

Isomeric Effects of Solution Processed Ladder-Type Non-Fullerene Electron Acceptors

Yongxi Li, Lian Zhong, Jiu-Dong Lin, Fu-Peng Wu, Hai-Jun Bin, Zhanjun Zhang, Lai Xu, Zuo-Quan Jiang, Zhi-Guo Zhang, Feng Liu, Thomas P. Russell, Yongfang Li, Liang-Sheng Liao, and Stephen R. Forrest*

The role of electronic structure and thin film morphology is investigated in determining charge transfer and electron coupling due to orbital interactions in two isomeric non-fullerene acceptors with the structure of acceptor-donor-acceptor. Differences are found in the distribution of electron density of the highest occupied molecular and lowest unoccupied molecular orbitals, whose bonding interactions result in improved intermolecular interactions and hence, molecular stacking. When combined with a large band gap polymer donor, solution-processed organic photovoltaic cells are demonstrated with power conversion efficiencies as high as $10.5 \pm 0.4\%$, and with absorption extending to wavelengths of 800 nm. Due to strong internal organization driven by the planar molecular structure and strong intermolecular interactions, no post-deposition processing such as solvent vapor or thermal annealing is required. To our knowledge, these are the highest efficiencies for as-cast solution-based devices employing non-fullerene acceptors.

Considerable effort has focused on developing NFAs to overcome deficiencies of the commonly used fullerenes, such as their relatively low optical absorption in the visible and near infrared (NIR), a limited ability to modify their ionization and gap energies, and high-cost synthesis and purification processes. The ability, however, to predict device function from the NFA structure is relatively limited due to a lack of understanding of the structure-property relationships of these emerging materials.

Isomeric effects play a significant role in governing electronic structure, molecular stacking and device performance,^[17–21] since they can determine the extent of conjugation along the molecular backbone. For example, naphthodithiophene (NDT) isomers used in organic field effect transistors have both linear- and angular-isomers with distinctly different electronic and crystal structures.^[17]

Furthermore, recent studies have considered a series of isomorphous small molecule donors.^[22–23] While their molecular structures are similar, the symmetrical and asymmetrical configurations of the conjugated skeletons result in large differences in molecular ordering, dipole direction and OPV efficiency.

1. Introduction

The power conversion efficiencies (PCE) of organic photovoltaics (OPVs) have surpassed 12% in single junctions,^[1] in part due to the development of non-fullerene acceptors (NFAs).^[2–16]

Dr. Y. Li, J.-D. Lin, F.-P. Wu, Dr. L. Xu,
Dr. Z.-Q. Jiang, Prof. L.-S. Liao
Jiangsu Key Laboratory for Carbon-Based
Functional Materials & Devices, Institute of
Functional Nano & Soft Materials (FUNSOM),
Soochow University, Suzhou, Jiangsu 215123,
China

Dr. Y. Li, Prof. S. R. Forrest
Department of Electrical Engineering and
Computer Science, University of Michigan, Ann
Arbor, Michigan 48109, USA
E-mail: stevefor@umich.edu

L. Zhong, H.-J. Bin, Z.-G. Zhang, Prof. Y. Li
Beijing National Laboratory for Molecular
Sciences, Institute of Chemistry, Chinese
Academy of Sciences, Beijing 100190, China

L. Zhong, Prof. Z. Zhang, Prof. Y. Li
University of Chinese Academy of Sciences,
Beijing 100049, China

Dr. L. Xu, Dr. Z.-Q. Jiang
Institute of Functional Nano & Soft Materials
(FUNSOM) Soochow University, Suzhou Jiangsu
215123, China E-mail: xulai15@suda.edu.cn;
zqjiang@suda.edu.cn

Prof. F. Liu
Department of Physics and Astronomy, and
Collaborative Innovation Center of IFSA
(CICIFSA), Shanghai Jiaotong University, Shanghai
200240, P. R. China E-mail: fengliu82@sjtu.edu.cn

Prof. F. Liu, Prof. T. P. Russell
Department of Polymer Science and Engineering,
University of Massachusetts, Amherst, MA 01003,
USA

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Here, we reveal the role that electronic structure plays in determining charge transfer and electron coupling due to orbital interactions by comparing the electronic structures both in the solid state and at the molecular levels of two isomers. One of the materials is a thiophene-fused ladder acceptor,^[6,24–32] represented by (4,4,9,9-tetrakis(4-hexylphenyl)-4,9-dihydro-s-indaceno[1,2-b:5,6-b']dithieno[3,2-b]thiophene-2,7-diyl) bis(2-(3-oxo-2,3-dihydroinden-1-ylidene)malononitrile) (IT-IC),^[33] which has attracted interest in view of its suitability for use in OPVs due to its visible light absorption and high charge carrier mobility.^[1,3,34–41] The core unit of IT-IC, indacenodithieno[3,2-b]thiophene (IDTT) is a highly planar conjugated structure. A similar ladder type molecule: dithienyl-[1,2-b:4,5-b']benzodithiophene (BDCPDT Scheme 1) that is isoelectronic with IDTT has also been predicted to be a promising core for organic semiconductors.^[18,42,43] We compare the electronic and morphological properties of IT-IC with a planar BDCPDT-based acceptor–donor–acceptor (a–d–a) NFA (BDT-IC, Scheme 1b, the details of synthesis see Supporting Information). The core structures of BDCPDT and IDTT are isomeric with four, linearly fused thiophenes, two cyclopentadienes and one benzene ring. We find differences in the distribution of electron density of the highest occupied molecular (HOMO) and lowest unoccupied molecular orbitals (LUMO) of IT-IC and BDT-IC. Antibonding interactions between atomic orbitals are found in IT-IC, which prevent substantial overlap between neighboring molecules, thereby resulting in disorder in the solid state. In contrast, BDT-IC exhibits a higher electron density and more compact crystal packing, leading to stronger inter- and intramolecular charge transfer with a reduced energy gap. Moreover, the improved order of BDT-IC combined with the large band gap polymer donor poly[2,6-(4,8-bis (5-(tripropylsilyl)thiophen-2-yl)benzo[1,2-b:4,5-b']dithiophene-alt-4,7-Bis(5-thiophen-2-yl)-2-(2-hexyldecyl)-5,6-difluoro-2H-benzo[d]-[1,2,3]triazole)](J71) result in OPVs with power conversion efficiencies of PCE = 10.5 ± 0.4%, which to our knowledge are the highest values for as-cast solution-based devices.

We point out that recently, Chen et al.^[43] also reported a high efficiency solar cell based on BDT-IC after solvent vapor annealing. In our work, the significant impacts of isomeric effects on electronic structures and crystal properties of the IT-IC and BDT-IC isomers are revealed, and the structure-property relationships thus presented improve our understanding of ladder-type structures.

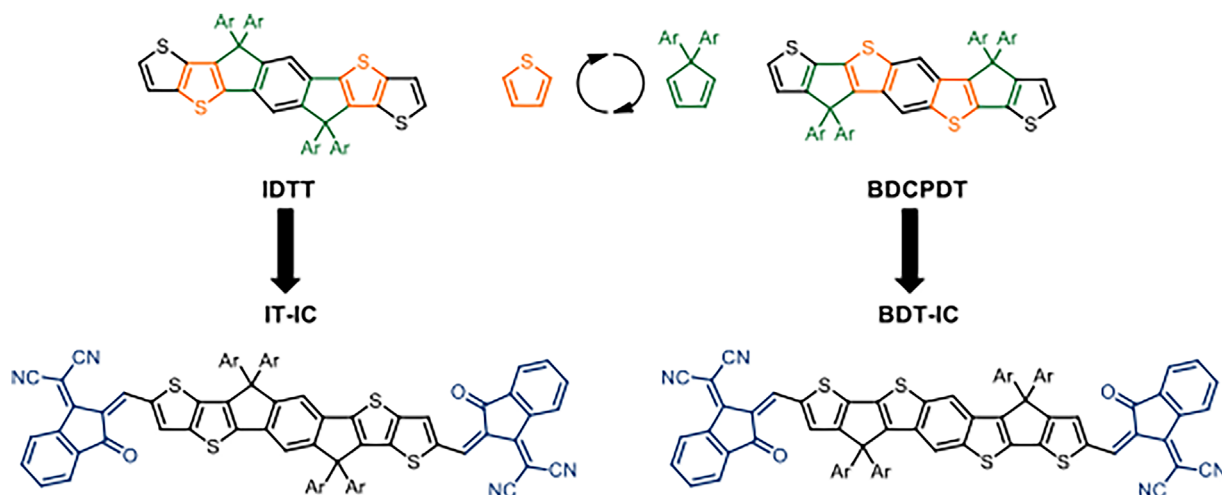
2. Results

2.1. Chemical/Thermal Properties

The chemical structural formulae of J71, IT-IC, and BDT-IC are shown in Figure 1a. BDT-IC is soluble in dichloromethane (DCM), chloroform (CF), chlorobenzene (CB), and *ortho*-dichlorobenzene (*o*-DCB), all at room temperature. Moreover, BDT-IC shows relatively good solubility in halogen-free solvents such as tetrahydrofuran (THF), *o*-xylene and 1,2,4-trimethylbenzene, which is beneficial for roll-to-roll printing of large-area modules due to the need for less toxic solvents. Thermal properties were investigated by thermal gravimetric analysis performed under nitrogen. BDT-IC has a decomposition temperature (5% weight loss) of 325 °C (Figure S1 in Supporting Information).

2.2. Photo-Physical and Electrochemical Properties

Thin film UV-Vis absorption spectra of J71, IT-IC and BDT-IC are shown in Figure 1b. BDT-IC has an absorption band spanning wavelengths between $\lambda = 550$ and 750 nm in dilute DCM solution, with a maximum extinction coefficient of $1.4 \times 10^5 \text{ M}^{-1} \text{ cm}^{-1}$ at $\lambda = 693$ nm (Figure S2 in Supporting Information). This is red-shifted and slightly higher compared to IT-IC ($1.3 \times 10^5 \text{ M}^{-1} \text{ cm}^{-1}$ at 664 nm^[33]). The red-shifted absorption is attributed to the increased internal charge transfer by the introduction of a stronger electron-donating moiety of the



Scheme 1. Molecular design of the BDCPDT-based non-fullerene acceptor, BDT-IC. BDCPDT is isoelectronic with IDTT by re-organizing the alignment of thiophene and cyclopentadiene units.

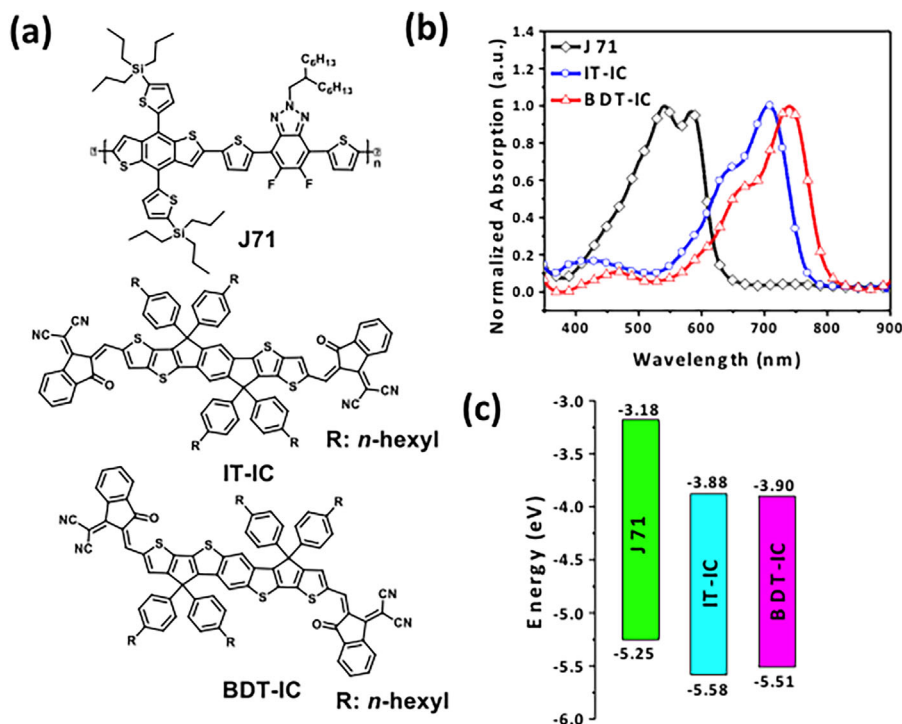


Figure 1. (a) Molecular structural formulae of J71, IT-IC, and BDT-IC; (b) Normalized UV-Vis absorption spectra of J71, IT-IC, and BDT-IC thin films; (c) Energy diagram of J71, IT-IC, and BDT-IC relative to vacuum obtained from cyclic voltammetry measurements.

BDCPDT unit in this a–d–a type electron acceptor. Notably, the thin film absorption spectrum of BDT-IC is red-shifted by 40 nm, which may be due to aggregation. The optical energy gap (E_g) of BDT-IC extracted from its long wavelength absorption edge in the solid-state is 1.53 ± 0.02 eV (Table 1), which is 0.06 eV smaller than for IT-IC (1.59 ± 0.02 eV). The absorption edge of J71 is located at 640 nm.

The energetics of J71, BDT-IC, and IT-IC were investigated using cyclic voltammetry to obtain HOMO and LUMO energies (E_{HOMO} and E_{LUMO} , respectively) listed in Table 1, with voltammetry data in Figure S3 in Supporting Information. As shown in Figure 1c, BDT-IC exhibits a higher HOMO energy but almost the same LUMO energy compared with IT-IC, implying that changes in the a–d–a molecular backbone affects E_{HOMO} but not E_{LUMO} . This allows for an increased cutoff wavelength without reducing the open circuit voltage (V_{oc}) in OPVs.

2.3. Molecular Orientation and Crystallinity

Grazing incidence X-ray diffraction (GIXD) was used to investigate the influence of isomerization on the thin film morphology (see in Figure 2). It is clear that the IT-IC thin film is disordered, with only weak diffraction peaks at 0.34 and 1.77 \AA^{-1} (summarized in Table 2). IT-IC diffraction shows the (100) peak in the in-plane (IP) direction at 0.34 \AA^{-1} (corresponding to an inter-planar spacing of $d_{100} = 18.5 \pm 0.4 \text{ \AA}$) and the (010) plane is in the out-of-substrate-plane (OOP) direction at 1.77 \AA^{-1} ($d_{010} = 3.5 \pm 0.1 \text{ \AA}$). Due to considerable disorder of IT-IC spin-cast films, we were unable to accurately estimate the crystal coherence length. On the other hand, The BDT-IC showed a narrow diffraction peak at 1.81 \AA^{-1} in the perpendicular direction, corresponding to ordered π -stacking at $3.4 \pm 0.1 \text{ \AA}$, and a crystal coherence length of $\zeta = 4.5 \pm 0.2 \text{ nm}$. Furthermore,

Table 1. Optical and electrochemical properties of non-fullerene acceptors.

Material	$\lambda_{\text{max}}^{\text{a}}$ (nm)	E_g^{b} (eV)	HOMO ^c (eV)	LUMO ^c (eV)	μ_e^{d} ($\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$)	λ^{e} (meV)
IT-IC	709	1.59	−5.58	−3.88	0.1	94
BDT-IC	739	1.53	−5.51	−3.90	0.5	93

^aThin film measurement.

^bOptical energy gap.

^cFrom cyclic voltammetry measurement.

^dFrom OTFT measurement.

^eReorganization energy.

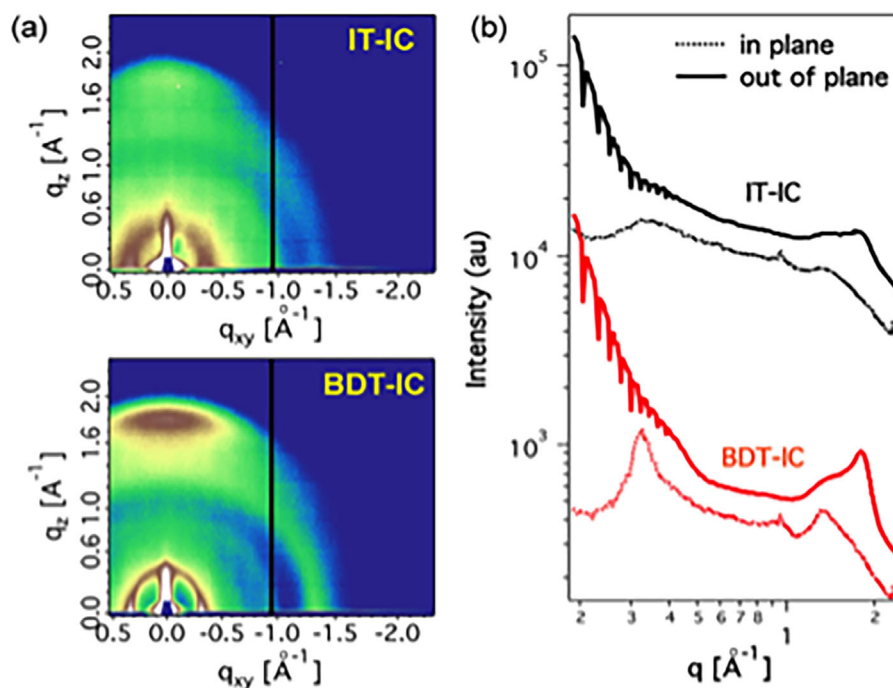


Figure 2. (a) Grazing incidence X-ray diffraction (GIXD) patterns of as-cast IT-IC and BDT-IC films; (b) In-plane (dotted line) and out-of-plane (solid line) X-ray scattering patterns extracted from the 2D GIXD images. Here, q is the scattering vector, q_{xy} is the in-plane direction and q_z is the out-of-plane direction.

the (100) peak at 0.32 \AA^{-1} was observed in the IP direction, corresponding to a lamellar distance of $19.6 \pm 0.5 \text{ \AA}$ and $\zeta = 22.0 \pm 0.6 \text{ nm}$. Therefore, we conclude that configurational isomerization rotates the molecule from upstanding to flat-lying on the substrate with increased crystallite size.

2.4. Organic Field Effect Transistor (OFET) Device Performance

Top-gate/bottom-source and drain-contact thin film field effect transistors (TFTs) with BDT-IC and IT-IC channels were used to evaluate the charge transport behavior. Both acceptors exhibited n-type transport (Figure S4 in Supporting Information) with well-defined linear and saturation regimes. The transfer

characteristics show high off-current at a large drain voltage due to the ambipolar charge transport properties of these two acceptors. The mobilities in the saturation regimes were 0.46 ± 0.03 and $0.11 \pm 0.01 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for BDT-IC and IT-IC, respectively. The different electron mobilities are attributed to the different electronic structures and crystal packing habits in the thin films.

2.5. Organic Solar Cell Performance

The performance of OPVs based on IT-IC and BDT-IC combined with a polymer donor (J71) are summarized in Table 3. The devices were fabricated from chloroform solution with the following structures: ITO/PEDOT:PSS (40 nm)/J71:BDT-IC or IT-IC

Table 2. Grazing incidence X-ray diffraction results of J71, IT-IC, and BDT-IC in neat and blend films.

Active layer	$q_{(100)}$ (\AA^{-1})	$d_{(100)}$ (\AA)	$q_{(010)}$ (\AA^{-1})	$d_{(010)}$ (\AA)	$\zeta_{(100)}$ (nm)	$\zeta_{(010)}$ (nm)
IT-IC	0.34	18.5 ± 0.4	1.77	3.5 ± 0.1	–	–
BDT-IC	0.32	19.6 ± 0.5	1.81	3.4 ± 0.1	22.0 ± 0.6	4.5 ± 0.2
J71	0.29	21.6 ± 0.5	1.67	3.8 ± 0.1	7.0 ± 0.2	2.9 ± 0.1
J71:IT-IC ^a	–	–	1.62	3.88	–	–
J71:BDT-IC	0.32	19.6 ± 0.5	1.76	3.5 ± 0.1	–	–

^a Results from ref.[36].

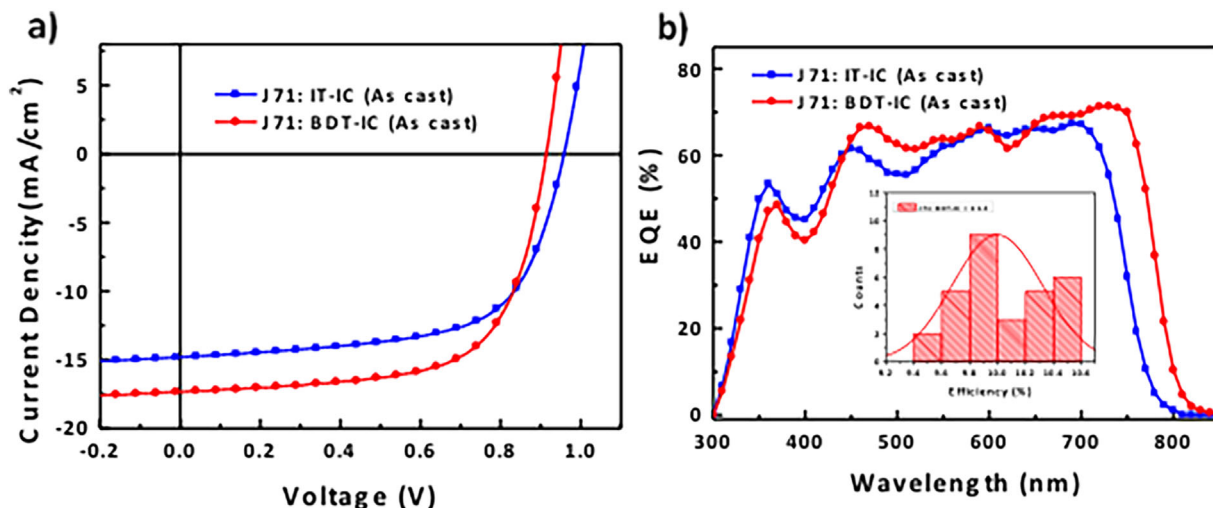


Figure 3. (a) Current-density-voltage characteristics, and (b) External quantum efficiency (EQE) spectra of the organic photovoltaic cells based on J71: IT-IC (1:1, w/w) and J71: BDT-IC (1:1.2, w/w). Inset: Histogram of efficiencies of organic photovoltaic cells (among 30 devices) based on J71: BDT-IC (1:1.2, w/w).

(100 nm) /PDINO (10 nm)/Al (100 nm). Here, PDINO (perylene-diimide functionalized with amino *N*-oxide) is the cathode buffer exciton blocking layer.^[44] The details of fabrication are found in Methods (Supporting Information). The optimized devices for J71: BDT-IC were spin-coated from chloroform solution with a 1:1.2 donor:acceptor (D:A) weight ratio. The optimized device had $PCE = 10.5 \pm 0.4\%$ with $V_{oc} = 0.92 \pm 0.01$ V, short circuit current $J_{sc} = 17.3 \pm 0.4$ mA cm⁻², and fill factor $FF = 0.66 \pm 0.03$ at 1 sun intensity, simulated AM1.5G illumination (see Table 3). The

efficiency histogram in **Figure 3** shows >70% of the devices (among 30 devices) have PCE higher than 9.8%. In contrast, the as-cast devices based on IT-IC exhibited $PCE = 9.0 \pm 0.3\%$ with $V_{oc} = 0.96 \pm 0.01$ V, $J_{sc} = 14.8 \pm 0.5$ mA cm⁻² and $FF = 0.64 \pm 0.03$.

The EQE versus wavelength for the ladder-type molecules are provided in Figure 3b. The long wavelength cut off of the BDT-IC based OPV is at $\lambda = 810$ nm, which is red-shifted by ~40 nm compared to the IT-IC based device. The EQE of the BDT-IC based device reaches 70%, between 450 nm and

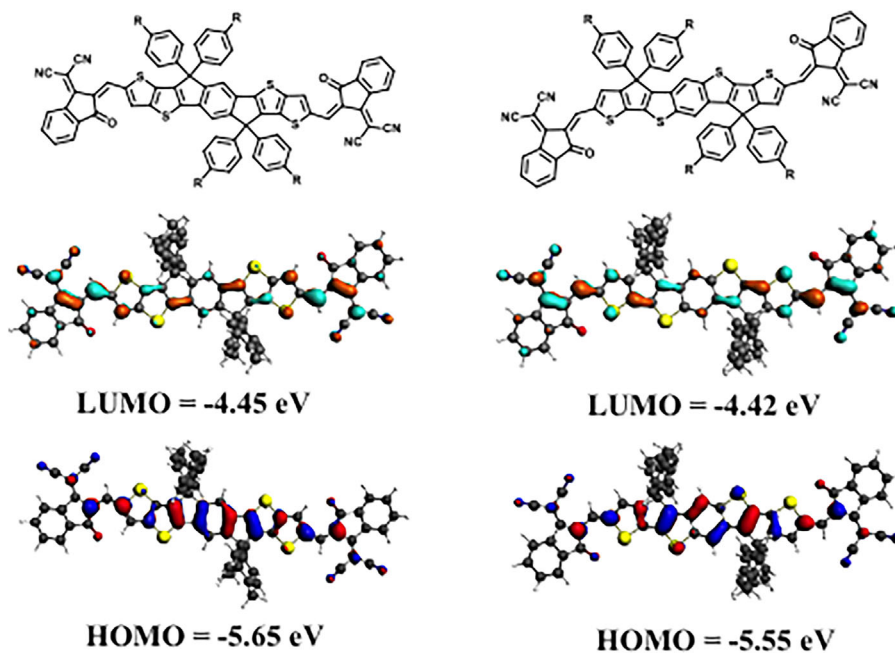


Figure 4. Calculated highest occupied molecular orbitals (HOMOs) and lowest unoccupied molecular orbitals (LUMOs) of IT-IC (left) and BDT-IC (right) from density functional theory using the TZP basis set at the GGA:PW91 level. Blue and red regions show the HOMOs along the of IT-IC and BDT-IC, the green and orange regions show the LUMOs along the of IT-IC and BDT-IC.

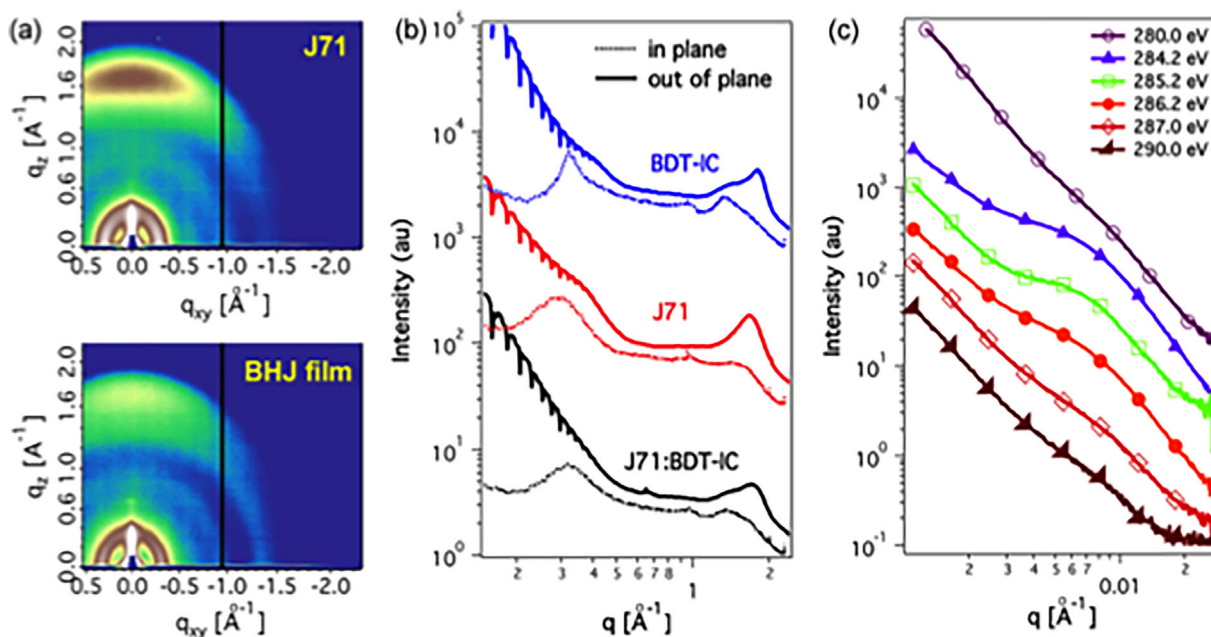


Figure 5. (a) Grazing incidence X-ray diffraction patterns; (b) patterns taken along the lines in (a), and (c) resonant soft x-ray scattering of J71, BDT-IC and J71:BDT-IC blends. Here, q is the scattering vector, q_{xy} is the in plane direction and q_z is the out of plane direction.

780 nm. The J_{sc} inferred from the integrated EQE is 16.5 mA cm^{-2} , which is within 4% of that measured from the simulated 1 sun, AM1.G solar illuminated photovoltaic J - V measurement.

3. Discussion

The different positions of cyclopentadiene result in crystalline morphologies that lead to differences in the charge mobility. We also speculate that the distributions of the electron densities of the HOMO and LUMO play an important role in differentiating their charge transport properties. Density functional theory (DFT) calculations at the GGA:PW91/TZP level were performed to investigate the geometric and electronic properties of IT-IC and BDT-IC.^[45] Interestingly, as shown in **Figure 4**, the distribution of electron densities of the HOMOs and LUMOs for IT-IC and BDT-IC are different. Previous studies have found that the larger atomic radius of sulfur (S) compared to carbon (C) results in increased intermolecular interactions in ladder-type molecules.^[17,46] As a result of antibonding interactions between atomic orbitals, the electron densities on the S atoms in IT-IC are

also small. On the other hand, the large electron densities on the S atoms in both the HOMO and LUMO levels of the BDT-IC provide substantial overlap between neighboring molecules in the solid state, giving rise to increased crystallinity, and thus improved charge carrier transport. This agrees with the observation that the reduced π -stacking distance of BDT-IC results in a five times higher electron mobility than IT-IC. We expect that the differences in the electronic structures of these two molecules originate from their fused-ring frameworks. Compared to the thieno[3,2-b]thiophene in IT-IC, it is easier for the S in the cyclopentadienedithiophene unit in BDT-IC to supply two π -electrons, resulting in a comparatively high electron density. Also, BDT-IC has a planar structure with torsion angles $< 2^\circ$ that facilitate π -electron delocalization.

The blends of J71:IT-IC and J71:BDT-IC were also characterized by GIXD, as shown in **Figure 5**. J71 in the pure film showed a prominent (100) diffraction peak at 0.29 \AA^{-1} in the IP direction. The lamellar distance is $21.6 \pm 0.5 \text{ \AA}$ and $\zeta = 7.0 \pm 0.2 \text{ nm}$ (Table 3). An intense π -stacking diffraction peak is seen in the OOP direction located at 1.67 \AA^{-1} , giving a π -stacking distance of $3.8 \pm 0.1 \text{ \AA}$ nm with $\zeta = 2.9 \pm 0.1 \text{ nm}$. In J71:BDT-IC blend thin films, the (100) diffraction peak is dominated by contributions from BDT-IC since

Table 3. Photovoltaic parameters of the as-cast OPVs based on J71 as a donor and IT-IC/BDT-IC as an acceptor under AM 1.5G, 100 mW cm^{-2} illumination.

Active layer ^a	V_{oc} (V)	J_{sc} (mA cm^{-2})	FF (%)	PCE ^b (%)
J71:IT-IC	0.96 ± 0.01	14.8 ± 0.5	63.6 ± 3.4	9.0 ± 0.3
J71:BDT-IC	0.92 ± 0.01	17.3 ± 0.4	65.5 ± 2.9	10.5 ± 0.4

^a As deposited value.

^b Values are average PCEs from 30 devices.

the (100) diffraction peak is located at 0.32 \AA^{-1} . In this case, the presence of BDT-IC guides the blend morphology. Furthermore, The BT-CIC shows improved π -stacking, as seen from the (010) diffraction peak at 1.76 \AA^{-1} in the perpendicular direction, which indicates a reduced π -stacking distance compared to the J71:IT-IC film (1.62 \AA^{-1}).^[36] Recently, studies by Ade and co-workers predicted that face-on orientation in the blend film is a general feature of high-performance bulk heterojunction organic solar cells due to the reduction of recombination of excitons during charge separation.^[47] The combination of the preferred face-on orientation and the tight π -stacking of the BDT-IC blends, can improve exciton dissociation and/or charge transport, thus enhancing performance.

From Table 3, the improvement of J_{sc} in the BDT-IC versus the IT-IC device is attributed to the red-shifted absorption that provides solar spectral response into the near-infrared (NIR). The length scale of phase separation of J71:BDT-IC was studied using resonant soft x-ray diffraction. A well-defined phase separation at 0.007 \AA^{-1} corresponding to a distance of 90 nm is observed (see Figure 5c). Such a large length scale in non-fullerene-based solar cells gives rise to a high J_{sc} . This result is also consistent with transmission electron microscopy and atomic force microscopy (AFM) images of the blends (see Figure S5). Fiber-like features are observed in the topographic image of the J71:BDT-IC blend with larger root-mean-square (RMS) roughness of 1.13 than J71:IT-IC blend (0.53). The roughness of the blend results in increased contact area between the active and interfacial layers, which may result in efficient exciton dissociation at the interface. We also observe an increased FF in the J71:BDT-IC based device. The hole and electron mobilities of the blend were estimated using the space charge limited current (SCLC) method (see Supporting Information) to be $1.13 \pm 0.06 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ and $1.90 \pm 0.09 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, respectively for J71:BDT-IC, corresponding to nearly balanced charge transport ($\mu_e/\mu_h = 1.7$).

Many high-performance solution-processed solar cells require often-complicated post-cast treatments or additives that can lead to irreproducibility and increased cost.^[39] However, the BDT-IC-based solar cells can achieve high performance without these treatments. Typically, for fullerene-based OPVs, additives or post-cast treatments are used to increase polymer crystallinity while reducing fullerene domain size.^[48] Since ladder-type NFAs have a high solubility in organic reagents compared with fullerenes, the blends can be uniform. However, there may be a residual phase separation when mixed with the donors, which reduces the charge separation efficiency. One approach to mitigate this issue is increasing the NFA crystallinity. This is why BDT-IC provides improved OPV performance compared with IT-IC without post-cast treatments or additives.

4. Conclusions

Two isomeric non-fullerene acceptors featuring a large fused ring ladder-type structural motif were designed and synthesized for OPVs. The isomer BDT-IC shows a smaller energy gap, stronger intermolecular interactions and more crystallinity than IT-IC. Specifically, BDT-IC exhibited five times higher electron mobility, which leads to a reduced charge recombination.

Moreover, as-cast OPVs without additional treatments or additives employing BDT-IC paired with the J71 donor exhibited $\text{PCE} = 10.5 \pm 0.4\%$. The significant impacts of isomeric effects on electronic structure and crystal properties of the IT-IC and BDT-IC isomers are revealed. These studies help to elucidate the structure-property relationships of high performance NFAs, providing insights for the future rational design of improved non-fullerene acceptor materials.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflicts of Interest

The author(s) declare no competing interest.

Keywords

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- [1] S. Li, L. Ye, W. Zhao, S. Zhang, S. Mukherjee, H. Ade, J. Hou, *Adv. Mater.* **2016**, *28*, 9423.
- [2] W. Zhao, D. Qian, S. Zhang, S. Li, O. Inganäs, F. Gao, J. Hou, *Adv. Mater.* **2016**, *28*, 4734.
- [3] Y. Yang, Z.-G. Zhang, H. Bin, S. Chen, L. Gao, L. Xue, C. Yang, Y. Li, *J. Am. Chem. Soc.* **2016**, *138*, 15011.
- [4] D. Meng, H. Fu, C. Xiao, X. Meng, T. Winands, W. Ma, W. Wei, B. Fan, L. Huo, N. L. Doltsinis, Y. Li, Y. Sun, Z. Wang, *J. Am. Chem. Soc.* **2016**, *138*, 10184.
- [5] Y. Li, L. Zhong, B. Gautam, H.-J. Bin, J.-D. Lin, F.-P. Wu, Z. Zhang, Z.-Q. Jiang, Z.-G. Zhang, K. Gundogdu, Y. Li, L.-S. Liao, *Energy Environ. Sci.* **2017**, *10*, 1610.
- [6] Y. Li, L. Zhong, F.-P. Wu, Y. Yuan, H.-J. Bin, Z.-Q. Jiang, Z. Zhang, Z.-G. Zhang, Y. Li, L.-S. Liao, *Energy Environ. Sci.* **2016**, *9*, 3429.

- [7] S. X. Li, W. Q. Liu, M. M. Shi, J. Q. Mai, T. K. Lau, J. H. Wan, X. H. Lu, C. Z. Li, H. Z. Chen, *Energy Environ. Sci.* **2016**, *9*, 604.
- [8] D. Baran, T. Kirchartz, S. Wheeler, S. Dimitrov, M. Abdelsamie, J. Gorman, R. S. Ashraf, S. Holliday, A. Wadsworth, N. Gasparini, P. Kaienburg, H. Yan, A. Amassian, C. J. Brabec, J. R. Durrant, I. McCulloch, *Energy Environ. Sci.* **2016**, *9*, 3783.
- [9] Y. Zhong, M. T. Trinh, R. Chen, G. E. Purdum, P. P. Khlyabich, M. Sezen, S. Oh, H. Zhu, B. Fowler, B. Zhang, W. Wang, C. Y. Nam, M. Y. Sfeir, C. T. Black, M. L. Steigerwald, Y. L. Loo, F. Ng, X. Y. Zhu, C. Nuckolls, *Nat. Commun.* **2015**, *6*, 8242.
- [10] C. B. Nielsen, S. Holliday, H.-Y. Chen, S. J. Cryer, I. McCulloch, *Acc. Chem. Res.* **2015**, *48*, 2803.
- [11] Y. Lin, X. Zhan, *Mater. Horiz.* **2014**, *1*, 470.
- [12] G. Zhang, G. Yang, H. Yan, J.-H. Kim, H. Ade, W. Wu, X. Xu, Y. Duan, Q. Peng, *Adv. Mater.* **2017**, *29*, 1606054.
- [13] J. Zhao, Y. Li, G. Yang, K. Jiang, H. Lin, H. Ade, W. Ma, H. Yan, *Nat. Energy* **2016**, *1*, 15027.
- [14] O. K. Kwon, M. A. Uddin, J.-H. Park, S. K. Park, T. L. Nguyen, H. Y. Woo, S. Y. Park, *Adv. Mater.* **2016**, *28*, 910.
- [15] W. Chen, Q. Zhang, *J. Mater. Chem. C* **2017**, *5*, 1275.
- [16] Y. Zang, C.-Z. Li, C.-C. Chueh, S. T. Williams, W. Jiang, Z.-H. Wang, J.-S. Yu, A. K. Y. Jen, *Adv. Mater.* **2014**, *26*, 5708.
- [17] S. Shinamura, I. Osaka, E. Miyazaki, A. Nakao, M. Yamagishi, J. Takeya, K. Takimiya, *J. Am. Chem. Soc.* **2011**, *133*, 5024.
- [18] Y.-L. Chen, J.-Y. Hsu, F.-Y. Lin, Y.-Y. Lai, H.-C. Chou, Y.-J. Cheng, *J. Org. Chem.* **2016**, *81*, 2534.
- [19] I. Osaka, T. Abe, S. Shinamura, K. Takimiya, *J. Am. Chem. Soc.* **2011**, *133*, 6852.
- [20] S. W. Cheng, D. Y. Chiou, C. E. Tsai, W. W. Liang, Y. Y. Lai, J. Y. Hsu, C. S. Hsu, I. Osaka, K. Takimiya, Y. J. Cheng, *Adv. Funct. Mater.* **2015**, *25*, 6131.
- [21] J. Lee, H.-F. Chen, T. Batagoda, C. Coburn, P. I. Djurovich, M. E. Thompson, S. R. Forrest, *Nat. Mater.* **2016**, *15*, 92.
- [22] C. J. Takacs, Y. Sun, G. C. Welch, L. A. Perez, X. Liu, W. Wen, G. C. Bazan, A. J. Heeger, *J. Am. Chem. Soc.* **2012**, *134*, 16597.
- [23] X. Che, C.-L. Chung, X. Liu, S.-H. Chou, Y.-H. Liu, K.-T. Wong, S. R. Forrest, *Adv. Mater.* **2016**, *28*, 8248.
- [24] Y. Li, X. Liu, F.-P. Wu, Y. Zhou, Z.-Q. Jiang, B. Song, Y. Xia, Z.-G. Zhang, F. Gao, O. Inganas, Y. Li, L.-S. Liao, *J. Mater. Chem. A* **2016**, *4*, 5890.
- [25] Y. Li, D. Qian, L. Zhong, J. D. Lin, Z. Q. Jiang, Z. G. Zhang, Z. Zhang, Y. Li, L. S. Liao, F. Zhang, *Nano Energy* **2016**, *27*, 430.
- [26] Y. Lin, Q. He, F. Zhao, L. Huo, J. Mai, X. Lu, C.-J. Su, T. Li, J. Wang, J. Zhu, *J. Am. Chem. Soc.* **2016**, *138*, 2973.
- [27] H. Yao, Y. Cui, R. Yu, B. Gao, H. Zhang, J. Hou, *Angew. Chem. Int. Ed.* **2017**, *56*, 3045.
- [28] S. Dai, F. Zhao, Q. Zhang, T.-K. Lau, T. Li, K. Liu, Q. Ling, C. Wang, X. Lu, W. You, *J. Am. Chem. Soc.* **2017**, *139*, 1336.
- [29] F. Liu, Z. Zhou, C. Zhang, T. Vergote, H. Fan, F. Liu, X. Zhu, *J. Am. Chem. Soc.* **2016**, *138*, 15523.
- [30] F. Liu, Z. Zhou, C. Zhang, J. Zhang, Q. Hu, T. Vergote, T. P. Russell, X. Zhu, *Adv. Mater.* **2017**, *29*, 1606574.
- [31] N. Qiu, H. Zhang, X. Wan, C. Li, X. Ke, H. Feng, B. Kan, H. Zhang, Q. Zhang, Y. Lu, *Adv. Mater.* **2017**, *29*, 4964.
- [32] H. Yao, Y. Chen, Y. Qin, R. Yu, Y. Cui, B. Yang, S. Li, K. Zhang, J. Hou, *Adv. Mater.* **2016**, *28*, 8283.
- [33] Y. Lin, J. Wang, Z.-G. Zhang, H. Bai, Y. Li, D. Zhu, X. Zhan, *Adv. Mater.* **2015**, *27*, 1170.
- [34] L. Gao, Z. G. Zhang, H. Bin, L. Xue, Y. Yang, C. Wang, F. Liu, T. P. Russell, Y. Li, *Adv. Mater.* **2016**, *28*, 8288.
- [35] H. Bin, Z.-G. Zhang, L. Gao, S. Chen, L. Zhong, L. Xue, C. Yang, Y. Li, *J. Am. Chem. Soc.* **2016**, *138*, 4657.
- [36] H. Bin, L. Gao, Z.-G. Zhang, Y. Yang, Y. Zhang, C. Zhang, S. Chen, L. Xue, C. Yang, M. Xiao, Y. Li, *Nat. Commun.* **2016**, *7*, 13651.
- [37] F. Yang, C. Li, W. Lai, A. Zhang, H. Huang, W. Li, *Mater. Chem. Front.* **2017**, *1*, 1389.
- [38] Y. Lin, F. Zhao, Q. He, L. Huo, Y. Wu, T. C. Parker, W. Ma, Y. Sun, C. Wang, D. Zhu, A. J. Heeger, S. R. Marder, X. Zhan, *J. Am. Chem. Soc.* **2016**, *138*, 4955.
- [39] Y. Lin, Q. He, F. Zhao, L. Huo, J. Mai, X. Lu, C.-J. Su, T. Li, J. Wang, J. Zhu, Y. Sun, C. Wang, X. Zhan, *J. Am. Chem. Soc.* **2016**, *138*, 2973.
- [40] H. Yao, L. Ye, J. Hou, B. Jang, G. Han, Y. Cui, G. M. Su, C. Wang, B. Gao, R. Yu, H. Zhang, Y. Yi, H. Y. Woo, H. Ade, J. Hou, *Adv. Mater.* **2017**, *29*, 1700254.
- [41] Y.-X. Xu, C.-C. Chueh, H.-L. Yip, F.-Z. Ding, Y.-X. Li, C.-Z. Li, X. Li, W.-C. Chen, A. K. Y. Jen, *Adv. Mater.* **2012**, *24*, 6356.
- [42] Y.-L. Chen, C.-Y. Chang, Y.-J. Cheng, C.-S. Hsu, *Chem. Mater.* **2012**, *24*, 3964.
- [43] B. Kan, H. Feng, X. Wan, F. Liu, X. Ke, Y. Wang, Y. Wang, H. Zhang, C. Li, J. Hou, Y. Chen, *J. Am. Chem. Soc.* **2017**, *139*, 4929.
- [44] Z.-G. Zhang, B. Qi, Z. Jin, D. Chi, Z. Qi, Y. Li, J. Wang, *Energy Environ. Sci.* **2014**, *7*, 1966.
- [45] G. t. Te Velde, F. M. Bickelhaupt, E. J. Baerends, C. Fonseca Guerra, S. J. van Gisbergen, J. G. Snijders, T. Ziegler, *J. Comput. Chem.* **2001**, *22*, 931.
- [46] K. Takimiya, S. Shinamura, I. Osaka, E. Miyazaki, *Adv. Mater.* **2011**, *23*, 4347.
- [47] J. R. Tumbleston, B. A. Collins, L. Yang, A. C. Stuart, E. Gann, W. Ma, W. You, H. Ade, *Nat. Photon.* **2014**, *8*, 385.
- [48] Z. Li, K. Jiang, G. Yang, J. Y. Lai, T. Ma, J. Zhao, W. Ma, H. Yan, *Nat. Commun.* **2016**, *7*, 13094.