on supported Ag nanoparticles (Ag/Vulcan XC72R) in the thin-film rotating-disk electrode (RDE) configuration to measure their oxygen reduction rate as a function of potential and rotation rate. The data in Figure 4 show the polarization behavior for representative Ag sphere and cube nanoparticle electrocatalysts. The polarization curves in Figure 4 were obtained by adjusting the kinetic current to an equal surface area basis (with a roughness factor of 1) as described in Ref. [8]. This was accomplished by first computing the raw kinetic current using

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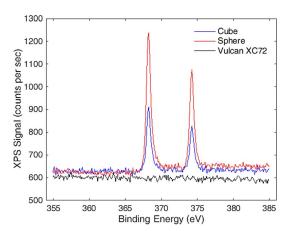


Figure 3. XPS spectra for S2p (a) and Ag3d (b) core states measured for various Ag nanoparticle catalysts and carbon support.

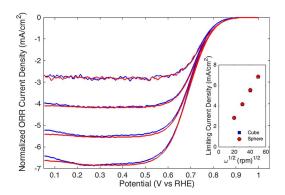


Figure 4. ORR current densities were measured for Ag nanospheres (——) and nanocubes (——) as a function of potential (V vs. RHE) at rotation rates of 400, 900, 1600, and 2500 rpm. Both particle morphologies exhibited nearly equivalent mass-transfer limiting-current densities and similar selectivity towards complete O_2 reduction to water. The slope of the Levich plot (inset) is consistent with a $4e^-$ transfer mechanism.

Koutecky–Levich equation: $|i_k(V)| = (1/|i(V)| -1/|i_L|)^{-1}$, where i is the raw measured current, i_k is the raw kinetic current, and i_L is the limiting current measured between 0.2 and 0.5 V_{RHE}

(V_{RHE} = potential relative to the reversible hydrogen electrode (RHE). The raw kinetic current was then normalized to the electrochemical surface area (ESCA). We describe further below how ESCA was measured. The normalized kinetic current density ($j_k = |i_k|$ /ECSA) was then used to regenerate the RDE current density (j) curves on a superficial area basis as: $|j| = (1/j_k + A_{disk}/|i_L|)^{-1}$.

At high potentials, ORR is limited by the intrinsic activity of the electrocatalysts and its loading, showing an exponential dependence of current on applied potential. At low potentials, the overall rate becomes limited by the transport of O_2 to the surface which is a function of the RDE rotation rate. The kinetic rate of ORR can be determined by accounting for the effects of O_2 transport to the electrode on the observed rate (current) as described in the Koutecky–Levich equation. The data in Levich plot (inset) shows the mass-transfer-limiting current density as a function of rotation rate. The slope of this line indicates that ORR proceeds through the $4\,\mathrm{e}^-$ pathway with very high selectivity to H_2O for both electrocatalysts.

To compare the relative kinetic performance of different electrocatalysts, the rate of ORR was normalized by the number of electrocatalytic sites, or ECSA. This required a rigorous measurement of electrochemical surface area, i.e., the area at the reactive interface between the liquid electrolyte and metal electrode. The ECSA of Ag electrocatalysts was measured by utilizing the phenomenon of Pb underpotential deposition (UPD) onto Ag. Previous work has shown that by holding an Ag electrode slightly above the Pb2+/Pb reduction potential, a submonolayer (\approx 2/3 monolayers) of Pb deposits onto the Ag surface.[13] By scanning to higher potentials, the underpotentially deposited Pb submonolayer is dissolved, generating an oxidative current. By measuring the oxidative current, it is possible to directly measure the Ag ECSA by integrating the Pb_{UPD} oxidization peak. The data in Figure 5 show the oxidative currents generated during the oxidation of Pb sub monolayers on the surface of representative Ag cube and sphere electrocatalyst. To generate the plots of the oxidative current in Figure 6, the background resulting from capacitive charging of the carbon support was removed. We note that at the poten-

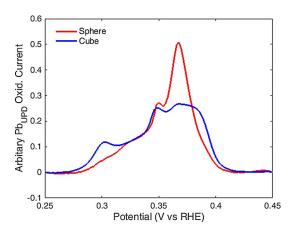
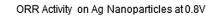
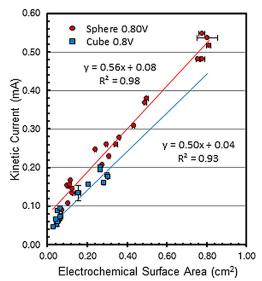


Figure 5. The background-corrected Pb UPD oxidization curves for silver nanocubes (——) and nanospheres (——) in 0.1 м NaOH show different features over the potential range of 0.25–0.45 V.





b ORR Activity on Ag Nanoparticles at 0.85V

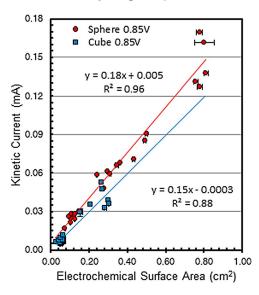


Figure 6. Alkaline ORR kinetic currents of supported silver nanosphere (●) and nanocube (■) electrocatalysts with different weight loadings at two different operating potentials: 0.80 V (a) and 0.85 V (b).

tial range where the oxidation and desorption of Pb from Ag surfaces takes place (0.25–0.45 V vs. RHE) there were no other electrochemical processes and therefore no further background corrections were required. Multiple peaks in the oxidative current curves are associated with the Pb oxidation and desorption from the different surface facets of Ag. The multiple peaks allow us to quantify the contribution of different Ag surface facets to the electrochemical surface area. We will revisit to this issue further below in the discussion section.

The data in Figure 6 show the net kinetic currents (not normalized by the ESCA) for different samples of Ag nanoparticle electrocatalysts plotted against their electrochemical surface area at two different operating potentials (0.80 and 0.85 V vs. RHE). We note that unlike for Pt electrodes, where it is customary to report currents at 0.9 V vs. RHE, Ag-based electrodes exhibit lower ORR activity and the currents are usually reported at 0.85 and 0.80 V. The kinetic current was obtained as described above using Koutecky–Levich equation. The data shows that oxygen reduction rate over this range of metal nanoparticle loadings scales linearly with the electrochemical surface area. This linear scaling suggests that the electrode design allows for a rigorous deconvolution and measurements of kinetic currents, which is critical if different electrocatalysts are to be compared using the RDE setup.

The slope of the lines in Figure 6 corresponds to the average specific activity (the ORR rate per electrochemical surface area) of the nanoparticles over this range of surface areas. By comparing these slopes at various potentials, it is possible to compare the performance of different Ag nanoparticle electrocatalysts. From the specific activities calculated for these silver nanoparticle samples, we observe that spheres are approximately $12\pm 8\,\%$ and $18\pm 10\,\%$ more active than cubes at 0.80 and 0.85 V on the reversible hydrogen electrode potential scale.

Data in Figure 6 show relatively small differences in ORR activity between Ag nanospheres and nanocubes. One plausible explanation is that nanoparticles undergo morphological changes in the electrochemical environment leading to similar surface structures and therefore similar activities. To address this possibility we performed in situ electrochemical adsorption/desorption measurements to investigate potential electrocatalyst restructuring during the course of the measurement. Whereas cyclic-voltammetry measurements on Ag nanoparticles in deaerated electrolyte did not yield any significant electrochemical adsorption features, both Ag structures exhibited distinct Pb_{UPD} stripping features, characteristic of the presence and persistence of different Ag surface facets.

The data in Figure 5 show the Pb_{UPD} oxidative stripping spectra for the two Ag nanoparticle samples as a function of electrochemical potential measured after a set of electrochemical measurements were performed on the samples. The Pb stripping spectra show that the Ag surfaces in the two electrocatalysts are different from each other. Previous measurements on well-defined Ag(111) and Ag(100) single crystal surfaces allow us to assign the stripping features to a particular Ag surface facet. [13] It has been demonstrated that the Ag(111) surface exhibits two sharp peaks located at \pm 130 and \pm 155 mV relative to Pb/Pb²⁺ reduction potential (corresponding to 0.34 and 0.37 $V_{\rm RHE}$), whereas the Ag(100) surface exhibits three peaks: a short broad feature located at \pm 50 mV (0.26 $V_{\rm RHE}$), a sharp peak located at \pm 100 mV (0.31 $V_{\rm RHE}$), and a broad peak located at approximately \pm 135 mV (0.36 $V_{\rm RHE}$). [13]

Utilizing the Pb_{UPD} stripping spectra, it is possible to estimate the relative abundance of the Ag(111) and Ag(100) sites on Ag cubes and spheres. To accomplish this, we deconvoluted the measured Pb stripping spectra in terms of the linear combination of the previously reported and discussed Ag(111) and Ag(100) Pb spectra.^[13a] The concentration of various surface facets on our nanostructures was determined by minimizing



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the sum of square errors between the deconvoluted and experimental spectra. Not surprisingly, we found that Ag nanocube samples exhibit largely (100) character ($\approx 80-95\%$) whereas Ag spheres contained significant fractions of both surfaces with Ag(100) surface facets covering approximately 40–45% of the surface.

The observation that the Ag spheres are more active that Ag cubes is in contradiction with our proposed hypothesis that Ag surfaces that bind oxygenated intermediates slightly more strongly than the clean Ag(111) should exhibit higher ORR activity than Ag(111). We postulate that the reason for this contradiction is that the Ag(100) surface binds oxygenated intermediates, such as OH*, too strongly and that under relevant high potential ORR conditions these sites are poisoned by these intermediates. This is supported by DFT calculations which showed that Ag(100) binds OH stronger than Ag(111) by \approx 0.5 eV, suggesting that Ag(100) is more likely to be poisoned at high potentials and therefore less catalytically active than the Ag(111) surface. [9a,14] We also note that Blizanac and coworkers demonstrated using single crystal Ag(111) and Ag(100) electrodes that Ag(111) was \approx 33 and 50% more active than Ag(100) at 0.80 and 0.85 V vs. RHE, respectively, for ORR performed in 0.1 M KOH. [4a] If we assume cubes are entirely terminated by Ag(100) sites and spheres have 45% Ag(100) and 55% Ag(111) surface coverage, using these previously reported ORR rate numbers we compute that the spheres are expected to be more active than cubes by $\approx 18-27\%$ at potentials between 0.80 and 0.85 V_{RHE} , which is relatively consistent with our findings. [4a] This work suggests the intrinsic ORR activity of Ag nanostructures cannot be improved by replacing Ag(111) with Ag(100) surface sites.

Conclusions

We have demonstrated that carbon-supported 40 nm Ag spheres are slightly more catalytically active towards alkaline oxygen reduction reaction (ORR) than carbon-supported 40 nm Ag cubes. Ex situ and electrochemical characterization have shown that Ag nanoparticles maintain their distinct morphologies throughout catalyst preparation and testing. We observed that the relative ORR activity enhancement of Ag spheres compared to Ag cubes is consistent with the abundance of the Ag(111) surface facets compared to Ag(100) on the spheres.

Experimental Section

Electrocatalyst preparation

Silver nanospheres were synthesized following a previously established procedure. [8f, 15] In general, ethylene glycol (10 mL, JT Baker) was added to a vial and heated for 45–60 min at 160 °C in a silicone oil bath. Then, 3 mL solutions of 600 mm PVP (Sigma–Aldrich; $M_{\rm w}=55\,000$) and 100 mm AgNO₃ (Sigma–Aldrich) in ethylene glycol were prepared and dropwise added to the heated vial. The vial was capped and after approximately 1 h removed from the oil bath and quenched with cool water.

Silver cubes were synthesized by a method developed by Xia. [10b] Ethylene glycol (5 mL) was heated to 140–150 °C. After 45 min,

30 mm Na_2S solution (80 $\mu L)$ was added to ethylene glycol. A few minutes later, $0.28\,\text{m}$ AgNO $_3$ solution (0.5 mL) and $0.18\,\text{m}$ PVP solution (1.5 mL) were quickly added to the vial. The vial was capped and the solution was allowed to react for approximately 45 min after which it was removed from the oil bath to inhibit the further nanoparticle growth.

The solutions containing Ag nanoparticles were washed four times with acetone to remove capping agents, solvent, and unreacted reagents. After each washing procedure, Ag nanoparticles were collected by centrifugation at 7000 rpm for 20 min. The effluent was discarded and Ag nanoparticle suspensions were reconstituted in a small volume (2–5 mL) of ethanol prior to the next acetone wash. After the final centrifugation, Ag nanoparticles were re-suspended in pure ethanol and deposited onto Vulcan XC72R powder (Cabot, pretreated for at least 2 h in $\rm H_2$ at 400 °C). The catalyst solutions were dried overnight in a heating oven.

Characterization

UV/Vis extinction spectroscopy experiments were performed using a Thermo Scientific Evolution 300 UV/Vis spectrophotometer with a Xenon lamp source to measure the extinction spectra of dilute Ag nanoparticle suspensions. UV/Vis spectra were collected between wavelengths of 300 and 800 nm at a rate of 600 nm min⁻¹ and were normalized by the peak signal intensity to facilitate comparison of samples with different concentrations. SEM was performed on Ag nanoparticles deposited directly on a clean Si chip (\approx 1 cm²; rinsed with piranha solution and dried) using a FEI NOVA Nanolab operating in secondary electron detection mode. Micrographs were obtained by using a TLD (through-the-lens detector) and the accelerating voltage was 15 kV. SERS experiments were measured on Ag nanoparticle on Si substrate using a Horiba Jobin-Yvon LabRAM HR system. SERS spectra were acquired over 300–2000 cm⁻¹ range using a 532 nm laser with 5 s acquisition time and by averaging the signal over 10 consecutive acquisitions. XPS experiments were performed on Ag nanoparticle catalysts on Vulcan XC72R support using a Kratos Axis Ultra XPS with monochromatic $Al_{K\alpha}$ (1486.69 eV) X-ray source. As-prepared Ag/C samples were heated overnight in an UHV system to remove residual moisture before being directly transferred to a vacuum chamber (pressure of $\approx 10^{-9}$ torr) for analysis. XPS spectra were measured at every 0.1 eV over the ranges of 158–172 eV, 277–300 eV, and 355– 385 eV to observe S2p, C1s, and Ag3d peaks. XPS signals were normalized by their acquisition times (10.6, 6.5, and 7.5 s, for S, C, and Ag peaks, respectively). Charging effects were compensated using a flood gun. XPS binding energies were calibrated based on positioning the main C1s peak at 284.8 eV, which originates from the carbon support.

Electrochemical measurements

Electrochemical measurements were performed at RT in a custom-built Teflon three-electrode cell with a Gamry Instruments Reference 3000 potentiostat/galvanostat/Frequency-Response-Analyzer. The reference (Hg/HgO in 1 $\rm M$ KOH, Radiometer Analytical) and counter electrodes (Pt wire, Alfa Aesar) were both in isolated compartments with long diffusion paths to the working electrode chamber. Working electrodes were prepared by sonicating the catalyst powders in 99.9% acetone (Fisher) at 1.50 $\rm mg\,mL^{-1}$ for more than an hour and dispersing 18 $\rm \mu L$ onto a 5 mm glassy carbon electrode insert (Pine Instruments). These electrodes were placed





into a Teflon RDE housing, which was attached to a rotator (Pine Instruments).

Electrolyte solutions of 0.1 M NaOH were prepared from ultrapure water (18.2 M Ω cm⁻¹ (Millipore)) and 99.99% sodium hydroxide (Alfa Aesar). The reaction rates were measured in the O2-saturated electrolyte using linear sweep voltammetry at rotation rates between 400-2500 rpm at a scan rate of 10 mV s⁻¹ to achieve steadystate rates. A consistent uncompensated resistance of \approx 40 Ω was measured via high-frequency impedance, and was corrected for in the polarization curves. All potentials are reported relative to the RHE ($V_{RHE} = V_{NHE} + 0.0591 \text{ pH}$ (NHE, normal hydrogen electrode)), calibrated by the H₂ oxidation equilibrium. Electrochemical surface areas (ESCA) of Ag electrocatalysts were measured by Pb underpotential deposition immediately following the rate measurements. In these measurements PbNO₃ was added to make 125 μm Pb²⁺ in 0.1 M NaOH and the cell was deaerated by bubbling Ar for 15-20 min. The electrode was repeatedly held at $0.20~V_{RHE}$ and swept to 0.6 $\rm V_{RHE}$ at 20 $\rm mV\,s^{-1}.$ The stable voltammograms were integrated assuming 280 $\mu C\,cm^{-2}\,Ag.^{[13]}$

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a) H. A. Gasteiger, S. S. Kocha, B. J. Sompalli, Appl. Catal. B 2005, 56, 9;
b) I. E. L. Stephens, A. S. Bondarenko, U. Grønbjerg, J. Rossmeisl, I. Chorkendorff, Energy Environ. Sci. 2012, 5, 6744.

- [2] a) H. A. Gasteiger, P. N. Ross, J. Phys. Chem. 1996, 100, 6715; b) J. Schmidt, V. Stamenkovic, P. N. Ross, Phys. Chem. Chem. Phys. 2003, 5, 400.
- [3] US Geological Survery, Mineral Commodities Summaries 2015 http://minerals.usgs.gov/minerals/pubs/mcs/2015/mcs2015.pdf.
- [4] a) B. B. Blizanac, P. N. Ross, N. M. Markovic, J. Phys. Chem. B 2006, 110, 4735; b) G. K. H. Wiberg, K. J. J. Mayrhofer, M. Arenz, Fuel Cells 2010, 10, 575.
- [5] A. Holewinski, J. Idrobo, S. Linic, Nat. Chem. 2014, 6, 828.
- [6] a) A. Holewinski, S. Linic, J. Electrochem. Soc. 2012, 159, H864; b) H. Xin, A. Holewinski, S. Linic, ACS Catal. 2012, 2, 12; c) H. Xin, A. Holewinski, N. Schweitzer, E. Nikolla, S. Linic, Top. Catal. 2012, 55, 376.
- [7] a) Y. Lu, N. Zhang, L. An, X. Li, D. Xia, J. Power Sources 2013, 240, 606;b) Y. Jin, F. Chen, Y. Lei, X. We, ChemCatChem 2015, 7, 2377.
- [8] a) C. L. Lee, H. P. Chiou, C. M. Syu, C. C. Wu, Electrochem. Commun. 2010, 12, 1609; b) J. Ohyama, Y. Okata, N. Watabe, M. Katagiri, A. Nakamura, H. Arikawa, K. I. Shimizu, T. Takeguchi, W. Ueda, A. Satsuma, J. Power Sources 2014, 245, 998; c) Q. Wang, X. CUI, W. Guan, L. Zhang, X. Fan, Z. Shi, W. Zheng, J. Power Sources 2014, 269, 152; d) Y. Lu, Y. Wang, W. Chen, J. Power Sources 2011, 196, 3033; e) P. Christopher, S. Linic, J. Am. Chem. Soc. 2008, 130, 11264; f) P. Christopher, S. Linic, ChemCatChem 2010, 2, 78; g) S. Linic, P. Christopher, ChemCatChem 2010, 2, 1061.
- [9] a) J. Greeley, J. Rossmeisl, A. Hellman, J. K. Nørskov, Z. Phys. Chem. 2007, 221, 1209; b) P. Singh, D. A. Buttry, J. Phys. Chem. C 2012, 116, 10656.
- [10] a) M. Rycenga, C. M. Cobley, J. Zeng, W. Li, C. Moran, Q. Zhang, D. Qin, Y. Xia, Chem. Rev. 2011, 111, 3669; b) X. Xia, J. Zeng, K. Oetien, L. Qingge, Y. Xia, J. Am. Chem. Soc. 2012, 134, 1793.
- [11] L. S. Taylor, F. W. Langkilde, G. Zografi, J. Pharm. Sci. 2001, 90, 888.
- [12] a) M. Liu, W. Chen, Nanoscale 2013, 5, 12558; b) Y. Lu, W. Chen, J. Power Sources 2012, 197, 107.
- [13] a) E. Herrero, L. J. Buller, H. D. Abruna, Chem. Rev. 2001, 101, 1897; b) E. Kirowa-Eisner, Y. Bonfil, D. Tzur, E. Gileadi, J. Electroanal. Chem. 2003, 552, 171; c) H. Bort, K. Juttner, W. J. Lorenz, J. Electroanal. Chem. 1978, 90, 413.
- [14] V. Viswanathan, H. A. Hansen, J. Rossmeisl, J. K. Nørskov, ACS Catal. 2012. 2, 1654.
- [15] A. Marimuthu, P. Christopher, S. Linic, J. Phys. Chem. C 2012, 116, 9824.

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