
Monodispersed bioactive glass nanoclusters with ultra-large pores and intrinsic exceptionally high miRNA loading for efficiently enhancing bone regeneration

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This is the author manuscript accepted for publication and has undergone full peer review but has not been through the copyediting, typesetting, pagination and proofreading process, which may lead to differences between this version and the [Version of Record](#). Please cite this article as [doi: 10.1002/advs.201700630](https://doi.org/10.1002/advs.201700630).

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Abstract

Bioactive glass nanoparticles (BGNs) have attracted much attention in drug delivery and bone tissue regeneration, due to the advantages including biodegradation, high bone-bonding bioactivity and facile large-scale fabrication. However, the wide biomedical applications of BGNs such as efficient gene delivery are limited due to their poor pore structure and easy aggregation. Herein, for the first time, we report novel monodispersed bioactive glass nanoclusters (BGNCs) with ultra-large mesopores (10-30 nm) and excellent miRNA delivery for accelerating critical-sized bone regeneration. BGNCs with different size (100-500 nm) were fabricated by using a branched polyethylenimine as the structure director and catalyst. BGNCs showed an excellent apatite-forming ability and high biocompatibility. Importantly, BGNCs demonstrated an almost 19 times higher miRNA loading than those of conventional BGNs. Additionally, BGNCs-miRNA nanocomplexes exhibited a significantly high anti-enzymolysis, enhanced cellular uptake and miRNA transfection efficiency, overpassing BGNs and commercial Lipofectamine 3000. BGNCs-mediated miRNA delivery significantly improved the osteogenic differentiation of bone marrow stromal stem cells (BMSCs) *in vitro* and efficiently enhanced bone formation *in vivo*. BGNCs could be a highly efficient non-viral vector for various gene therapy applications. Our study may provide a novel strategy to develop

highly gene-activated bioactive nanomaterials for simultaneous tissue regeneration and disease therapy.

Keywords: Silica-based biomaterials; Bioactive glass; Nanoclusters; miRNA delivery; Bone regeneration

1 Introduction

The traumas and diseases resulted bone tissue loss has become huge social problems seriously affecting the human quality of life.^[1] Although bone tissue possesses the self-repair ability, the large critical-size bone defect cannot regenerate without bone grafting and 5%-10% fractures show nonunion or delayed healing.^[2] As the non-coding RNAs (~22nt), microRNAs (miRNAs) can regulate the fundamental physiological process in various organisms through degrading and repressing the expressions of target RNAs.^[3] In recent years, various miRNAs have shown the capacity to positively regulate the osteogenic genes expressions and osteogenesis *in vitro* and *in vivo* through multiple signaling pathways.^[4] Unfortunately, naked miRNAs are difficult to enter cells and also easy to be degraded by the nuclease under physiological environment. Different gene vectors have been synthesized and used to deliver miRNA for enhanced bone tissue formation.^[5]

Viral vectors exhibited good miRNA transfection efficiency but also showed some negative effects such as infection and immunogenicity.^[6] To avoid the problems of viral vectors, non-viral vectors including cationic polymers and liposomes were widely investigated for miRNA delivery *in vitro* and *in vivo*.^[7-9] However, the miRNA delivery of cationic polymers and liposomes was always

unsatisfactory due to the cytotoxicity, low transfection efficiency and serum stability.^[10] It is

necessary to develop next generation non-viral vectors with good osteogenic biocompatibility, high miRNA loading and protecting ability, excellent miRNA delivery efficiency.

As compared to conventional inorganic biomaterials, bioactive glasses (BGs) have inspired intensive research interest in bone tissue regeneration due to their remarkable osteoblast biocompatibility, bioactivity including the chemical bone-bonding ability and gene activation properties.^[11-16] Therefore, BG nanoparticles (BGNs) have a great potential to be employed as novel bioactive non-viral gene vectors for tissue regeneration and gene therapy. Up to now, few studies reported the gene delivery applications based on BGNs. For instance, Kim used amine-functionalized mesoporous BGNs to deliver small interfere RNA (siRNA) and 5.18% loading efficiency was obtained.^[17] Chen and coworkers developed an amine-modified radial mesoporous BGNs which showed good DNA delivery efficiency and protection ability.^[18] Although some positive results of BGNs in DNA and siRNA delivery were achieved, the amine-functionalization was necessary to make BGNs positively charged for loading genes. The limited surface positive charge density usually resulted in the low loading and poor transfection efficiency. Additionally, the charged surface of nanoparticles may also cause the non-specific protein adsorption and cytotoxicity *in vivo* applications.^[19]

Recently our group demonstrated that BGNs could bind miRNA directly without any surface cationic modification and release them in cellular environment, suggesting the great potential of

BGNs as the next generation non-viral vectors for enhanced gene therapy.^[20] Through the strong interaction between calcium in BGNs and phosphates in miRNA, BGNs showed a significantly high miRNA loading and intracellular transfection efficiency, as compared to conventional silica nanoparticles.^[20] Even so, as an ideal gene vector, the small mesopore size (<5 nm) and pore volume (<0.04 m³/g) of BGNs was still not satisfied in biomedical applications.^[21, 22] It is very worthwhile to develop BGNs with the novel nanostructure for enhanced gene delivery applications.

Branched polyethyleneimine (PEI) is a common functional macromolecule that usually used for gene delivery, but is rarely employed as template to direct the formation of nanostructure.^[23] Based on the fact that the abundant amino in PEI can catalyze the hydrolysis of silica precursor and the branched structure may direct the formation of large pores, here, for the first time, we developed monodispersed bioactive glass nanoclusters (BGNCs) with ultra-large pores and controlled size. We also demonstrated their highly efficient miRNA delivery, activated osteogenic genes expression *in vitro* and enhanced critical-sized bone defect regeneration *in vivo*, as compared to conventional BGNs.

2. Experimental Section

2.1. Materials

Polyethylenimine (PEI, Mn 25000), tetraethyl orthosilicate (TEOS), triethylphosphate (TEP), calcium nitrate tetrahydrate (CN), ethyl alcohol (99% ETOH), tris, agarose, boric acid were purchased from Sigma-Aldrich. Opti-MEM, AlamarBlue cell viability kit, LIVE/DEAD mammalian cell viability

stain kit, Hoechst 33342 and 4-(2-hydroxyethyl)-1-piperazine-N'-2-ethanesulfonic acid (HEPES) were obtained from Thermo Fisher. Normal miRNA and FAM labeled miRNA (miR-5106) were bought from Genepharma (China). Deionized water (DW) was used for all experiments. All chemicals were used as received without further purification.

2.2. Synthesis and characterizations of monodispersed BGNCs

Monodispersed BGNCs with ultra-large pores were fabricated via a sol-gel method using PEI as the catalyst and structure-directing agent. Briefly, PEI was dissolved in 500 mL of ethyl alcohol/water solution (3/1= v/v) under magnetically stirring, followed by addition of 10 mL TEOS, and the resulting mixture was stirred for 30 min. Then TEP and CN were added to the reaction system in sequence and the reaction was kept at 40 °C for 3 h. The resulted product was collected by centrifugation and purification by washing with ETOH/DW for several times. The nanoparticle size of BGNCs was controlled through adding different amount of PEI. Specifically, the mass ratio of TEOS to PEI were 1:0.06, 1:0.1, 1:0.2 and 1:0.3, and the final products were denoted as 6% BGNCs, 10% BGNCs, 20% BGNCs and 30% BGNCs. Additionally, conventional bioactive glass nanoparticles (BGNs) with the same chemical composition as BGNCs were prepared according to our previous report and used as a control material in this study.^[24]

The nanostructure and morphology of BGNCs were observed by a transmitted electronic microscopy (TEM, HT-7700, Hitachi). The phase, chemical structure and composition of BGNCs were investigated by X-ray diffraction (XRD, DMAX/A, Rigaku), Fourier transform infrared spectroscopy

(FTIR, Nicolet 6700, Thermo Sci.) and X-ray energy dispersive spectroscopy (EDS, Quanta 250 FEG, FEI). Nitrogen adsorption-desorption measurement and Brunauer-Emmet-Teller (BET) method was carried out to analyze the specific surface area and pore size distributions of BGNCs (Quadrascorb SI-3, Quantachrome). UV-vis absorption spectra were recorded on a UV-vis spectrophotometer (Lambda 35, PerkinElmer) at room temperature.

2.3. Apatite-forming bioactivity evaluation of BGNCs

The apatite-forming investigation of BGNCs was performed in simulated body fluids (SBF) for 3 days, according to our previous report.^[24] In a typical process, BGNCs was soaked into SBF at a concentration of 10 mg/mL and shaken at 100 rpm at 37 °C for predetermined times. The morphology and structures of apatite formed on the surface of BGNC was observed by TEM, and further analyzed by FTIR and XRD.

2.4. Formation and stability analysis of BGNC@miRNA complexes

Based on the optimum monodispersed size and controlled morphology, 10% BGNCs were used in further miRNA loading, delivery and biocompatibility studies. The miRNA (miR-5106 from GenePharma, Shanghai) loading capacity and stability of BGNCs was analyzed using a gel retardation assay,^[25] BGNs were used as a control. Briefly, the BGNC@miRNA or BGN@miRNA complexes at different mass ratios ranging from 0:1 to 64:1 were prepared as the following procedure. Different amount of BGNCs/BGNs were mixed with 13 µg/mL miRNA in HEPES and incubated for 30 min at 37 °C. Then the resulting complexes were loaded on 2% agarose gel to conduct the gel retardation

assay (110V for 15 min) on an electrophoresis system (HE120, Tanon). The mean intensity of miRNA bands and pictures-taking were investigated by a gel imaging system (Gel Doc XR⁺, Bio-rad). The binding percentage was calculated as the following equation:

$$Bp=(Da-Db)/(Dc-Db) \times 100\%,$$

(1)

where Bp is the miRNA binding percentage, Da is the densities of miRNA bands in the wells with BGNC/BGN, Dc represents the densities of miRNA bands in the wells without BGNC/BGN and Db is the density of background, respectively.

The stability investigation of BGNC@miRNA complexes was performed in the presence of serum, using BGNs and commercial Lipofectamine™ 3000 (Lipo) as the controls. Briefly, BGNC@miRNA complexes solutions were added with 25% fetal bovine serum (FBS) and incubated for predetermined times (0~24 h). Afterwards, ribonuclease inhibitor (RRI, Tiangen) was added to the solutions to inhibit the further degradation of miRNA. Then the samples were collected and treated with heparin and subsequently loaded on 2% agarose gel for gel retardation assay, the mean intensity of miRNA bands were analyzed by the gel imaging system.

2.5. Cytotoxicity investigation

The cellular biocompatibility of BGNCs was carried out by using the bone marrow stromal stem cells (BMSCs from Cell Bank, Chinese Academy of Sciences, China), according to previous report.^[26]

The live cellular morphology and proliferation was evaluated by using a Live-Dead kit (Life Science) and alamar blue assay kit (Life Science) respectively. Briefly, cells were seeded in a 96 well plate at a density of 4000 cells/well and incubated in a growth medium (DMEM, 10% FBS, Life Sciences) in a humidified incubator at 37 °C with 5% CO₂. After attachment for 24 h, the medium was removed, followed by adding the growth medium containing various BGNCs concentrations (40~160 µg/mL). The cells were cultured with BGNCs for 1, 3, 5 days. At the predetermined time point, the cells were washed by PBS for three times and incubated in 100 µL of Live-Dead solution for 45 min at 37 °C. The live cell morphology was observed via an inverted fluorescent microscope (Olympus, IX53). After 1 day, 3 days and 5 days incubation, the live cell viability was measured by using an Alamar Blue kit in accordance with the manufacture's instruction. The live cell fluorescence was measured at an excitation/emission wavelength of 530/600 nm on a microreader (SpectraMax[®], Molecular Services). Tissue culture plate (TCP) without materials was used as a control. At least five species per sample was tested.

2.6. Transfection efficiency investigations

For demonstrating the delivery efficiency of miRNA by BGNCs, the cellular uptake and intracellular fluorescence expression of BGNC@miRNA complexes were evaluated by laser confocal scanning microscope (CLSM, FV1200, Olympus) and a flow cytometer (CytoFLEX, Coulter Beckman), using BGNs and Lipo as controls. For CLSM observation, cells were seeded on glass slides at a density of 5000 cells/well in a growth medium. After 24 h, cells were rinsed by PBS and incubated with the

fluorescent labeled miRNA complexes (BGNC@miRNA, BGN@miRNA or Lipo@miRNA) in a transfection medium. After culture for 24 h, the medium was removed and cells were washed by PBS for three times. The cell nucleus was stained as blue by Hoechst (Life Technologies). Then the transfected cell images were observed under the CLSM.

Flow cytometer (CytoFLEX, Coulter Beckman) was used to evaluate the cellular uptake of miRNA-loaded 10%BGNC/BGN/Lipo. Typically, BMSCs transfected with miRNA-loaded 10%BGNC/BGN/Lipo were washed by Dulbecco's phosphate buffered saline (DPBS, Thermo Fisher Scientific) for two times, after trypsinization and centrifugation process, the cells were finally suspended in 100 μ L PBS supplemented with 2% FBS and 100 μ L of 4% paraformaldehyde. After incubated for 2 h at 4 $^{\circ}$ C, the percentage of cells successfully transfected were analyzed by detecting the fluorescence intensity of samples through flow cytometry and analyzed by CytExpert software (Beckman Coulter).

2.7. Osteogenic genes expressions examination

The BMSCs were transfected with different miRNA complexes and cultured for different times (7 days and 14 days) for evaluating osteogenic genes expressions (*Alp*, *Runx2*, and *Opn*) using a quantitative real-time polymerase chain reaction (qRT-PCR, Bio-rad). The α -MEM (Sigma) supplemented with 10% FBS (Gibco), 50 mg mL⁻¹ ascorbic acid, 10 mM α -glycerophosphate (Sigma) and 10 nM dexamethasone (Sigma) was used as osteogenic inductive medium. 10% BGNCs were used to load miRNA for osteogenic differentiation experiment. Typically, cells were seeded in a

24-well plate at a density of 2×10^4 cells/well and cultured in DMEM supplemented with 10% FBS (v/v) for 24 h. Then the cells were rinsed by PBS and treated with different samples (BGNC@miRNA, BGN@miRNA, BGNC and BGN). After incubated for 24 h, the medium were changed to osteogenic inductive medium. The medium were replaced by fresh medium every 3 days. At every specific time point, cells were collected and total RNA were extracted using Tripure (Roche), then single-stranded cDNA was synthesized using a RevertAid First Strand cDNA Synthesis kit (Roche) according to manufacturer's instructions. qRT-PCR was carried out by employing iTaq universal SYBR Green supermix (Applied Biosystems 7500, Bio-rad) with the gene-specific primers listed in Table S1. The qRT-PCR amplifications were conducted under 95 °C for 5 min, 40 cycles of 95 °C for 10 s, 60 °C for 30 s, and 70 °C for 10 min.

2.8. Critical-sized bone regeneration evaluation *in vivo*

The *in vivo* bone regeneration ability of BGNCs and BGNC@miRNA were investigated by a rat cranial critical sized defect model (5 mm in diameter, Sprague Dawley rats, 6 weeks, male, 200–220 g). Animal surgeries and euthanasia work was conducted in compliance with the animal care and use guidelines of National Institutes of Health and the animal research committee of Xi'an Jiaotong University. Briefly, the rats was narcotized with 10 wt% chloral hydrate under a rigorous aseptic condition (n=6 per group), 5 mm diameter defects were made at the both sides of calvarium. Before implantation, samples were mixed with biomedical fibrin glue (Porcine Fibrin Sealant Kit, Hangzhou Puji Pharmaceutical Technology) at a weight ratio of 1:1. The as-prepared hybrid solution could be

formed to hydrogel in defect site once implantation. The critical sized defect was filled with fibrin glue hydrogel containing BGN, BGN@miRNA, BGNC, BGNC@miRNA, using pure fibrin glue hydrogel as a negative control (NC). After implantation for 4 weeks, the rats were sacrificed, the implanted materials/surrounding native bones were dissected and fixed in 10% neutral buffered formalin for microCT analysis and histological evaluation. MicroCT analysis was performed on a three-dimensional micro-focus X-ray microcomputed tomography system (Y.CHEETAH*, YXLON) to quantitatively analyze the newly-formed bone within the defects. For the histology evaluation, bone samples were decalcified in 10% EDTA and embedded in paraffine. Then the samples were sectioned (5 μm thick) and stained with hematoxylin and eosin (H&E, Sigma) and masson's trichrome (Sigma), respectively. The bone volume and trabecular thickness was calculated based on the microCT analysis. At least three rats per group were evaluated.

2.9. Statistical Analysis

All results were expressed as mean value \pm standard deviation (SD). SPSS Statistics software (version 13.0) was utilized for all the statistical analysis, and Student's *t*-test was carried out to evaluate the statistically significant differences between two groups. It was considered that the differences were significant when **p* values < 0.05 or ***p*< 0.01. All experiments were performed at least in triplicate.

3.1. Preparation and characterization of monodispersed BGNCs

In this work, we have developed a facile and simple method to fabricate the monodispersed BGNCs by employing a branched macromolecules PEI as structure director and reaction catalyst. The synthesize process, reaction mechanism and potential miRNA delivery application is illustrated in Figure 1. On one hand, the amine-based polymer PEI could serve as an in situ catalyst for silica precursor hydrolysis and condensation. On the other hand, PEI could self-assemble into the framework during condensation reactions, therefore act as a pore expansion agent to generate ultra-large pores (Figure 1A). Based on the strong gene binding ability and ultra-large pores, BGNCs could serve as a high efficient non-viral vector to deliver miRNA for accelerating bone tissue regeneration (Figures 1B-C).

The final nanostructure and size of the synthesized BGNCs could be tuned by the concentration of PEI during the reaction. As shown in Figures 2A-D, BGNCs showed a typical nanoclusters morphology and large mesopores. Specifically, the morphology and size of BGNCs could be effectively tailored by the mass ratio of PEI. 10%BGNCs, 20%BGNCs, and 30%BGNCs exhibited a size around 100 nm, 200 nm and 450 nm, respectively (Figure S1). Generally, the monodispersed size of BGNCs was increased from 100 nm to 450 nm with the increase of PEI contents. However, when the mass ratio of PEI decreased to 6%, BGNCs presented a non-uniform size with a wide range from 50 nm to 500 nm (Figure S2). Under the high concentration of PEI, the rapid hydrolysis and condensation of silanol groups took place, leading to the fast formation of silica network. Then the smaller nanoclusters would aggregate and produce a larger one after further reaction. However, at the low concentration of PEI, the incomplete hydrolysis and condensation of silanol groups would

result in the non-uniform size and irregular morphologies of BGNCs. Through varying the amount of PEI, the size, microstructure and morphology of BGNCs could be controlled facilely. In particular, through increasing the amount of precursors and solvents, the large-scale production of BGNCs can be achieved, and their morphology/ size distribution were almost unchanged (Figure S3).

BGNCs exhibit a strong and broad band at $2900\text{ cm}^{-1}\sim 3500\text{ cm}^{-1}$ and 1600 cm^{-1} which were attributed to the stretching and bending vibration of hydroxyls (Figure 2E). The characteristic peaks of BGNCs can be found at 1060 cm^{-1} , 800 cm^{-1} and 480 cm^{-1} .^[27] The strong absorbance at 1060 cm^{-1} is assigned to the stretch vibration of Si-O-Si, while the peaks at 800 cm^{-1} and 480 cm^{-1} were owing to the bending vibration of Si-O-Si. The broad peak in XRD patterns indicated the typical amorphous structure of BGNCs (Figure 2F). The EDS analysis confirmed the chemical composition of BGNC containing Si and Ca (Figure 2G). To investigate the pore structure and property of BGNCs, the specific surface area, pore volume, and pore size distribution of BGNCs and BGNs were determined (Figure 2H and Figure S4). The BET surface area of 10%BGNCs and BGNs are calculated to be $213.611\text{ m}^2\text{ g}^{-1}$ and $236.620\text{ m}^2\text{ g}^{-1}$ respectively (Table S2). The 10%BGNC showed a total pore volume of $1.138\text{ cm}^3\text{ g}^{-1}$ and average pore size of 20.056 nm , which were almost 30 and 6 times higher than those of BGNs ($0.0342\text{ cm}^3\text{ g}^{-1}$ and 3.468 nm) (Table S2). The ultra-large porous structure may endow BGNCs the high gene storage and delivery ability for efficient bone tissue regeneration.

3.2. Apatite-forming evaluation of BGNCs

BGs have been successfully exploited as a kind of highly bioactive material for bone regeneration because of its improved apatite-forming ability as compared to other conventional bioactive ceramics. The well-formed apatite layer on BGs in physiological environment can bond to bone and even stimulate new bone growth.^[28,29] In this work, the apatite-forming ability of the synthesized BGNCs was studied in detail. After soaking in SBF for 3 days, the flake-like nanocrystals were observed clearly on BGNCs (Figure 3A). Further FTIR spectra indicated the presence of three new peaks at 550, 605 and 950 cm^{-1} which are associated to the vibration of crystalline P-O in crystalline PO_4^{3-} (Figure 3B).^[30] The characteristic peaks of hydroxyapatite nanocrystals (JCPDS 09-432) could also be observed from the XRD pattern at 26° (002), 32° (211) and 46° (222), which further implied the formation of hydroxyapatite layer (Figure 3C).^[30] By investigating the bioactivity of BGNC utilizing FTIR, XRD and TEM, we can reasonably conclude that the obtained BGNCs possess excellent apatite-forming ability, and the superior bioactivity make it can be widely used in bone tissue repair and regeneration.

3.3. Ultra-high miRNA loading and stability of BGNCs

Compared with BGNs and conventional mesoporous materials, BGNCs possess ultra-large mesopores and may have high miRNA loading ability. Figures 4A-C reveals the miRNA loading ability of BGNCs at different weight ratios (0-64 $\text{mg}/\mu\text{g}$). The miRNA was retarded completely by BGNCs at a weight ratio of 4:1 (Figure 4A), overpassing that of BGNs (8:1) (Figure 4B). The quantitative analysis showed that BGNCs demonstrated significantly high miRNA loading efficiency (3.8 $\mu\text{g}/\text{mg}$) which was

almost 19 times high compared to BGNs (0.2 $\mu\text{g}/\text{mg}$) at the same nanoparticles concentration (40 $\mu\text{g}/\text{mL}$). The high loading ability of BGNCs was probably due to two reasons. On one hand, the Ca^{2+} in BGs framework has a strong binding affinity to the phosphates groups in miRNA structure, which contributed to their loading ability.^[31] This strong gene binding mechanism of BGs through a metal ion complex process has been demonstrated by our previous report.^[20] In addition to the calcium-miRNA complex interaction, the ultra-large porous structure of BGNCs should also contribute to their high miRNA loading amount. BGNCs have a 30 times and 6 times high pore volume and mean pore size comparing with BGNs, thus showed the ultra-high miRNA loading efficiency. The large pore volume and size derived high gene or drug loading performance was reported everywhere, especially for mesoporous inorganic nanomaterials.^[32, 33] The anti-nuclease degradation ability of miRNA complexes (BGNC@miRNA, BGN@miRNA) was investigated by incubating them with serum, Lipo and naked miRNA were chose as controls (Figures 4D-E). It can be found that both BGNCs and BGNs could protect the miRNA from the enzymolysis by nuclease within 24 h, and almost no significant degradation of miRNA could be detected (Figure 4D). Naked miRNA was degraded completely after 3 h (Figure 4E). However, after 24 h incubation, more than 80% of the intact miRNA could be detected in BGNC@miRNA group, but only 50% and 35% of intact miRNA could be detected in BGN@miRNA and Lipo@miRNA group, suggesting the excellent miRNA protection ability of BGNCs (Figure 4E). The excellent miRNA protecting ability of BGNC was mainly owing to the large pore size and volume of BGNCs. On one hand, the large pores created more chances for the interaction between Ca^{2+} and the guest genes, therefore significantly enhanced the

binding forces to miRNA and avoided the fast release of miRNA. On the other hand, the large pore volume of BGNCs could effectively protect miRNA from enzymolysis. The ultra-high miRNA loading and excellent anti-nuclease degradation ability of BGNCs enabled their promising application in gene delivery.

3.4. *In vitro* cell biocompatibility, miRNA transfection and osteogenic differentiation

The BMSCs viability and proliferation behavior was investigated through incubating with BGNCs at different concentrations (40-160 $\mu\text{g}/\text{mL}$). After culture with BGNCs for 1 to 5 days, cells viability showed a great increase and no significant difference was observed compared with TCP control, indicating their high biocompatibility (Figure 5A). Further Live-Dead staining fluorescent images exhibited that cells showed good attachment morphology after incubated with BGNCs at different concentrations and no dead cells were observed after 1 day (Figures 5B-D). These results were consistent with previous findings that BG-based bioactive materials have a good cell biocompatibility especially the osteoblasts.^[34]

BMSCs were used as model cells to investigate the miRNA transfection efficiency of BGNCs. After transfected for 48 h, the significant green fluorescence of miRNA in the cells was observed clearly in BGNCs group compared with Lipo/BGN group, suggesting the high efficient miRNA intracellular delivery of BGNCs (Figure 6A). The magnified CLSM images of cells showed that miRNA was mainly distributed around the cell nucleus, implying that miRNA was successfully transported into the cells. Moreover, quantitative analysis based on CLSM images showed that the mean

fluorescent intensity of BGNCs group was significantly higher than that of Lipo and BGNs group (Figure 6B). The flow cytometry examination further demonstrated the difference of miRNA transfection efficiency for various groups (Figure 6C). Almost negligent transfection ability (2.3%) in BMSCs was found in naked FAM-labelled miRNA group (NC). BGNCs group demonstrated significantly high miRNA transfection efficiency (45.3%), as compared to BGNs group (40.1%) and commercial Lipo group (35.1%). The excellent miRNA delivery and transfection efficiency of BGNCs should be attributed to their ultra-large pore structure and ultra-high miRNA loading efficiency, as well as strong miRNA protection capacity.

In this study, the used miRNA-5106 (miR-5106) has been shown to have positive osteogenic inductive ability through upregulating several osteogenic genes expressions.^[26] Here, we investigate the effect of BGNC-based miR-5106 delivery on osteogenic differentiation markers expressions (*Runx2*, *Opn* and *Alp*) in BMSCs by qRT-PCR analysis (Figure 7). As compared to pure BGNs and BGNCs group, BGN@miRNA and BGNC@miRNA group showed the significantly high *Runx2*, *Opn* and *Alp* expression level during the culture periods of 7 days and 14 days, which suggested that BGNs and BGNCs could efficiently deliver miR-5106 to enhance osteogenic differentiation of cells. Additionally, *Runx2*, *Opn* and *Alp* expression increased gradually in all groups during 14 days. At day 14, the *Runx2*, *Opn* and *Alp* expression in BGNC@miRNA group was 1.5 times, 2.0 times and 1.8 times higher than those in BGN@miRNA group. These results further indicated that BGNCs could efficiently deliver miRNA and enhance the osteogenic differentiation of BMSCs, and may have a great potential for *in vivo* bone regeneration as a new gene-activated bioactive materials.

3.5. *In vivo* bone tissue regeneration

The promising *in vitro* results of BGNCs, including miRNA transfection and osteogenic differentiation, encouraged us to investigate the effect of BGNCs on *in vivo* bone repair and regeneration. The bone regeneration ability of BGNCs and BGNC@miRNA was investigated by a rat critical sized cranial defect model. After implantation in defect site for 4 weeks, the bone tissue samples were removed and were scanned by an X-ray microCT systems to evaluate the newly-formed bone tissue. As shown in Figure 8A, blank NC group shows little new bone formation near the defects border, but newly-formed bone tissue could be clearly observed in BGNCs and BGNs group. In addition, large amount of new bone formation could be found after treated by BGNC@miRNA group, which was significantly higher than BGNC/BGN/BGN@miRNA group. Further quantitative bone tissue analysis exhibited that BGNC@miRNA group demonstrated a significantly high new bone volume and trabecular thickness, compared with BGNCs, BGN@miRNA and BGNs group (Figures 8B-C).

The newly-formed bone tissues were further examined through H&E and Masson's trichrome staining (Figure 9). H&E staining showed the results of inflammation response and new bone formation (NB in red). All groups exhibited the low inflammation response in defect site. There was minimal new bone formation in NC group and significantly increased bone tissue was observed in BGN-based groups. BGNC@miRNA group induced the best bone formation after 4 weeks implantation in defect site. Masson trichrome staining was used to identify the collagen deposition

in bone defect site after repair. Significantly high and dense collagen deposition (dark site) could be found in BGNCs and BGNC@miRNA group. Limited positive staining in NC group suggested their low bone collagen deposition. The histological results were in accordance with those of microCT, demonstrating the rapid bone regeneration ability of BGNC@miRNA.

In bone tissue regeneration, conventional bioactive materials including bioactive glass and calcium phosphate ceramics-based nanocomposites have shown promising results in enhancing bone tissue formation.^[35, 36] However, the gene-activated bioactive materials with intrinsic gene loading capacity are few demonstrated, although they are rather important for tissue regeneration and disease therapy. In this paper, the developed BGNCs exhibited ultrahigh miRNA loading and excellent delivery performance for enhanced osteogenic differentiation and bone formation. Inorganic nanoparticles with ultra-large pores have shown special advantages in drug and gene delivery applications.^[37] As novel non-viral gene vectors, compared with silica nanoparticles, BGNs present inherent gene binding ability through their unique metal-organic complex mechanism.^[20] Unfortunately, the gene may be only loaded on the surface in BGNs due to their small pore structure (below 5 nm). Herein, by using branched PEI as a template, BGNCs with a monodispersed size and ultra-large porous structure were successfully fabricated. Then, the ultra-high miRNA loading efficiency, significantly enhanced osteogenic genes expressions and bone formation were achieved through BGNCs-mediated delivery. As compared to conventional non-viral vectors such as liposomes and cationic polymers, BGNCs possess several advantages. Firstly, large-scale fabrication of BGNCs could be achieved facilely, which is very important for wide clinical applications. Secondly, BGNCs

show high biocompatibility and intrinsic osteogenic activity, which may have synergetic effect with miRNA delivery in bone formation. Based on these excellent properties, BGNCs may have great potential as next-generation non-viral vector to deliver genes for tissue regeneration and disease therapy.

4. Conclusion

In summary, we developed a novel and simple strategy for large-scale synthesis of monodispersed bioactive glass nanoclusters (BGNCs) with ultra-large mesopores (~20 nm) and tailored nanosize (100~500 nm) using branched PEI as the template and catalyst. BGNCs show an excellent apatite-forming ability due to their high surface area and pore structure. BGNCs can support and enhance BMSC attachment and proliferation, indicating their good cellular compatibility. Significantly, BGNCs demonstrate ultra-high miRNA loading efficiency which is 19 times higher than conventional BGNs. BGNCs can protect, deliver miR-5106 to enhance the osteogenic differentiation of BMSCs and significantly promote the new bone formation in critical sized cranial defects in rats. Therefore, based on the excellent pore structure, unique gene binding ability and biosafety, BGNCs would be a favorable vehicle to deliver various genes for enhancing tissue regeneration and cancer therapy. As a new-generation gene vector, further *in vivo* tissue regeneration and cancer therapy applications based on BGNCs should be demonstrated.

Acknowledgements

We acknowledge the valuable comments of potential reviewers. This work was supported by the State Key Laboratory for Mechanical Behavior of Materials (No. 20161801), the Fundamental Research Funds for the Central Universities (No. XJJ2014090), and National Natural Science Foundation of China (No. 51502237, No. U1501245).

References

- [1] G. A. Rodan, T. J. Martin, *Science* **2000**, *289*, 1508.
- [2] C.-H. Lu, Y.-H. Chang, S.-Y. Lin, K.-C. Li, Y.-C. Hu, *Biotechnol. Adv.* **2013**, *31*, 1695.
- [3] B. Peng, Y. Chen, K. W. Leong, *Adv. Drug Delivery Rev.* **2015**, *88*, 108.
- [4] L. Zheng, Q. Tu, S. Meng, L. Zhang, L. Yu, J. Song, Y. Hu, L. Sui, J. Zhang, M. Dard, *J. Cell. Physiol.* **2017**, *232*, 182.
- [5] M. Sriram, R. Sainitya, V. Kalyanaraman, S. Dhivya, N. Selvamurugan, *Int. J. Biol. Macromol.* **2015**, *74*, 404.
- [6] N.-f. Sun, Z.-a. Liu, W.-b. Huang, A.-l. Tian, S.-y. Hu, *Crit. Rev. Oncol./Hematol.* **2014**, *89*, 352.
- [7] H. Wang, Y. Jiang, H. Peng, Y. Chen, P. Zhu, Y. Huang, *Adv. Drug Delivery Rev.* **2015**, *81*, 142.
- [8] H. Yin, R. L. Kanasty, A. A. Eltoukhy, A. J. Vegas, J. R. Dorkin, D. G. Anderson, *Nat. Rev. Genet.* **2014**, *15*, 541.
- [9] A. Ganju, S. Khan, B. B. Hafeez, S. W. Behrman, M. M. Yallapu, S. C. Chauhan, M. Jaggi, *Drug Discovery Today* **2017**, *22*, 424.
- [10] M. Ramamoorth, A. Narvekar, *J. Clin. Diagn. Res.* **2015**, *9*, GE01.

-
- [11] F. Baino, S. Fiorilli, C. Vitale-Brovarone, *Acta Biomater.* **2016**, *42*, 18.
- [12] A. L. Maçon, T. B. Kim, E. M. Valliant, K. Goetschius, R. K. Brow, D. E. Day, A. Hoppe, A. R. Boccaccini, J. Y. Kim, C. Ohtsuki, *J. Mater. Sci.: Mater. Med.* **2015**, *26*, 1.
- [13] B. Lei, L. Wang, X. Chen, S. K. Chae, *J. Mater. Chem. B.* **2013**, *1*, 5153.
- [14] L. L. Hench, N. Roki, M. B. Fenn, *J. Mol. Struct.* **2014**, *1073*, 24.
- [15] B. Lei, X. Chen, Y. Wang, N. Zhao, C. Du, L. Fang, *Biomed. Mater.* **2010**, *5*, 054103.
- [16] L. L. Hench, J. M. Polak, *Science* **2002**, *295*, 1014.
- [17] A. El-Fiqi, T.-H. Kim, M. Kim, M. Eltohamy, J.-E. Won, E.-J. Lee, H.-W. Kim, *Nanoscale* **2012**, *4*, 7475.
- [18] X. Li, X. Chen, G. Miao, H. Liu, C. Mao, G. Yuan, Q. Liang, X. Shen, C. Ning, X. Fu, *J. Mater. Chem. B* **2014**, *2*, 7045.
- [19] F. Chen, G. Wang, J. I. Griffin, B. Brenneman, N. K. Banda, V. M. Holers, D. S. Backos, L. Wu, S. M. Moghimi, D. Simberg, *Nat. Nanotechnol.* **2016**, *12*, 387.
- [20] M. Yu, Y. Xue, P. X. Ma, C. Mao, B. Lei, *ACS Appl. Mater. Interfaces* **2017**, *9*, 8460.
- [21] C. Vichery, J.-M. Nedelec, *Materials* **2016**, *9*, 288.
- [22] J. Wei, Z. Sun, W. Luo, Y. Li, A. A. Elzatahry, A. M. Al-Enizi, Y. Deng, D. Zhao, *J. Am. Chem. Soc.* **2017**, *139*, 1706.
- [23] X. Guan, Z. Guo, T. Wang, L. Lin, J. Chen, H. Tian, X. Chen, *Biomacromolecules* **2017**, *18*, 1342.
- [24] Y. Xue, Y. Du, J. Yan, Z. Liu, P. X. Ma, X. Chen, B. Lei, *J. Mater. Chem. B* **2015**, *3*, 3831.

-
- [25] X. Du, B. Shi, Y. Tang, S. Dai, S. Z. Qiao, *Biomaterials* **2014**, *35*, 5580.
- [26] M. Yu, B. Lei, C. Gao, J. Yan, P. X. Ma, *Nano Res.* **2017**, *10*, 49.
- [27] Q. Hu, Y. Li, G. Miao, N. Zhao, X. Chen, *RSC Adv.* **2014**, *4*, 22678.
- [28] L. L. Hench, H. Paschall, *J. Biomed. Mater. Res., Part A* **1973**, *7*, 25.
- [29] T. Kokubo, H.-M. Kim, M. Kawashita, *Biomaterials* **2003**, *24*, 2161.
- [30] A. L. Macon, T. B. Kim, E. M. Valliant, K. Goetschius, R. K. Brow, D. E. Day, A. Hoppe, A. R. Boccaccini, J. Y. Kim, C. Ohtsuki, T. Kokubo, A. Osaka, M. Vallet-Regi, D. Arcos, L. Fraile, A. J. Salinas, A. V. Teixeira, Y. Vueva, R. M. Almeida, M. Miola, C. Vitale-Brovarone, E. Verne, W. Holand, J. R. Jones, *J. Mater. Sci.: Mater. Med.* **2015**, *26*, 115.
- [31] S. J. Quinn, A. R. Thomsen, J. L. Pang, L. Kantham, H. Brauner-Osborne, M. Pollak, D. Goltzman, E. M. Brown, *Am. J. Physiol.* **2013**, *304*, E310.
- [32] J. Tu, A. L. Boyle, H. Friedrich, P. H. Bomans, J. Bussmann, N. A. Sommerdijk, W. Jiskoot, A. Kros, *ACS Appl. Mater. Interfaces* **2016**, *8*, 32211.
- [33] M. Wu, Q. Meng, Y. Chen, L. Zhang, M. Li, X. Cai, Y. Li, P. Yu, L. Zhang, J. Shi, *Adv. Mater.* **2016**, *28*, 1963.
- [34] G. Kaur, S. G. Waldrop, V. Kumar, O. P. Pandey, N. Sriranganathan, *Biocompatible Glasses* **2016**, *53*, 19.
- [35] M. N. Rahaman, D. E. Day, B. S. Bal, Q. Fu, S. Jung, L. Bonewald, A. Tomsia, *Acta Biomater.* **2011**, *7*, 2355.

[36] M. Mastrogiacomo, S. Scaglione, R. Martinetti, L. Dolcini, F. Beltrame, R. Cancedda, R.

Quarto, *Biomaterials* **2006**, 27, 3230.

[37] D. Niu, Z. Liu, Y. Li, X. Luo, J. Zhang, J. Gong, J. Shi, *Adv. Mater.* **2014**, 26, 4947.

Figure captions

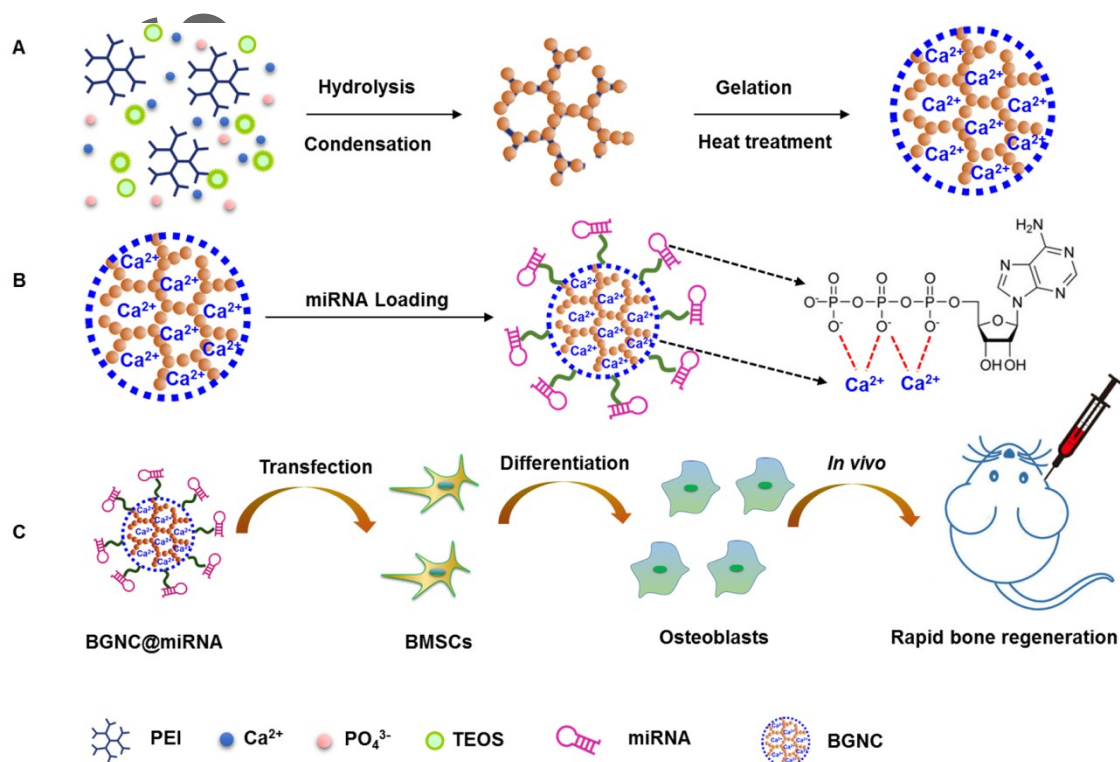


Figure 1. Schematic illustration showing the fabrication and miRNA delivery application of BGNCs.

(A) Synthesis of BGNCs by a branched PEI-assisted sol-gel process; (B) Ultra-high miRNA loading ability and binding mechanism; (C) BGNC-based miRNA delivery to regulate osteogenic activity of stem cells *in vitro* and bone regeneration *in vivo*.

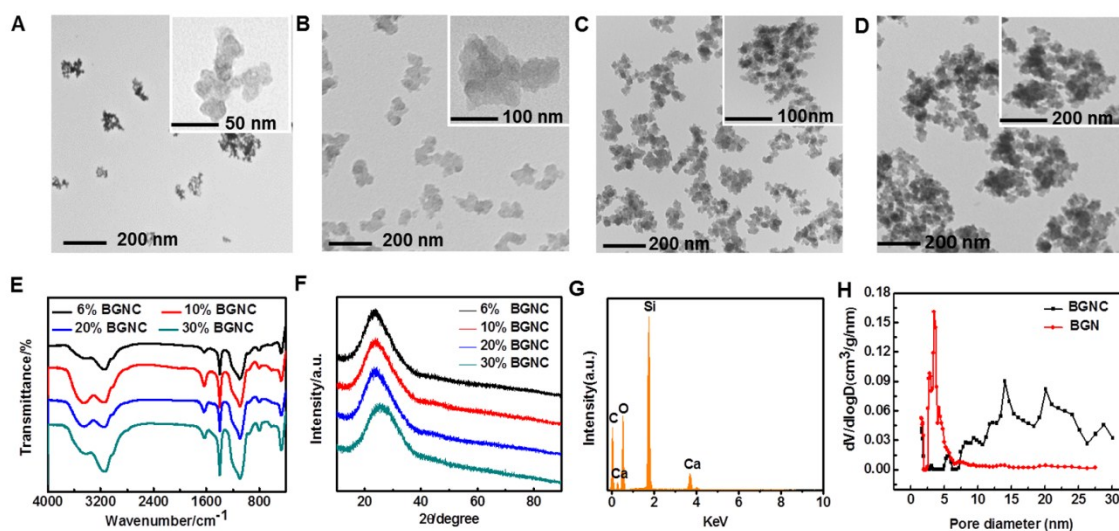


Figure 2. Structure and characterizations of BGNCs. (A-D) TEM images showing the monodispersed cluster-like morphology and nanoscale size of 6% BGNC (A), 10% BGNC (B), 20% BGNC (C), 30% BGNC (D); (E) FTIR spectra exhibiting chemical structure of BGNCs; (F) XRD patterns presenting the typical amorphous structure of BGNCs; (G) EDS analysis identifying the element composition of BGNCs; (H) Pore size distributions of BGNs and BGNCs based on N₂ adsorption-desorption analysis.

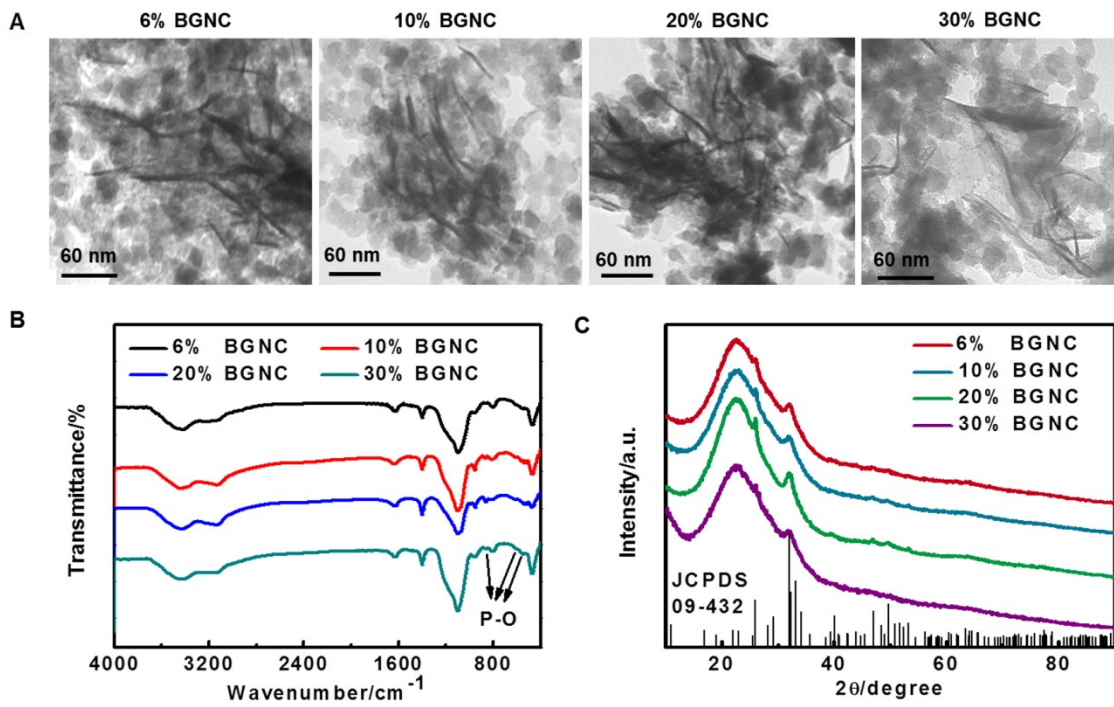


Figure 3. Apatite-forming ability evaluation of BGNCs after soaking in SBF for 3 days. (A) TEM image showing the apatite nanocrystals formed on surface of BGNCs; **(B)** FTIR spectra indicating the chemical structure of new formed hydroxyapatite nanocrystals on BGNCs; **(C)** XRD patterns exhibiting the characteristic peaks of hydroxyapatite nanocrystals on BGNCs (JCPDS No 09-432).

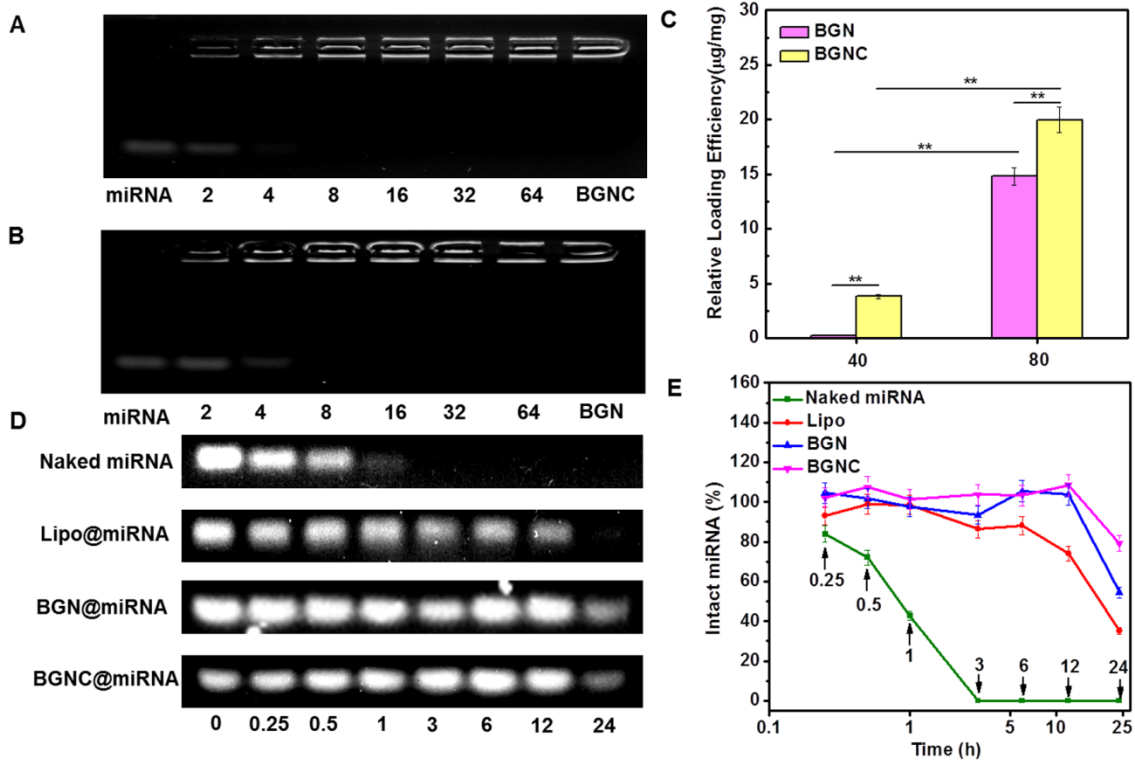


Figure 4. Ultra-high miRNA loading and protection capacity of BGNCs. (A-B) Gel retardation pictures showing the miRNA binding ability of BGNCs (A) and BGNs (B) with different weight ratios to miRNA (2 to 64 mg/μg); (C) Quantitative miRNA loading efficiency of BGNC and BGN at 40 μg/mL and 80 μg/mL (**p<0.01, n=5); (D) Anti-nuclease degradation ability of various miRNA complexes after soaking in serum solution for 24 h; (E) Quantitative miRNA intact percentage of different samples.

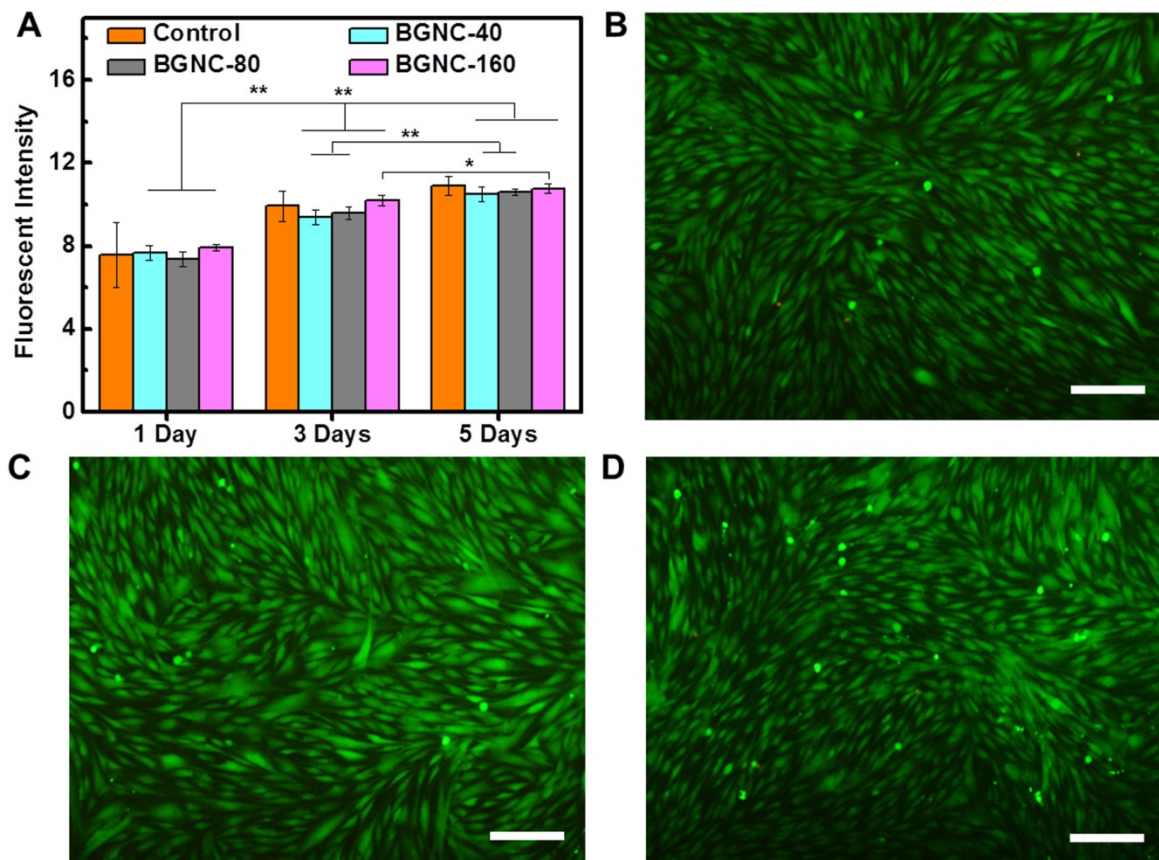


Figure 5. Cellular biocompatibility evaluation. (A) BMSCs viability and proliferation of cells after treated with BGNCs at concentrations of 40 mg mL^{-1} , 80 mg mL^{-1} and 160 mg mL^{-1} (* $p < 0.05$ and ** $p < 0.01$, $n = 5$); (B-D) Live-Dead staining fluorescence images of cells after culture with BGNCs for 1 day at (B) 40 mg mL^{-1} , (C) 80 mg mL^{-1} and (D) 160 mg mL^{-1} (scale bar: $200 \mu\text{m}$).

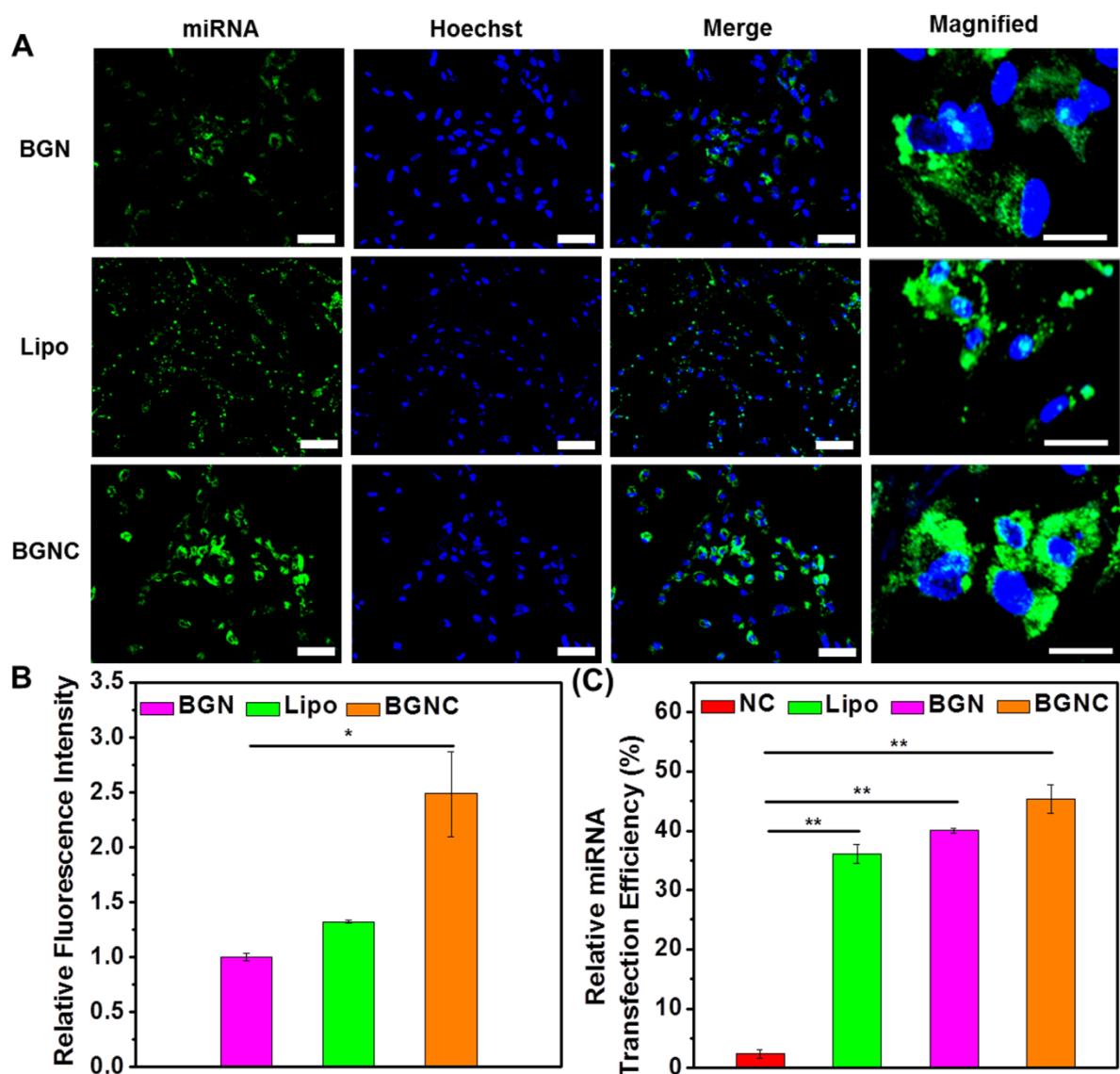


Figure 6. MiRNA transfection efficiency evaluation of BGNCs. (A) CLSM images of BMSCs after transfected with BGNC@miRNA for 48 h, using BGN and Lipo as controls (scale bar: 100 μ m). The miRNA was labeled as green by FAM and cell nuclear was stained as blue by Hoechst; (B) Quantitative fluorescent intensity analysis of cells based on CLSM images after transfected by various miRNA complexes; (C) Flow cytometry analysis of cells after transfected by various miRNA complexes, demonstrating the miRNA transfection efficiency (* P <0.05, ** P <0.01, n =5).

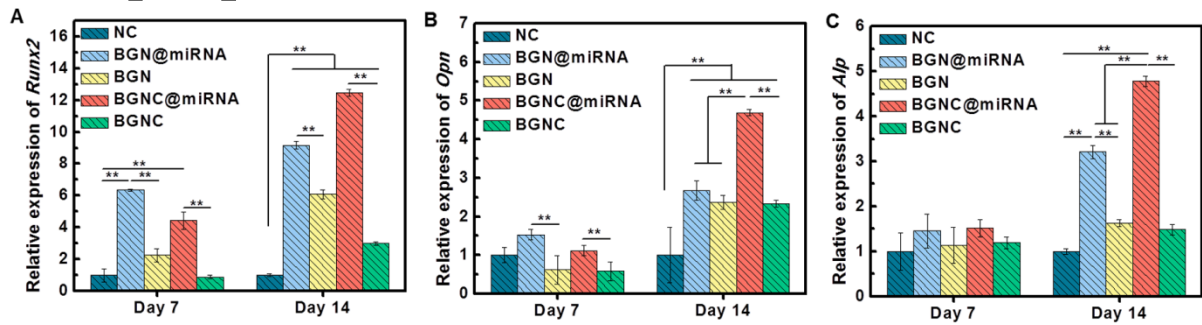


Figure 7. Osteogenic gene maker expressions of cells after induced for 7 and 14 days by various miRNA-5106 complexes. (A) *Runx2*; (B) *Opn*; (C) *Alp*. Significantly enhanced *Runx2*, *Opn*, *Alp*

expressions were observed at day 14 in BGCs group, as compare to other groups. The tissue culture plate (TCP) with naked miRNA-5106 was used as negative control (NC) (* $P < 0.05$, ** $P < 0.01$, $n = 3$).

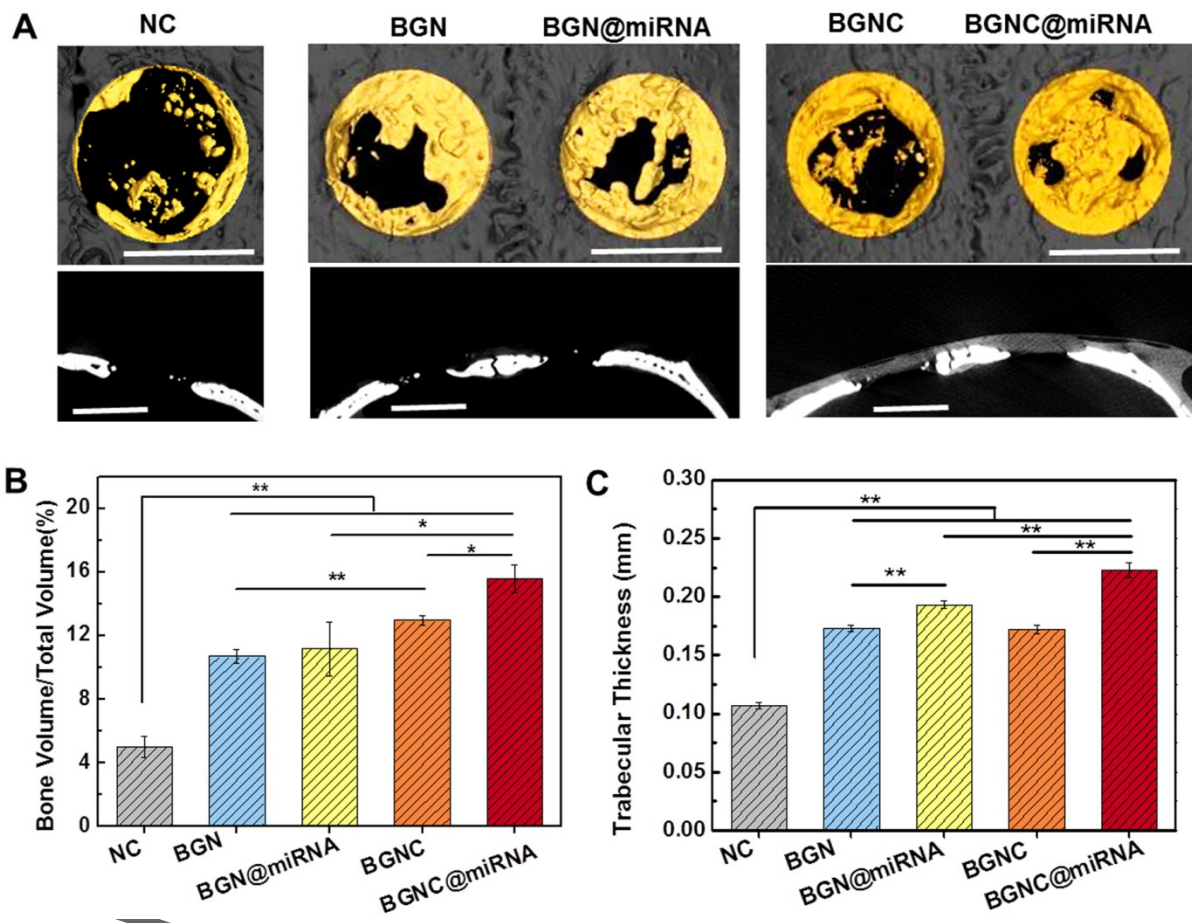


Figure 8. *In vivo* bone tissue regeneration of BGNs, BGNCs, BGN@miRNA, BGNC@miRNA after implantation in critical-sized defect for 4 weeks. (A) MicroCT images showing the repair and reconstruction of frontage (up) and cross-section (down) in cranial defects (Scale bars=5 mm); (B-C) Quantitative analysis of new bone volume fraction (B) and trabecular thickness (C) calculated from the micro-CT images (* $P < 0.05$, ** $P < 0.01$, $n = 3$). Significantly high new bone formation was observed in BGNC@miRNA group.

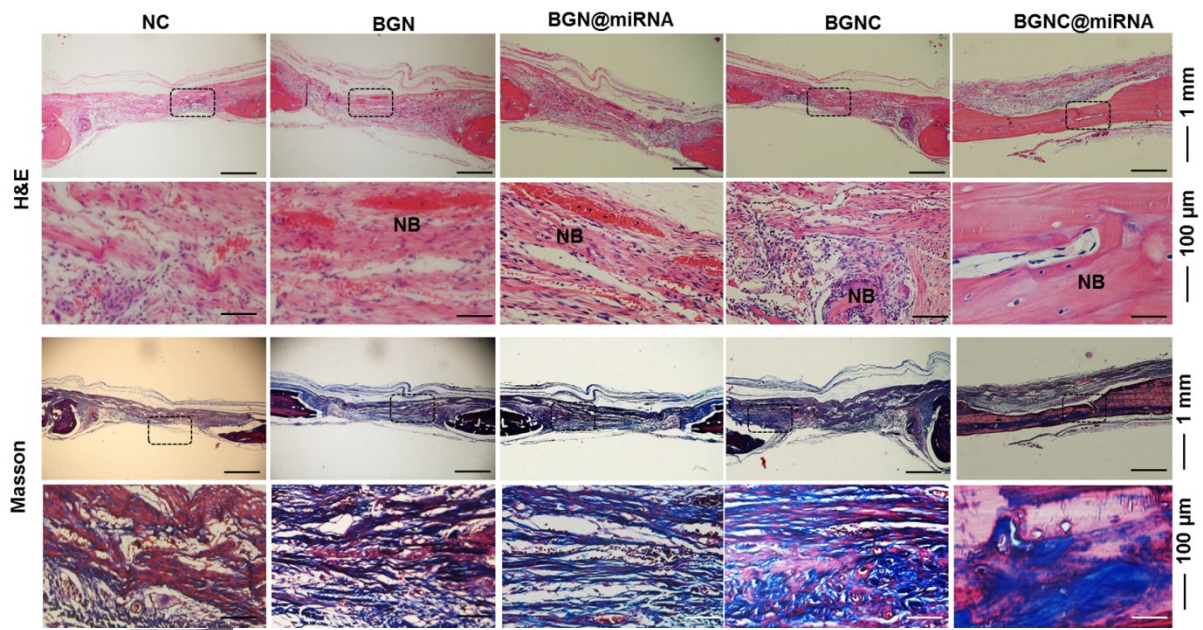
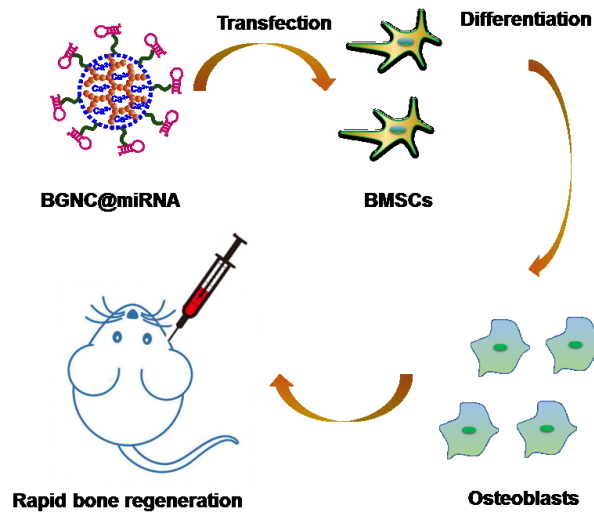


Figure 9. Histological analysis of new bone formation after 4 w implantation in defect site. H&E staining demonstrates the new bone formation (NB) dependent on various samples; Masson staining exhibits the collagen deposition (dark blue) in newly-formed bone tissue. BGN@miRNA, BGNC and BGNC@miRNA group significantly enhanced bone formation and collagen deposition *in vivo*.

TOC



Monodispersed bioactive glass nanoclusters ultra-large mesopores (10-30 nm) are developed for highly efficient miRNA delivery to enhance bone tissue regeneration. BGNCs demonstrated an almost 19 times higher miRNA loading than those of conventional BGNs. Additionally, BGNCs-miRNA nanocomplexes exhibited a significantly high anti-enzymolysis, enhanced cellular uptake and miRNA transfection efficiency, overpassing commercial Lipofectamine 3000. BGNCs-mediated miRNA delivery significantly improved the osteogenic differentiation of bone marrow stromal stem cells (BMSCs) *in vitro* and efficiently enhanced bone formation *in vivo*.