

Nanofabrication and Characterization of Nanoelectronic Biosensors
Based on Emerging Layered Semiconductors

by

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Table of Contents

Acknowledgments	ii
List of Figures	viii
List of Appendices	xxi
Abstract	xxii
Chapter 1 Introduction	1
1.1 Emerging Two-Dimensional (2D) Atomically Layered Material.....	1
1.2 Attractive Biosensing Properties of Semiconducting Transition Metal Dichalcogenides	3
1.3 Need of New Biosensing Technologies for Enabling Rapid Detection of Low- Concentration Illness-Related Biomarker Molecules	6
1.4 Nanoscale Field-Effect-Transistor Biosensors for Realizing Rapid, Label-Free, Multiplexing Biomarker Quantification at fM-Level Detection Limits and New Opportunities of Emerging 2D Layered Semiconductors	8
1.5 Summary of Dissertation	10
Chapter 2 MoS₂ Transistors Fabricated via Plasma-Assisted Nanoprinting of Few-Layer MoS₂ Flakes into Large-Area Arrays	13
2.1 Introduction.....	14
2.2 Methods and Materials	17
2.2.1 Prepatterning of Bulk MoS ₂ Stamps	17
2.2.2 Transfer Printing of Prepatterned MoS ₂ Flakes onto Substrates	17

2.2.3	Fabrication of Field-Effect Transistors Using Printed MoS ₂ Flakes ...	18
2.2.4	Plasma-Assisted Transfer Printing of Prepatterned Graphene	
	Nanoribbons	18
2.3	Results and Discussion	19
2.3.1	Experimental Results of Plasma-Assisted Transfer Printing	18
2.3.2	Maxwell Stress Tensor Calculation	29
2.3.3	Field-Effect Transistor (FETs) Made from Printed MoS ₂ Flakes.....	31
2.4	Summary.....	40
Chapter 3	Multiple MoS₂ Transistor for Sensing Molecule Interaction	
Kinetics.....		41
3.1	Introduction.....	43
3.2	Methods and Materials	47
3.2.1	Fabrication and Characterization of MoS ₂ Transistor Biosensors.....	50
3.2.2	Bio-Functionalization of MoS ₂ Transistor Biosensors	50
3.2.3	Quantification of the Time-Dependent Association/Dissociation	
	Kinetics of the Antibody-(TNF- α) Pair	50
3.3	Results and Discussion	50
3.3.1	Static Measurement Results of Few-Layer MoS ₂ Transistor Biosensors	
	50
3.3.2	Time-Dependent Measurement Results of Few-Layer MoS ₂ Transistor	
	Biosensors	62
3.4	Summary.....	68

Chapter 4	Two Different Device Physics Principles for Operating MoS₂	
	Transistor Biosensors with Femtomolar-Level Detection Limits.....	70
4.1	Introduction.....	70
4.2	Methods and Materials	72
4.2.1	Fabrication and Characterization of MoS ₂ Transistor Biosensors	72
4.2.2	Bio-Functionalization of MoS ₂ Transistor Biosensors	73
4.3	Results and Discussion	74
4.3.1	Comparison of Insulating-Layer-Coated and Insulating-Layer-Free MoS ₂ Transistor Biosensors	74
4.3.2	Simulation of Electrostatic Potential Distribution in Insulating-Layer- Coated and Insulating-Layer-Free MoS ₂ Channels.....	77
4.3.3	Static Measurement: Sensor Responses Measured Using a MoS ₂ FET Sensor with a 5 nm Thick HfO ₂ Insulating Layer	79
4.3.4	Repeatability of the Sensor Response Behaviors Using MoS ₂ FET Sensors.....	81
4.4	Summary.....	85
Chapter 5	Cycle-Wise Operation of MoS₂ and WSe₂ Transistor Biosensors for	
	Rapid Biomolecule Quantification at Femtomolar Levels.....	86
5.1	Introduction.....	88
5.2	Methods and Materials.....	90
5.2.1	Fabrication of MoS ₂ and WSe ₂ FET Biosensors.....	90
5.2.2	Bio-Functionalization of TMDC FET Biosensors.....	91

5.3 Results and Discussion.....	92
5.3.1 Cycle-Wise Time-Dependent Biodetection Method for Operating TMDC-Based FET Biosensors.....	92
5.3.2 Time-Dependent Sensor Responses with 30 fM Pure Streptavidin (MoS ₂ FET Biosensor).....	94
5.3.3 Comparison of the Time-Dependent Response Signals Obtained Using IFDM Cycle-Wise and Continuous Detection Methods.....	97
5.3.4 Detection Specificity of the TMDC FET Biosensors Operated by the IFDM Cycle-Wise Method.....	99
5.3.5 Sensor Response Signal Measured at Different Streptavidin Concentrations.....	103
5.4 Summary.....	110
Chapter 6 Conclusions and Future Work.....	112
6.1 Summary of Thesis.....	112
6.2 Future Research.....	113
APPENDICES	117
Bibliography	122

List of Figures

Figure 1.1 Three-dimensional schematic illustration of the MoS ₂ structure. Courtesy of reference [25]	4
Figure 1.2 Unit cell structures of 2H-MX ₂ and 1T-MX ₂ . Courtesy of reference [41].....	5
Figure 1.3 The role of cytokines during the pathogenic invasion. Courtesy of reference [47].....	7
Figure 1.4 Schematic figure of MoS ₂ -based FET biosensor. Courtesy of reference [43].....	7
Figure 2.1 (a) Schematic flowchart of transfer printing of prepatterned few-layer MoS ₂ flakes, which includes (1) initial bulk MoS ₂ with a pristine surface; (2) photolithography for patterning device features; (3) formation of Ti masks by metal deposition followed by lift-off; (4) plasma etching of underlying MoS ₂ ; (5) removal of Ti masks and finalization of a bulk MoS ₂ stamp bearing relief features; (6) plasma treatment of the SiO ₂ substrate; (7) direct transfer printing of prepatterned few-layer MoS ₂ flakes onto the substrate. (b) SEM images of a bulk MoS ₂ stamp prestructured with 5 μm size periodic pillars.....	19
Figure 2.2 (a) SEM image of arrays of 10 μm size MoS ₂ flake pixels printed onto a pristine SiO ₂ substrate. (b) Stacked column chart of the average thickness data	

collected from 100 as-printed MoS₂ pixels. The thickness data were obtained from MoS₂ pixels printed over a ~1cm² area by using an AFM. (c) SEM images of periodic arrays of 10 μm size MoS₂ flakes printed onto a pristine SiO₂ surface, which were acquired from different locations over the printed area, as indicated by the red arrows..... 21

Figure 2.3 (a, b) Secondary-electron SEM images of MoS₂ pixel arrays printed onto an O₂ plasma-charged substrate, which exhibit clear, well-defined edge profiles faithfully correlated to the edge profiles of pillar features on the bulk MoS₂ stamps. (c, d) Backscattered SEM images of MoS₂ pixel arrays, which show the presence of thin inner MoS₂ flakes within each printed pixel. (e) Backscattered image of MoS₂ pixels with broken inner films.....23

Figure 2.4 (a) SEM images of periodic arrays of MoS₂ patterns printed onto a plasma-charged SiO₂ surface, which were acquired from different locations over the printed area, as indicated by the red arrows. (b) SEM images of MoS₂ pixels with various feature sizes. All the MoS₂ features printed on the plasma-charged substrates exhibit clear, well-defined edge profiles that are faithfully duplicated from the edge profiles of the pillar features on the bulk MoS₂ stamps.....24

Figure 2.5 X-ray energy dispersive spectrometer (EDS) spectra of (a) the outer edge ribbon of a printed MoS₂ pixel and (b) the inner flake of a MoS₂ pixel. The dashed red squares in the inset SEM images indicate the locations on the samples where the EDS spectra were obtained. The EDS results further confirms the presence of sulfur and molybdenum in both the edge and inner portions of printed MoS₂ pixels.....25

Figure 2.6 Optical micrographs of printed MoS₂ pixel arrays on a 330 nm thick SiO₂ substrate: (a) a low-magnification view; (b) a zoomed view of printed pixels with continuous inner films; (c) a zoomed view of printed pixels with broken inner films. All printed pixels exhibit regular edge profiles faithfully duplicated from prepatterned pillars on the bulk MoS₂ stamps.....26

Figure 2.7 (a) AFM image of a 10 μm size MoS₂ pixel printed on a plasma-charged SiO₂ substrate. The solid line indicates a scanning trace across the pixel, which is explicitly plotted in (b). (c) Stacked column chart of the average thickness data collected from 100 as-printed MoS₂ pixels. Here, the thickness data acquired from the inner flakes (solid columns) and the outer edge ribbons (hatched columns) of these MoS₂ pixels are separately plotted.....27

Figure 2.8 SEM images of (a) a HOPG stamp prepatterned with 100 nm half-pitch relief gratings by using nanoimprint lithography followed with plasma etching and (b) graphene nanoribbons printed onto a plasma-charged SiO₂ substrate.....29

Figure 2.9 (a) Illustration of the 2-D model for Maxwell stress tensor calculation of surface charge-induced electrostatic attractive stress between the bulk MoS₂ stamp and the dielectric substrate. (b) Calculated attractive stress plotted as a function of positions. (c) Zoomed view of attractive stress distribution within a single MoS₂ mesa in contact with a SiO₂ surface. It is found that the attractive stress acting on a microscale MoS₂ mesa is uniform in the central region of the MoS₂ mesa but is significantly increased along the mesa edges due to the fringe effect. 31

Figure 2.10 (a) BSE image of an exemplary back-gated FET made from the inner flake of a printed MoS₂ pixel with flake thickness of ~5 nm, channel width of ~3.7 μm, channel length of ~5.4 μm, and gate dielectric (SiO₂) thickness of 330 nm, in which Ti/Au contacts were deposited as drain (*D*) and source (*S*) contacts and the p⁺ silicon substrate serves as a back gate. (b) *I*_{DS}-*V*_{DS} characteristics under different gate voltages (*V*_G) ranging from -75 to 100 V. (c) Semilogarithmic plot of an *I*_{DS}-*V*_G characteristic curve under a fixed drain-source voltage *V*_{DS} = 10 V, which exhibits an ON/OFF current ratio (*I*_{ON}/*I*_{OFF}) ≈ 10⁷. The inset graph shows the linear plot of the same *I*_{DS}-*V*_G curve, and the transconductance (*dI*_{DS}/*dV*_G) is obtained by fitting the linear region of the *I*_{DS}-*V*_G curve, as indicated by the red line. The field-effect mobility is subsequently extracted to be μ = 22 cm²/(V s) for this FET.....34

Figure 2.11 (a) SEM image of an exemplary back-gated FET made from the outer edge ribbon of a printed MoS₂ pixel with channel width of ~300 nm, channel length of ~500 nm, and gate dielectric (SiO₂) thickness of 330 nm. *I*_{DS}-*V*_{DS} (b) and *I*_{DS}-*V*_G (c) characteristics of this edge ribbon-based FET exhibit p-type conduction and field-effect mobility of ~0.27 cm²/(V s). (d) *I*_{DS}-*V*_G characteristics of a p-type FET made from a MoS₂ flake blank-treated by SF₆ plasma.....35

Figure 2.12 Semi-logarithmic *I*_{DS}-*V*_G characteristic curves of 14 back-gated FETs made from the inner flakes of MoS₂ pixels printed on a single substrate, which exhibit ON/OFF current ratios (*I*_{ON}/*I*_{OFF}) ranging from 10⁵ to 10⁷. The inset graphs

display the linear plots of the I_{DS} - V_G curves, and the field-effect mobility (μ) can be extracted by fitting the linear regions of I_{DS} - V_G curves. μ ranges from 15 to 24 cm^2/Vs37

Figure 2.13 (a) ON/OFF current ratio and (b) field-effect mobility data extracted from the FET characteristics listed in Figure 2.12, which are plotted as a function of the MoS_2 flake thickness.....39

Figure 3.1 Flow chart for fabricating a MoS_2 transistor biosensor: (a) printing of a few-layer MoS_2 flake onto a p^+ - Si/SiO_2 substrate; (b) fabrication of Ti/Au D/S contacts; (c) ALD growth of the HfO_2 effective layer on top of the MoS_2 channel and coating of D/S contacts with thick SiO_x layers; (d) integration of a PDMS liquid reservoir on top of the MoS_2 transistor for measuring sensor responses from different $\text{TNF-}\alpha$ concentrations under thermodynamic equilibrium condition and determining the affinity of the antibody-($\text{TNF-}\alpha$) pair; (e) integration of a microfluidic inlet/outlet tubing kit driven by a motorized syringe pump on top of the transistor for quantifying the association-dissociation kinetics of the antibody-($\text{TNF-}\alpha$) pair; (f) functionalization of the HfO_2 effective layer with antibody receptors and subsequent $\text{TNF-}\alpha$ detection.....47

Figure 3.2 Protocol for functionalizing a MoS_2 transistor sensor with anti-human $\text{TNF-}\alpha$ antibody receptors for detecting $\text{TNF-}\alpha$ molecules.....49

Figure 3.3 Optical micrographs or photographs of (a) an exemplary MoS_2 transistor with channel length (L) and width (W) of 5 and 6 μm , respectively; (b) an as-fabricated MoS_2 transistor biosensor integrated with a cylindrical liquid reservoir, which is drilled into a PDMS block and is ~ 4 mm deep and ~ 1 mm in diameter; (c) a

transistor biosensor integrated with a microfluidic channel system connected with an inlet/outlet tubing kit, which is driven by a motorized syringe pump. 50

Figure 3.4 Sensor responses measured in the linear transport regimes of MoS₂ transistor biosensors: (a) transfer characteristics of an exemplary MoS₂ transistor sensor measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM; (b) a set of calibrated linear-regime responses (S) measured from five different MoS₂ transistor sensors with respect to TNF- α concentration (n). These S - n relationships can be well fitted with Langmuir isotherms and the dissociation constant (K_D) of the antibody-(TNF- α) pair is extracted to be 369 ± 48 fM. 51

Figure 3.5 Linear-regime sensor responses at the equilibrium state: The transfer characteristics of five different MoS₂ transistor sensors measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM.....54

Figure 3.6 Sensor responses measured in the subthreshold regimes of MoS₂ transistor biosensors: (a) transfer characteristics of an exemplary MoS₂ transistor sensor measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM (Here I_{DS} data are plotted in the logarithm scale, and the subthreshold regimes are emphasized); (b) a set of calibrated subthreshold-regime responses (S) measured

from five different MoS₂ transistor sensors with respect to TNF- α concentration (n). These S - n relationships can be well fitted with Langmuir isotherms and the dissociation constant (K_D) of the antibody-(TNF- α) pair is extracted to be 424 ± 70 fM.....56

Figure 3.7 Subthreshold-regime sensor responses at the equilibrium state: The transfer characteristics of five different MoS₂ transistor sensors measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM.....59

Figure 3.8 Sensitivity data acquired from (a) the linear-regime I_{DS} signals measured from the five sensors shown in Figure 3.5 and (b) the subthreshold-regime I_{DS} signals measured from the five sensors shown in Figure 3.7. All differential sensitivities were evaluated at TNF- α concentration of $n = 60$ fM. 60

Figure 3.9 A negative control test of the detection specificity of MoS₂ transistor biosensors: The transfer characteristics of a control sensor measured at stages of (1) bare transistor, (2) antibody functionalization (still functionalized with anti-human TNF- α antibody receptors), and inputs of IL-6 solutions with concentrations of (3) 600 fM and (4) 6 pM. 61

Figure 3.10 Time-dependent association kinetics of the antibody-(TNF- α) pair: (a) real-time sensor responses of antibody-(TNF- α) binding measured under different TNF- α concentrations ($n = 60$ fM, 600 fM, 3 pM, and 6 pM). Each of the response curves was measured from a different MoS₂ transistor sensor and all responses were normalized using Equation (5). The rise parts of the binding

response curves can be fitted with Equation (6). (b) The equilibrium-state responses (S_{eq}) extracted from this fit plotted as a function of TNF- α concentration, which can be further fitted with Langmuir isotherm. The equilibrium constant (K_D) is extracted to be 326 ± 37 fM. (c) The extracted ($k_{on}n + k_{off}$) data plotted as a function of TNF- α concentration (n). The linear fitting of this ($k_{on}n + k_{off}$)-versus- n graph results in rate constants of $k_{on} = (5.03 \pm 0.16) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$ and $k_{off} = (3.44 \pm 0.15) \times 10^{-4} \text{ s}^{-1}$ 63

Figure 3.11 Transfer characteristics of four different MoS₂ transistor biosensors measured before the input of TNF- α samples, from which the subthreshold-swing (SS) parameters were acquired for normalizing the real-time subthreshold-regime sensor responses (Equation (5)). These sensors were utilized to quantify the real-time kinetics of antibody-(TNF- α) binding under different TNF- α concentrations (n) of (a) 60 fM, (b) 600 fM, (c) 3 pM, and (d) 6 pM. The operation points (OP, *i.e.*, the fixed V_G and V_{DS} values, under which a real-time response curve was measured) are also labeled by the red arrows... 64

Figure 3.12 Time-dependent dissociation kinetics of the antibody-(TNF- α) pair measured from two MoS₂ transistor sensors that were incubated in solutions with TNF- α concentrations of $n = 600$ fM and 3 pM for about 2 hours and subsequently rinsed with the pure buffer liquid flow... 66

Figure 3.13 Sensor response measured in the subthreshold regime of a MoS₂ transistor biosensor with a 60 nm thick HfO₂ effective layer (*i.e.*, $t_{HfO_2} = 60$ nm): (a) transfer characteristics of the MoS₂ transistor sensor with $t_{HfO_2} = 60$ nm, which were measured from a set of incremental TNF- α concentrations (*i.e.*, $n = 0, 60$ fM, 300

fM, 600 fM, 3 pM, and 6 pM; (b) The calibrated subthreshold-regime responses (S) measured from this sensor (labeled as red stars) with respect to TNF- α concentration (n)..... 68

Figure 4.1 Illustrations (a) and optical micrographs (b) of insulating-layer-coated (left) and insulating-layer-free (right) MoS₂ FET biosensors..... 73

Figure 4.2 Comparison of sensor responses measured from MoS₂ FET biosensors with and without an insulating layer: (a) transfer characteristics of a sensor with a 30 nm HfO₂ insulating layer, measured at various TNF- α concentrations of $n = 0, 60, 300, 600, 3000,$ and 6000 fM. We can make these transfer characteristic curves coincide by using a horizontal shift along the V_G -axis, as shown in (b). (c) transfer characteristics of an insulating-layer-free sensor measured at $n = 60, 600, 6000,$ and $60\ 000$ fM. We can make these transfer characteristic curves coincide by multiplying them with different factors, as shown in (d)..... 76

Figure 4.3 Simulation of the electrostatic potential distribution in the MoS₂ channel of (a) an insulating-layer-coated sensor, in which charged molecules are bonded to the HfO₂ insulating layer, and (b) an insulating-layer-free sensor, in which charge molecules are directly bonded to the MoS₂ surface..... 78

Figure 4.4 Sensor responses measured using a MoS₂ FET sensor with a 5 nm thick HfO₂ insulating layer: (a) transfer characteristics of this sensor measured at various TNF- α concentrations of $n = 0, 60, 300, 600, 3000,$ and 6000 fM. We can make these transfer characteristic curves coincide by horizontally shifting them along the V_G -axis and subsequently multiplying them with different factors, as shown in (b) and (c)..... 80

Figure 4.5 Transfer characteristics of five effective-layer-coated MoS ₂ FET biosensors ($t_{HfO_2} = 30$ nm). Each of them is subjected to a set of incremental TNF- α concentrations (<i>i.e.</i> , $n = 0, 60$ fM, 300 fM, 600 fM, 3 pM, and 6 pM).....	82
Figure 4.6 Transfer characteristics of five effective-layer-free MoS ₂ FET biosensors ($t_{HfO_2} = 0$). Each of them is subjected to a set of incremental TNF- α concentrations (<i>i.e.</i> , $n = 0, 60$ fM, 600 fM, 6 pM, and 60 pM).....	83
Figure 4.7 The standard curves for TNF- α quantification, acquired from (a) a set of insulating-layer-coated sensors (the ones listed in Figure 4.5), and (b) a set of insulating-layer-free sensors (the ones listed in Figure 4.6).....	84
Figure 4.8 Simulation of the electrostatic potential distribution in the MoS ₂ channel of an insulating-layer-coated sensor. In this device, the HfO ₂ insulating layer has a surface roughness of ~ 10 nm (average peak-to-valley magnitude).	84
Figure 5.1 Experimental method: (a) illustration of a TMDC-based FET sensor, which is integrated with a PDMS microfluidic channel for enabling the cycle-wise time-dependent detection capability; illustration of a few-layer TMDC FET channel that is functionalized with (b) biotin receptors; (c) IL1 β antibody	91
Figure 5.2 Biosensor fabrication and setup: (a) SEM image of several printed few-layer-MoS ₂ FET channels; (b) optical micrograph of a representative few-layer-MoS ₂ FET biosensor; (c) photograph of a WSe ₂ FET sensor integrated with a microfluidic channel system. In this work, the few-layer-TMDC (MoS ₂ or WSe ₂) FET channels were functionalized with biotin receptors, and we experimentally demonstrated time-dependent quantification of streptavidin-biotin binding kinetics.....	93

Figure 5.3 Time-dependent sensor responses to a 30 fM solution of pure streptavidin measured using a pair of representative MoS₂ and WSe₂ FET biosensors: (a) I_{DS} - V_G curves of the MoS₂ sensor measured at different accumulative incubation times t ranging from 0 to 110 min (these curves were measured with both positive and negative V_G sweep directions); (b) I_{DS} - t response signals captured from the t -dependent I_{DS} - V_G curves in (a) at $V_G = 100$ V; (c) and (d) t -dependent I_{DS} - V_G curves of the WSe₂ sensor measured with positive and negative V_G sweep directions, respectively; (e) and (f) I_{DS} - t response signals captured from the p-type (at $V_G = -100$ V) and n-type ($V_G = 100$ V) branches of the t -dependent I_{DS} - V_G curves of the WSe₂ FET, respectively.....95

Figure 5.4 Comparison between a MoS₂ FET biosensor operated using the IFDM cycle-wise method and another similar sensor operated using the regular continuous detection method in their responses to a 30 fM solution of pure streptavidin: (a) t -dependent I_{DS} - V_G characteristic curves of the sensor operated using the cycle-wise method; (b) the S - t response signals obtained using the IFDM cycle-wise method (solid circles) and the regular continuous detection method (the solid line).....99

Figure 5.5 IFDM cycle-wise responses ($T_i = 10$ min) of four MoS₂ FET biosensors that were fabricated in the same fabrication batch: t -dependent I_{DS} - V_G characteristic curves of Sensors #1 (a) and #2 (b), which were measured in a pure solution of 30 fM streptavidin, and Sensors #3 (c) and #4 (d), which were measured in a mixed solution containing 30 fM streptavidin and 60 fM tumor necrosis factor alpha (TNF- α).....101

Figure 5.6 Sensor responses to targeted streptavidin molecules in different solutions: (a) $S-t$ responses, measured from four n-type MoS₂ FET sensors, to 30 fM streptavidin in a pure solution (sensors #1 and #2) and a solution also containing 60 fM TNF- α (sensors #3 and #4); (b) p-type and (c) n-type $S-t$ responses, measured from two ambipolar WSe₂ FET sensors, to 30 fM streptavidin in a pure solution (sensor #1) and a serum solution (sensor #2).....102

Figure 5.7 IFDM cycle-wise responses, measured from two ambipolar WSe₂ FET biosensors, to streptavidin molecules in a pure solution and a serum solution: (a) p-type and (b) n-type branches of the t -dependent $I_{DS}-V_G$ characteristic curves measured from the sensor (Sensor #1) in a pure solution of 30 fM streptavidin; (c) p-type and (d) n-type branches of the t -dependent $I_{DS}-V_G$ characteristic curves measured from the sensor (Sensor #2) in a serum solution containing 30 fM streptavidin.....103

Figure 5.8 t -dependent $I_{DS}-V_G$ characteristic curves of a set of MoS₂ FET sensors, which were measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM).....105

Figure 5.9 p-type branches of the t -dependent $I_{DS}-V_G$ characteristic curves of a set of WSe₂ FET sensors, which were measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM).....106

Figure 5.10 $S-t$ response curves measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM), which are captured from (a) the n-type branches of the t -dependent $I_{DS}-V_G$ curves of a set of MoS₂ sensors and (b) the p-type branches of the $I_{DS}-V_G$ curves of a set of WSe₂ sensors; (c) and (d) plots the

<i>dS/dt - nstreptavidin</i> curves extracted from the <i>S-t</i> response curves measured from MoS ₂ and WSe ₂ sensors, respectively.....	108
Figure 5.11 <i>t</i> -dependent I_{DS} - V_G characteristic curves of a set of MoS ₂ FET sensors, which were measured at different IL1 β concentrations ($n_{IL1\beta} = 0$ fM, 1 fM, 4 fM, 20 fM, 100 fM, and 500 fM).....	109
Figure 5.12 (a) <i>S-t</i> response curves measured at different IL1 β concentrations ($n_{IL1\beta} = 0$ fM, 1 fM, 4 fM, 20 fM, 100 fM, and 500 fM), which are captured from the <i>t</i> -dependent I_{DS} - V_G curves of a set of MoS ₂ sensors at $V_G = 100$ V, $V_{DS} = 1$ V; (b) plots the $dS/dt - n_{IL1\beta}$ curve extracted from the <i>S-t</i> response curves shown in (a)	110
Figure A1 (a) Simulation model based on finite element analysis (FEA) for calculating the effective gate capacitance of the edge ribbon of a MoS ₂ pixel, which takes into account the fringe effect at the feature edges. (b) Cross-sectional viewgraph of simulated electric field around the MoS ₂ edge ribbon. (c) Cross-sectional viewgraph of simulated equipotential lines of 6 electric field around the MoS ₂ ribbon.....	116
Figure B1 A schematic illustration of the dual-gate thin-film transistor biosensor model.....	118

List of Appendices

Appendix A	Finite Element Analysis for Simulating Electric Field around the Edge ribbon of a MoS ₂ Pixel and Calculating Effective Gate Capacitance of Edge-Based FETs (C_g)	116
Appendix B	The Dual-Gate Thin-Film Transistor Biosensor Model.....	118

Abstract

Many important biomedical and clinical applications, such as early-stage cancer diagnosis, autoimmune disease treatment, and real-time monitoring of patients' immune status, demand new integrated multiplexing nanoelectronic/microfluidic biosensors. These biosensors are anticipated to enable fast (minute-scale) quantification of illness-related biomarkers, unprecedented detection sensitivity, fM-level limit-of-detection (LOD), and point-of-care capability. However, these new highly desirable biosensing capabilities have not been realized yet.

Atomically layered transition metal dichalcogenides (TMDCs) have gained a lot of attention because of their excellent electronic and structural properties. Especially, semiconducting TMDCs can serve as an essential complement to zero-band-gap graphene and enable novel semiconductor-related applications, such as thin-film transistors, phototransistors, and various types of sensors. More importantly, such TMDCs hold significant potential to be exploited for making new electronic biosensors and realize the highly desirable biosensing capabilities mentioned above. The research presented in this thesis sought to advance the scientific and technical knowledge for fabricating and

operating new TMDC-based electronic/microfluidic-integrated biosensors and realizing rapid fM-level quantification of biomarkers.

The first part (*i.e.*, the second chapter) is mainly focused on developing a top-down nanofabrication approach for producing orderly arranged, pristine few-layer MoS₂ flakes, which holds significant potential to be developed into a upscalable nanomanufacturing technology. The second part (*i.e.*, the third-to-fifth chapters) presents a systematic study on the biosensing characteristics of the TMDC-based transistor sensors fabricated using our nanoprinting techniques. First, multiple sets of MoS₂-based transistor biosensors were fabricated using our plasma-assisted nanoprinting method. Second, we studied the underlying device physics governing the response characteristics of TDMC transistor biosensors. Third, we further studied a cycle-wise method for operating MoS₂/WSe₂-based transistor biosensors to enable rapid, low-noise, highly specific biomolecule quantification at femtomolar levels.

The presented research has leveraged the superior electronic properties of emerging layered semiconductors for biosensing applications and advances label-free biosensing techniques toward realizing fast real-time immunoassay for low-abundance biomolecule detection. Moreover, the nanofabrication approaches developed in this research can be generally utilized for making other nanoelectronic devices based on emerging 2D layered materials, and the obtained device physics knowledge is anticipated to greatly leverage the excellent electronic and structural properties of TMDCs for other relevant sensing applications.

Chapter 1

Introduction

1.1 Emerging Two-Dimensional (2D) Atomically Layered Materials

Since Novoselov and Geim exfoliated graphene layers from highly oriented pyrolytic graphite (HOPG) for the first time using the mechanical cleavage method in 2004, great possibilities of two-dimensional atomically layered materials have opened up for research [1]. Graphene is a monolayer of carbon atoms with a two-dimensional (2D) honeycomb lattice, which has almost the same crystal energy as diamond [2]. Additionally, graphene has various extraordinary and attractive properties, such as ultrahigh carrier mobility at room temperature ($\sim 10,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$) [1], high Young's modulus ($\sim 1 \text{ TPa}$) [3], and excellent thermal conductivity ($3000\text{-}5000 \text{ W m}^{-1} \text{ K}^{-1}$) [4]. These peculiar properties of graphene have motivated many researchers to develop a variety of applications in a broad range of device applications, such as electronics/optoelectronics [5-7], biomedicine [8, 9], sensors [10-16], and more.

Moreover, the success of graphene has inspired people to investigate other atomically layered materials, such as transition metal dichalcogenides [17], topological insulators [18], hexagonal boron nitrides [19], and layered metal oxides [20].

In comparison to the solid materials with zero-dimensional (0D), one-dimensional (1D), or three-dimensional (3D) morphologies, 2D nanomaterials provide several superior and unique characteristics. Especially in single-layer and few-layer forms, 2D layered materials exhibit a strong quantum confinement effect, which results in excellent electronic properties [21], such as Van Hove singularity points for high light absorption. Furthermore, 2D layered materials have other superb electronic properties. For example, owing to the high-quality crystal structure of graphene, the massless Dirac electrons in graphene can travel submicrometer distances without scattering, resulting in ultrahigh charge carrier mobility as mentioned above [22]. In addition, the atomically thin nature of 2D materials offers superb mechanical properties and high specific surface area. For example, the breaking strength and the ultimate tensile strength of graphene layers are 42 N m^{-1} and 120 GPa , respectively [23]. These standout properties of graphene indicate that graphene is one of the strongest materials today [3]. Moreover, the maximum Brunauer–Emmett–Teller (BET) surface area of graphene is $2630 \text{ m}^2 \text{ g}^{-1}$, leading to very high electrocatalytic activity and ultrahigh loading capacity [24].

Another representative material is MoS_2 , one of the transition metal dichalcogenides, which has mostly been explored recently. MoS_2 has a direct bandgap of $\sim 1.8 \text{ eV}$ when it is in the single-layer form, and has an indirect bandgap of $\sim 1.2 \text{ eV}$ in the bulk form [25]. Therefore, both monolayer and multilayer MoS_2 flakes can be exploited for semiconductor-related device applications and serve as a valuable complement to zero

bandgap graphene structures. Additionally, the breaking strength, the in-plane stiffness, and the Young's modulus of MoS₂ layers are 15 N m⁻¹, 180 N m⁻¹, and 270 GPa, respectively, which are comparable to those of steel [26]. Further, MoS₂ has a high specific surface area of as high as 412 m² g⁻¹ [27]. Exploiting these superior properties of 2D materials, there has been a broad range of potential device applications, including flexible electronics [28], solar cells [29], and surface-active applications owing to their high sensitivities from external stimuli, such as sensors [30-33], catalysis [34, 35], energy storage and conversion, and supercapacitors [36], *etc.*

1.2 Attractive Biosensing Properties of Semiconducting Transition Metal

Dichalcogenides

All layered transition metal dichalcogenides (TMDCs) have a stratified crystal structure. The chemical formula of TMDCs can be generally expressed as MX₂, where M is the transition metal (*e.g.*, Mo and W), and X is a chalcogen atom (*e.g.*, S, Se, and Te), as illustrated in Figure 1.1 [37-39]. In a TMDC quintuple layer, the sub-layer consisting of transition metal atoms is sandwiched by two sub-layers consisting of chalcogen atoms that are covalently bonded together to form the quintuple layer. In a bulk TMDC material, multiple such quintuple layers are stacked together through the van der Waals interaction [40]. Multilayer MoS₂ has two types of stacking configurations: 2H-MoS₂ and 1T-MoS₂, as illustrated in Figure 1.2. The trigonal prismatic (2H) MoS₂ structure is thermodynamically stable and semiconductive, while the octahedral (1T) MoS₂ is thermodynamically unstable and metallic [41]. Attracted by their atomically thin

structures and semiconducting properties (*e.g.*, MoS₂ and WSe₂), recently, many researchers have attempted to explore the applicability of semiconducting TMDCs for making biosensing devices [42-45].

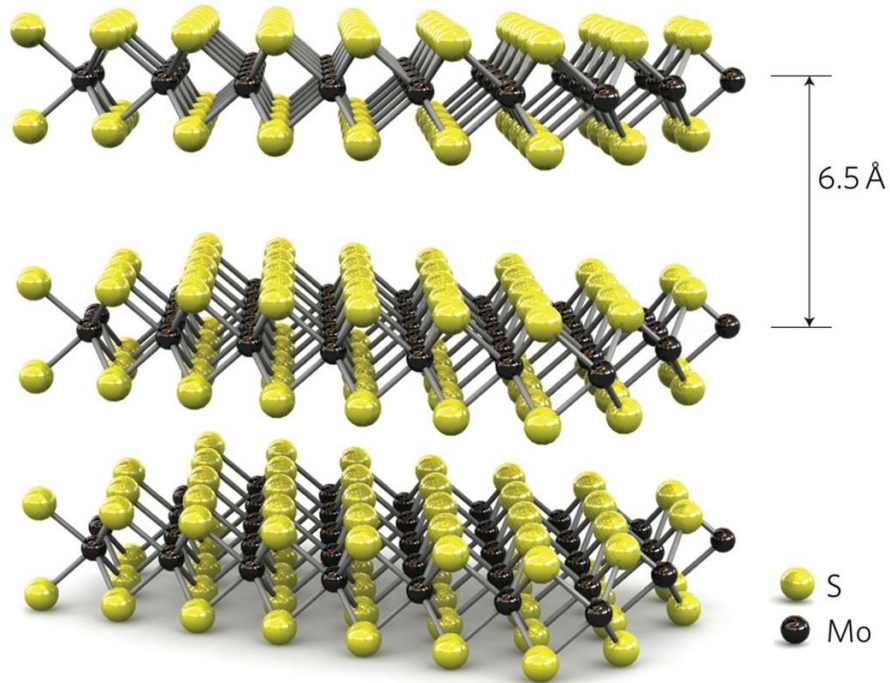


Figure 1.1 Three-dimensional schematic illustration of the MoS₂ structure. Courtesy of reference [25]

TMDCs have several appealing properties for biosensing applications. For example, pristine MoS₂ structures have high-quality 2D surfaces with an extremely low areal density of out-of-plane dangling bonds. Such 2D surfaces lead to very low probabilities for charged carriers to be scattered and trapped by surface defects, which results in a low electronic noise level and therefore a high signal-to-noise ratio for detecting the signals associated with biological events [43]. Additionally, due to the sizable bandgap and high specific surface area of monolayer and few-layer MoS₂

structures, MoS₂-based biosensors are anticipated to have a better biodetection sensitivity as compared to those based on bulk semiconductors and graphene. Moreover, MoS₂ and other TMDC layers have a very large mechanical strength as well as a superior flexibility, which could be exploited for making wearable and implantable biosensor devices. For example, a curvature radius down to 0.75 mm can be obtained in MoS₂ thin-film transistors with no observable degradation of their electrical performance [28]. Lastly, TMDCs show an excellent biocompatibility. In particular, through a series of biocompatibility tests, people have confirmed that MoS₂ exhibits a very low toxicity to most biological species as well as a high cell viability [46]. Such superior biocompatibility of MoS₂ is critical for making safe in-vivo TMDC-based biosensors in the near future.

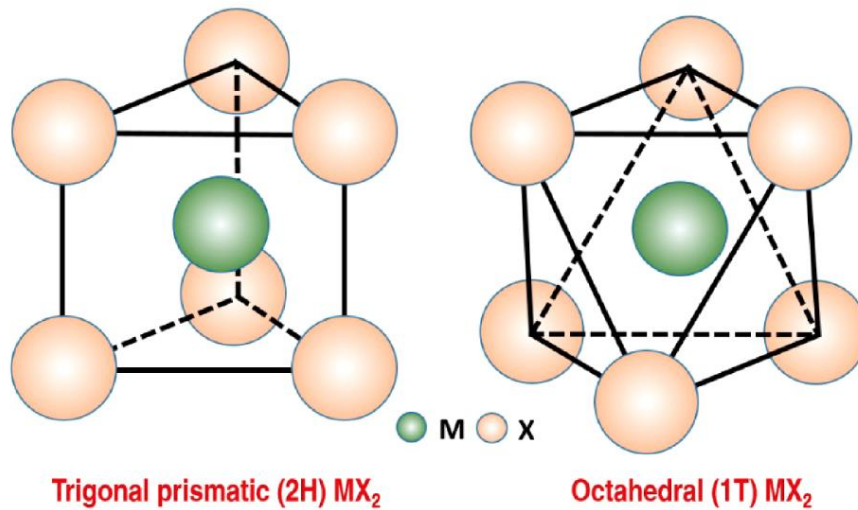


Figure 1.2 Unit cell structures of 2H-MX₂ and 1T-MX₂. Courtesy of reference [41]

1.3 Need of New Biosensing Technologies for Enabling Rapid Detection of Low-Concentration Illness-Related Biomarker Molecules

Cytokines are small secreted protein biomarkers released by immune cells, and they play very important roles in cell signaling and inflammatory response in the immune system (Figure 1.3). When pathogenic viruses or bacteria invade the immune system, the immune cells are activated and start to produce cytokines [47, 48]. Consequently, the concentration of different biomarkers indicates the development progress of target diseases. Therefore, the capability to detect and quantify low-concentration (femtomolar or single-molecule-level) biomarkers is critical for monitoring the development of immune responses at the early stage of a disease.

The current gold standard for detecting and quantifying biomarkers is enzyme-linked immunosorbent assay (ELISA). ELISA has commonly been used for both clinical and research purposes owing to its advantages, such as strong specificity, relatively high sensitivity, and safe environment for biomolecules. This method utilizes a plate-based assay with attached target biomolecules tagged with labels, and the signals from the labels are detected for quantification of the biomolecules. These labeling techniques require a long assay time because of a long sample preparation time and are not applicable for real-time detection of target biomolecules [48]. Additionally, ELISA processes need bulky instrument setups with large sample volumes, which hardly realize the point-of-care capability. To solve the listed challenges above, there is a strong need

for new technology for enabling rapid label-free detection of low-concentration illness-related biomarker molecules.

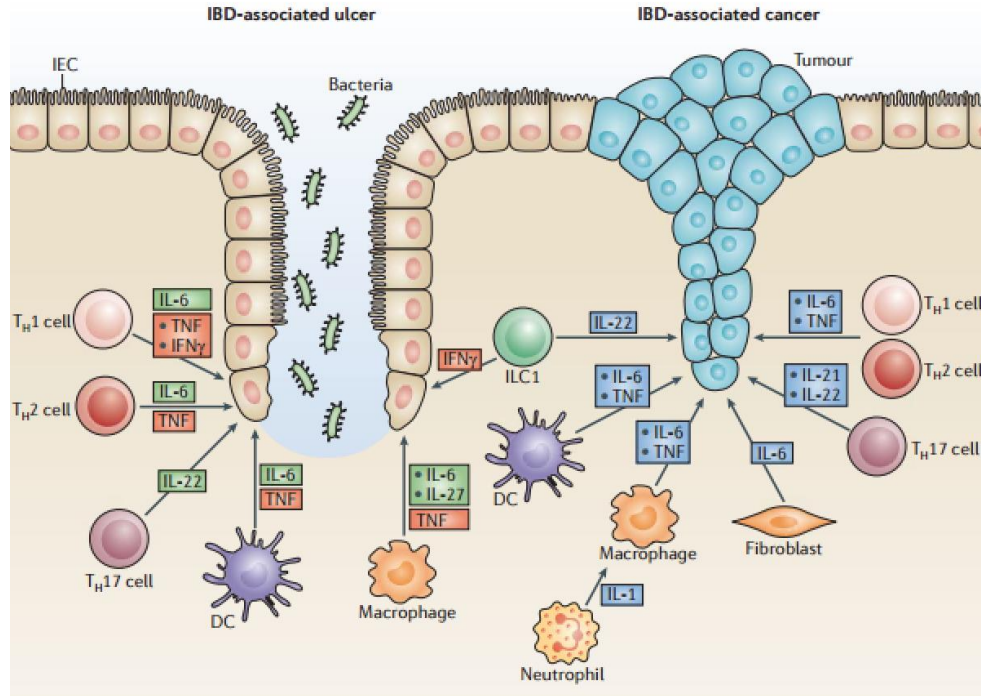


Figure 1.3 The role of cytokines during the pathogenic invasion. Courtesy of reference [47]

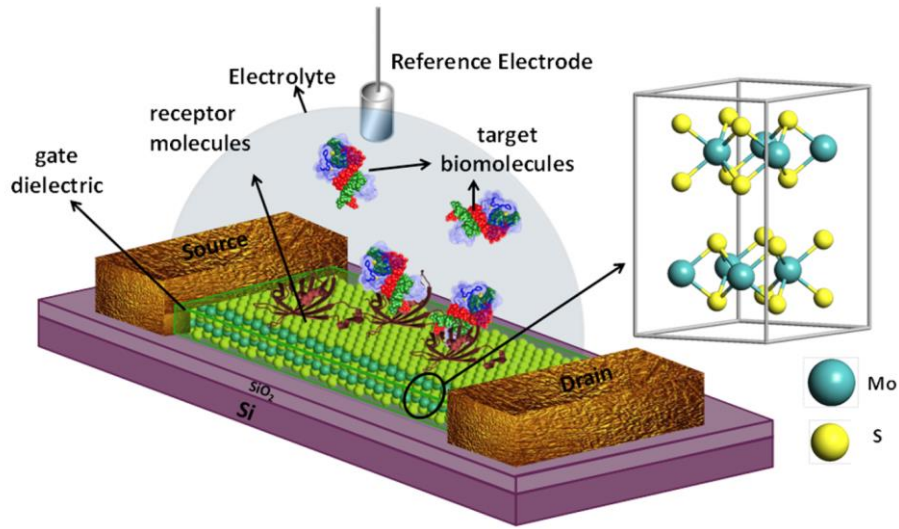


Figure 1.4 Schematic figure of MoS₂-based FET biosensor. Courtesy of reference [43]

1.4 Nanoscale Field-Effect-Transistor Biosensors for Realizing Rapid, Label-Free, Multiplexing Biomarker Quantification at fM-Level Detection Limits and New Opportunities of Emerging 2D Layered Semiconductors

Field-effect-transistor (FET) biosensors have gained a great deal of attention for several years, and they have been demonstrated to be able to enable high biodetection sensitivity as well as label-free quantification capability. Additionally, FET biosensors can be integrated with portable electrical circuits and therefore enable point-of-care capability [49].

Among various types of nanomaterial structures, one-dimensional (1D) FET biosensors, such as silicon nanowires (Si NWs) [50] and carbon nanotubes (CNTs) [51], have been widely studied due to their high biodetection sensitivities. Such high sensitivities are attributed to the electrostatically gated nanoscale FET channels, which

are highly sensitive to biomolecule conjugation reactions. However, cost-effective manufacturing of orderly-arranged (or addressable) NW or CNT-based FET sensors is still a critical challenge, which has significantly hindered their practical applications. Specifically, using top-down nanolithography methods (*e.g.*, electron-beam lithography (EBL) and nanoimprint lithography (NIL)), people can produce uniform arrays of Si NW FET biosensors on silicon-on-insulator (SOI) substrates. However, such sensor fabrication processes are too expensive to be used for practical biosensing applications. The bottom-up methods for directly synthesizing 1D FET biosensor structures still suffers from poor ordering and uniformity of synthesized 1D nanostructures over large areas. In addition, the bottom-up fabrication of CNT-based FET biosensors needs special techniques capable of selecting semiconducting nanotubes and eliminating metallic ones (*i.e.*, the purity of semiconducting CNTs need to be higher than 99% for scale-up applications), which also significantly increases the complexity and cost of the fabrication of CNT FET biosensors [43].

To address the aforementioned challenges, researchers have started to investigate the applicability of two-dimensional (2D) nanomaterials, such as graphene and TMDCs, for making FET biosensors. Due to its atomically thin structure, graphene can be used to make highly sensitive sensing FETs, and the fabrication process is highly compatible with the state-of-the-art planar lithographic techniques, which is potentially suitable for scale-up applications. However, although many studies have been performed for developing graphene FET biosensors, the lack of a bandgap in graphene fundamentally leads to a low carrier modulation for graphene FET biosensors [43]. Therefore, new efforts have been performed to create and characterize FET biosensors based on

semiconducting TMDCs (Figure 1.4), which also have atomically thin FET channel structures [42, 43, 49]. TMDC-based FETs (*e.g.*, those based on MoS₂ and WSe₂) exhibit high On/Off current ratios up to 10⁸, which, in combination with the advantages already discussed in chapter 1.2, can be exploited for creating label-free, highly sensitive FET biosensors with low limit-of-detection (LOD). Wang *et al.* and Sarkar *et al.* recently fabricated MoS₂ FET biosensors with 100-400 fM LODs for a cancer marker protein, Prostate-specific antigen (PSA) and the standard streptavidin–biotin conjugation reaction, respectively [42, 43]. Although individual MoS₂ FET biosensors have been fabricated and exhibited very high biodetection sensitivities, the fabrication of multiple such sensors with consistent sensor response characteristics has not been realized. Additionally, in order to achieve the multiplexing biomarker quantification capability, people need to develop new nanofabrication approaches capable of incorporating uniform TMDC device structures into addressable device sites or ordered arrays.

1.5 Summary of Dissertation

To address the scientific and technical gaps discussed in chapter 1.4, I have finished a series of nanofabrication and device-oriented projects, which are presented in this dissertation. The whole dissertation is divided into two main parts: (1) new nanoprinting processes for generating arrays of few-layer TMDC device structures (Chapter 2), and (2) fabrication of multiple TMDC FET biosensors for rapid, low-noise, multiplexing biomarker quantification at fM-level detection limits (Chapter 3, 4, and 5).

In chapter 2, I present a novel approach for producing orderly arranged, pristine few-layer MoS₂ flakes, which holds significant potential to be developed into a generic nanomanufacturing technology for producing device structures based emerging layered materials. In this work, I have successfully demonstrated the transfer printing of MoS₂ flakes into ordered arrays over cm²-scale areas. Further, the FETs made from as-printed MoS₂ flakes exhibit excellent transport properties (*e.g.*, On/Off current ratio: 10⁵-10⁷, field-effect mobility on SiO₂ gate dielectrics: 6 to 44 cm²/(V s)) as well as a good uniformity of such transistor parameters over large areas.

In chapter 3, I present the demonstration of MoS₂-based transistor biosensors capable of detecting tumor necrosis factor – alpha (TNF- α) with detection limits as low as 60 fM. All sets of transistors exhibit consistent calibrated responses with respect to TNF- α concentrations, and they result in a standard curve, from which the equilibrium constant of the antibody-(TNF- α) binding pair is extracted to be $K_D = 369 \pm 48$ fM. Based on this calibrated sensor model, the time-dependent binding kinetics is also presented and the association/dissociation rates of the antibody-(TNF- α) pair are extracted to be $(5.03 \pm 0.16) \times 10^8$ M⁻¹s⁻¹ and $(1.97 \pm 0.08) \times 10^{-4}$ s⁻¹, respectively.

In chapter 4, two different physics principles for operating MoS₂ transistor biosensors are experimentally identified, which depend on antibody functionalization locations. If antibodies are functionalized on an insulating layer coated on a MoS₂ transistor, antibody-antigen binding events mainly modify the transistor threshold voltage, which can be explained by the conventional capacitor model. If antibodies are directly grafted on the MoS₂ transistor channel, the binding events mainly modulate the ON-state

transconductance of the transistor, which is attributed to the antigen-induced disordered potential in the MoS₂ channel.

In chapter 5, a cycle-wise method for operating emerging MoS₂/WSe₂-based transistor biosensors is studied to enable rapid, low-noise, highly specific biomolecule quantification at femtomolar levels. The cycle-wise detection approach can physically separate incubation, flushing, and electrical measurement steps in the device chamber and therefore avoid the liquid-solution-induced electrochemical damage, screening, and nonspecific adsorption to the sensor. This capability can improve the transistor sensor's durability, sensitivity, and specificity. Furthermore, the time-dependent response signals measured by the cycle-wise method exhibit lower noise as compared to those measured by the continuous detection method. These advantages in combination with the inherent high sensitivity of MoS₂/WSe₂-based biosensors allow for rapid biomolecule quantification at femtomolar levels by analyzing the initial slopes of time-dependent response curves. We demonstrate the cycle-wise quantification of streptavidin and interleukin 1 beta in pure and complex solutions with detection limit ~ 1 fM and a total incubation time less than 20 min.

Chapter 2

MoS₂ Transistors Fabricated via Plasma-Assisted Nanoprinting of Few-Layer MoS₂ Flakes into Large-Area Arrays

2.1 Introduction

Molybdenum disulfide (MoS₂) belongs to the family of layered transition metal dichalcogenides [37]. It has been broadly used as a dry lubricant and as a catalyst for desulfurization in petroleum refineries [52]. Recently, MoS₂ attracted a great deal of attention because of its attractive electronic, optoelectronic, and mechanical properties [25, 28, 53, 54]. In the bulk form, MoS₂ is an indirect band-gap semiconductor with an energy gap of ~1.2 eV [54]. In the monolayer form, MoS₂ has a large direct band gap (~1.8 eV) [54]. Therefore, MoS₂ can serve as an important complement to zero-band-gap graphene and enable new semiconductor-related applications of two-dimensional (2-D) materials such as thin-film transistors [25, 55], phototransistors [56], chemical sensors [57], integrated circuits [58], and thin-film light-emitting diodes [54, 59]. As a 2-D nanoelectronic material, MoS₂ is advantageous over bulk Si for suppressing the undesirable tunneling between drain and source regions at the scaling limit of transistors

and therefore provides benefits for miniaturization of electronic devices beyond Moore's law. In addition, bulk (or multilayer) MoS₂ shows relatively high in-plane carrier mobility comparable to that of crystalline Si [60], as well as robust mechanical and chemical properties, which makes it an attractive material for making flexible electronic devices with high performance and long lifetime [28, 58, 61]. A broad variety of prototype devices based on few layer MoS₂, such as high-performance field effect transistors [25, 55], phototransistors [56], sensors [33, 57], and integrated circuits [58], have been fabricated and extensively studied in research laboratories. However, the scale-up applications of MoS₂, especially the mass production of commercially feasible products, require large arrays of orderly arranged MoS₂ structures. This requirement breaks down into two critical challenges in nanomanufacturing: (1) incorporating pristine MoS₂ films over large areas and (2) patterning MoS₂ into ordered micro- and nanostructures over large areas to obtain both desirable electronic properties and required functionality. Several approaches have been attempted to produce MoS₂ materials for large-area applications, including Scotch tape exfoliation [1, 25, 61], liquid phase exfoliation in an organic solvent [62, 63], intercalation followed by forced hydration [59, 64, 65], transition metal sulfurization [66, 67], thermal decomposition of thiosalts [68], chemical vapor deposition (CVD) [69, 70], van der Waals epitaxial growth [71], etc. So far, a few efforts have been developed on the lithographic patterning of MoS₂ sheets and the deposition of MoS₂ crystals into ordered arrays [72-74]. All of these technologies for producing MoS₂ structures still suffer from one or more obstructions that prevent the creation of ordered, pristine MoS₂ device arrays over large areas. In particular, Scotch tape-based or liquid-phase exfoliation-based processes usually create a poor yield of few-

layer MoS₂ flakes [1, 25, 61, 62]. Chemical intercalation methods can produce a relatively high yield of monolayer or few-layer MoS₂ flakes in colloidal solutions, but cannot arrange them into ordered arrays over large areas [59, 64, 65]. CVD and epitaxial methods are promising to generate very thin MoS₂ flakes over large areas in the future, but the present as-grown MoS₂ samples still feature randomly distributed micro- or nanoflakes with 10's-100's nm scale crystalline domains that are much smaller than the crystalline domains achieved in geographic MoS₂ materials (typically, 1 to 100's μm size) [37, 69-71]. Obviously, a nanomanufacturing technology capable of being scaled up and producing ordered and pristine few-layer MoS₂ patterns would have a transformative impact on future manufacturing of MoS₂ electronic and optoelectronic devices and systems.

In this chapter, we present novel transfer-printing processes for generating large-area arrays of prepatterned few-layer MoS₂ features. In this work, bulk MoS₂ films are prestructured with relief patterns by using lithographic techniques and subsequently serve as stamps for printing out MoS₂ flakes on pristine and plasma-charged SiO₂ substrates. Here, SiO₂ is chosen as the substrate material, because SiO_x-based substrates are widely used for electronic applications. Therefore, the approaches developed in this work can be generally applied to other SiO_x-based substrates such as glass, fused silica, and flexible silicone rubber. The few-layer MoS₂ flake pixels printed on such SiO_x substrates can be used to create working transistors showing excellent performance. In the future, the presented printing approaches in combination with other nanolithography techniques, precise deposition and etching processes, and new high-k gate dielectrics can potentially be employed for producing high performance MoS₂-based large-scale integrated circuits.

2.2 Methods and Materials

2.2.1 Pre patterning of Bulk MoS₂ Stamps

To fabricate a bulk MoS₂ stamp, a piece of bulk MoS₂ is mechanically exfoliated from a MoS₂ source sample (SPI, Inc., size ~1 cm²) and firmly attached onto a bendable copper tape. The exfoliated MoS₂ film has a fresh and pristine surface with total area of ~1 cm². To pattern microscale relief features on the bulk MoS₂, a 1.4 μm thick photoresist layer is spun onto the MoS₂ surface and exposed on a Karl Suss MA6 photoaligner. The photomask used for this work bears periodic pillar patterns with pillar diameters ranging from 3 to 10 μm. After development, 100 nm thick Ti masks are formed on the MoS₂ surface by using electron-beam evaporation followed with lift-off in acetone. Afterward, the Ti mask patterns are etched onto the underlying MoS₂ using a SF₆- based RIE recipe (i.e., SF₆ flow rate 20 sccm, pressure 20 mTorr, power 200 W) with an etching rate of ~100 nm/min. Finally, the Ti masks are removed by soaking the MoS₂ film in a diluted HF acid solution.

2.2.2 Transfer Printing of Prepatterned MoS₂ Flakes onto Substrates

Prepatterned few-layer MoS₂ flakes are transfer printed onto SiO₂/Si substrates that are cleaned by using the standard RCA process. To perform a transfer printing, a

bulk MoS₂ stamp and a SiO₂/Si substrate are firmly pressed to each other by using a lab-made pressing system that can generate a gauge pressure up to 3 MPa for contact printing. To improve the bonding strength between the MoS₂ flakes and the SiO₂ surface, an O₂-based plasma recipe (i.e., O₂ flow rate 50 sccm, pressure 25 mTorr, power 100 W, time duration 2 min) is used to treat the SiO₂ surface before the printing step. Such a plasma treatment is expected to introduce uniformly distributed electric charges on the SiO₂ surface that can provide additional electrostatic attractive stress for exfoliating prepatterned few-layer MoS₂ flakes from the bulk stamp [75]. After the printing process, the printed MoS₂ patterns are imaged by using a scanning electron microscope in secondary electron and backscattered modes as well as an optical microscope. An atomic force microscope is employed to measure the thickness of printed MoS₂ features in the tapping mode. Furthermore, an X-ray energy dispersive spectrometer integrated with a SEM system is used to confirm the presence of MoS₂ features within the printed areas.

2.2.3 Fabrication of Field-Effect Transistors Using Printed MoS₂ Flakes

To fabricate back-gated MoS₂ FETs, the metallic drain/source contacts (5 nm Ti/55 nm Au) are fabricated by photolithography or electron-beam lithography followed by metal deposition and lift-off. Especially, photolithography is used for fabricating FETs based on the inner flakes of MoS₂ pixels, and EBL is specifically used for fabricating FETs based on the outer edge ribbons of MoS₂ pixels. Finally, another metallic contact is made onto the p⁺-Si substrate, which serves as a back gate contact. The device

characteristic curves of FETs are measured using an Agilent-4145B semiconductor parameter analyzer.

2.2.4 Plasma-Assisted Transfer Printing of Prepatterned Graphene Nanoribbons

First, a highly oriented pyrolytic graphite stamp (SPI, Inc.) is prepatterned with 100 nm half-pitch, 100 nm deep relief gratings. Such grating features are replicated from a master mold (Nanonex, Inc.) by using thermal nanoimprint lithography (T-NIL) followed by O₂-based plasma etching [76]. The T-NIL process was performed on a Specac thermal pressing tool equipped with a cooling-water system. The fabricated HOPG stamp is subsequently used to stamp out graphene nanoribbons onto plasma-charged SiO₂ substrates. The plasma-assisted printing process performed here is the same as the process used for printing microscale MoS₂ pixel arrays.

2.3 Results and Discussion

2.3.1 Experimental Results of Plasma-Assisted Transfer Printing

Figure 2.1 (a) schematically illustrates our approach for transfer printing prepatterned MoS₂ flakes. The fabrication process includes the following steps. (1) The process starts with a piece of pristine bulk MoS₂. (2) Photolithography is performed to pattern a photoresist layer spin-coated on top of the MoS₂ surface. (3) Arrays of metal masks are created by depositing 100 nm Ti followed with lift-off in acetone. (4) SF₆-

based reactive ion etching (RIE) is performed to transfer the Ti mask pattern onto underlying MoS₂ [74, 77, 78]. (5) Ti masks are completely removed in hydrofluoric (HF) acid, and a bulk MoS₂ stamp is created. (6) A SiO₂ substrate is treated with O₂ plasma to generate electric charges on the surface [75]. (7) Finally, the bulk MoS₂ stamp is used for

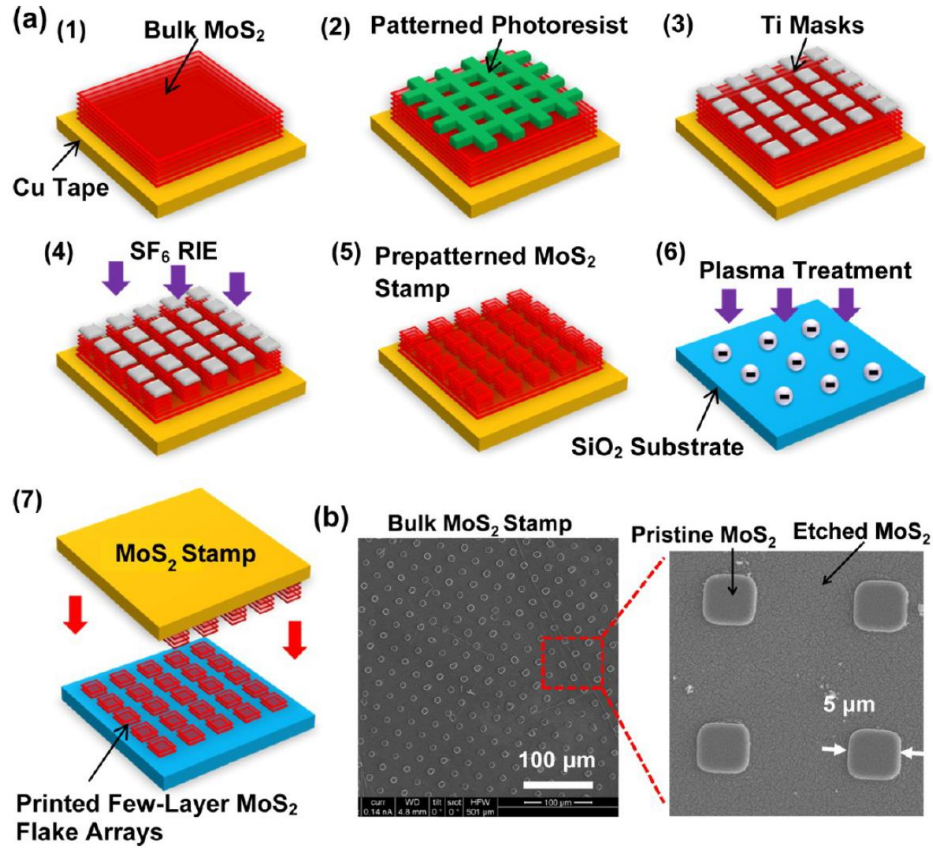


Figure 2.1 (a) Schematic flowchart of transfer printing of prepatterned few-layer MoS₂ flakes, which includes (1) initial bulk MoS₂ with a pristine surface; (2) photolithography for patterning device features; (3) formation of Ti masks by metal deposition followed by lift-off; (4) plasma etching of underlying MoS₂; (5) removal of Ti masks and finalization of a bulk MoS₂ stamp bearing relief features; (6) plasma treatment of the SiO₂ substrate; (7) direct transfer printing of prepatterned few-layer MoS₂ flakes onto the substrate. (b) SEM images of a bulk MoS₂ stamp prestructured with 5 μm size periodic pillars.

printing out MoS₂ flake arrays onto the SiO₂ substrate. This process can create MoS₂ device patterns directly from pristine geographic MoS₂ materials that have the largest

crystalline domains (typically, 1 to 100's μm size) and the best electronic properties to date [37, 69-71]. Moreover, this approach can be generalized for manufacturing other emerging atomically layered nanomaterials such as graphene [79-81], boron nitride [82, 83], and topological insulator thin films [84]. It should be noted that such a plasma-assisted transfer-printing process is significantly advantageous over voltage-based electrostatic exfoliation methods previously developed by Liang *et al.* for generating atomically layered materials [80], in terms of application scope and printing uniformity.

In particular, plasma-assisted printing can be applied to any substrates with a dielectric layer, whereas voltage-based exfoliation processes can only be utilized for conductive substrates. Furthermore, plasma-induced surface charges are usually immobilized and uniformly distributed over dielectric substrates and, therefore, result in uniform attractive stress for printing MoS_2 features over large areas. However, voltage-generated free charges are movable in the conductive substrate, and they tend to accumulate at locations with the smallest stamp/substrate gap, resulting in nonuniform electrostatic printing stress and a high risk of electrical leakage [80]. Figure 2.1 (b) shows scanning electron micrographs (SEMs) of an exemplary MoS_2 stamp prepatterned with 5 μm size, 600 nm high pillars. The zoomed view in Figure 2.1 (b) shows that the SF_6 plasma-etched area exhibits a relatively high roughness, which is attributed to plasma etching or ion bombardment. However, the raised pillar mesas protected by the Ti masks are still as smooth as a pristine MoS_2 surface. This should yield a conformal contact with the flat substrate during a mechanical printing process and therefore a high transfer-printing efficiency of MoS_2 flakes.

Figure 2.2 (a) and (c) display SEM images of MoS₂ flakes printed on a pristine SiO₂ surface. These images, captured over a large printed area (~1 cm²), show that the mechanical printing process can create large-area, orderly paved arrays of MoS₂ pixel features. However, most of the printed pixels have relatively irregular edge profiles that

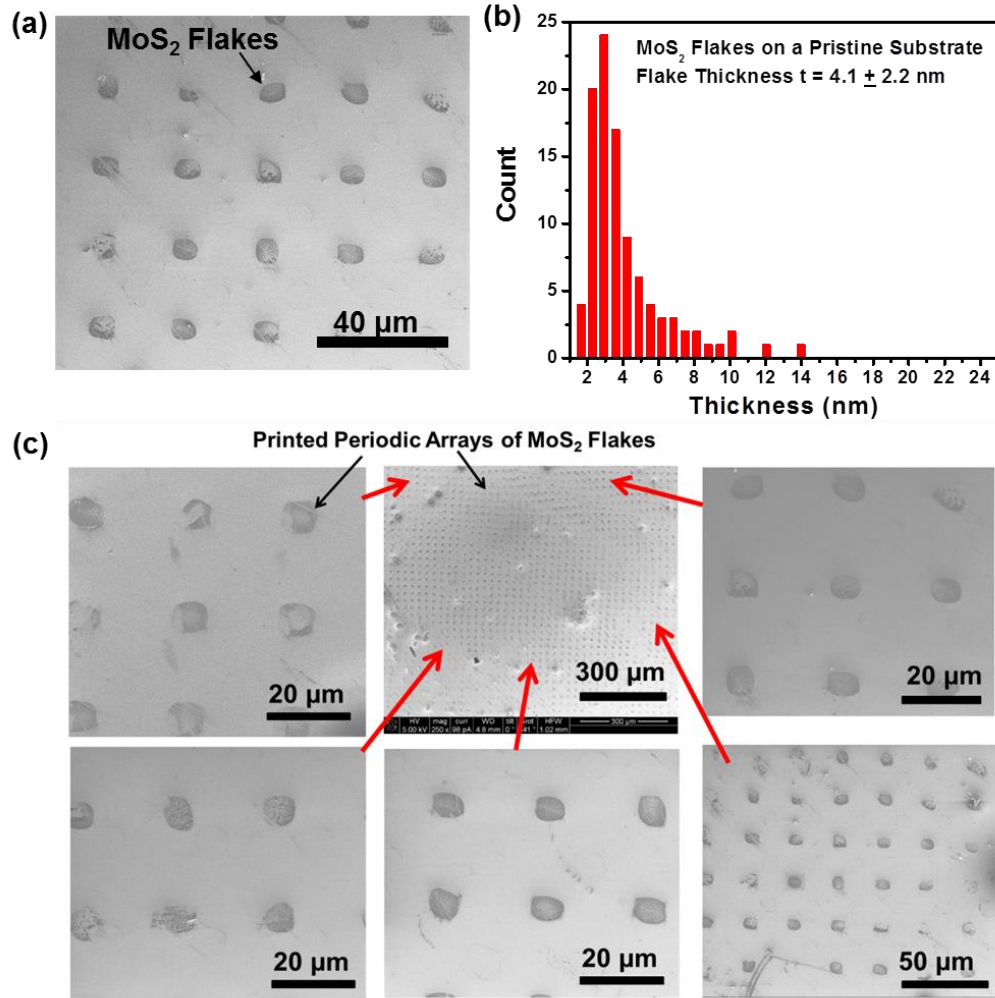


Figure 2.2 (a) SEM image of arrays of 10 μm size MoS₂ flake pixels printed onto a pristine SiO₂ substrate. (b) Stacked column chart of the average thickness data collected from 100 as-printed MoS₂ pixels. The thickness data were obtained from MoS₂ pixels printed over a ~1 cm² area by using an AFM. (c) SEM images of periodic arrays of 10 μm size MoS₂ flakes printed onto a pristine SiO₂ surface, which were acquired from different locations over the printed area, as indicated by the red arrows.

are not faithfully correlated to the edge profiles of pillars prestructured on the bulk stamp. The thickness data of MoS₂ pixels were obtained by using an atomic force microscope (AFM). For each of the printed MoS₂ pixels, the average flake thickness was extracted from AFM topographic data. Figure 2.2 (b) plots the statistical distribution of the average thickness data of 100 MoS₂ flake pixels of 10 μm size produced in a single transfer-printing cycle. Figure 2.2 (b) shows that the overall average flake thickness is measured to be 4.1 nm (~6 monolayers), and the standard deviation is 2.2 nm (~3 monolayers) over a ~1 cm² area. About 95% and 80% of printed MoS₂ flakes are thinner than 10 and 5 nm, respectively.

Figure 2.3 (a) shows an SEM image of MoS₂ patterns printed onto an O₂ plasma-charged SiO₂ substrate. Figure 2.4 lists more SEM images of MoS₂ patterns with various dimension sizes, which were captured from different locations over the printed area. These images show that the printing process on plasma-charged substrates can create large-area, orderly arranged arrays of MoS₂ flake pixels with a higher uniformity of pixel profiles in comparison with the printing result on a pristine substrate. In particular, MoS₂ pixel patterns feature clear, well-defined edge profiles that are faithfully correlated to the edge profiles of pillars prestructured on the bulk stamp. The zoomed image in Figure 2.3 (b) reveals that the clear edge profile of a MoS₂ pixel is indeed made up of a ring-shaped MoS₂ ribbon. Such outer edge ribbons of MoS₂ pixels have widths ranging from 200 to 400 nm. Besides these edge ribbon features, there are indeed thinner MoS₂ films or flakes located in the inner regions of printed pixels enclosed by the edge ribbons. These inner

MoS₂ flakes typically show a poor feature contrast in secondary-electron images. To enhance the SEM contrast, printed MoS₂ pixels were also imaged by detecting backscattered electrons (BSEs), as shown in Figure 2.3 (c-e), which are often used to detect contrast between areas with different chemical compositions. The BSE image contrast in Figure 2.3 (c-e) suggests the presence of thin MoS₂ flakes within each of the pixels.

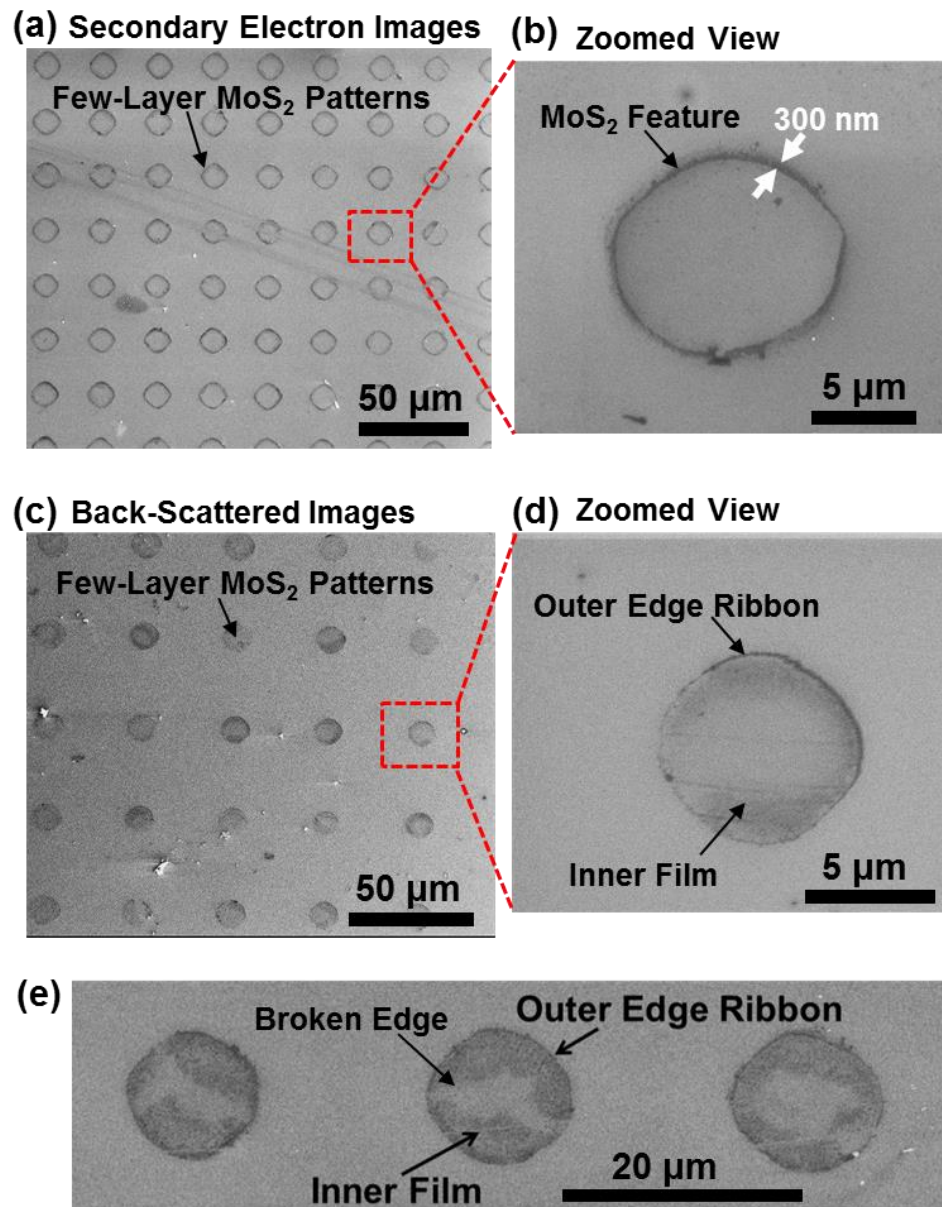


Figure 2.3 (a, b) Secondary-electron SEM images of MoS₂ pixel arrays printed onto an O₂ plasma-charged substrate, which exhibit clear, well-defined edge profiles faithfully correlated to the edge profiles of pillar features on the bulk MoS₂ stamps. (c, d) Backscattered SEM images of MoS₂ pixel arrays, which show the presence of thin inner MoS₂ flakes within each printed pixel. (e) Backscattered image of MoS₂ pixels with broken inner films.

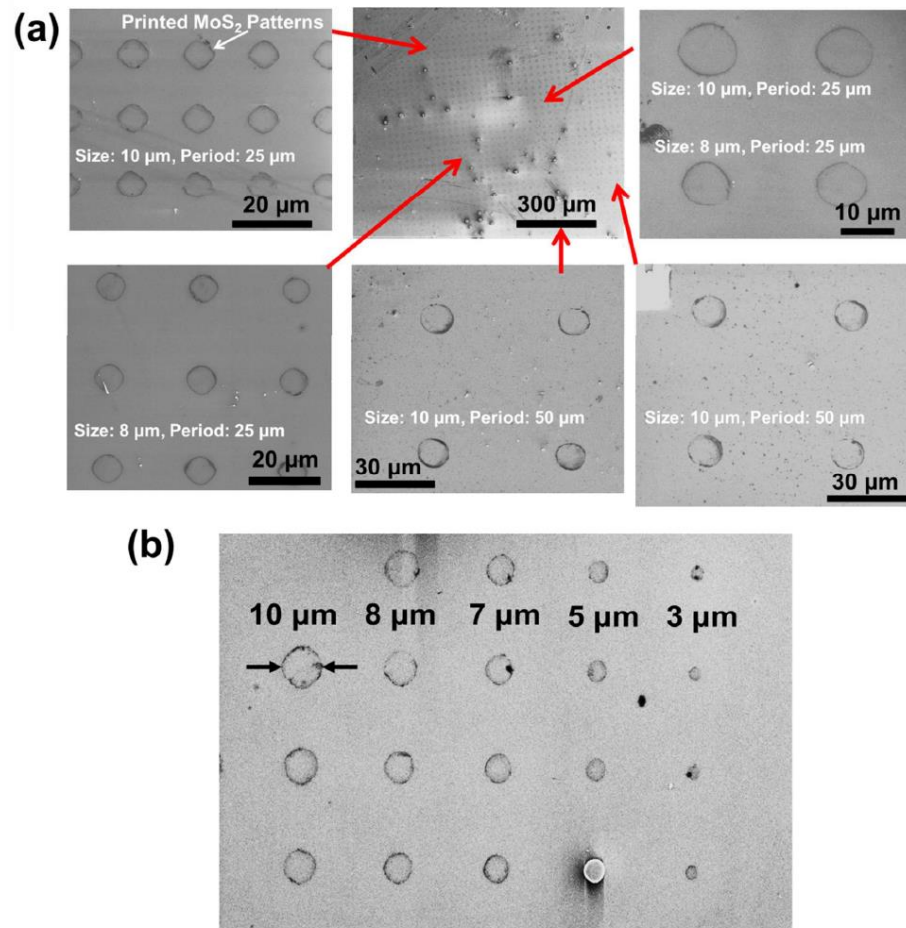


Figure 2.4 (a) SEM images of periodic arrays of MoS₂ patterns printed onto a plasma-charged SiO₂ surface, which were acquired from different locations over the printed area, as indicated by the red arrows. (b) SEM images of MoS₂ pixels with various feature sizes. All the MoS₂ features printed on the plasma-charged substrates exhibit clear, well-defined edge profiles that are faithfully duplicated from the edge profiles of the pillar features on the bulk MoS₂ stamps.

X-ray energy dispersive spectrometer (EDS) spectra were captured from the edge ribbons as well as the inner films of MoS₂ pixels, as shown in Figure 2.5. The EDS results confirm the presence of sulfur and molybdenum in both the edge and inner portions of printed pixels. Additionally, Figure 2.6 exhibits optical micrographs of printed MoS₂ pixels on a 330 nm thick SiO₂ substrate, and the feature contrast further confirms the presence of inner flakes within printed MoS₂ pixels. The spatial variation of BSE image contrast and the EDS intensity of MoS₂-associated peaks propose an interpixel variation of MoS₂ flake thickness over the printed substrate. To obtain the inner flake thickness data, MoS₂ pixels with partially broken inner films (*e.g.*, the pixels shown in Figure 2.3 (e) and Figure 2.6 (c)) were imaged by using AFM, and the thickness of an inner MoS₂ flake was measured from its broken edges.

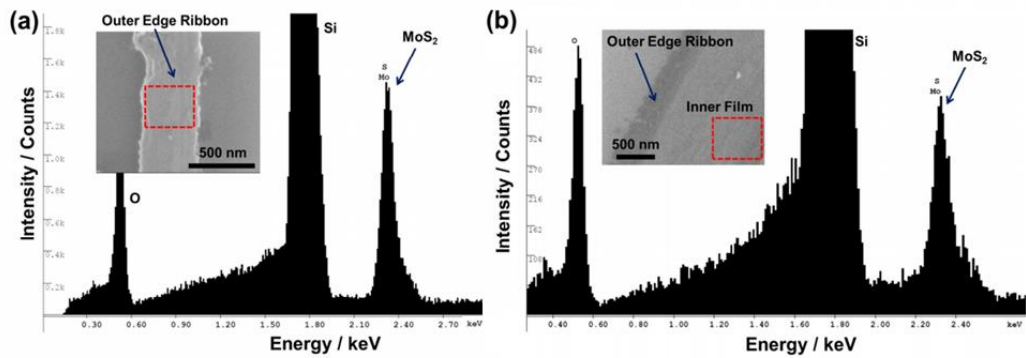


Figure 2.5 X-ray energy dispersive spectrometer (EDS) spectra of (a) the outer edge ribbon of a printed MoS₂ pixel and (b) the inner flake of a MoS₂ pixel. The dashed red squares in the inset SEM images indicate the locations on the samples where the EDS spectra were obtained. The EDS results further confirms the presence of sulfur and molybdenum in both the edge and inner portions of printed MoS₂ pixels.

Figure 2.7 (a) displays an AFM image of an exemplary MoS₂ pixel consisting of a relatively thick edge ribbon and broken inner flakes. The scan line denoted by the solid

line and accordingly plotted in Figure 2.7 (b) explicitly exhibits that the thickness values measured at the left and the right sides of this outer edge ribbon are 7 nm (~11 monolayers) and 8 nm (~12 monolayers), respectively; the thickness of the broken inner flakes is measured to be 2.4 nm (~4 monolayers). The thickness data acquired from 10 scan lines are used to calculate the average thickness of the inner film and the outer edge

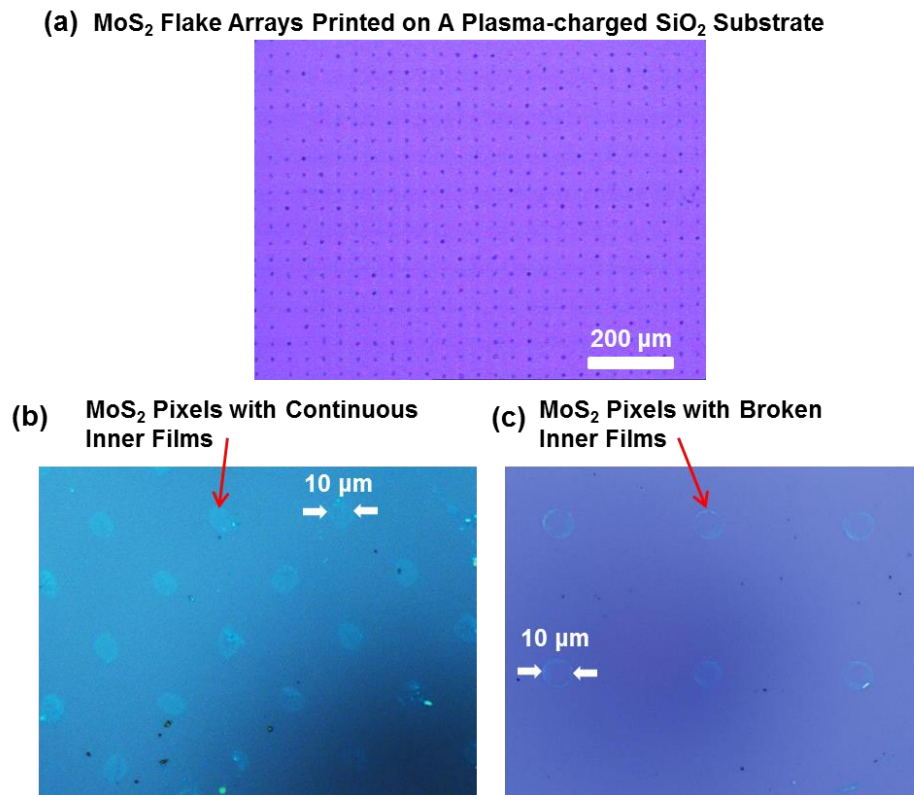


Figure 2.6 Optical micrographs of printed MoS₂ pixel arrays on a 330 nm thick SiO₂ substrate: (a) a low-magnification view; (b) a zoomed view of printed pixels with continuous inner films; (c) a zoomed view of printed pixels with broken inner films. All printed pixels exhibit regular edge profiles faithfully duplicated from prepatterned pillars on the bulk MoS₂ stamps.

ribbon of an individual MoS₂ pixel. Figure 2.7 (c) plots the statistical distribution of the average thickness data of 100 MoS₂ pixels printed on a plasma-charged substrate. Here, the thickness data of inner films (solid columns) and outer edge ribbons (blue hatched columns) of MoS₂ pixels are separately plotted. Figure 2.7 (c) shows that the overall average thickness of outer edge ribbons is 17 nm (~26 monolayers) with a standard deviation of 3 nm (~5 monolayers), whereas the overall average thickness of inner films is 3.0 nm (~5 monolayers) with a standard deviation of 1.9 nm (~3 monolayers). About 90% of inner flakes of MoS₂ pixels printed on a plasma-charged SiO₂ substrate are thinner than 5 nm (~8 monolayers). On the basis of such SEM, AFM, EDS, and optical micrograph characterizations, it is concluded that microscale MoS₂ pixels printed on a plasma-charged SiO₂ surface feature relatively thinner inner films or flakes enclosed by relatively thicker ring-shaped edge ribbons and a higher percentage yield of few-layer MoS₂ flakes thinner than 5 nm in comparison with pixels printed on a pristine SiO₂ substrate.

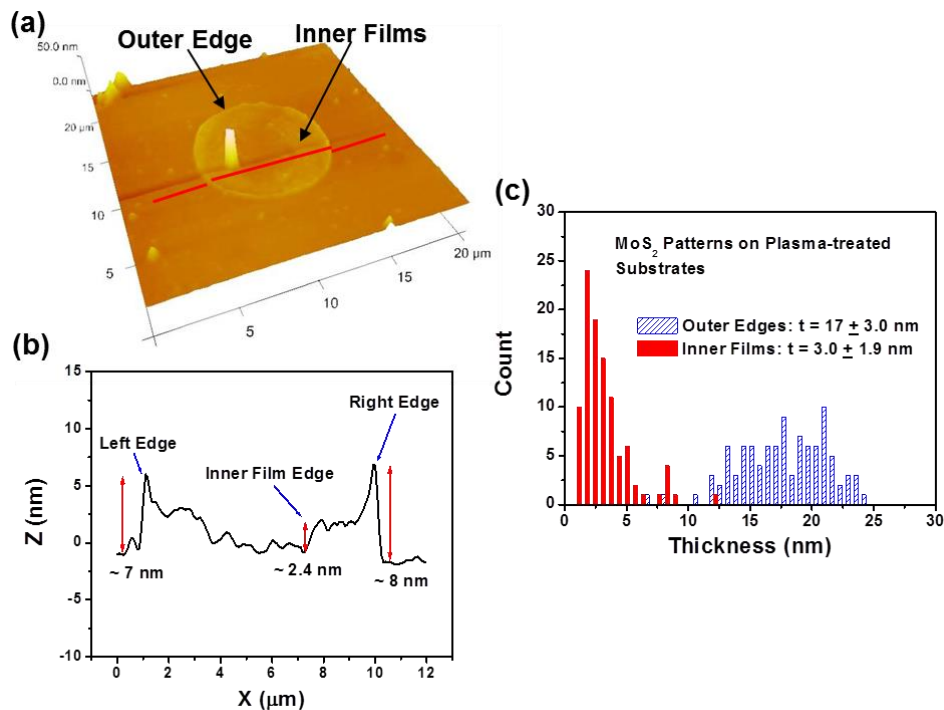


Figure 2.7 (a) AFM image of a 10 μm size MoS_2 pixel printed on a plasma-charged SiO_2 substrate. The solid line indicates a scanning trace across the pixel, which is explicitly plotted in (b). (c) Stacked column chart of the average thickness data collected from 100 as-printed MoS_2 pixels. Here, the thickness data acquired from the inner flakes (solid columns) and the outer edge ribbons (hatched columns) of these MoS_2 pixels are separately plotted.

Although plasma-assisted printing can produce large-area arrays of microscale MoS_2 pixels with regular edge profiles, many pixels have broken areas in their central regions, as shown in Figure 2.3 (e) and Figure 2.6 (c). This can be attributed to several possible reasons, including the limited size of crystalline domains in bulk MoS_2 , nonuniformity of attractive stress within a microscale MoS_2 pixel mesa, as discussed in the simulation analysis, and the paradigm rule that the direct exfoliation of a large-area atomic layer (e.g., a complete microscale MoS_2 pixel film free of defects) is thermodynamically unfavorable. Such an analysis suggests that it is indeed desirable to prepattern bulk MoS_2 stamps with densely arranged nanostructures that can enhance the

printing fidelity and eliminate the defects in the middle of printed patterns. In addition, such relief nanostructures are expected to generate a higher fringe field during printing processes because of the higher density of sharp feature edges and result in the higher transfer-printing efficiency of MoS₂ features. So far we have not developed a scalable process for patterning nanostructures on bulk MoS₂ because of its overly rough surface. However, we and other groups have successfully realized the nanopatterning of highly oriented pyrolytic graphite (HOPG) stamps with 100 nm half-pitch gratings using nanoimprint lithography, as shown in Figure 2.8 (a) [76, 85]. Figure 2.8 (b) shows SEM images of 100 nm half-pitch graphene nanoribbons (GNRs) produced by using plasma-assisted transfer printing. The printed GNRs exhibit a high degree of uniformity in ribbon widths over large areas and do not exhibit any visible defects in the middle of individual ribbons. The thickness of GNRs was measured to be 2.0 ± 1.0 nm by using an AFM. This work demonstrates that (1) nanoscale defect-free atomic layer patterns can be more easily produced by using plasma-assisted printing in comparison with microscale ones; (2) plasma-assisted printing can be generalized for producing high quality nanostructures of other atomically layered materials.

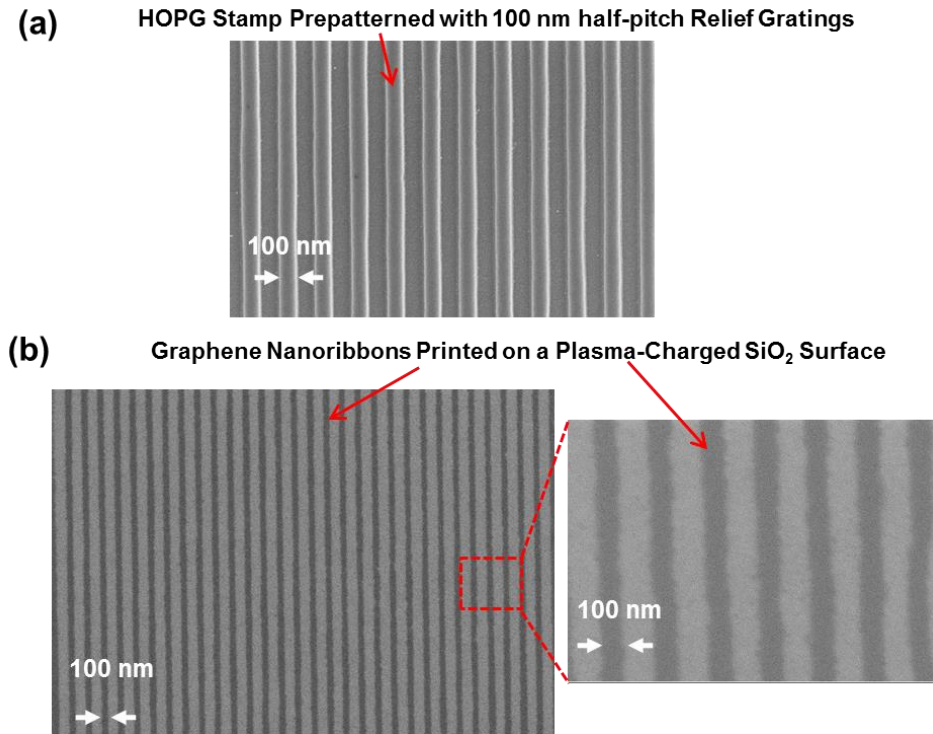


Figure 2.8 SEM images of (a) a HOPG stamp prepatterned with 100 nm half-pitch relief gratings by using nanoimprint lithography followed with plasma etching and (b) graphene nanoribbons printed onto a plasma-charged SiO₂ substrate.

2.3.2 Maxwell Stress Tensor Calculation

To obtain a preliminary understanding of plasma-assisted printing mechanisms responsible for the resultant morphology of MoS₂ pixels, Maxwell stress tensor calculation was performed and used for evaluating the distribution of surface charge-induced electrostatic attractive stress between the bulk MoS₂ stamp and the dielectric substrate [86, 87]. Figure 2.9 (a) illustrates the 2-D model for the calculation, in which a plasma-charged SiO₂ substrate is in contact with a bulk MoS₂ stamp and the surface charge density is arbitrarily set to 0.05 C/m² (currently we lack experimentally measured

data of surface charge densities). Figure 2.9 (b) plots the calculated attractive stress exerted by the plasma-charged SiO_2 substrate on the bulk MoS_2 stamp as a function of positions. Figure 2.9 (c) shows the zoomed view of the attractive stress distribution within a single MoS_2 mesa in contact with a SiO_2 surface. It is found that the attractive stress acting on a microscale MoS_2 mesa is uniform in the central region of the mesa but is significantly increased along the mesa edges due to the fringe effect. During a transfer printing process, such high attractive stress at the mesa edges is expected to result in the exfoliation of MoS_2 flake pixels with well-defined edges, as experimentally demonstrated. In addition, the strong electric field at the $\text{MoS}_2/\text{SiO}_2$ interface is expected to influence dispersion and dipole interactions of atoms there and therefore change the cohesive energy of MoS_2 layers close to the SiO_2 surface [88, 89]. This could lead to a dependence of the number of printed MoS_2 monolayers on the field magnitude, which could qualitatively explain our experimental result that for MoS_2 pixels printed on plasma-charged substrates the edge portions are statistically thicker than the inner flakes, as shown in Figure 2.7 (c). Our future theoretical simulation work will incorporate quantum mechanics, molecular simulations, and experimentally measured surface charge data to quantitatively analyze the effects of electric field on the number of printed MoS_2 monolayers.

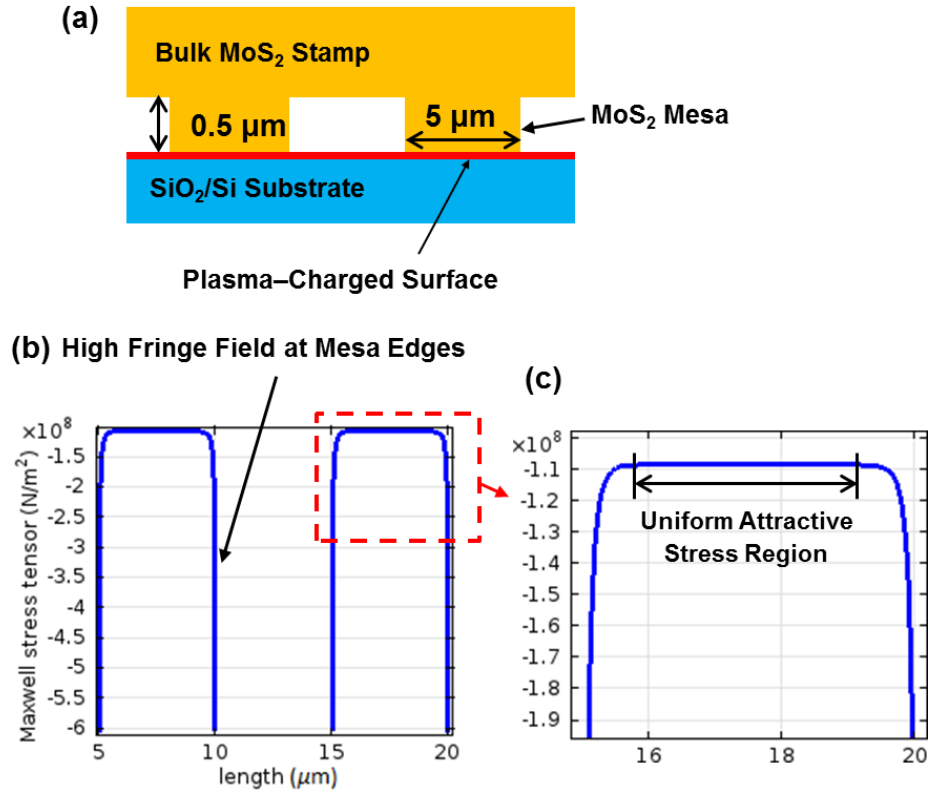


Figure 2.9 (a) Illustration of the 2-D model for Maxwell stress tensor calculation of surface charge-induced electrostatic attractive stress between the bulk MoS₂ stamp and the dielectric substrate. (b) Calculated attractive stress plotted as a function of positions. (c) Zoomed view of attractive stress distribution within a single MoS₂ mesa in contact with a SiO₂ surface. It is found that the attractive stress acting on a microscale MoS₂ mesa is uniform in the central region of the MoS₂ mesa but is significantly increased along the mesa edges due to the fringe effect.

2.3.3 Field-Effect Transistors (FETs) Made from Printed MoS₂ Flakes

To evaluate the electronic properties of printed MoS₂ flakes, we fabricated field-effect transistors (FETs) using MoS₂ pixels printed on plasma-charged SiO₂/p⁺ Si substrates. Figure 2.10 (a) shows a BSE image of an inner flake of a MoS₂ pixel that was used to fabricate a back-gated FET with flake thickness of ~5 nm, channel length of $L =$

5.4 μm , average channel width of $W \approx 3.7 \mu\text{m}$, and gate dielectric thickness of $d = 330 \text{ nm}$. Figure 2.10 (b) plots drain-source current (I_{DS}) versus drain-source voltage (V_{DS}) characteristics of this exemplary FET under different gate voltages (V_G) ranging from -75 to 100 V. Figure 2.10 (c) plots the I_{DS} - V_G characteristics under a fixed drain-source voltage ($V_{DS} = 10 \text{ V}$). As shown in Figure 2.10 (b) and (c), this FET displays n-type conduction with an ON/OFF current ratio (I_{ON}/I_{OFF}) of $\sim 10^7$. The transconductance at the linear region of the I_{DS} - V_G characteristic curve was obtained as $dI_{DS}/dV_G = 1.60 \mu\text{S}$ by the linear fitting (denoted with the red solid line in the inset of Figure 2.10 (c)). The field-effect mobility was estimated to be $\mu = 22 \text{ cm}^2/(\text{V s})$ by using equation (1) (valid for the linear region of MoS₂-based FETs with microscale channel widths), where ϵ_0 is the vacuum permittivity, $\epsilon_r \approx 3.9$ is the dielectric constant of SiO₂, C_{ox} is the gate capacitance, and W/L is the width/length ratio of the MoS₂ flake channel. The field-effect mobility values obtained from other FETs made from the inner flakes of printed MoS₂ pixels range from 6 to 44 $\text{cm}^2/(\text{V s})$, which are comparable to the highest mobility values previously reported for MoS₂ FETs using SiO₂ as the gate dielectric [61, 90]. This indicates that our transfer-printing approaches can produce high-quality MoS₂ features and are suitable for practical nanoelectronic applications.

$$\mu = \frac{\Delta I_{DS}}{C_{ox} \frac{W}{L} V_{DS} \Delta V_G} \quad C_{ox} = \frac{\epsilon_0 \epsilon_r}{d} \quad (1)$$

We also fabricated FETs using the ring-shaped edge ribbons of MoS₂ pixels as the semiconducting channels. To make an edge ribbon-based FET, electron-beam lithography (EBL) followed by metal evaporation and lift-off was performed to fabricate

drain and source electrodes precisely aligned to the specific segment of the edge ribbon of a MoS₂ pixel. In EBL, the overlay alignment was carefully performed to avoid incorporating any inner pixel flakes into the FET channel. Figure 2.11 (a) shows an SEM image of an exemplary edge ribbon-based FET with a channel width of $W \approx 300$ nm, channel length of $L \approx 500$ nm, and average MoS₂ thickness of ~ 10 nm. Figure 2.11 (b) and (c) displays $I_{DS}-V_{DS}$ and $I_{DS}-V_G$ characteristics, respectively, which show that this edge ribbon-based FET exhibits p-type conduction for $V_G = -100$ to 100 V. The transconductance at the linear region of the $I_{DS}-V_G$ characteristic curve was obtained as $dI_{DS}/dV_G = -1.74$ nS by the linear fitting (denoted with the red solid line in the inset of Figure 2.11 (c)). The field-effect mobility was estimated to be $\mu = 0.27$ cm²/(V s) by using equation (2), where C_g is the average gate capacitance associated with a single MoS₂ edge ribbon per unit channel length [unit: F/m]. Here, C_g is calculated by using a simulation model based on finite element analysis that takes into account the fringe effect at the MoS₂ nanoribbon edges, as shown in Appendix A. Such a fringe effect can significantly affect the values of C_g for MoS₂ FETs with nanoscale channel widths. The field-effect mobility values measured from other edge ribbon-based FETs range from 0.1 to 1.0 cm²/(V s). The p-type conduction is generally observed in other edge ribbon-based FETs, and it is attributed to the chemical doping to the edge portions of MoS₂ pixels, which might be induced during the SF₆ RIE process for patterning pillars in bulk MoS₂ stamps.

$$\mu = \frac{L}{C_g V_{DS}} \left(\frac{dI_{DS}}{dV_G} \right) \quad (2)$$

Our current printing approaches can generate few-layer MoS₂ flakes with the change of thickness mainly in the range 0.7-10 nm on a pristine SiO₂ substrate (Figure 2.2 (b)) or 0.7-5 nm on a plasma-charged substrate (Figure 2.7 (c)). As-printed MoS₂ pixels with such flake thickness distributions over large areas are still suitable for scale-up transistor-based electronic applications (if not for all applications), because previous works have demonstrated that FETs made from multilayer MoS₂ flakes with thicknesses

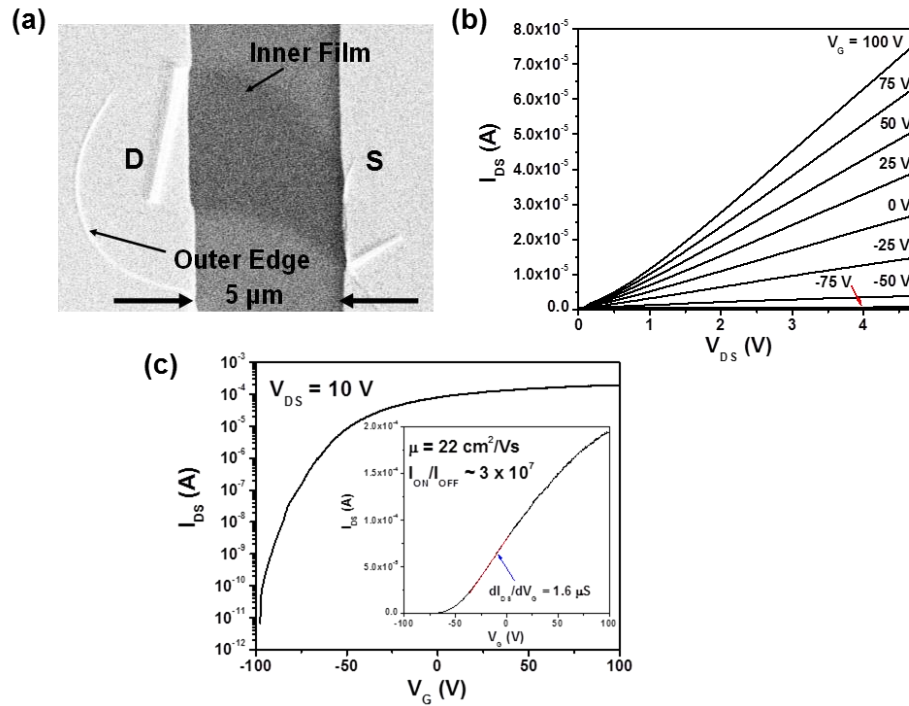


Figure 2.10 (a) BSE image of an exemplary back-gated FET made from the inner flake of a printed MoS₂ pixel with flake thickness of ~ 5 nm, channel width of ~ 3.7 μm , channel length of ~ 5.4 μm , and gate dielectric (SiO₂) thickness of 330 nm, in which Ti/Au contacts were deposited as drain (*D*) and source (*S*) contacts and the p⁺ silicon substrate serves as a back gate. (b) I_{DS} - V_{DS} characteristics under different gate voltages (V_G) ranging from -75 to 100 V. (c) Semilogarithmic plot of an I_{DS} - V_G characteristic curve under a fixed drain-source voltage $V_{DS} = 10$ V, which exhibits an ON/OFF current ratio (I_{ON}/I_{OFF}) $\approx 10^7$. The inset graph shows the linear plot of the same I_{DS} - V_G curve, and the transconductance (dI_{DS}/dV_G) is obtained by fitting the linear region of the I_{DS} - V_G curve, as indicated by the red line. The field-effect mobility is subsequently extracted to be $\mu = 22$ cm²/(V s) for this FET.

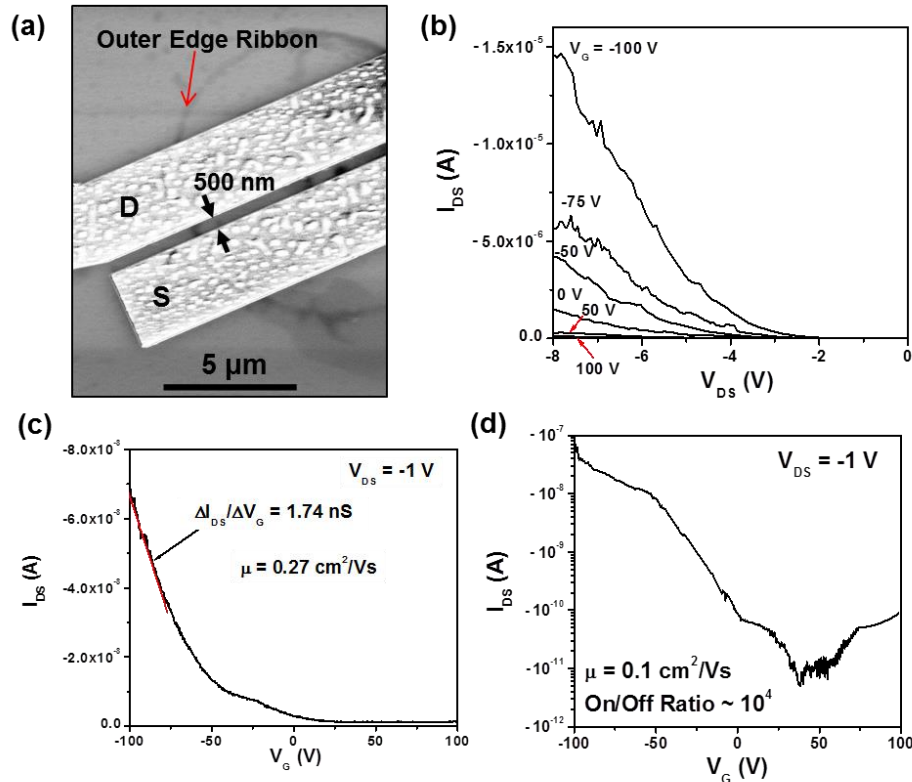
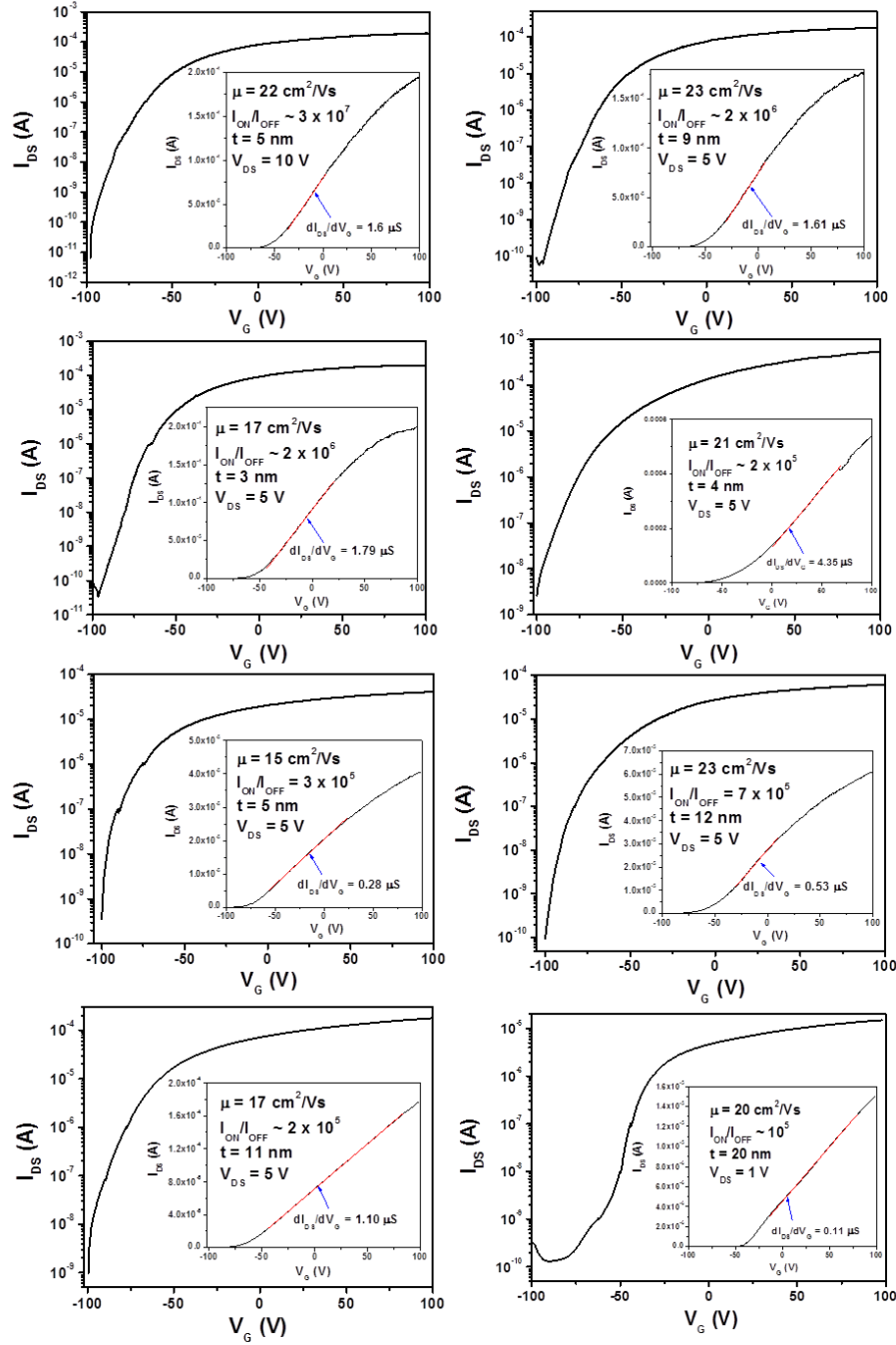


Figure 2.11 (a) SEM image of an exemplary back-gated FET made from the outer edge ribbon of a printed MoS₂ pixel with channel width of ~300 nm, channel length of ~500 nm, and gate dielectric (SiO₂) thickness of 330 nm. I_{DS} - V_{DS} (b) and I_{DS} - V_G (c) characteristics of this edge ribbon-based FET exhibit p-type conduction and field-effect mobility of ~0.27 cm²/(V s). (d) I_{DS} - V_G characteristics of a p-type FET made from a MoS₂ flake blank-treated by SF₆ plasma.

ranging from 2 to 50 nm exhibit excellent and stable transport properties (*i.e.*, high ON/OFF ratios ranging from 10⁴ to 10⁷, high field-effect mobility values on the order of 10s cm²/(V s) on SiO₂-based dielectrics and 100's cm²/(V s) on high-k dielectrics, as well as subthreshold slopes of 60-70 mV/decade for top-gated FETs) [91-93]. To evaluate the potential scalability of our printing approaches especially for future scale-up transistor-based applications, we studied the uniformity of the transport characteristics of MoS₂

FETs produced by plasma-assisted printing as well as the dependence of FET characteristics on the change of the MoS₂ thickness. Figure 2.12 shows the transport characteristics of 14 FETs produced in a single printing cycle. Figure 2.13 displays (a) ON/OFF current ratio and (b) field-effect mobility data measured from these FETs, which are plotted as a function of the MoS₂ thickness. These FETs have different flake thicknesses ranging from 3 to 20 nm. It should be noted that although the thickness values of most as-printed MoS₂ flakes can be controlled to be less than 10 nm, as shown in Figures 2.2 (b) and 2.7 (c), relatively thicker flakes (*i.e.*, thickness ~10-20 nm) were intentionally chosen for making FETs in order to extend the investigation range of the flake thickness values and analyze the degree of redundancy in the control of the MoS₂ thickness. Figure 10 shows that in spite of the variation of the MoS₂ flake thickness in the range 3-20 nm, all the FETs exhibit high ON/OFF ratios in the range 10⁵-10⁷, reasonably high field-effect mobility values on SiO₂ gate dielectrics ranging from 15 to 24 cm²/(V s), and uniform threshold gate voltages around $V_{th} \approx -50$ V. Such results preliminarily demonstrate that the MoS₂ FETs produced by plasma-assisted printing do not show a sensitive dependence of FET characteristics on the change of the MoS₂ thickness in the range 3-20 nm. This also proposes that our printing approaches can generate a high yield of electronic-grade MoS₂ flakes with an acceptable degree of uniformity in FET characteristics and hold significant potential to be further developed into a manufacturing process for making arrays of working FETs. In addition, using HfO₂-based high-k gate dielectrics, the mobility values of our FETs are expected to be further improved by at least 1 order of magnitude [25], which is attributed to the dielectric screening effect [94].

The large arrays of such high-performance FETs produced by printing processes are expected to significantly facilitate the future scale-up electronic applications of few-layer



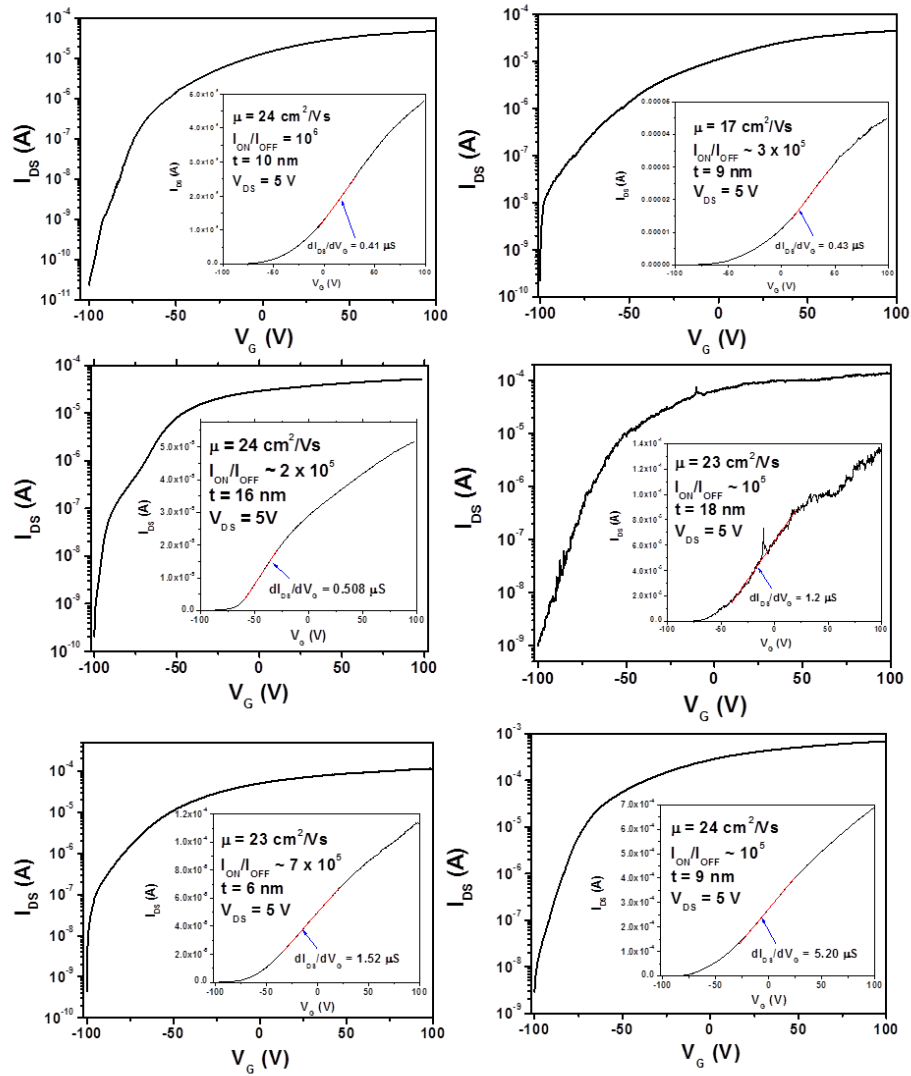


Figure 2.12 Semi-logarithmic I_{DS} - V_G characteristic curves of 14 back-gated FETs made from the inner flakes of MoS₂ pixels printed on a single substrate, which exhibit ON/OFF current ratios (I_{ON}/I_{OFF}) ranging from 10^5 to 10^7 . The inset graphs display the linear plots of the I_{DS} - V_G curves, and the field-effect mobility (μ) can be extracted by fitting the linear regions of I_{DS} - V_G curves. μ ranges from 15 to 24 cm²/Vs.

MoS₂. In our future research, we will integrate the top gates with sub-5 nm thick gate dielectrics into MoS₂ FETs in order to study the uniformity of other important FET parameters (e.g., subthreshold slopes) and further explore the potential of our printing processes for manufacturing high-performance MoS₂-based electronic devices.

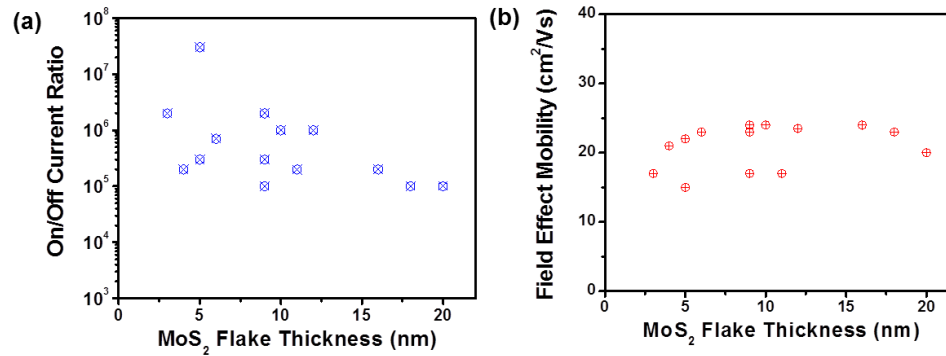


Figure 2.13 (a) ON/OFF current ratio and (b) field-effect mobility data extracted from the FET characteristics listed in Figure 2.12, which are plotted as a function of the MoS₂ flake thickness.

There are device applications requiring a more demanding control of the MoS₂ thickness. For example, optoelectronic applications usually require MoS₂ monolayers to obtain a large direct band gap [6]. For such applications, our printing approaches can potentially serve as a useful technique for transforming initial raw materials of few-layer MoS₂ into arrays of active device sites. These orderly formed MoS₂ flakes can be subsequently tailored through a series of postprinting processes to achieve a higher degree of uniformity in thickness and feature profile. For example, the laser-thinning technology with a self-termination mechanism recently developed by Castellanos-Gomez *et al.* could be used as a postprinting process for thinning as-printed MoS₂ flakes to increase the percentage yield of MoS₂ monolayers [72]. Furthermore, a postprinting lithography step (e.g., photolithography and nanoimprint) followed by plasma etching can be easily implemented to trim as-printed MoS₂ flakes into functional nanopatterns with specific shapes.

2.4 Summary

In conclusion, we demonstrate a novel approach for producing ordered arrays of few-layer-MoS₂ device features. In this process, the relief structures are prepatterned onto a bulk MoS₂ film, which serves as a stamp for printing out orderly arranged MoS₂ pixel patterns over cm²-scale areas on both pristine and plasma-charged SiO₂ substrates. MoS₂ pixels printed on plasma-charged substrates feature a higher degree of uniformity in pattern profiles and a narrower distribution of the MoS₂ flake thickness (i.e., 3 ± 1.9 nm) in comparison with those printed on pristine substrates. This is attributed to the strong fringe field around the feature edges that is induced by plasma-introduced electric charges. We demonstrate that such printing approaches can be generalized for producing other emerging atomically layered nanostructures (e.g., graphene nanoribbons). The printed MoS₂ flakes can be used to build working n-type FETs with superior electronic properties (i.e., $I_{ON}/I_{OFF} \approx 10^5$ - 10^7 , mobility $\mu \approx 6$ - 44 cm²/(V s)). Using additional plasma treatment processes, as-printed MoS₂ flakes can be doped to create p-type FETs. Finally, we systematically study the thickness-dependent characteristics of MoS₂ FETs and show that our printing processes can produce a high yield of electronic-grade MoS₂ flakes with an acceptable degree of uniformity in transport properties.

Chapter 3

Multiple MoS₂ Transistors for Sensing Molecule Interaction Kinetics

3.1 Introduction

Using field-effect transistor (FET)-based biosensors produced from nanowires (NWs) and carbon nanotubes (CNTs), researchers have demonstrated detection of cancer biomarkers from nM to fM range in serum [95-101], in vitro detection of nM proteins in cell growth systems [102, 103], and quantification of the affinities/kinetics of the protein interactions with fM-level sensitivities [104]. The fM-level limit-of-detection (LOD) achieved by such nanoscale FET biosensors for monitoring biomarker concentrations would enable label-free, single-molecule-level detection of trace-level amount biomarkers. The arrays of such biosensors with consistent transistor responses would serve as reliable lab-on-a-chip platforms for precisely determining the kinetics of various biomolecule interactions. However, serious constraints imposed on nanofabrication severely prohibit the reliable manufacturing of the affordable biosensor chips utilizing such one dimensional (1D) nanostructures [95, 99, 100]. In particular, high-quality, small-size NWs and CNTs are required to make biosensors with fM-level LOD for

concentration monitoring (or single-molecule-level LOD for trace-level amount detection) [105]. Especially, for trace-level amount detection, the critical dimensions of the sensing channels need to be comparable to the impact dimensions of charged molecules to maximize the gating effect due to the charged molecules and achieve very low LOD [98, 106, 107]. CNTs and many NWs are usually produced by using bottom-up synthesis methods (*e.g.*, chemical vapor deposition (CVD)). The community currently lacks proper top-down planar nanofabrication processes to produce ordered arrays of such nanostructures, which makes it very challenging to realize parallel high-throughput assay using biosensors made from these nanostructures. High-quality Si NW biosensor arrays can be made using top-down lithographic techniques [104]. However, the fabrication of such Si NW arrays usually requires expensive semiconductor-on-insulator (SOI) substrates and exquisite nanolithographic tools, which can result in a high processing cost and is not very suitable for manufacturing affordable (even disposable) assay chips for practical clinical biosensing applications. Emerging two-dimensional (2D) atomically layered materials, such as graphene, topological insulators (TIs), and TMDCs, recently attracted a great deal of interest because of their attractive electronic/ optoelectronic properties, large abundance, and compatibility to planar nanofabrication processes [25, 28, 53, 54, 58, 59, 61, 108-111]. Due to their atomically thin structures, the transport properties of 2D layers are highly sensitive to the external stimuli, which can enable new ultrasensitive 2D FETs suitable for biosensing applications [112-116]. Especially, in comparison with the thin film transistors made from conventional bulk semiconductors (*e.g.*, Si and III-V compounds), the transistors based on MoS₂ and other atomically layered semiconductors are expected to show much more sensitive electrical responses to

antigen-antibody binding events. Moreover, all 2D layers have an extremely low density of dangling bonds on their surfaces, which can result in high-quality FET channels with low densities of scattering centers (and hence low Flicker noise level), and enable highly sensitive, low-noise-level detecting of biomolecules [113, 117-120]. Novoselov *et al.* have demonstrated graphene-based FET sensors capable of detecting individual gas molecules absorbed on the graphene channels [10, 121]. In contrast to zero-bandgap graphene, semiconducting TMDCs (*e.g.*, MoS₂) have sizable bandgaps. Therefore, TMDC-based FETs show high On/Off current ratios up to 10⁸, which, in combination with their atomically thin structures, can potentially enable the higher detection sensitivities for gas, chemical, and biological sensing applications in comparison with graphene FETs [43, 57, 122]. Wang *et al.* and Sarkar *et al.* recently demonstrated that FET biosensors made from microscale few-layer-MoS₂ flakes exhibit 100-400 fM LODs for detecting cancer-related biomarkers [42, 43]. These previous works strongly imply that such TMDC-based FET biosensors may not require sensing channels of nanoscale width to achieve fM-level LODs for concentration monitoring applications, and the fabrication of such biosensors would not need exquisite nanolithographic tools. Additionally, several recent nanomanufacturing-related works suggest that monolayer/few-layer TMDC structures and other relevant atomically layered materials hold significant potential to be produced over large areas on low-cost substrates (*e.g.*, glass, plastic, or rubber) by using cost-efficient processes such as CVD followed with roll-to-roll transfer [123], addressable deposition [124], and microscale stamping [80, 125, 126]. Therefore, it is very promising to realize cost-efficient manufacturing of multiplexing assays based on TMDC transistor arrays in the future. Toward such

envisaged bio-assay capability, additional device-oriented research is required for quantitatively calibrating the sensor responses measured from multiple sets of TMDC FET biosensors, so that the calibrated response signals are consistent with each other and can synergistically enable precise quantification of biomarker concentrations (or amounts) as well as the affinities/kinetics of biomolecule interactions. Although individual MoS₂ FET biosensors have been fabricated and exhibited very high biodetection sensitivity [42, 43], the utilization of multiple such devices for quantifying the biomolecule interactions has not been attempted.

In this chapter, we fabricated multiple sets of MoS₂-based transistor biosensors and demonstrated that these devices can be synergistically utilized to measure the concentrations of analyte solutions as well as the affinity and kinetic properties of the analyte-receptor pair. The biomolecule under study, TNF- α , is a pro-inflammatory cytokine and a key biomarker associated with host defense and immunosurveillance [127-130]. Researchers have shown that TNF- α secreted from immune cells stimulated with lipopolysaccharide (LPS) - an endotoxin causing septic shock due to severely pronounced immune response of the human body - reflects a functioning innate immune response [131, 132]. All our biosensors showed a TNF- α detection limit as low as 60 fM despite the small molecular size of the cytokine biomarker (~17 kDa) that renders its label-free detection at high sensitivity significantly challenging. Such a low detection limit was achieved in both linear and subthreshold regimes of the transfer characteristics of MoS₂ transistors. In both transport regimes, the electrically measured sensor responses were calibrated into signal quantities independent of the transistor performance. All sets of transistor biosensors showed consistent relationships between calibrated sensor

responses and TNF- α concentrations. They created a standard curve, from which the equilibrium constant of the antibody-(TNF- α) pair was extracted to be $K_D = 369 \pm 48$ fM. Based on this calibrated sensor model, the time-dependent association-dissociation kinetics of the antibody-(TNF- α) pair was further studied and the association/dissociation rates of the antibody-(TNF- α) pair were measured to be $(5.03 \pm 0.16) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$ and $(1.97 \pm 0.08) \times 10^{-4} \text{ s}^{-1}$, respectively.

3.2 Methods and Materials

3.2.1 Fabrication and Characterization of MoS₂ Transistor Biosensors

The MoS₂ transistors were fabricated using a microprinting method previously reported [126]. Few-layer-MoS₂ channel thicknesses were specifically controlled to be 15-20 nm. Such a MoS₂ thickness range has been demonstrated to result in the optimal field-effect mobility values for MoS₂ transistors [93, 133]. The transistor channel lengths (L) were $\sim 5 \mu\text{m}$ and the channel widths (W) ranged from 5 to 8 μm . Ti (5 nm)/Au (50 nm) electrode pairs served as drain (D) and source (S) contacts, which were produced using photolithography followed with metal deposition and lift-off. The p⁺-Si substrates were used as the back gates (G). Thermally grown SiO₂ layers (300 nm thick) were used as the back-gate dielectrics. Such 300 nm thick SiO₂ layers can enable a simple color coding method for us to quickly identify MoS₂ flakes with appropriate thicknesses (*i.e.*, 15-20 nm) [134]. All electrical measurements were performed using an HP-4145B semiconductor parameter analyzer.

3.2.2 Bio-Functionalization of MoS₂ Transistor Biosensors

First, an as-fabricated transistor biosensor is immersed in 5% (3-Aminopropyl) triethoxysilane (APTES, purchased from Sigma-Aldrich Co. LLC.) in ethanol for 1 hour. After the incubation, the sensor is rinsed with phosphate buffered saline (PBS) and blown dry by nitrogen gas. After this step, the HfO₂ effective layer is silanized with an APTES monolayer. The device is subsequently immersed in 5% gluteraldehyde (GA) (purchased from Sigma-Aldrich Co. LLC.) in PBS for 2 hours followed by rinsing with PBS. Afterwards, anti-human TNF- α antibody (from eBioscience, Inc.) of 50 $\mu\text{g/ml}$ concentration in DI water is dropped on the sensor and incubated for 1 hour. For studying the equilibrium-state sensor responses, the as-functionalized sensor is incubated with TNF- α solutions with incremental concentrations (i.e., $n = 60 \text{ fM}, 300 \text{ fM}, 600 \text{ fM}, 3 \text{ pM},$ and 6 pM ; the incubation time for each of the concentrations: ~ 2 hours). The incubation is performed using the setup illustrated in Figure 3.1 (d). After each incubation process, the transfer characteristics of the transistor sensor are measured.

3.2.3 Quantification of the Time-Dependent Association/Dissociation Kinetics of the Antibody-(TNF- α) Pair

An as-functionalized MoS₂ transistor biosensor is covered with a polydimethylsiloxane (PDMS) block bearing a microfluidic channel (10 mm in length, 200 μm in width, 50 μm in height), as illustrated in Figure 3.1 (e). A motorized syringe pump is used for driving the analyte flows into and out of the microfluidic channel through an inlet/outlet tubing kit (tube diameter: 0.75 mm). At the beginning of the

measurement of a real-time sensor response curve associated with antibody-(TNF- α) binding, deionized (DI) water is injected into the sensor with flow rate of 5 $\mu\text{L}/\text{min}$. At the same time, the MoS₂ transistor is biased under a given set of V_G and V_{DS} . After the I_{DS} value is stabilized, the analyte solution with a specific TNF- α concentration is inserted into the sensor.

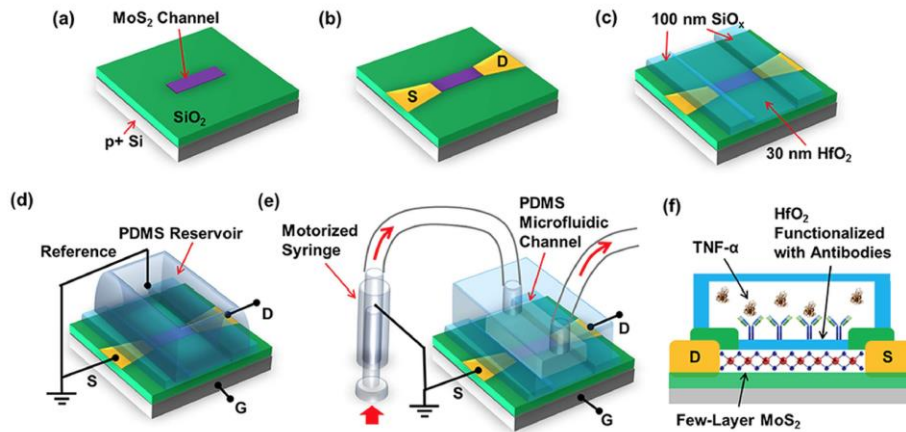


Figure 3.1 Flow chart for fabricating a MoS₂ transistor biosensor: (a) printing of a few-layer MoS₂ flake onto a p⁺-Si/SiO₂ substrate; (b) fabrication of Ti/Au D/S contacts; (c) ALD growth of the HfO₂ effective layer on top of the MoS₂ channel and coating of D/S contacts with thick SiO_x layers; (d) integration of a PDMS liquid reservoir on top of the MoS₂ transistor for measuring sensor responses from different TNF- α concentrations under thermodynamic equilibrium condition and determining the affinity of the antibody-(TNF- α) pair; (e) integration of a microfluidic inlet/outlet tubing kit driven by a motorized syringe pump on top of the transistor for quantifying the association-dissociation kinetics of the antibody-(TNF- α) pair; (f) functionalization of the HfO₂ effective layer with antibody receptors and subsequent TNF- α detection.

3.3 Results and Discussion

3.3.1 Static Measurement Results of Few-Layer MoS₂ Transistor Biosensors

Figure 3.1 displays the steps for fabricating a transistor biosensor with a few-layer MoS₂ sensing channel. First, a pristine few-layer MoS₂ flake is printed onto a p⁺-doped Si substrate coated with 300 nm thick SiO₂ (Figure 3.1 (a)). This printing process is the same as the method previously reported by us [126]. The thickness of the MoS₂ flake chosen for making a biosensor is specifically controlled to be 15-20 nm, aiming to achieve relatively high field-effect mobility values ($\mu = 20$ to $30 \text{ cm}^2/\text{V s}$) [93, 133]. After the MoS₂ printing, metallic drain/source (D/S) contacts (5 nm Ti/50 nm Au) are fabricated using photolithography followed with metal deposition and lift-off, and a back-gated MoS₂ transistor is subsequently formed (Figure 3.1 (b)). To enable a capacitive coupling between the microfluidic reservoir (or channel) and the MoS₂ transistor channel, a 30 nm thick HfO₂ layer is deposited on top of the MoS₂ channel using atomic layer deposition (ALD) (Figure 3.1 (c)). This HfO₂ layer also serves as an effective layer for biofunctionalization. Afterwards, additional 100 nm thick SiO_x is sputtered on D/S contacts to minimize the leakage current between D/S contacts and microfluidic components (Figure 3.1 (c)). Before the TNF- α detection, anti-human TNF- α antibody is functionalized on the HfO₂ effective layer. The detailed antibody functionalization procedure is illustrated in Figure 3.2. To measure the MoS₂ transistor sensor responses from different TNF- α concentrations under the thermodynamic equilibrium condition and determine the affinity of the antibody-(TNF- α) pair, a large open liquid reservoir made from polydimethylsiloxane (PDMS) is integrated on top of the MoS₂ transistor (Figure

3.1 (d)). Such a setup is very simple and enables the quick loading of various analyte solutions. To measure the association-dissociation kinetics of the antibody-(TNF- α) pair, a microfluidic channel is integrated on top of the transistor sensor, and a motorized syringe pump is used for driving the TNF- α solution flow into and out of the microfluidic

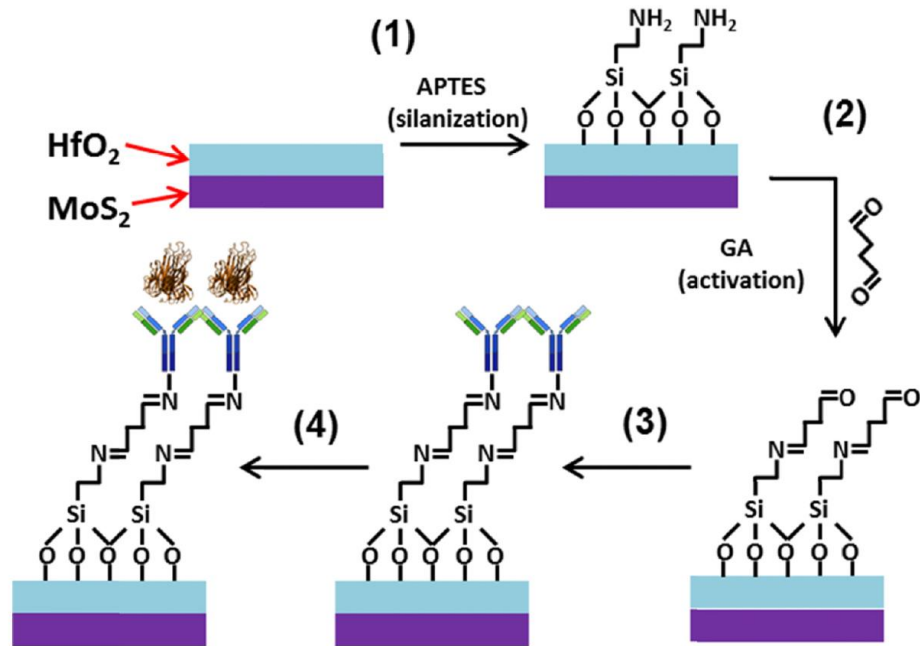


Figure 3.2 Protocol for functionalizing a MoS₂ transistor sensor with anti-human TNF- α antibody receptors for detecting TNF- α molecules

channel through an inlet/outlet tubing kit (Figure 3.1 (e)). Such a setup can enable stable laminar flows of analyte solutions and minimize the noise induced by the liquid loading processes, which is needed to precisely analyze the real-time kinetic processes of antibody-(TNF- α) binding. Figures 3.1 (d) and (e) also illustrate the circuit setups for measuring the transistor sensor responses. In addition, Figure 3.1 (f) illustrates the cross-sectional view of a MoS₂ transistor sensor in the TNF- α detection operation.

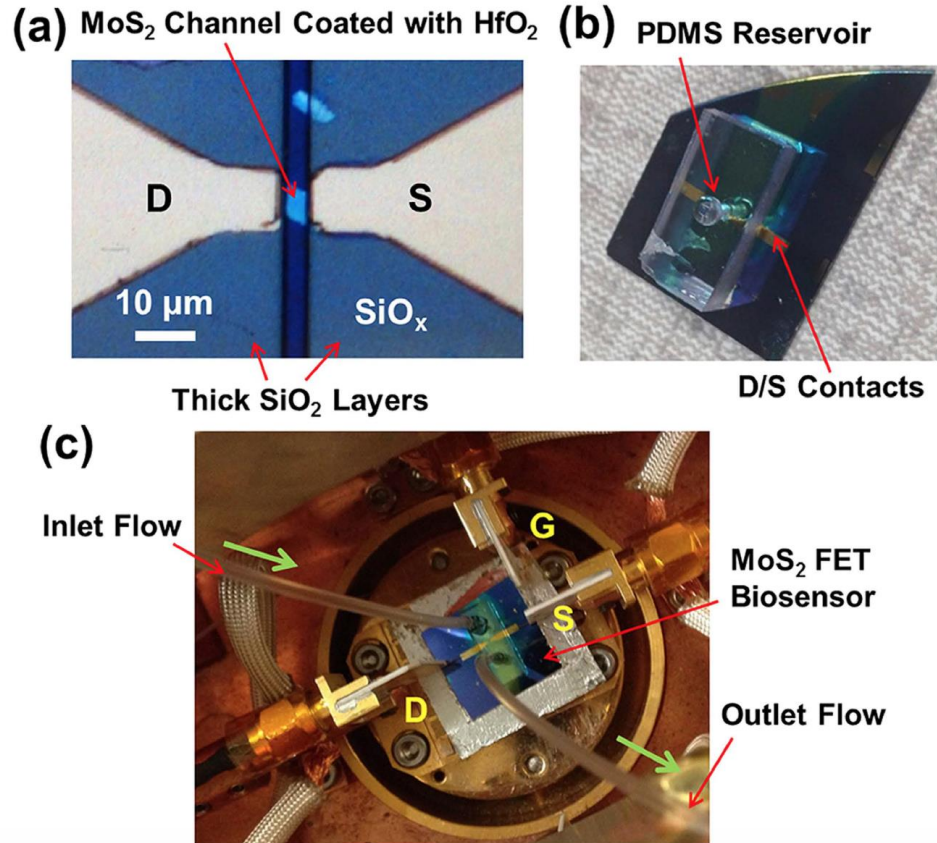


Figure 3.3 Optical micrographs or photographs of (a) an exemplary MoS₂ transistor with channel length (L) and width (W) of 5 and 6 μm, respectively; (b) an as-fabricated MoS₂ transistor biosensor integrated with a cylindrical liquid reservoir, which is drilled into a PDMS block and is ~4 mm deep and ~1 mm in diameter; (c) a transistor biosensor integrated with a microfluidic channel system connected with an inlet/outlet tubing kit, which is driven by a motorized syringe pump.

Figure 3.3 (a) displays the optical micrograph (OM) of an exemplary MoS₂ transistor with channel length (L) and width (W) of 5 and 6 μm, respectively. Figure 3.3 (b) shows the photograph of an as-fabricated MoS₂ transistor biosensor integrated with a PDMS liquid reservoir. The reservoir is ~4 mm deep and is ~1 mm in diameter, which is drilled into a PDMS block with length, width, thickness of 2, 1, and 0.4 cm, respectively. Figure 3.3 (c) displays the photograph of a biosensor integrated with a microfluidic channel connected with an inlet/outlet tubing kit.

First we measured the sensor responses from different TNF- α concentrations under the thermodynamic equilibrium condition. The biosensor setup shown in Figure 3.3 (b) was used for this measurement. For each transistor biosensor, the static transfer characteristics (*i.e.*, drain-source current (I_{DS}) – back gate voltage (V_G) curves acquired under a fixed drain-source voltage (V_{DS}) were measured at each of the biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM. To remove the effect of the I_{DS} - V_G hysteresis, all I_{DS} - V_G curves were measured by sweeping V_G from -100 V to 100 V with a sweep rate of 10 V/s.

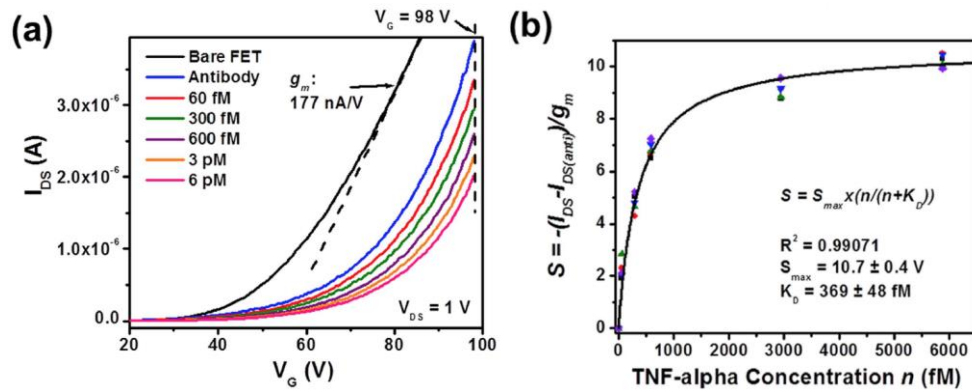


Figure 3.4 Sensor responses measured in the linear transport regimes of MoS₂ transistor biosensors: (a) transfer characteristics of an exemplary MoS₂ transistor sensor measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM; (b) a set of calibrated linear-regime responses (S) measured from five different MoS₂ transistor sensors with respect to TNF- α concentration (n). These S - n relationships can be well fitted with Langmuir isotherms and the dissociation constant (K_D) of the antibody-(TNF- α) pair is extracted to be 369 ± 48 fM.

Figure 3.4 shows the sensor responses measured in the linear transport regimes of MoS₂ transistor sensors. Specifically, Figure 3.4 (a) shows the transfer characteristics of

an exemplary sensor measured at various biodetection stages. Here, I_{DS} data are plotted in the linear scale. The transfer characteristics of this sensor show a strong dependence on TNF- α concentrations, and the TNF- α detection limit is estimated to be ~ 60 fM. We choose a fixed V_G within the linear regimes of all I_{DS} - V_G curves (e.g., $V_G = 98$ V, as denoted by the dashed vertical line in Figure 3.4 (a)). The I_{DS} values measured under this V_G vary according to different biodetection states and such I_{DS} data could be used as a sensor response signal. However, such a response signal is highly dependent on the transistor performance parameters (e.g., transconductance (g_m) and threshold voltage (V_T)). Therefore, in the analysis of a given biodetection state, the I_{DS} signals obtained by different MoS₂ transistors may exhibit a poor device-to-device consistency due to the nonuniformity of MoS₂ transistors. Although such an issue could be alleviated through optimizing the material deposition and device fabrication processes, a calibrated sensor response quantity independent of the device performance is highly desirable.

$$I_{DS} = g_m \left(V_G - V_T - \frac{V_{DS}}{2} \right) \quad (1)$$

$$S = -\frac{I_{DS} - I_{DS(anti)}}{g_m} = \Delta V_T = \frac{qd_{SiO_2}\sigma_{TNF}}{K_{SiO_2}\epsilon_0} \quad (2)$$

The linear regime of an I_{DS} - V_G characteristic curve measured from a microscale MoS₂ transistor sensor in a specific biodetection state can be expressed as Equation (1). In our experiments, it is observed that for a given transistor sensor, the g_m values extracted from different I_{DS} - V_G curves that correspond to different biodetection states are

very close and can be approximated as a constant for this sensor. For example, the g_m value of the sensor shown in Figure 3.4 (a) is extracted to be ~ 177 nS at $V_{DS} = 1$ V. Based on this observation and Equation (1) as well as the implication from previous works done by Duan *et al.* and Ishikaw *et al.* [104, 135], a calibrated sensor response quantity (S) is derived and expressed in Equation (2), where $I_{DS(anti)}$ is the I_{DS} value measured in the “antibody functionalization” state of a sensor biased under a set of fixed V_{DS} and V_G , and $I_{DS} - I_{DS(anti)}$ indicates the I_{DS} variation induced by the introduction of TNF- α molecules. Such an I_{DS} variation normalized by the g_m of this sensor results in a sensor response quantity directly associated with the change in the V_T of the sensor (*i.e.*, ΔV_T). It should be noted that although ΔV_T is assumed to be completely induced by the charge brought to the HfO₂ effective layer on top of the transistor channel through antibody-(TNF- α) binding events, ΔV_T is not exactly the binding-event-induced potential change ($\Delta\Phi$) on the effective layer. This is because in this work, ΔV_T is the change in the V_T measured from the back gate. However, ΔV_T and $\Delta\Phi$ can be related by $\Delta V_T = (C_{HfO_2}/C_{SiO_2})\Delta\Phi$, where C_{SiO_2} and C_{HfO_2} are the capacitances of the SiO₂ back gate dielectric and the HfO₂ effective layer, respectively. More detailed discussion about these potential parameters can be found from the dual-gate transistor model illustrated in Appendix B. Based on this model, ΔV_T can be evaluated using $\Delta V_T = qd_{SiO_2}\sigma_{TNF}/k_{SiO_2}\epsilon_0$, where q is the effective charge carried by a TNF- α molecule (the screening effect due to the buffer liquid has been incorporated into q); d_{SiO_2} and k_{SiO_2} are the thickness and dielectric constant of the SiO₂ back-gate dielectric layer, respectively; ϵ_0 is the vacuum permittivity; and σ_{TNF} is the areal density of TNF- α molecules bound to the antibody receptors functionalized on the effective layer. Therefore, such a calibrated response quantity (S) is proportional to the

antibody receptor occupancy at the equilibrium state and it is also independent of the MoS₂ transistor performance. These two conditions are important for the subsequent Langmuir isotherm analysis.

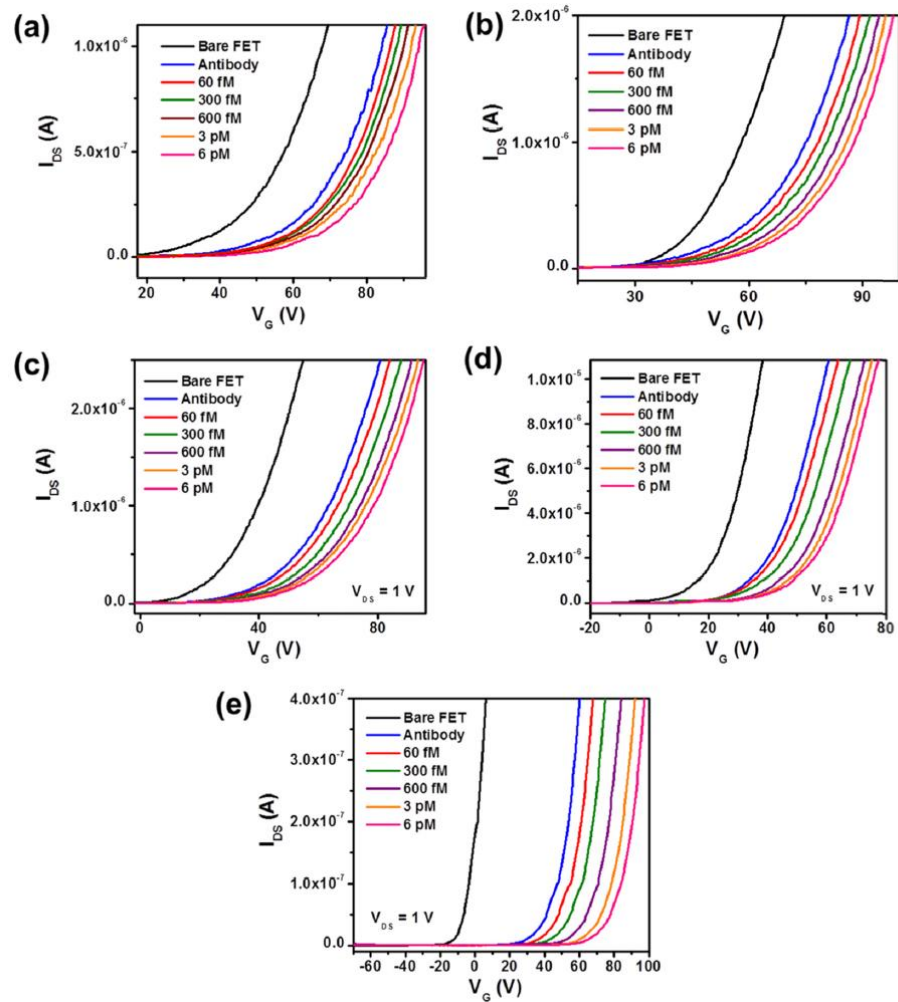


Figure 3.5 Linear-regime sensor responses at the equilibrium state: The transfer characteristics of five different MoS₂ transistor sensors measured at various biotransduction stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM.

Figure 3.4 (b) plots the calibrated responses measured from the linear transport regimes of five different sensors with respect to TNF- α concentration (n). The detailed transfer characteristics of these five devices measured at various biodetection stages are showed in Figure 3.5. Although Figure 3.5 shows that the transfer characteristics of these five sensors exhibit significant difference in V_T , I_{DS} and g_m , Figure 3.4 (b) shows that the calibrated responses from these sensors are consistent with each other and can serve as a standard curve (*i.e.*, a generic S - n curve) for TNF- α detection. This standard curve can be well fitted with Langmuir isotherms (Equation (3)) and the affinity equilibrium (or dissociation) constant (K_D) of the antibody-(TNF- α) pair is extracted to be 369 ± 48 fM; the maximum sensor response (S_{max}) is extracted to be 10.7 ± 0.4 V.

$$S = S_{max} \frac{n}{n + K_D} \quad (3)$$

Alternatively, sensor responses can also be measured from the subthreshold regimes of MoS₂ transistor sensors. In the subthreshold regime of a transistor sensor, the sensitivity of I_{DS} to the variation of electrical potential (or charge) at the effective layer is much higher than that in the linear transport regime of this sensor. Therefore, the responses from the subthreshold regimes of transistor sensors are expected to result in the higher biodetection sensitivity in comparison with those from the linear regimes. Figure 3.6 (a) displays the transfer characteristics of another exemplary MoS₂ transistor sensor, which were measured at various biodetection stages. Here I_{DS} data are plotted in the logarithm scale, and the subthreshold regimes are emphasized. We choose a fixed V_G within the subthreshold regimes of all I_{DS} - V_G curves (*e.g.*, $V_G = 29$ V denoted by the

vertical dashed line in Figure 3.6 (a)). The I_{DS} values measured under this V_G clearly vary according to different biodetection states and exhibit a strong dependence on TNF- α concentration. Here, the TNF- α detection limit is estimated to be at least as low as 60 fM. Similarly, such I_{DS} data acquired in the subthreshold regimes of transistor sensors cannot be directly used as standard sensor responses. A calibrated subthreshold-regime response quantity independent of the transistor performance is required.

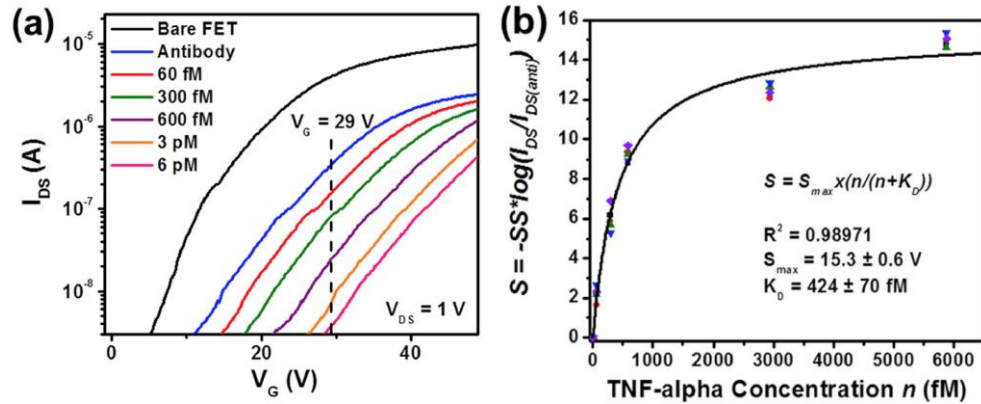


Figure 3.6 Sensor responses measured in the subthreshold regimes of MoS₂ transistor biosensors: (a) transfer characteristics of an exemplary MoS₂ transistor sensor measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM (Here I_{DS} data are plotted in the logarithm scale, and the subthreshold regimes are emphasized); (b) a set of calibrated subthreshold-regime responses (S) measured from five different MoS₂ transistor sensors with respect to TNF- α concentration (n). These S - n relationships can be well fitted with Langmuir isotherms and the dissociation constant (K_D) of the antibody-(TNF- α) pair is extracted to be 424 ± 70 fM.

In the subthreshold regime of a microscale MoS₂ transistor sensor, the I_{DS} - V_G relationship measured from a specific biodetection state can be approximately expressed by Equation (4), where I_T is the I_{DS} value measured at $V_G=V_T$ under a given V_{DS} ; SS is the subthreshold swing. As observed in our experiments, although the functionalization of a

transistor sensor with antibody receptors (*i.e.*, the transition from “bare transistor” to “antibody functionalization” states) can result in an observable reduction of the SS of this sensor, the SS value does not significantly vary among the subsequent biodetection states, including the inputs of TNF- α samples with incremental concentrations. Therefore, for a given as-functionalized transistor sensor, SS can be approximated as a constant. Based on this observation, a calibrated subthreshold-regime sensor response quantity (S) is derived from Equation (4) and expressed in Equation (5), in which $I_{DS(anti)}$ is the drain-source current measured in the “antibody functionalization” state of a sensor biased under a set of fixed V_{DS} and V_G ; and I_{DS} is the drain-source current measured from a subsequent biodetection state (*i.e.*, a specific TNF- α concentration). Similar to the calibrated linear-regime response quantity expressed in Equation (2), this subthreshold counterpart is also directly related to ΔV_T , independent of the transistor performance, and proportional to σ_{TNF} .

$$I_{DS} = I_T \times 10^{(V_G - V_T)/SS} \quad (4)$$

$$S = -SS \times \log\left(\frac{I_{DS}}{I_{DS(anti)}}\right) = \Delta V_T = \frac{q d_{SiO_2} \sigma_{TNF}}{k_{SiO_2} \epsilon_0} \quad (5)$$

Figure 3.6 (b) shows the calibrated subthreshold-regime responses (S) measured from five different sensors with respect to TNF- α concentration (n). The detailed transfer characteristics of these five devices measured at various biodetection stages are showed in Figure 3.7. As shown in Figure 3.7, the transfer characteristics of these five sensors show significant difference in V_T , I_{DS} and SS parameters. However, as shown in Figure

3.6 (b), the calibrated S - n curves measured from these devices are consistent with each other and can be well fitted with Langmuir isotherms (Equation (3)). Here, the equilibrium constant (K_D) of the antibody-(TNF- α) pair is extracted to be 424 ± 70 fM, which is consistent with the K_D value extracted from the linear-regime sensor responses (*i.e.*, 369 ± 48 fM). The S_{max} parameter is fitted to be 15.3 ± 0.6 V, which is about 40% larger than that extracted from the linear-regime responses (*i.e.*, 10.7 ± 0.4 V). This observable discrepancy has not been fully understood. However, this could be temporarily attributed to the different back-gate V_G levels required for biasing sensors in subthreshold and linear regimes, which could result in different magnitudes of electric field penetrating through few-layer MoS₂ channels as well as HfO₂ effective layers and leaking into the analyte solution. This could lead to different degrees of the modification of electrical-double-layers around sensors and therefore different degrees of the screening of the charges brought through analyte-receptor binding pairs.

Although the I_{DS} signals measured from both linear and subthreshold regimes can be mathematically normalized to consistent device-independent response quantities using Equations (2) and (5), the physical limit-of-detection of a transistor biosensor is indeed determined by the sensitivity of I_{DS} to the variation of analyte concentration (dn) as well as the noise level of electrically measured I_{DS} signals. This I_{DS} sensitivity is quantitatively defined as the relative change in I_{DS} per change in n (*i.e.*, Sensitivity = $\frac{dI_{DS}}{I_{DS}}/dn$). Figure 3.8 displays and compares the sensitivity data acquired from (a) the linear-regime I_{DS} signals measured from the five sensors shown in Figure 3.5 and (b) the subthreshold-regime I_{DS} signals from the five sensors shown in Figure 3.7. All differential sensitivity values are evaluated at TNF- α concentration of $n = 60$ fM. This can offer critical

information about the sensitivity required for obtaining fM-level detection limits. Figure 3.8 shows that the subthreshold-regime I_{DS} sensitivities ($0.52 \pm 0.3\%/fM$) are statistically

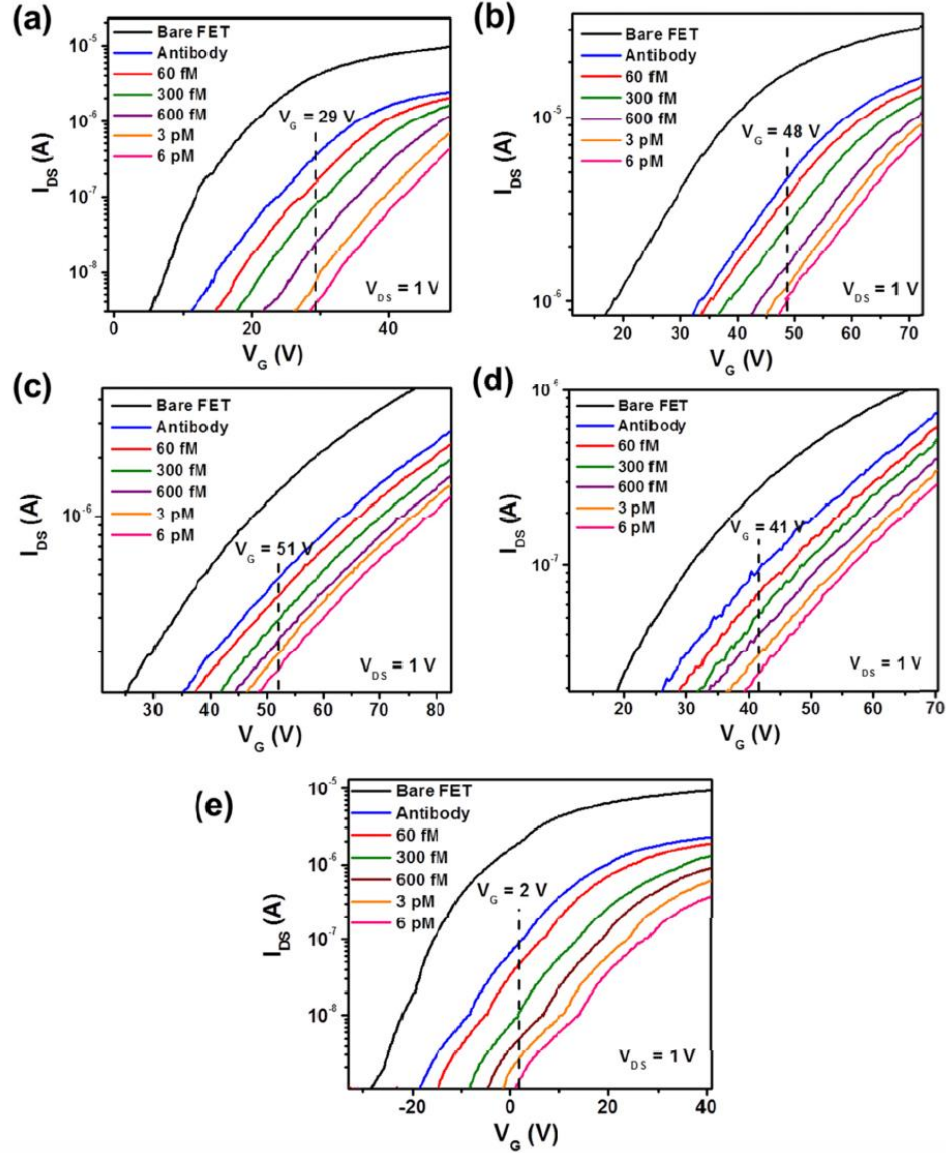


Figure 3.7 Subthreshold-regime sensor responses at the equilibrium state: The transfer characteristics of five different MoS₂ transistor sensors measured at various biodetection stages, following the sequence of (1) bare transistor, (2) antibody functionalization, and inputs of TNF- α solutions with concentrations of (3) 60 fM, (4) 300 fM, (5) 600 fM, (6) 3 pM, and (7) 6 pM.

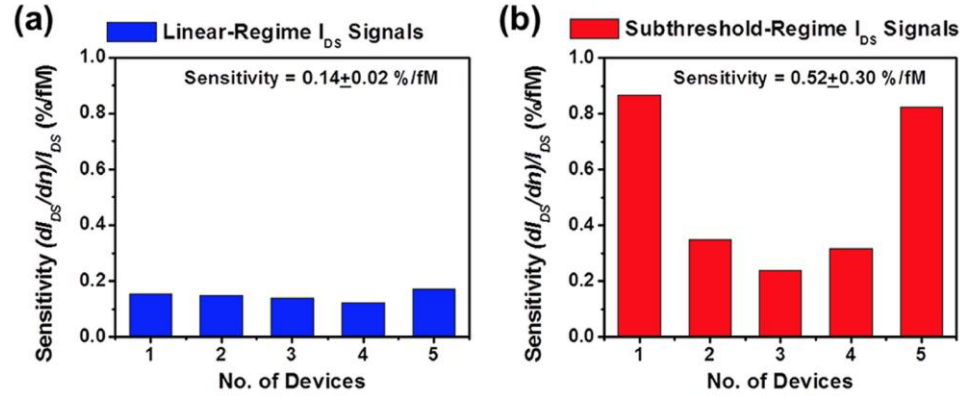


Figure 3.8 Sensitivity data acquired from (a) the linear-regime I_{DS} signals measured from the five sensors shown in Figure 3.5 and (b) the subthreshold-regime I_{DS} signals measured from the five sensors shown in Figure 3.7. All differential sensitivities were evaluated at TNF- α concentration of $n = 60$ fM

higher than the linear-regime I_{DS} sensitivities ($0.14 \pm 0.02\%/fM$). Therefore, subthreshold-regime sensor responses are more desirable in achieving high detection sensitivity. However, it should be noted that the ultimate detection limit of a transistor sensor is also limited by the signal-to-noise ratios of electrically measured I_{DS} signals. In addition, for detecting low-abundance molecules, the non-specific adsorption of target molecules could also strongly affect the detection limit. The further analysis of these aspects is beyond the scope of this work but will be addressed in the future research. Finally, it is also noted that the sensitivity data listed in Figure 3.8 (b) show the larger device-to-device variation in comparison with those listed in Figure 3.8 (a). This is probably because of that the subthreshold swing properties of MoS₂ transistors are more sensitive to the fabrication-introduced defects than their linear-regime transconductance properties. Therefore, for our current MoS₂ transistors, their linear-regime I_{DS} signals (or linear-regime transconductances) exhibit the higher device-to-device consistency than their subthreshold-regime I_{DS} signals.

To evaluate the detection specificity of our MoS₂ transistor sensors, a sensor functionalized with anti-human TNF- α antibody is used for detecting interleukin-6 (IL-6) cytokine. Figure 3.9 shows the transfer characteristics of this sensor measured at various stages, including (1) bare transistor, (2) antibody functionalization, and inputs of IL-6 solutions with concentrations of (3) 600 fM and (4) 6 pM. Figure 3.9 displays that the presence of IL-6 that is not specific to TNF- α antibody cannot result in prominent change in the transfer characteristics. Such experimentally observed weak sensor responses to IL-6 indicate a negligible nonspecific adsorption of IL-6 molecules on the sensor surface, which may be effectively screened by the densely-packed self-assembled monolayers of (3-Aminopropyl) triethoxysilane (APTES) on HfO₂ effective layers.

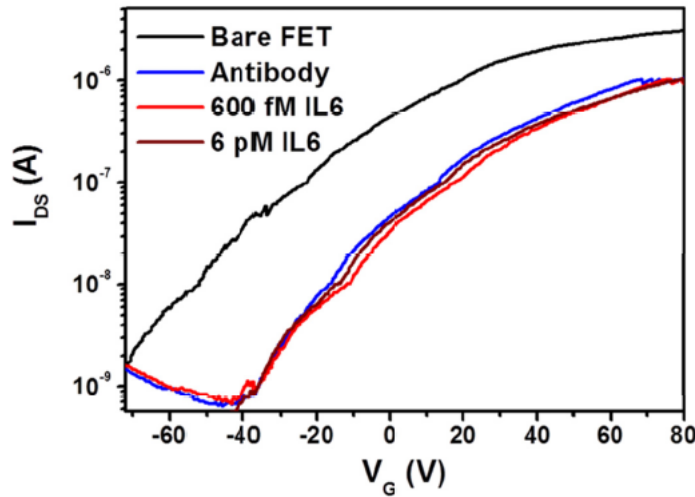


Figure 3.9 A negative control test of the detection specificity of MoS₂ transistor biosensors: The transfer characteristics of a control sensor measured at stages of (1) bare transistor, (2) antibody functionalization (still functionalized with anti-human TNF- α antibody receptors), and inputs of IL-6 solutions with concentrations of (3) 600 fM and (4) 6 pM.

3.3.2 Time-Dependent Measurement Results of Few-Layer MoS₂ Transistor Biosensors

The biosensor setup shown in Figure 3.3 (c) is used for measuring the time-dependent association/dissociation kinetics of the antibody-(TNF- α) pair. Figure 3.10 (a) shows real-time sensor responses of antibody-(TNF- α) binding measured under different TNF- α concentrations (*i.e.*, $n = 60$ fM, 600 fM, 3 pM, and 6 pM). Each of the time-dependent response curves was measured from a different MoS₂ transistor sensor and all as-measured I_{DS} responses were normalized using $S = -SS \times \log(I_{DS}/I_{DS(anti)})$ (*i.e.*, Equation (5) for calibrating subthreshold-regime responses). Figure 3.11 presents the detailed transfer characteristics of these transistor sensors measured before the input of TNF- α samples, from which the SS parameters required for normalizing I_{DS} responses were acquired. In addition, the operation points (*i.e.*, the fixed V_G and V_{DS} values, under which a real-time response curve was measured) are also labeled in Figure 3.11. In Figure 3.10 (a), the red arrow indicates the onset time, at which the solutions with specific TNF- α concentrations were filled into the respective biosensors. The real-time response curves in Figure 3.10 (a) exhibit that the association rate of the antibody-(TNF- α) pair increases with increasing TNF- α concentration. The rise segment of each real-time response curve can be well fitted with the first-order absorption equation (*i.e.*, Equation (6)) [104]. In Equation (6), S_{eq} is the sensor response at the final equilibrium state; k_{on} and k_{off} are association and dissociation rates, respectively; $k_{on}n + k_{off}$ relates to the rising slope of the linear regime of the response curve. Table 1 lists the fitting results of S_{eq} and $(k_{on}n + k_{off})$ parameters for $n = 60$ fM, 600 fM, 3 pM, and 6 pM. These S_{eq} values extracted from the real-time binding responses are consistent with the sensor responses directly measured at

the equilibrium state, that is, after a long incubation time of ~ 2 hours (e.g., the equilibrium-state response data shown in Figure 3.6 (b)). In particular, Figure 3.10 (b) plots the extracted S_{eq} data as a function of TNF- α concentration, which can be also fitted

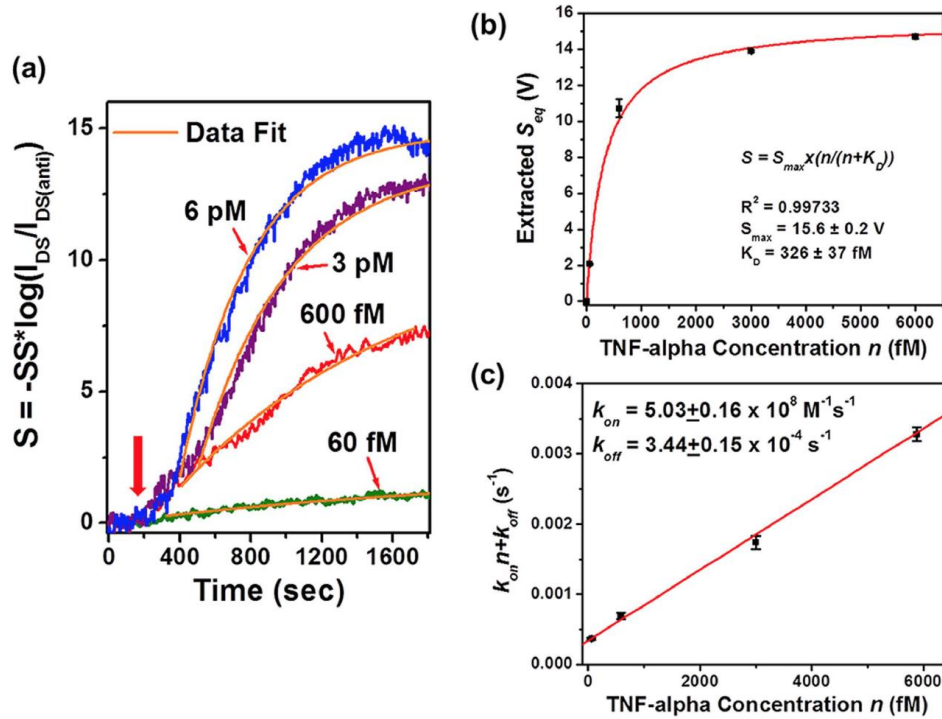


Figure 3.10 Time-dependent association kinetics of the antibody-(TNF- α) pair: (a) real-time sensor responses of antibody-(TNF- α) binding measured under different TNF- α concentrations ($n = 60$ fM, 600 fM, 3 pM, and 6 pM). Each of the response curves was measured from a different MoS₂ transistor sensor and all responses were normalized using Equation (5). The rise parts of the binding response curves can be fitted with Equation (6). (b) The equilibrium-state responses (S_{eq}) extracted from this fit plotted as a function of TNF- α concentration, which can be further fitted with Langmuir isotherm. The equilibrium constant (K_D) is extracted to be 326 ± 37 fM. (c) The extracted ($k_{on}n + k_{off}$) data plotted as a function of TNF- α concentration (n). The linear fitting of this ($k_{on}n + k_{off}$)-versus- n graph results in rate constants of $k_{on} = (5.03 \pm 0.16) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$ and $k_{off} = (3.44 \pm 0.15) \times 10^{-4} \text{ s}^{-1}$.

with Langmuir isotherm. Here, the equilibrium constant (K_D) is extracted to be 326 ± 37 fM and the maximum response (S_{max}) is extracted to be 15.6 ± 0.2 V, which are consistent

with those extracted from the equilibrium-state subthreshold-regime responses shown in Figure 3.6 (b).

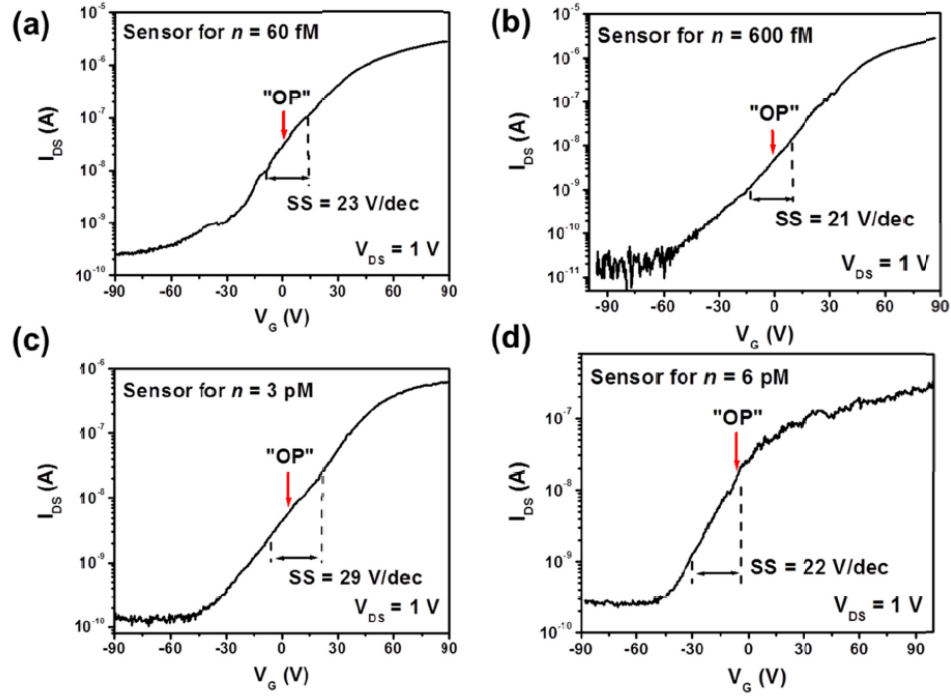


Figure 3.11 Transfer characteristics of four different MoS₂ transistor biosensors measured before the input of TNF- α samples, from which the subthreshold-swing (SS) parameters were acquired for normalizing the real-time subthreshold-regime sensor responses (Equation (5)). These sensors were utilized to quantify the real-time kinetics of antibody-(TNF- α) binding under different TNF- α concentrations (n) of (a) 60 fM, (b) 600 fM, (c) 3 pM, and (d) 6 pM. The operation points (OP, *i.e.*, the fixed V_G and V_{DS} values, under which a real-time response curve was measured) are also labeled by the red arrows.

$$S = S_{eq}(1 - e^{-(k_{on}n + k_{off})t}) \quad (6)$$

	n=60 fM	600 fM	3 pM	6 pM
S_{eq} (V)	2.07 ± 0.03	10.7 ± 0.5	13.9 ± 0.06	14.7 ± 0.13
$(k_{on}n \pm k_{off})$ (s^{-1})	$(3.68 \pm 0.15) \times 10^{-4}$	$(6.94 \pm 0.46) \times 10^{-4}$	$(1.74 \pm 0.10) \times 10^{-3}$	$(3.28 \pm 0.10) \times 10^{-3}$

Table 1 The fitting results of the real-time sensor response curves shown in Figure 3.8 (a) that are fitted with Equation (6). The table lists the extracted S_{eq} and $(k_{on}n + k_{off})$ parameters for $n = 60$ fM, 600 fM, 3 pM, and 6 pM.

To evaluate k_{on} and k_{off} parameters, the extracted $(k_{on}n + k_{off})$ data are plotted as a function of TNF- α concentration (n) (see Figure 3.10 (c)). The linear fitting results in rate constants of $k_{on} = (5.03 \pm 0.16) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$ and $k_{off} = (3.44 \pm 0.15) \times 10^{-4} \text{ s}^{-1}$. It should be noted that this fit is not sensitive to the dissociation rate (k_{off}) because of its small numerical value. To achieve a more precise quantification of k_{off} , we directly measured the real-time dissociation kinetics of the antibody-(TNF- α) pair. Specifically, two as-functionalized MoS₂ transistor biosensors were incubated in solutions with TNF- α concentration of 600 fM and 3 pM, respectively. The incubation time was more than 2 hours so that antibody-(TNF- α) association/dissociation processes reached to the equilibrium state. Afterwards, these fully incubated sensors were rinsed with pure buffer liquid flow and the calibrated sensor responses were recorded as a function of the lapsed time, as displayed in Figure 3.12. Figure 3.12 displays that the sensor responses decreased with time, which was attributed to the unbinding events. The response curve measured from the device incubated with TNF- α concentration of 600 fM can be well fitted with a monoexponential decay function (*i.e.*, the desorption equation expressed in Equation (7)). In Equation (7), S_r represents the sensor response corresponding to the areal density of bound molecule residues after the desorption process. This fit results in $k_{off} = (1.97 \pm 0.08) \times 10^{-4} \text{ s}^{-1}$, from which the affinity equilibrium constant K_D can be also

estimated to be $K_D = k_{off}/k_{on} = 392$ fM. This K_D value is also consistent with those extracted from the equilibrium-state sensor responses (*i.e.*, K_D values extracted in Figure 3.4 and 3.6). From this fit, S_r is extracted to be 3.0 ± 0.2 V and S_{eq} is 9.2 ± 0.4 V. This implies that $\sim 30\%$ of bound TNF- α molecules are expected to remain absorbed on the sensor even after a long rinsing process.

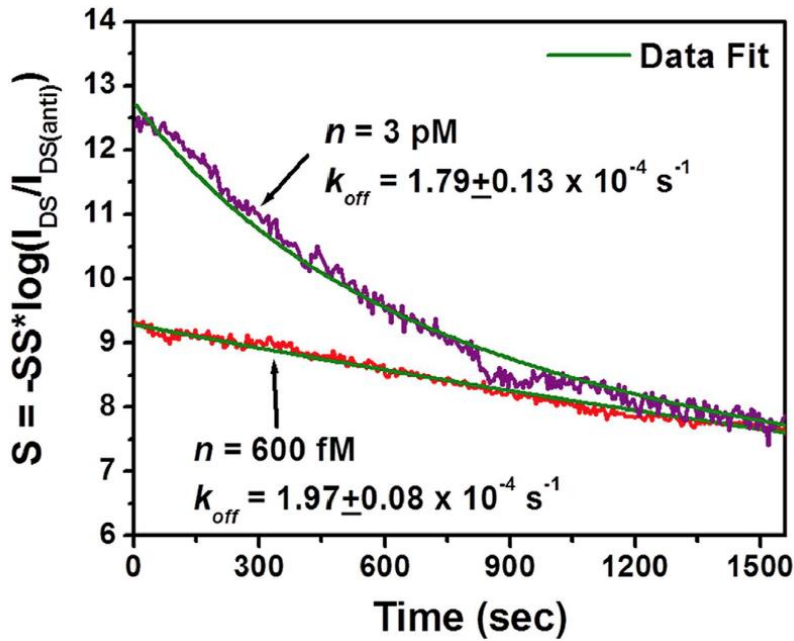


Figure 3.12 Time-dependent dissociation kinetics of the antibody-(TNF- α) pair measured from two MoS₂ transistor sensors that were incubated in solutions with TNF- α concentrations of $n = 600$ fM and 3 pM for about 2 hours and subsequently rinsed with the pure buffer liquid flow.

$$S = (S_{eq} - S_r)e^{-k_{off}t} + S_r \quad (7)$$

$$S = (S_{eq} - S_2 - S_r)e^{-k_{off}t} + S_2e^{-k_2t} + S_r \quad (8)$$

The response curve measured from the device incubated with TNF concentration of 3 pM can be hardly fitted with monoexponential Equation (7). We notice that it can be fitted with a bi-exponential decay equation (Equation (8)). This fit results in $S_{eq} = 13.6 \pm 1.0$ V, $S_2 = 4.5 \pm 0.2$ V, $S_r = 2.9 \pm 0.3$ V, $k_2 = (2.0 \pm 0.16) \times 10^{-3}$ s⁻¹, and $k_{off} = (1.79 \pm 0.13) \times 10^{-4}$ s⁻¹. As reported by several previous works [104, 136, 137], such a bi-exponential behavior of sensor responses is probably due to the multivalent antigen-antibody binding, which may become more prominent with increasing the analyte concentration. This explanation is reasonable because the antibody used in this work is polyclonal. To fully understand the association/dissociation kinetics of multivalent binding/unbinding processes, a more complicated model for describing antibody-(TNF- α) binding is required.

Finally, it should be noted that for our current MoS₂ transistor sensors, the calibrated sensor responses do not explicitly depend on HfO₂ layer thickness (t_{HfO_2}). All sensors discussed above have 30 nm thick HfO₂ effective layers. To further experimentally verify that the sensor responses of our sensors do not strongly depend on HfO₂ layer thickness (t_{HfO_2}), we fabricated additional sensors with $t_{HfO_2} = 60$ nm. Figure 3.13 (a) shows the transfer characteristics of an exemplary sensor with $t_{HfO_2} = 60$ nm, which were measured from a set of incremental TNF- α concentrations. From such transfer characteristics, we extracted calibrated subthreshold-regime responses (S) at $V_G = -25$ V ($V_T \sim -10$ V) and plotted them as a function of TNF- α concentration (n) (see the red stars shown in Figure 3.13 (b)). This S - n relationship is consistent with those

measured from the sensors with $t_{\text{HfO}_2} = 30$ nm. This result proves that the calibrated sensor response values do not strongly depend on the HfO₂ effective layer thickness.

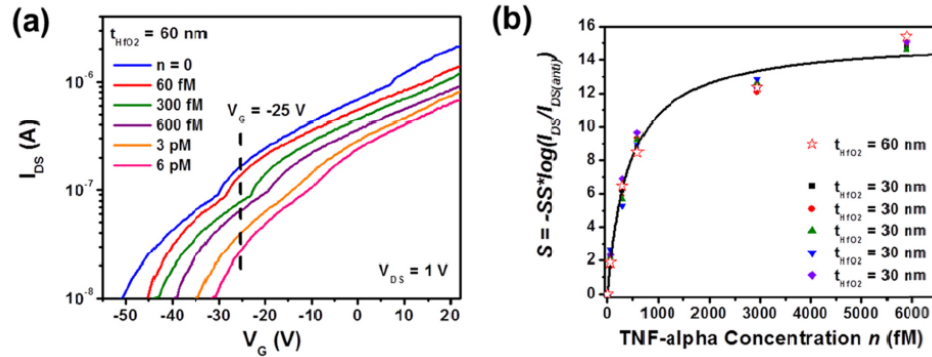


Figure 3.13 Sensor response measured in the subthreshold regime of a MoS₂ transistor biosensor with a 60 nm thick HfO₂ effective layer (*i.e.*, $t_{\text{HfO}_2} = 60$ nm): (a) transfer characteristics of the MoS₂ transistor sensor with $t_{\text{HfO}_2} = 60$ nm, which were measured from a set of incremental TNF- α concentrations (*i.e.*, $n = 0$, 60 fM, 300 fM, 600 fM, 3 pM, and 6 pM; (b) The calibrated subthreshold-regime responses (S) measured from this sensor (labeled as red stars) with respect to TNF- α concentration (n).

3.4 Summary

In conclusion, we presented significant device physics and metrics for calibrating the responses of MoS₂ transistor biosensors and demonstrated that multiple such sensors can be utilized to enable quantification of low-abundance biomarker molecules as well as the affinities and kinetics of antibody-mediated binding events. In particular, our biosensors exhibited a TNF- α detection limit at least as low as 60 fM. Such a low detection limit can be obtained in both linear and subthreshold regimes of MoS₂ transistors. We further observed that the sensors operated in the subthreshold regime showed the higher current sensitivities in comparison with those in the linear regime.

Such high subthreshold-regime sensitivities hold significant potential to further lower the TNF- α detection limit. In both transport regimes, the measured current signals can be normalized into response quantities independent of the transistor performance, which can effectively reduce the effect of sensor-to-sensor variation on biodetection results. Based on this calibration method, all sets of our biosensors can generate very consistent sensor responses with respect to TNF- α concentration and therefore a standard curve for TNF- α quantification. From this standard curve, the equilibrium constant of the antibody-(TNF- α) pair was extracted to be $K_D = 369 \pm 48$ fM from linear-regime responses (or $K_D = 424 \pm 70$ fM from subthreshold-regime responses). Furthermore, the real-time association/dissociation processes of the antibody-(TNF- α) pair were also quantified using multiple sensors. The association/dissociation rates were extracted to be $k_{on} = (5.03 \pm 0.16) \times 10^8 \text{ M}^{-1} \text{ s}^{-1}$ and $k_{off} = (1.97 \pm 0.08) \times 10^{-4} \text{ s}^{-1}$, respectively. This work laid an important foundation for leveraging the excellent electronic properties of emerging atomically layered semiconductors in bio-assay applications as well as advanced the critical research capability in analyzing the biomolecule interactions with fM-level detection sensitivities. Notably, such capability would enable selection of antibodies with a high binding constant with respect to a specific target biomarker molecule, thereby offering a means to further enhance the selectivity and fidelity of immunoassay.

Chapter 4

Two Different Device Physics Principles for Operating MoS₂ Transistor Biosensors with Femtomolar-Level Detection Limits

4.1 Introduction

MoS₂, one of transition metal dichalcogenides (TMDCs), has been studied as an attractive nanoelectronic material for making field-effect transistor (FET) biosensors [42, 43], because the transport characteristics of monolayer or few-layer MoS₂ FET channels are extremely sensitive to external stimulations, such as antigen-antibody binding events; 2D MoS₂ surfaces have an extremely low density of electron scattering centers, which can result in a low detection noise; and also semiconducting MoS₂ FETs show much higher On/Off ratios and therefore the higher detection sensitivities in comparison with semi-metallic graphene FETs [53, 138, 139]. Therefore, MoS₂ FET biosensors hold the significant potential to enable single-molecule-level (or fM-level) detection of illness-related biomarkers. To leverage such superior sensing capability of MoS₂ FETs for cost-efficient immunoassay applications, large-scale MoS₂ FET arrays need to be

manufactured at an affordable cost. To minimize the manufacturing cost, the sensor structure should be as simple as possible while meeting requirements unique to biosensing. In a conventional way of fabricating a typical FET biosensor based on MoS₂ or other layered materials [140, 141], a dielectric insulating layer (e.g., SiO₂, Al₂O₃, and HfO₂) is deposited on top of the MoS₂ channel for hosting the antibody receptors [42]. The growth of such insulating layers requires exquisite atomic layer deposition (ALD) tools. Moreover, due to the hydrophilic nature of these insulating layers, extra chemical processes need to be performed to introduce additional chemical linkers (e.g., 3-Aminopropyl)triethoxysilane (APTES)) for interfacing the dielectric surface and hydrophobic antibodies [142]. These two concerns could significantly increase the manufacturing cost of MoS₂-based immunoassay chips. Recently, Lee *et al.* demonstrated that the direct antibody functionalization on MoS₂ surface is possible because of the hydrophobicity of MoS₂ layers, which could lower the fabrication complexity and the cost of MoS₂ biosensors [142]. Notwithstanding such a progress, the underlying physics of the device operation still remains unexplored. Additional work is needed for understanding the difference between insulating-layer-coated and insulating-layer-free MoS₂ sensors in their operation principles.

In this chapter, we report different physics principles governing the sensor responses measured from insulating-layer-coated and insulating-layer-free MoS₂ FET sensors. For an insulating-layer-coated sensor, antibody receptors are functionalized on the insulating layer. The charge introduced through antigen-antibody binding is capacitively coupled with the MoS₂ channel and shifts the threshold voltage without significantly changing the transconductance (or mobility). On the other hand, for an

insulating-layer-free sensor, in which antibodies are directly grafted on the bare MoS₂ channel, antigen-antibody binding events mainly modulate the ON-state transconductance, which is attributed to the disordered potential formed in MoS₂ layers.

4.2 Methods and Materials

4.2.1 Fabrication and Characterization of MoS₂ Transistor Biosensors

Figure 4.1 (a) displays insulating-layer-coated (left) and insulating-layer-free (right) MoS₂ biosensors. To fabricate such devices, first few-layer MoS₂ flakes (5-7 nm thick) are exfoliated from a bulk MoS₂ ingot and printed onto the device substrates (*i.e.*, p⁺-Si substrates coated with 300 nm thermally grown SiO₂) using our previously reported method [125, 126]. After MoS₂ exfoliation, metallic drain (*D*) and source (*S*) contacts are formed using photolithography followed with metal deposition (10 nm Ti/50 nm Au) and lift-off. To produce an insulating-layer-coated sensor, 30 nm HfO₂ is deposited on the MoS₂ channel using ALD. HfO₂ and other high-k dielectric materials can result in less surface scattering as compared to SiO₂ and can also host the antibody functionalization [25, 94]. We also deposit HfO₂ on D/S electrodes to minimize the leakage currents from the analyte solutions to the electrodes. To create an insulating-layer-free sensor, the antibodies are directly functionalized on the bare MoS₂ channel.

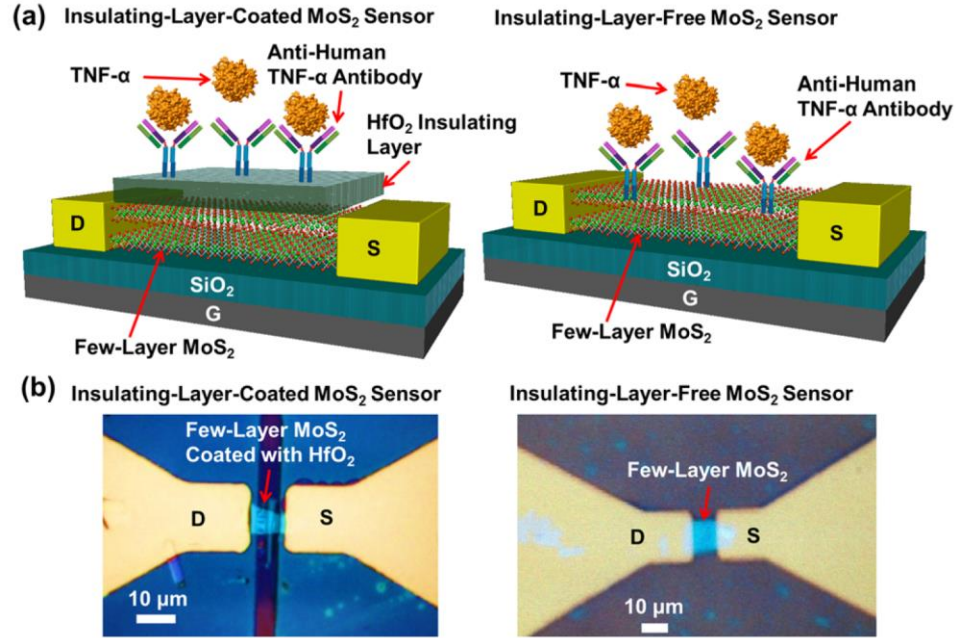


Figure 4.1 Illustrations (a) and optical micrographs (b) of insulating-layer-coated (left) and insulating-layer-free (right) MoS₂ FET biosensors.

4.2.2 Bio-Functionalization of MoS₂ Transistor Biosensors

Figure 4.1 (b) shows the optical micrographs of two exemplary biosensors with (left) and without (right) a 30 nm HfO₂ layer. In this work, we functionalized our sensors with Anti-Human TNF- α antibody receptors for detecting TNF- α molecules. The purified Anti-Human TNF- α (Clone: monoclonal antibody (MAb1) and Isotype: Immunoglobulin G (IgG1)) was purchased from Ebioscience, Inc. The MAb1 antibody has been used for capturing human TNF- α , immunoblotting, and neutralizing the biological activities of TNF- α . The antibody functionalization is described as follow. The sensors are submerged in 5% APTES in ethanol for 1 hour, and then they are washed with phosphate buffered saline (PBS) and blown dry. Afterwards, the sensors are submerged in 5% gluteraldehyde in PBS for 2 hours and rinsed with PBS [42]. Finally, the sensors are incubated with a

solution of 50 $\mu\text{g/ml}$ anti-TNF- α antibody for 45 min and subsequently blown dry by N_2 gas. For the TNF- α quantification at the equilibrium state of the (TNF- α)-antibody reaction, we perform the following steps: (1) incubate the sensor in a target TNF- α solution (solvent is 1 \times PBS; ionic strength is 162.7 mM) for a duration of 20-30 min to assure that the (TNF- α)-antibody reaction has reached to the equilibrium state; (2) briefly rinse away unreacted TNF- α molecules with DI water; and (3) blow dry the device and measure its transfer characteristics (*i.e.*, drain-source current (I_{DS})-gate voltage (V_G) curves) using a semiconductor analyzer. Such an equilibrium-state measurement is performed in the air, and the screening effect is negligible.

4.3 Results and Discussion

4.3.1 Comparison of Insulating-Layer-Coated and Insulating-Layer-Free MoS_2 Transistor Biosensors

Figure 4.2 (a) displays the transfer characteristics of an exemplary insulating-layer-coated sensor, which were measured from a set of incremental TNF- α concentrations (*i.e.*, $n = 0, 60 \text{ fM}, 300 \text{ fM}, 600 \text{ fM}, 3 \text{ pM}, \text{ and } 6 \text{ pM}$). With increasing n , the I_{DS} - V_G characteristic curve of this sensor shifted toward the positive V_G direction without changing the shape. Such a shape-retaining transition can be further verified by the fact that we can make all I_{DS} - V_G curves from different TNF- α concentrations coincide by using a horizontal shift of them along the V_G -axis, as shown in Figure 4.2 (b). This result indicates that the charges introduced by TNF- α molecules electrostatically affect

the free carrier concentration in the MoS₂ channel and induce a shift of the threshold voltage (*i.e.*, ΔV_T measured at the back gate), but these charges do not cause a noticeable change of the ON-state transconductance ($g_m = dI_{DS}/dV_G$) (or the field-effect mobility) of the FET. For such insulating-layer-coated sensors, the response behavior can be well described by the conventional capacitor model [105], and a calibrated sensor response quantity (S) is expressed in Equation (1) [104, 135], where $I_{DS(n=0)}$ is the I_{DS} measured from an as-functionalized sensor (*i.e.*, $n = 0$) biased under a set of fixed V_{DS} and V_G , and $I_{DS} - I_{DS(n=0)}$ indicates the change of I_{DS} induced by the introduction of TNF- α solutions with various concentrations. $I_{DS} - I_{DS(n=0)}$ normalized by the g_m of this sensor creates a sensor response quantity (S) associated with ΔV_T . Here, ΔV_T can be calculated using $\Delta V_T = qd_{SiO_2}\sigma_{TNF}/k_{SiO_2}\epsilon_0$, where q is the effective charge introduced by a TNF- α molecule; k_{SiO_2} and d_{SiO_2} are the dielectric constant and thickness of the SiO₂ back-gate dielectric, respectively; ϵ_0 is the vacuum permittivity; and σ_{TNF} is the areal density of TNF- α molecules bound to the sensor area.

Figure 4.2 (c) displays the transfer characteristics of an exemplary insulating-layer-free sensor, measured from $n = 0$, 60 fM, 600 fM, 6 pM, and 60 pM. This sensor exhibited a very different response behavior from insulating-layer-coated sensors. With increasing n , the $I_{DS} - V_G$ curve of this sensor did not significantly shift along the V_G -axis, but exhibited a proportional scale-down along the I_{DS} -axis (*i.e.*, the I_{DS} (or g_m) values proportionally decreased with increasing n). Such a proportional reduction of I_{DS} (or g_m) values can be further verified by the fact that we can make all $I_{DS} - V_G$ curves from different TNF- α concentrations coincide by multiplying their I_{DS} values with appropriate factors, as shown in Figure 4.2 (d). This result indicates that when TNF- α molecules are

bound to an insulating-layer-free sensor, they can result in reduction of the g_m (or the mobility) of the FET. Such a response behavior cannot be described by the simple capacitor model. For such insulating-layer-free sensors, a tentative calibrated response quantity is the relative change of the ON-state I_{DS} under a given V_G , as expressed by the Equation (2).

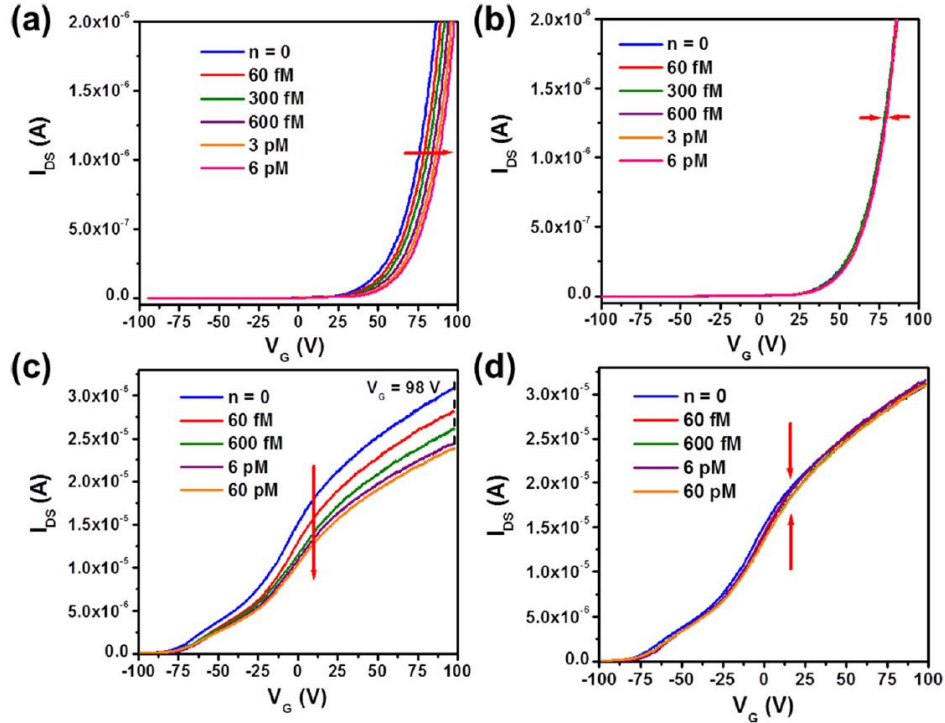


Figure 4.2 Comparison of sensor responses measured from MoS₂ FET biosensors with and without an insulating layer: (a) transfer characteristics of a sensor with a 30 nm HfO₂ insulating layer, measured at various TNF- α concentrations of $n = 0, 60, 300, 600, 3000,$ and 6000 fM. We can make these transfer characteristic curves coincide by using a horizontal shift along the V_G -axis, as shown in (b). (c) transfer characteristics of an insulating-layer-free sensor measured at $n = 60, 600, 6000,$ and $60\ 000$ fM. We can make these transfer characteristic curves coincide by multiplying them with different factors, as shown in (d).

$$S = -\frac{I_{DS} - I_{DS(n=0)}}{g_m} = \Delta V_T \quad (1)$$

As demonstrated in Figure 4.2, both insulating-layer-coated and insulating-layer-free sensors can result in a limit-of-detection (LOD) not higher than 60 fM for TNF- α detection. However, the employment of insulating-layer-free sensors can simplify the fabrication of future MoS₂ FET immunoassay chips. In addition, bare MoS₂ sensing areas are naturally hydrophobic and could enable the direct functionalization of a broad variety of antibodies without using complicated linkers, therefore simplifying the functionalization processes [142].

$$S = -\frac{I_{DS} - I_{DS(n=0)}}{I_{DS(n=0)}} \quad (2)$$

4.3.2 Simulation of Electrostatic Potential Distribution in Insulating-Layer-Coated and Insulating-Layer-Free MoS₂ Channels

Despite the simplicity of insulating-layer-free sensors, to ultimately realize biomarker quantification using such sensors, ones need to understand the device physics governing their response behavior. We tentatively attribute the antigen/ antibody-binding-induced reduction of the g_m values of our insulating-layer-free sensors (Figure 4.2 (c)) to the biomolecule-induced disordered potential in the MoS₂ channels, which can reduce the FET mobility [143-146]. To support this hypothesis, we quasi-quantitatively simulated the electrostatic potential distribution in the MoS₂ channels, which is induced by the charged antigen-antibody pairs. The simulation was performed using an electromagnetics simulation software (MaxwellTM) based on finite-element analysis. In this simulation, the antigen-antibody pairs were modeled as charged pillars randomly and discretely

distributed on the sensors (pillar diameter: 4 nm, height: 10 nm, and minimum spacing between pillars: 10 nm). Such a simple model of antigen-antibody pairs is approximately consistent with the geometric dimensions and spatial spacings of real antibody-(TNF- α) pairs bound to a solid sensor surface [147].

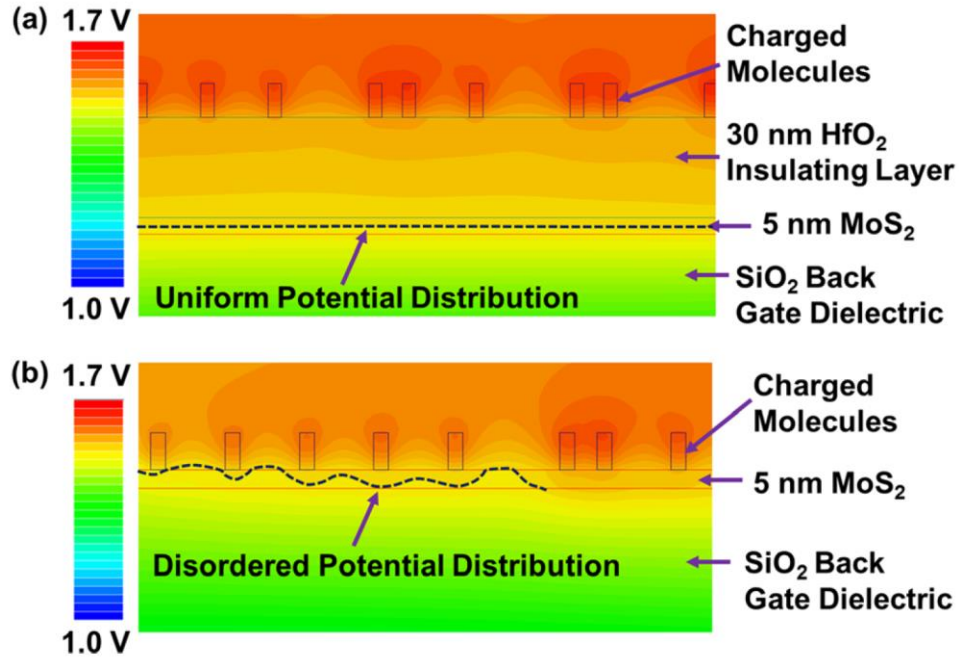


Figure 4.3 Simulation of the electrostatic potential distribution in the MoS₂ channel of (a) an insulating-layer-coated sensor, in which charged molecules are bonded to the HfO₂ insulating layer, and (b) an insulating-layer-free sensor, in which charge molecules are directly bonded to the MoS₂ surface.

Figure 4.3 displays the simulated potential distributions in the MoS₂ channels of (a) an insulating-layer-coated sensor and (b) an insulating-layer-free sensor. For both sensors, the charged molecules induce disordered and corrugated equipotential surfaces in the proximity of molecules. For the insulating-layer-coated sensor, the presence of a 30 nm HfO₂ insulating layer can effectively buffer such a disordered potential distribution

and result in relatively flat equipotential surfaces in the MoS₂ channel. Such a uniform field distribution in the MoS₂ channel does not significantly modulate the carrier mobility (or g_m) but only induces a ΔV_T . For the insulating-layer-free sensor, the molecules directly create a disordered/corrugated potential distribution in the MoS₂ channel, which is expected to significantly enhance the scattering and localization probabilities of moving carriers, therefore degrading the carrier mobility (and g_m) of the FET [143-146]. In 2D monolayer or quasi-2D few-layer MoS₂ FET channels, such a disordered potential effect is expected to be more prominent than that in bulk semiconductors because of the 2D confinement of carriers. This simulation result can qualitatively explain the response behavior of our insulating-layer-free sensors.

4.3.3 Static Measurement: Sensor Responses Measured Using a MoS₂ FET Sensor with a 5 nm Thick HfO₂ Insulating Layer

Additionally, our simulation implies a critical HfO₂ layer thickness of $t_c \sim 5$ nm. When the HfO₂ layer thickness (t_{HfO_2}) is thinner than t_c , the biomolecule-induced potential disorder is prominent in the MoS₂ channel and the FET is expected to function as a g_m -modulated sensor, as demonstrated in Figure 4.2 (c) and 4.2 (d). When t_{HfO_2} is thicker than t_c , the potential distribution in the MoS₂ channel is uniform and the FET is expected to function as a V_T -modulated sensor, as demonstrated in Figure 4.2 (a) and 4.2 (b). It is further speculated that when t_{HfO_2} is close to t_c , the sensor may exhibit a hybrid response behavior involving both g_m and V_T modulations. To test this speculation, we fabricated additional sensors with $t_{HfO_2} = 5$ nm $\sim t_c$. Figure 4.4 (a) displays the transfer

characteristics of one of such sensors. With increasing TNF- α concentration, the I_{DS} - V_G curve of this sensor indeed underwent a shift toward the positive V_G direction in concurrent with a drop of the g_m . This indicates that the sensors with $t_{HfO_2} \sim t_c$ show a hybrid g_m/V_T -modulated character. Such a response behavior can be further confirmed by

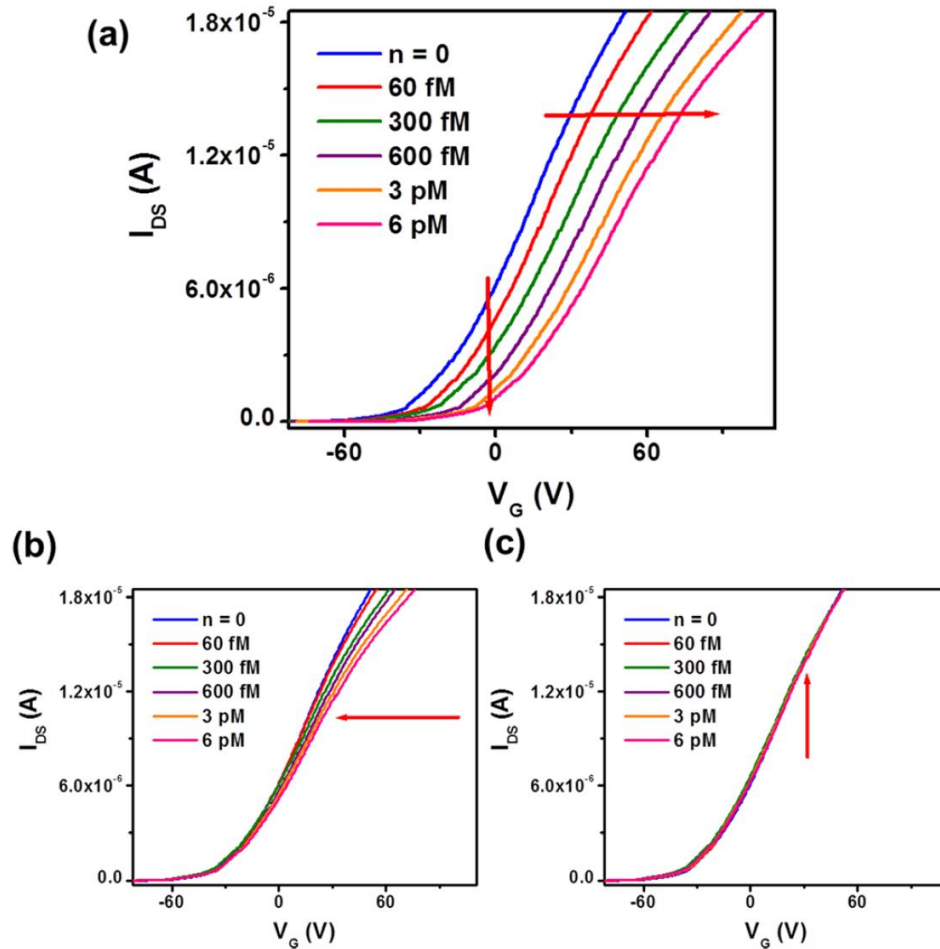


Figure 4.4 Sensor responses measured using a MoS₂ FET sensor with a 5 nm thick HfO₂ insulating layer: (a) transfer characteristics of this sensor measured at various TNF- α concentrations of $n = 0, 60, 300, 600, 3000,$ and 6000 fM. We can make these transfer characteristic curves coincide by horizontally shifting them along the V_G -axis and subsequently multiplying them with different factors, as shown in (b) and (c).

the fact that we can make all I_{DS} - V_G curves coincide by shifting them along the V_G -axis and subsequently multiplying them with different factors, as shown in Figure 4.2 (b) and 4.2 (c). This result strongly supports our simulation model and further implies that HfO_2 layer thickness strongly affects the potential disorder in the MoS_2 channels and determines the sensor response characters.

4.3.4 Repeatability of the Sensor Response Behaviors Using MoS_2 FET Sensors

To evaluate the repeatability of the sensor response behaviors, we identified, fabricated, and characterized additional insulating-layer-coated and insulating-layer-free sensors. Figures 4.5 and 4.6 display the transfer characteristics of all these devices. All insulating-layer-coated sensors exhibit a V_T -modulated behavior, and all insulating-layer-free ones exhibit a g_m -modulated behavior. Their sensor response behaviors are highly consistent with those of the representative devices, as shown in Figure 2. Figure 4.7 plots the response quantity (S)-TNF- α concentration (n) standard curves measured from all (a) insulating-layer-coated and (b) insulating-layer-free sensors. Here, the S values for insulating-layer-coated and insulating-layer-free sensors are calculated using Equation (1) and (2), respectively. The I_{DS} values used for the calculation are measured at $V_G = 98$ V. Figure 4.7 shows that our sensors exhibit a reasonably good device-to-device consistency in their S - n curves. Here, we do not use the Langmuir isotherm model to fit the S - n curves in Figure 4.7. This is because the antigen-antibody binding kinetics on our MoS_2 transistor biosensors may not follow the Langmuir isotherm model. In particular, for the insulating-layer-free sensors, we still lack a suitable physics model to correlate the sensor

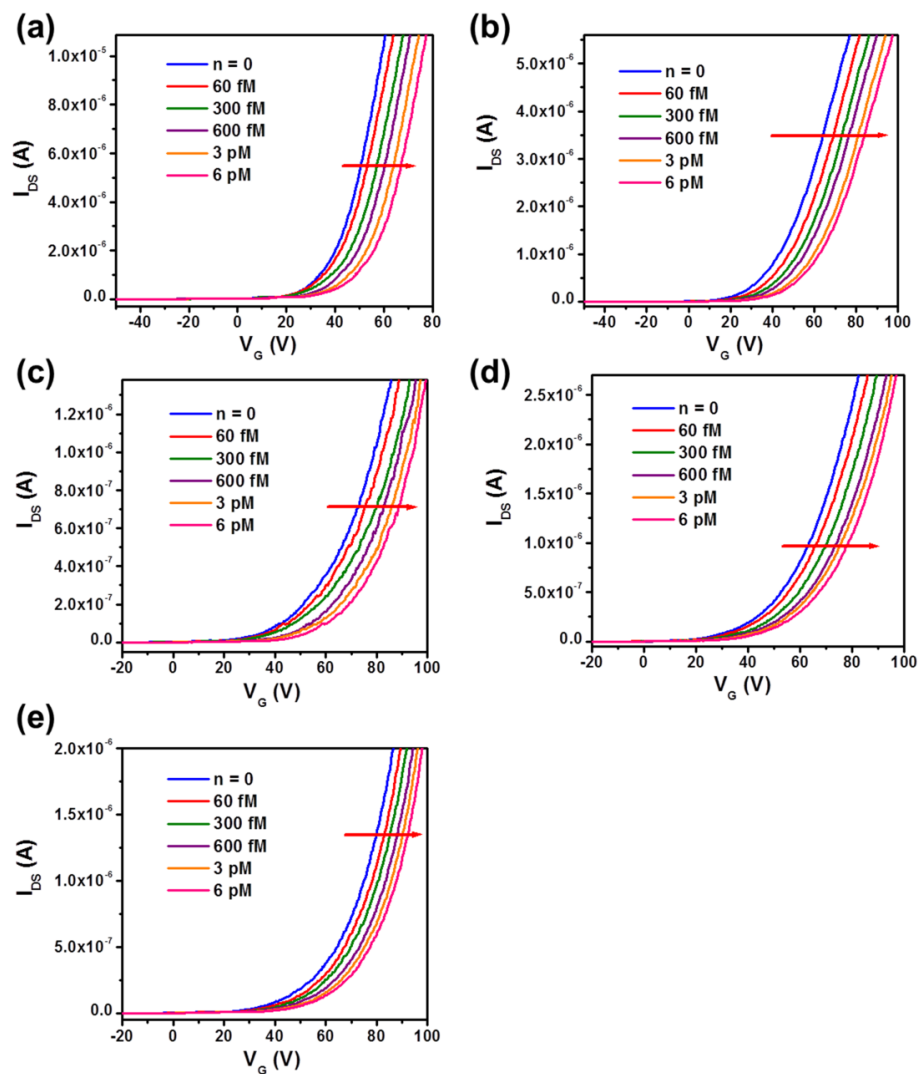


Figure 4.5 Transfer characteristics of five effective-layer-coated MoS_2 FET biosensors ($t_{\text{HfO}_2} = 30$ nm). Each of them is subjected to a set of incremental TNF- α concentrations (i.e., n = 0, 60 fM, 300 fM, 600 fM, 3 pM, and 6 pM).

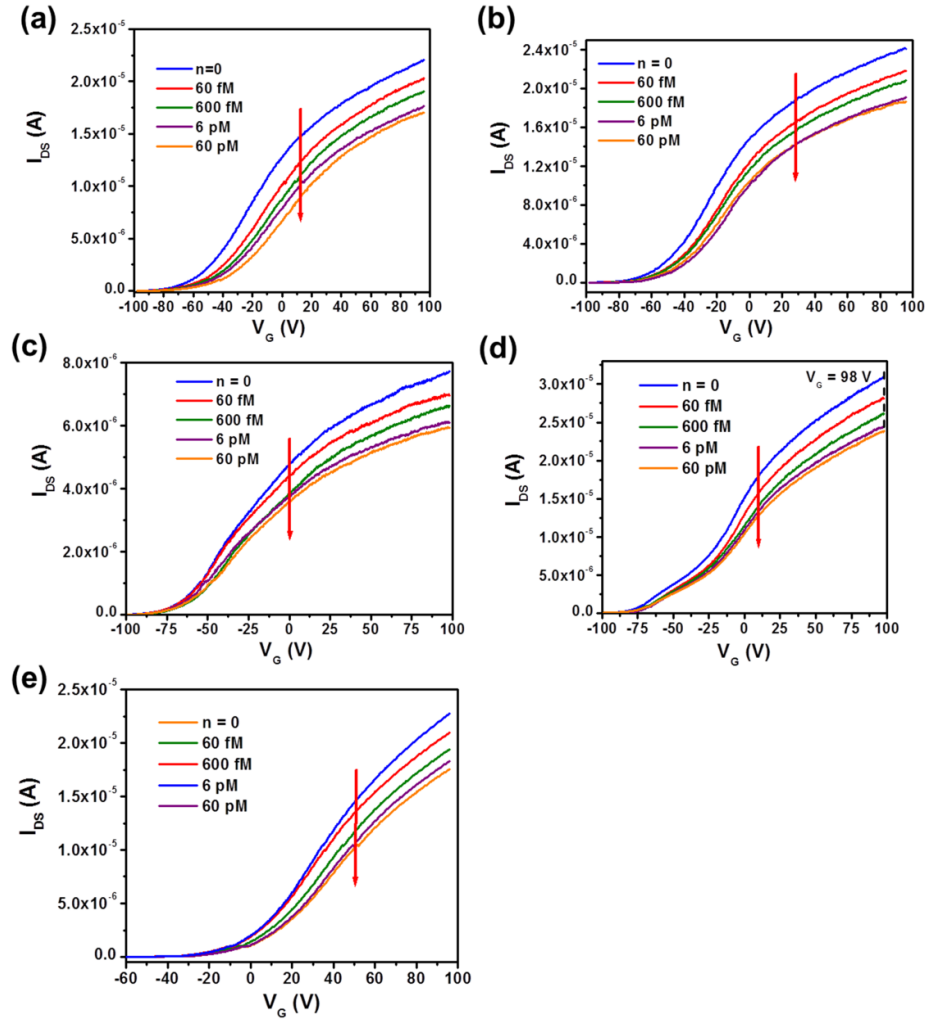


Figure 4.6 Transfer characteristics of five effective-layer-free MoS₂ FET biosensors ($t_{HfO_2} = 0$). Each of them is subjected to a set of incremental TNF- α concentrations (*i.e.*, $n = 0$, 60 fM, 600 fM, 6 pM, and 60 pM).

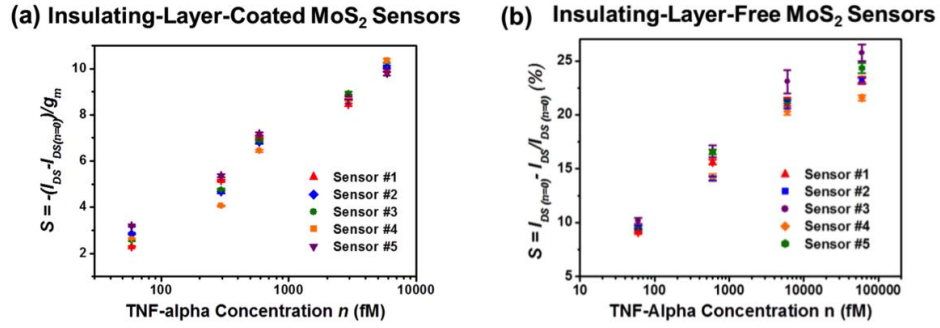


Figure 4.7 The standard curves for TNF- α quantification, acquired from (a) a set of insulating-layer-coated sensors (the ones listed in Figure 4.5), and (b) a set of insulating-layer-free sensors (the ones listed in Figure 4.6).

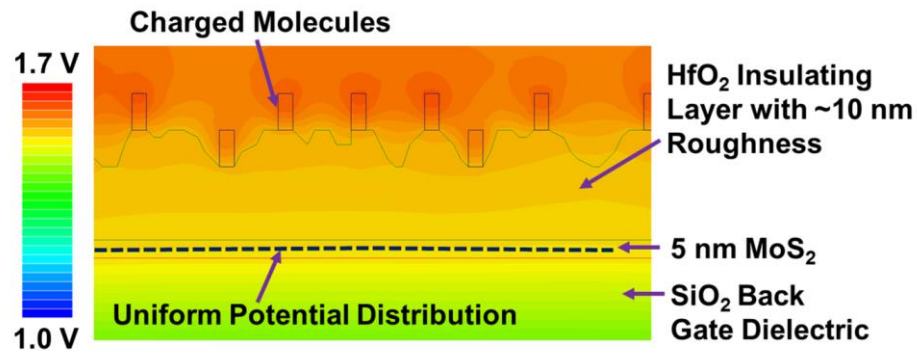


Figure 4.8 Simulation of the electrostatic potential distribution in the MoS₂ channel of an insulating-layer-coated sensor. In this device, the HfO₂ insulating layer has a surface roughness of ~10 nm (average peak-to-valley magnitude).

response quantity (S) with the areal occupancy of target biomolecules. For the insulating-layer-coated sensors, although we have a sensor response quantity (S) literally proportional to the areal density of bound molecules (see Equation (1)), previous works have shown that the ALD-grown HfO₂ films on 2D layer surfaces show a large roughness, which can make the antigen-antibody binding kinetics deviate from the Langmuir isotherm model [148-150]. Such a new justification about the effect of insulating layer

roughness on the antigen-antibody binding kinetics does not undermine our main conclusion, *i.e.*, the introduction of an HfO₂ insulating layer (even a rough layer) can effectively smooth out the antigen-induced disordered potential disturbances in the MoS₂ channel. Specifically, we further simulated the potential distribution in a sensor with HfO₂ roughness of ~10 nm. Figure 4.8 displays that the potential distribution in the MoS₂ channel is still uniform in spite of the HfO₂ roughness.

4.4 Summary

In summary, insulating-layer-coated and insulating-layer-free MoS₂ FET biosensors show V_T -modulated and g_m -modulated responses to antigen-antibody binding events, respectively. Our work proposes that such a difference in the sensor response character could be attributed to the effect of insulating layer thickness on the potential disorder in the MoS₂ FET channels. We also identify a critical insulating layer thickness, around which the sensors exhibit both V_T and g_m -modulated responses. This work advances the critical device physics for understanding and simplifying the FET biosensors based on emerging atomically layered semiconductors.

Chapter 5

Cycle-Wise Operation of MoS₂ and WSe₂ Transistor Biosensors for Rapid Biomolecule Quantification at Femtomolar Levels

5.1 Introduction

Field-effect transistor (FET) biosensors made from emerging layered transition metal dichalcogenides (TMDCs), such as MoS₂ and WSe₂, have exhibited attractive characteristics, such as high biodetection sensitivity, low limit-of-detection (LOD), and good compatibility with planar nanofabrication processes, potentially enabling multiplexing sensor arrays. Specifically, Wang *et al.* and Sarkar *et al.* experimentally demonstrated MoS₂ FET biosensors with femtomolar (fM)-level detection limits [42, 43]. Nam *et al.* established the device physics for calibrating the sensor response signals measured from MoS₂ FET sensors with insulating layers and preliminarily demonstrated the fabrication of MoS₂ FET arrays [49, 151]. Lee *et al.* demonstrated that antibody receptors can be directly grafted on MoS₂ FET channels without using insulating layers,

which is due to the hydrophobicity of the MoS₂ surface [142]. Such an insulating-layer-free sensor structure can lower the fabrication complexity as well as the cost of MoS₂ biosensors. Nam *et al.* further found that insulating-layer-free MoS₂ sensors exhibit a higher sensitivity than insulating-layer-coated ones [152]. This was attributed to that two mechanisms, target-molecule-induced doping and target-molecule-induced surface scattering, synergistically contribute to the response signal of an insulating-layer-free sensor, but only target-molecule-induced doping contributes to the response signal of an insulating-layer-coated sensor [152, 153]. Nam *et al.* also demonstrated that WSe₂ FET sensors exhibit a higher sensitivity than MoS₂ FET sensors, which is attributed to the ambipolar transfer characteristics of WSe₂ FETs [153].

In spite of the progress mentioned above, MoS₂ and WSe₂ FET biosensors still suffer seriously from a series of challenges that hinder their practical biosensing applications. Especially, a FET sensor needs to be continuously subjected to a liquid reagent environment for quantifying the time-dependent reaction kinetics of analyte-receptor binding. Such a time-dependent detection capability is a key to realize rapid biomolecule quantification based on sensor response readings at the non-equilibrium states of binding reactions, which could result in very short lapsed times and enable fast real-time immunoassay for quantifying extreme low-abundance biomolecules (*e.g.*, fM-level biomarkers) [154]. However, the electrical signal acquisition process is not generally compatible with such a liquid environment in the sensor. During a typical time-dependent detection process for quantifying the binding reaction kinetics, the sensing FET is electrically biased and constantly exposed to the analyte solution. Here, the applied biases include a gate voltage (V_G) that is used to bias the FET to the linear regime

of the transfer characteristic to obtain a high sensitivity as well as a consistent sensor response behavior [49, 153], and a drain-source voltage (V_{DS}) for driving the sensing channel current (I_{DS}). Such continuous electrical stresses in the liquid reagent environment can result in undesirable leakage currents flowing to electrodes and therefore generate high-level electronic noises and false sensor responses, and even electrochemically damage the sensor. It should be noted that even the FET sensor coated with an insulating layer can hardly survive in a long continuous time-dependent detection process (*e.g.*, >1 hour) because of the electric-field-enhanced permeation of reactive ions into the insulating layer [154, 155]. The analyte solutions with different ionic strengths also cause different degrees of the screening effect to the interaction between target molecules and the sensing channel, resulting in inconsistent sensor response signals. Such a screening effect also degrades the biodetection sensitivity of the sensor [156-159]. Another serious issue associated with TMDC-based FET sensors is the hysteretic behavior of the transfer characteristics of such sensors, which is attributed to the gate-modulated charges trapped at TMDC/dielectric interfaces [105]. Such a hysteretic behavior makes the I_{DS} value measured under a given V_G highly dependent on the V_G sweep range, sweep direction, sweep time and loading history, therefore resulting in inconsistent sensor readings. To rule out such indeterminacy of sensor readings due to the hysteresis effect, ones need to measure the complete transfer characteristic curves (*i.e.*, I_{DS} - V_G curves) by sweeping V_G along both positive and negative directions, which is similar to the way for characterizing a FET memory [160-162]. However, this electrical measurement process is not applicable during a continuous time-dependent detection process. Another expedient solution is to constantly bias the FET sensor to the linear (or

highly conductive) transport regime, where the indeterminacy of I_{DS} values is relatively smaller than that in the subthreshold regime. However, as discussed above, such a constantly applied gate bias can easily damage the sensor. In addition, during a continuous time-dependent detection process, the nonspecific adsorption of untargeted molecules on the FET sensing channel can also generate false signals. All of these issues seriously limit the applicability of TMDC-based FET biosensors for realizing rapid, highly specific biomolecule quantification using time-dependent sensor response signals or non-equilibrium-state sensor response readings.

In this chapter, we present a cycle-wise time-dependent biodetection method for operating MoS₂/WSe₂-based FET biosensors. Such a cycle-wise process alternately sets the FET sensor into incubation, flushing, and electrical measurement steps and therefore avoids the liquid-solution-induced electrochemical damage, screening, and nonspecific adsorption to the sensor. This method can also rule out the indeterminacy of sensor readings due to the hysteretic behaviors of TMDC FETs. This cycle-wise detection method in combination with the superior sensitivity and limit-of-detection (LOD) of MoS₂/WSe₂-based biosensors can enable rapid, low-noise, highly specific biomolecule quantification with fM-level LODs. In this work, we have experimentally achieved time-dependent responses associated with the reaction kinetics of streptavidin-biotin and IL1 β -IL1 β antibody binding and demonstrated fM-level streptavidin and IL1 β quantification in pure as well as complex solutions (*e.g.*, serum) with detection limit of ~ 1 fM and total incubation time less than 20 min.

5.2 Methods and Materials

5.2.1 Fabrication of MoS₂ and WSe₂ FET Biosensors

The TMDC FETs are produced onto p⁺-Si substrates with 300 nm thermally grown SiO₂ layers using a printing approach previously reported by us [151]. The SiO₂ layers are used as the back-gate dielectrics. Such 300 nm thick SiO₂ layers can also enable a simple color coding method for us to quickly identify and locate the MoS₂ and WSe₂ flakes with suitable thicknesses [163]. The thicknesses of printed Few-layer-TMDC channels are specifically controlled to be 5-15 nm. Such TMDC channel thicknesses can result in the optimal field-effect mobility and On/Off ratio values for TMDC FETs [20, 133, 134]. The p⁺-Si substrates served as the back gates (*G*). Ti (5 nm)/Au (50 nm) drain (*D*) and source (*S*) contacts to the TMDC channels are created using photolithography followed with metal deposition and lift-off. The channel lengths and widths of as-fabricated TMDC FETs are ~5 μm and ~15 μm, respectively. All electrical measurements are performed using an HP-4145B semiconductor parameter analyzer, which is connected to a LakeshoreTM probe station. Each of the TMDC FETs selected for making a biosensor is integrated with a polydimethylsiloxane (PDMS) block bearing a microfluidic channel (10 mm in length, 200 μm in width, 50 μm in height), as illustrated in Figure 5.1 (a). A motorized syringe pump is used for driving the reagent fluid flows into and out of the microfluidic channel through an inlet/outlet tubing kit (tube diameter: 0.75 mm).

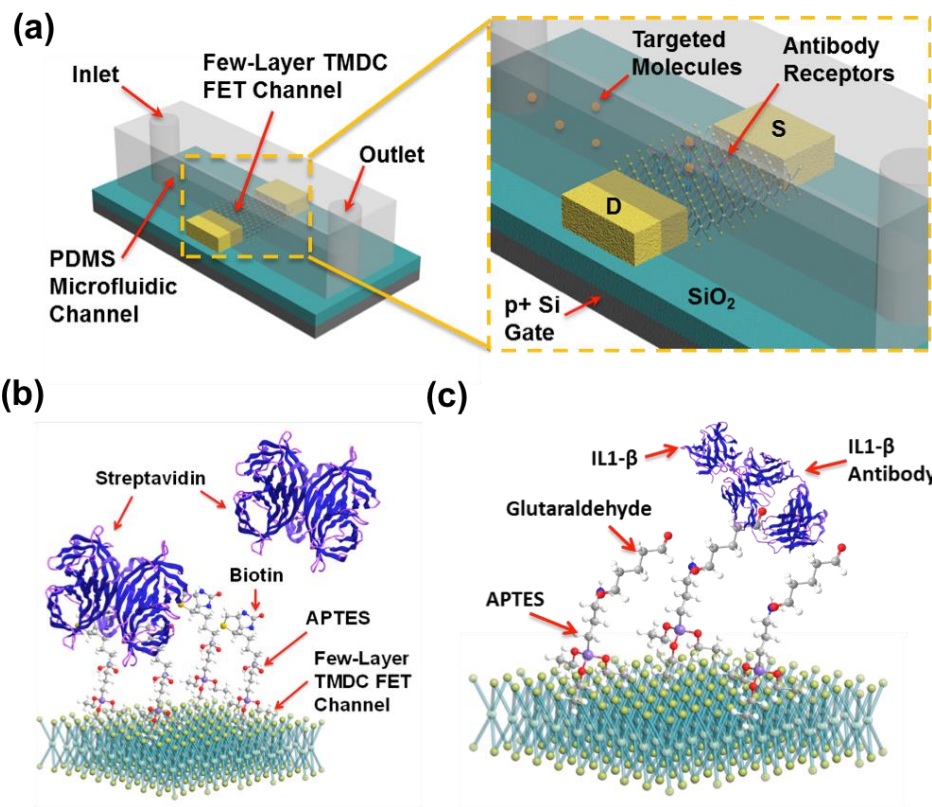


Figure 5.1 Experimental method: (a) illustration of a TMDC-based FET sensor, which is integrated with a PDMS microfluidic channel for enabling the cycle-wise time-dependent detection capability; illustration of a few-layer TMDC FET channel that is functionalized with (b) biotin receptors; (c) IL1 β antibody

5.2.2 Bio-Functionalization of TMDC FET Biosensors

An as-fabricated MoS₂ or WSe₂ FET biosensor is incubated in 5% (3-Aminopropyl) triethoxysilane (APTES) in ethanol for 1 hour. After this incubation process, the sensor is rinsed with phosphate buffered saline (PBS) and blown dry by nitrogen gas. For the IL1 β antibody functionalization, the sensor is subsequently immersed in 5% glutaraldehyde in PBS for 2 hours followed by rinsing with PBS. Then 0.1 mg/ml NHS-biotin or IL1 β antibody in PBS is subsequently incubated on the MoS₂ or WSe₂ channel area for 30 min followed by rinsing with PBS and N₂ blow-dry. During a

biodetection process, the integrated PDMS microfluidic system on the FET sensor is used to deliver a series of timely sequenced reagent fluids, including analyte streptavidin or IL1 β solutions, pure buffers, and air flows, to the active sensing FET channel.

5.3 Results and Discussion

5.3.1 Cycle-Wise Time-Dependent Biodetection Method for Operating TMDC-Based FET Biosensors

Figure 5.1 (a) illustrates the TMDC-based back-gated FET biosensor under study, which is integrated with a polydimethylsiloxane (PDMS)-based microfluidic channel. A pair of inlet and outlet tubes is connected with the microfluidic channel for delivering various reagent fluids to the FET sensor. Before a biosensing process, the sensing FET channel made from few-layer TMDC (MoS₂ or WSe₂) is functionalized with specific antibody receptors for detecting the target molecules in the analyte solution. During a cycle-wise biodetection and quantification process, the microfluidic structure is operated to deliver a series of timely sequenced reagent fluids to the sensing FET channel and periodically set the sensor into four biodetection stages, including (i) incubation of the sensing channel in the analyte solution containing target molecules for a short but deterministic duration (typically 5-10 min), (ii) flushing the sensing channel using a pure buffer to eliminate unreacted target molecules and untargeted background molecules, (iii) drying the sensing channel using an air flow, and (iv) under such dry condition, electrically measuring the transfer characteristics (*i.e.*, I_{DS} - V_G curves) of the sensing FET

by sweeping V_G along both positive and negative directions. Repetitive performance of multiple such incubation-flushing-drying-measurement (IFDM) cycles can acquire a time-dependent response signal associated with the reaction kinetics of analyte-receptor binding.

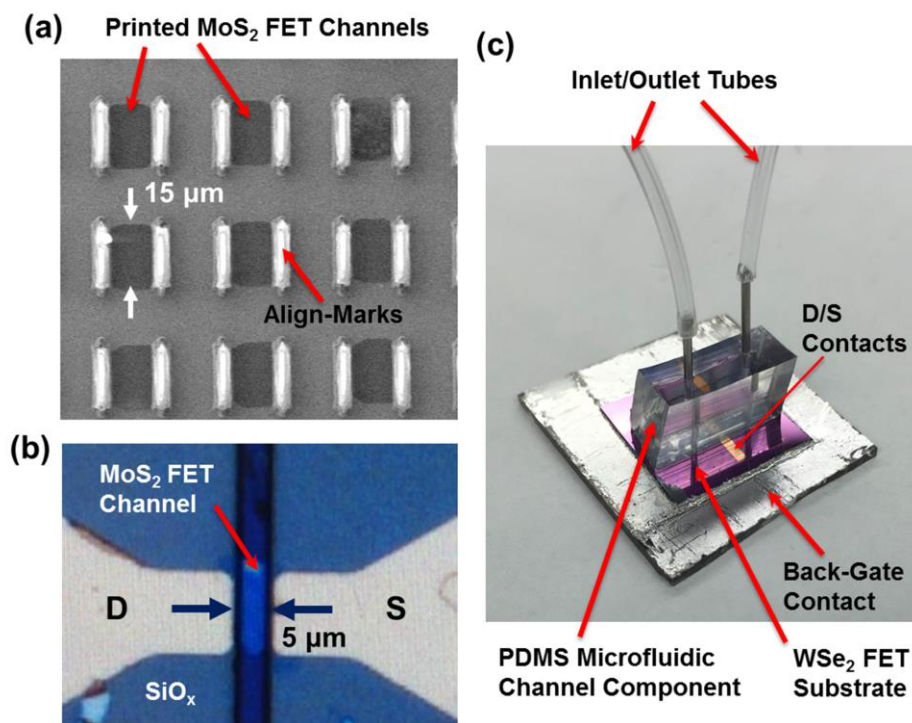


Figure 5.2 Biosensor fabrication and setup: (a) SEM image of several printed few-layer-MoS₂ FET channels; (b) optical micrograph of a representative few-layer-MoS₂ FET biosensor; (c) photograph of a WSe₂ FET sensor integrated with a micro-fluidic channel system. In this work, the few-layer-TMDC (MoS₂ or WSe₂) FET channels were functionalized with biotin receptors, and we experimentally demonstrated time-dependent quantification of streptavidin-biotin binding kinetics.

Figure 5.2 (a) shows the scanning electron micrograph (SEM) of several as-printed few-layer-MoS₂ FET channels. Each of the printed channels is associated with a pair of Ti/Au align-marks, which are used for locating the FET channel specifically selected for making a working biosensor. Figure 5.2 (b) displays the optical micrograph

of a representative TMDC FET biosensor consisting of a few-layer MoS₂ channel (thickness ~ 5 nm), a 300 nm thick SiO₂ back gate dielectric layer, a p⁺-Si back gate, and a pair of metallic drain/source (*D/S*) contacts (10 nm Ti/50 nm Au). The FET channel length and width are 5 and 15 μm, respectively. Figure 5.2 (c) shows the photograph of a WSe₂ FET sensor integrated with a PDMS microfluidic channel. This microfluidic channel is connected with a motorized syringe pump (not shown in Figure 5.2 (c)) through a pair of inlet/outlet tubes. As demonstrated in Figure 5.2 (c), we can operate this pump/tubing system to deliver timely sequenced reagent fluids to the FET sensing channel for performing IFDM detection cycles.

5.3.2 Time-Dependent Sensor Responses with 30 fM Pure Streptavidin (MoS₂ FET Biosensor)

Figure 5.3 shows the time-dependent sensor responses to a 30 fM solution of pure streptavidin, which were measured using a representative n-type MoS₂ sensor (Figure 5.3 (a) and (b)) and a representative ambipolar WSe₂ sensor (Figure 5.3 (c)-(f)). Both sensors were operated using the cycle-wise detection method discussed above, and for each IFDM cycle, the incubation time was $T_i = 10$ min. In the following discussion, the sensor responses are analyzed as a function of accumulative incubation time (t), which is defined as $t = nT_i$ (here, n is the number of already performed IFDM cycles). Specifically, Figure 5.3 (a) shows the transfer characteristic (*i.e.*, back-gate voltage (V_G) - drain-source current (I_{DS})) curves of the MoS₂ sensor measured at different t values ranging from 0 to 110 min. For each t , two transfer characteristic curves were measured. One was measured

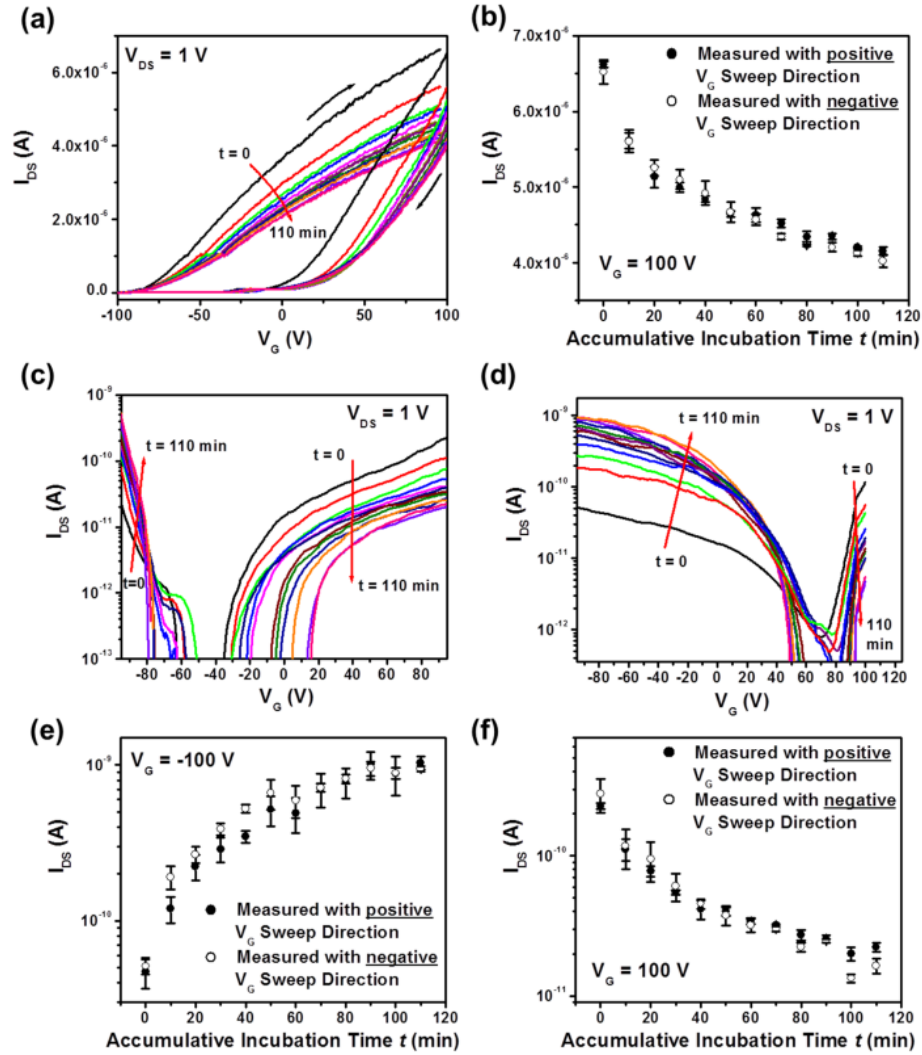


Figure 5.3 Time-dependent sensor responses to a 30 fM solution of pure streptavidin measured using a pair of representative MoS₂ and WSe₂ FET biosensors: (a) $I_{DS}-V_G$ curves of the MoS₂ sensor measured at different accumulative incubation times t ranging from 0 to 110 min (these curves were measured with both positive and negative V_G sweep directions); (b) $I_{DS}-t$ response signals captured from the t -dependent $I_{DS}-V_G$ curves in (a) at $V_G = 100$ V; (c) and (d) t -dependent $I_{DS}-V_G$ curves of the WSe₂ sensor measured with positive and negative V_G sweep directions, respectively; (e) and (f) $I_{DS}-t$ response signals captured from the p-type (at $V_G = -100$ V) and n-type ($V_G = 100$ V) branches of the t -dependent $I_{DS}-V_G$ curves of the WSe₂ FET, respectively.

with positive V_G sweep direction (*i.e.*, from -100 to 100 V, which is referred to as “positively-scanned curves” below), and the other was measured with negative V_G sweep

direction (*i.e.*, from 100 to -100 V, which is referred to as “negatively-scanned curves” below). Although this MoS₂ sensor exhibits a prominent hysteretic behavior in its transfer characteristics, both sets of the positively and negatively-scanned I_{DS} - V_G curves measured from this sensor show similar I_{DS} -reduced response behaviors with increasing accumulative incubation time t . To quantify the time-dependent reaction kinetics of biotin-streptavidin binding, the I_{DS} values measured at a given V_G are captured and plotted as a function of accumulative incubation time t . To minimize the indeterminacy of such $I_{DS} - t$ response signals due to the hysteresis of I_{DS} - V_G curves, $I_{DS} - t$ signals should be obtained in the highly conductive regime of I_{DS} - V_G characteristic curves (*e.g.*, $V_G = 100$ V for our n-type MoS₂ FETs). Figure 5.3 (b) plots the $I_{DS} - t$ response signals captured from the t -dependent I_{DS} - V_G curves in Figure 5.3 (a) (at $V_G = 100$ V). Figure 5.3 (b) shows that the $I_{DS} - t$ signals obtained from positively and negatively-scanned I_{DS} - V_G curves are consistent with each other, and the hysteresis-induced indeterminacy of I_{DS} - t signals can be ruled out.

Figures 5.3 (c) and (d) display the positively and negatively-scanned t -dependent I_{DS} - V_G curves of the WSe₂ sensor, respectively. For both sets of the I_{DS} - V_G curves, the t values range from 0 to 110 min. The WSe₂ FET exhibits an ambipolar transport property. In the p-type branch of I_{DS} - V_G curves, the WSe₂ FET exhibits an I_{DS} -enhanced response behavior with increasing t , whereas in the n-type branch, it exhibits an I_{DS} -reduced response behavior with increasing t . Reliable I_{DS} - t signals with the minimal indeterminacy due to the hysteretic behavior of I_{DS} - V_G curves can be obtained from both highly conductive p-type and n-type regimes of I_{DS} - V_G curves (*e.g.*, $V_G = -100$ V and 100 V for this WSe₂ FET). Figures 5.3 (e) and (f) plots the I_{DS} - t response signals captured at

$V_G = -100$ V (*i.e.*, p-type I_{DS} - t responses) and 100 V (*i.e.*, n-type I_{DS} - t responses), respectively. Figures 5.3 (e) and (f) also shows that for either p-type or n-type sensor response cases, the I_{DS} - t signals obtained from positively and negatively-scanned I_{DS} - V_G curves are highly consistent with each other. This work demonstrates that the IFDM cycle-wise method for operating MoS₂ and WSe₂ FET sensors can generate reliable time-dependent sensor response signals for quantifying the kinetics of binding reactions and rule out the indeterminacy of sensor readings due to the hysteresis of the transfer characteristics of such FETs.

5.3.3 Comparison of the Time-Dependent Response Signals Obtained Using IFDM Cycle-Wise and Continuous Detection Methods

We characterized two MoS₂ FET sensors fabricated in the same fabrication batch for a direct comparison of the time-dependent response signals measured using the cycle-wise method and the regular continuous detection method. Figure 5.4 shows these two sensors' responses to the pure 30 fM streptavidin solution. Figure 5.4 (a) shows the t -dependent I_{DS} - V_G characteristic curves measured using one of the MoS₂ sensors, which was operated using the cycle-wise method ($T_i = 10$ min). An I_{DS} - t signal can be captured from the highly conductive n-type transport regime (*i.e.*, $V_G = 100$ V) of the I_{DS} - V_G curves plotted in Fig. 5.4 (a). To further minimize the effect of the device-to-device variation in absolute I_{DS} values on the comparison, we used the relative change in I_{DS} (*i.e.*, $S = 100\% \times [I_{DS}(t) - I_{DS}(t = 0)] / I_{DS}(t = 0)$) with respect to the accumulative incubation time (t) as a calibrated time-dependent sensor response signal. Here, S can also serve as the

measure of the biodetection sensitivity. Figure 5.4 (b) plots the S - t response signal measured from the MoS₂ sensor that was operated using the cycle-wise method (solid circles). The other MoS₂ FET sensor was used to continuously measure the electrical signal associated with the time-dependent reaction kinetics of biotin-streptavidin binding. In particular, the sensor was continuously biased with a back-gate voltage (V_G) of 100 V and a drain-source voltage (V_{DS}) of 1 V. Since the 30 fM streptavidin solution was introduced into the sensor, the instant I_{DS} signal had been being continuously recorded as a function of t . This I_{DS} - t signal was also calibrated into the signal of t -dependent relative change in I_{DS} (*i.e.*, the S - t signal), which was also plotted in Figure 5.4 (b) (the solid line). In comparison with this S - t signal obtained using the continuous detection method, the S - t signal obtained using the IFDM cycle-wise method exhibits the larger signal magnitude (*i.e.*, the higher sensitivity) and the smaller relative error of the signal magnitude (*i.e.*, the higher signal-to-noise ratio) at a given accumulative incubation time. The relatively weaker magnitude of the S - t signal obtained using the continuous method is attributed to the Debye screening effect of the liquid solution, which can weaken the electrostatic coupling between the charges of target molecules and the FET channel, resulting in the lower sensitivity [105, 157-159]. The IFDM cycle-wise method can address this issue through physically isolating the incubation and measurement stages during an IFDM cycle. This cycle-wise method can also eliminate the electrical noise from the liquid solution and therefore result in the lower LOD. Additionally, as shown in Figure 5.4 (b), the MoS₂ sensor operated using the continuous method failed at $t \sim 50$ min. Our other TMDC FET sensors, when operated using the continuous method, also usually fail within one hour of the operation. Such a poor durability is attributed to the net effect of the

electrical stresses continuously applied to the FET channel and the continuous exposure of the sensor to the liquid environment, which can lead to electrochemical or thermal damage to the sensing channel. However, the similar TMDC sensors, if operated using the IFDM cycle-wise method (T_i : 5-10 min), can properly work for at least five hours without damage. Therefore, this comparison work indicates that the TMDC FET sensors operated using the cycle-wise method exhibit the better durability, the higher sensitivity, and the higher signal-to-noise ratio in comparison with those operated using the continuous detection method.

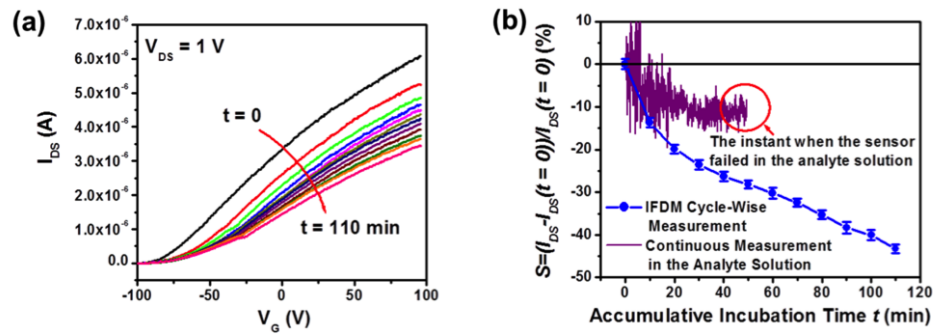


Figure 5.4 Comparison between a MoS₂ FET biosensor operated using the IFDM cycle-wise method and another similar sensor operated using the regular continuous detection method in their responses to a 30 fM solution of pure streptavidin: (a) t -dependent I_{DS} - V_G characteristic curves of the sensor operated using the cycle-wise method; (b) the S - t response signals obtained using the IFDM cycle-wise method (solid circles) and the regular continuous detection method (the solid line).

5.3.4 Detection Specificity of the TMDC FET Biosensors Operated by the IFDM Cycle-Wise Method

To evaluate the detection specificity of the TMDC FET biosensors operated by the IFDM cycle-wise method, we further performed two experiments to test time-

dependent quantification of biotin-streptavidin binding reaction kinetics in complex solutions. In the first experiment, we characterized four MoS₂ FET biosensors, which were fabricated in the same batch and have very similar transfer characteristics. Two of them (sensors #1 and #2) were used to quantify 30 fM streptavidin in a pure solution. The other two sensors (sensors #3 and #4) were used to quantify 30 fM streptavidin in a solution that also contained 60 fM tumor necrosis factor alpha (TNF- α). The processing parameters of the IFDM cycle-wise method are the same as those used for the experiments discussed above. Figure 5.5 shows the t -dependent I_{DS} - V_G characteristic curves of these four MoS₂ FET sensors, which were measured using the IFDM cycle-wise method ($T_i = 10$ min). Figure 5.6 (a) displays four sets of S - t signals respectively captured from the t -dependent I_{DS} - V_G curves of these four MoS₂ sensors (all I_{DS} and S data were captured at $V_G = 100$ V). These four sets of S - t signals are consistent with each other. This indicates that the presence of 60 fM TNF- α in the analyte solution does not significantly affect the t -dependent sensor response to 30 fM streptavidin. The slight difference among these four S - t curves is attributed to the slight sensor-to-sensor variation in transport characteristics.

In the second experiment, two WSe₂ FET biosensors were fabricated in the same batch. One sensor (sensor #1) was used to quantify 30 fM streptavidin in a pure solution, and the other sensor (sensor #2) was used to quantify 30 fM streptavidin in a serum solution (Merck KGaA, Serum matrix (Contains 0.08% Sodium Azide), 20 times dilution in DI water). Figure 5.7 displays the t -dependent I_{DS} - V_G characteristic curves of these two WSe₂ FET sensors, which were measured using the IFDM cycle-wise method ($T_i = 10$

min). Figures 5.6 (b) and (c) displays the S - t response signals captured from the p-type ($V_G = -100$ V) and n-type ($V_G = 100$ V) regimes of the t -dependent I_{DS} - V_G curves of these

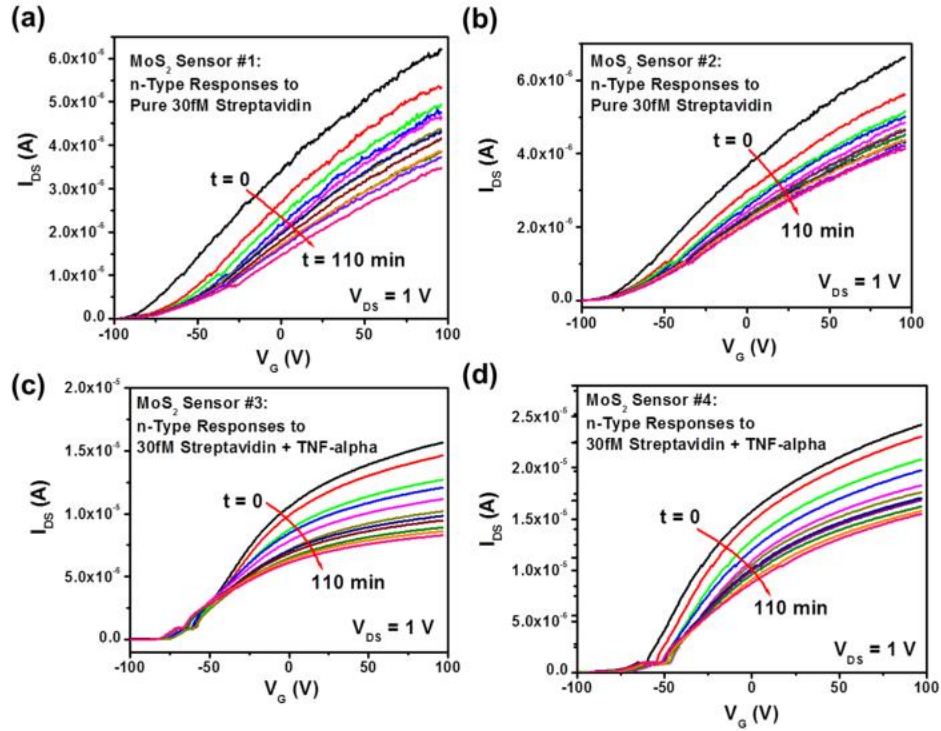


Figure 5.5 IFDM cycle-wise responses ($T_i = 10$ min) of four MoS₂ FET biosensors that were fabricated in the same fabrication batch: t -dependent I_{DS} - V_G characteristic curves of Sensors #1 (a) and #2 (b), which were measured in a pure solution of 30 fM streptavidin, and Sensors #3 (c) and #4 (d), which were measured in a mixed solution containing 30 fM streptavidin and 60 fM tumor necrosis factor alpha (TNF- α).

two WSe₂ sensors, respectively. Figures 5.6 (b) and (c) shows that for both p-type and n-type sensor readings, the S - t response to 30 fM streptavidin in the serum solution is highly consistent with the S - t response to 30 fM streptavidin in the pure streptavidin solution. The complex protein background in the serum solution does not significantly affect the t -dependent sensor readings associated with the specific concentrations of target molecules. These two comparison experiments have demonstrated that the MoS₂

and WSe₂ FET biosensors operated using the IFDM cycle-wise method exhibit a high biodetection specificity. This is attributed to the flushing step in an IFDM cycle, which can effectively remove non-specifically adsorbed molecules from the FET sensing channel, therefore minimizing the false sensor responses. Furthermore, two-dimensional TMDC surfaces usually have very low friction coefficients and exhibit very low adhesion

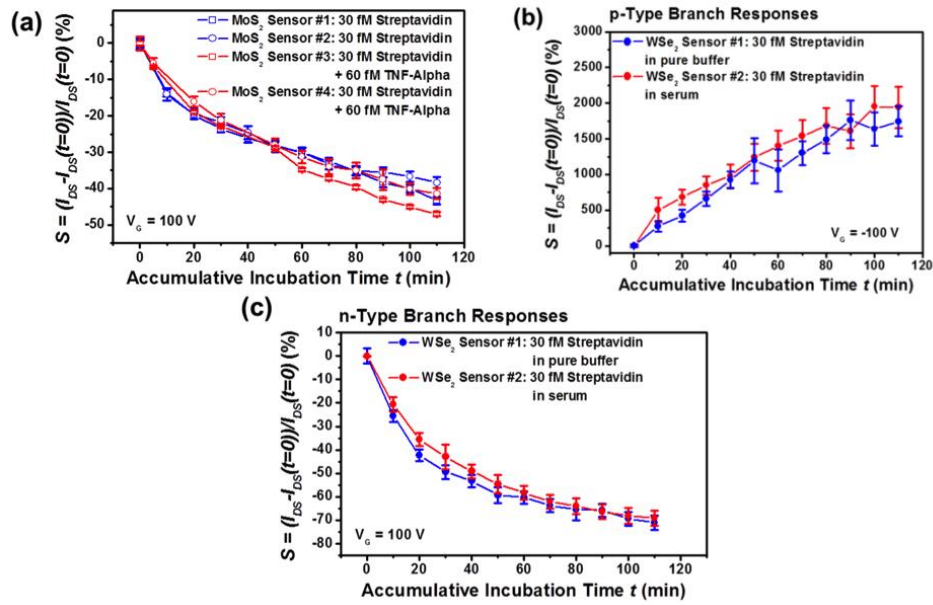


Figure 5.6 Sensor responses to targeted streptavidin molecules in different solutions: (a) S - t responses, measured from four n-type MoS₂ FET sensors, to 30 fM streptavidin in a pure solution (sensors #1 and #2) and a solution also containing 60 fM TNF- α (sensors #3 and #4); (b) p-type and (c) n-type S - t responses, measured from two ambipolar WSe₂ FET sensors, to 30 fM streptavidin in a pure solution (sensor #1) and a serum solution (sensor #2).

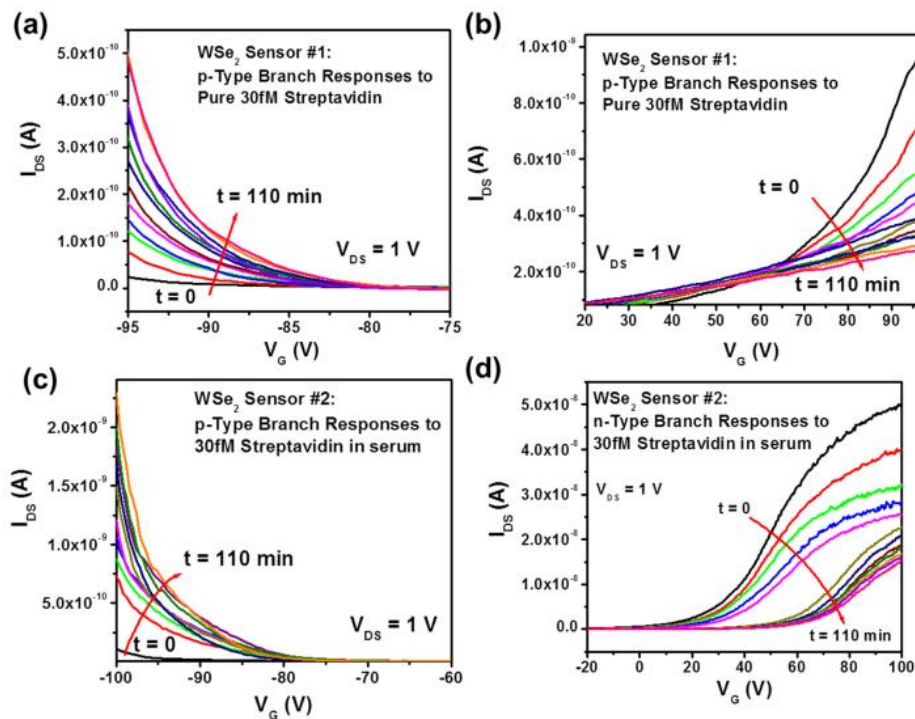


Figure 5.7 IFDM cycle-wise responses, measured from two ambipolar WSe_2 FET biosensors, to streptavidin molecules in a pure solution and a serum solution: (a) p-type and (b) n-type branches of the t -dependent I_{DS} - V_G characteristic curves measured from the sensor (Sensor #1) in a pure solution of 30 fM streptavidin; (c) p-type and (d) n-type branches of the t -dependent I_{DS} - V_G characteristic curves measured from the sensor (Sensor #2) in a serum solution containing 30 fM streptavidin.

to other materials, which is expected to make the flushing of non-specifically adsorbed molecules on TMDC FET channels much easier than that on the FET channels made from conventional semiconductors [164].

5.3.5 Sensor Response Signals Measured at Different Streptavidin Concentrations

We further used MoS_2 and WSe_2 FET sensors operated by the IFDM cycle-wise method to obtain standard curves for streptavidin quantification (*i.e.*, calibrated sensor response (R) versus analyte concentration (n) curves) and to enable rapid biomolecule quantification at femtomolar levels. Specifically, we fabricated a set of MoS_2 sensors and

a set of WSe₂ sensors in the same fabrication batch. These sensors were used to quantify 30 fM, 300 fM, and 3 pM solutions of pure streptavidin. Figures 5.8 and 5.9 display the t -dependent I_{DS} - V_G characteristic curves of these MoS₂ and WSe₂ sensors, respectively, which were measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM). Figures 5.10 (a) and (b) show the S - t response curves captured from the n-type branches of the I_{DS} - V_G curves of MoS₂ FET sensors (captured at $V_G = 100$ V) and the p-type branches of the I_{DS} - V_G curves of WSe₂ FET sensors (captured at $V_G = -100$ V), respectively. These S - t response curves were measured at $n_{streptavidin} = 30$ fM, 300 fM, and 3 pM. For both MoS₂ and WSe₂ sensors, the initial slopes of their S - t response curves exhibit a dependence on streptavidin concentrations ($n_{streptavidin}$) and can be exploited for rapid streptavidin quantification. The determination of such initial slopes of S - t curves does not need a long accumulative incubation time until the analyte-receptor binding reaction reaches to the equilibrium state. Therefore, the initial slope of an S - t curve (*i.e.*, dS/dt at $t \sim 0$) can serve as a derived sensor response quantity for enabling rapid biomolecule quantification. Here, the specific dS/dt value is extracted from the S - t response curve through linearly fitting the S data measured within the first three IFDM cycles (typically, the accumulative incubation time is shorter than 20 min for $n_{streptavidin} \sim 30$ fM, and the total assay time is less than 23 min). Figures 5.10 (c) and (d) plots the $dS/dt - n_{streptavidin}$ curves extracted from the S - t response curves measured from MoS₂ and WSe₂ sensors, respectively. Such $dS/dt - n_{streptavidin}$ curves can serve as standard curves for rapid streptavidin quantification. To evaluate the repeatability of such streptavidin quantification using the initial varying slopes of S - t response curves, another set of two MoS₂ FET sensors fabricated in the same batch were used to generate additional two S - t

response curves at $n_{streptavidin} = 3 \text{ pM}$, which were plotted in Figure 5.10 (a). All three $S-t$ response curves measured at $n_{streptavidin} = 3 \text{ pM}$ appear to have very similar initial slopes, which can be well correlated to $n_{streptavidin} = 3 \text{ pM}$. Additionally, another WSe₂ sensor was

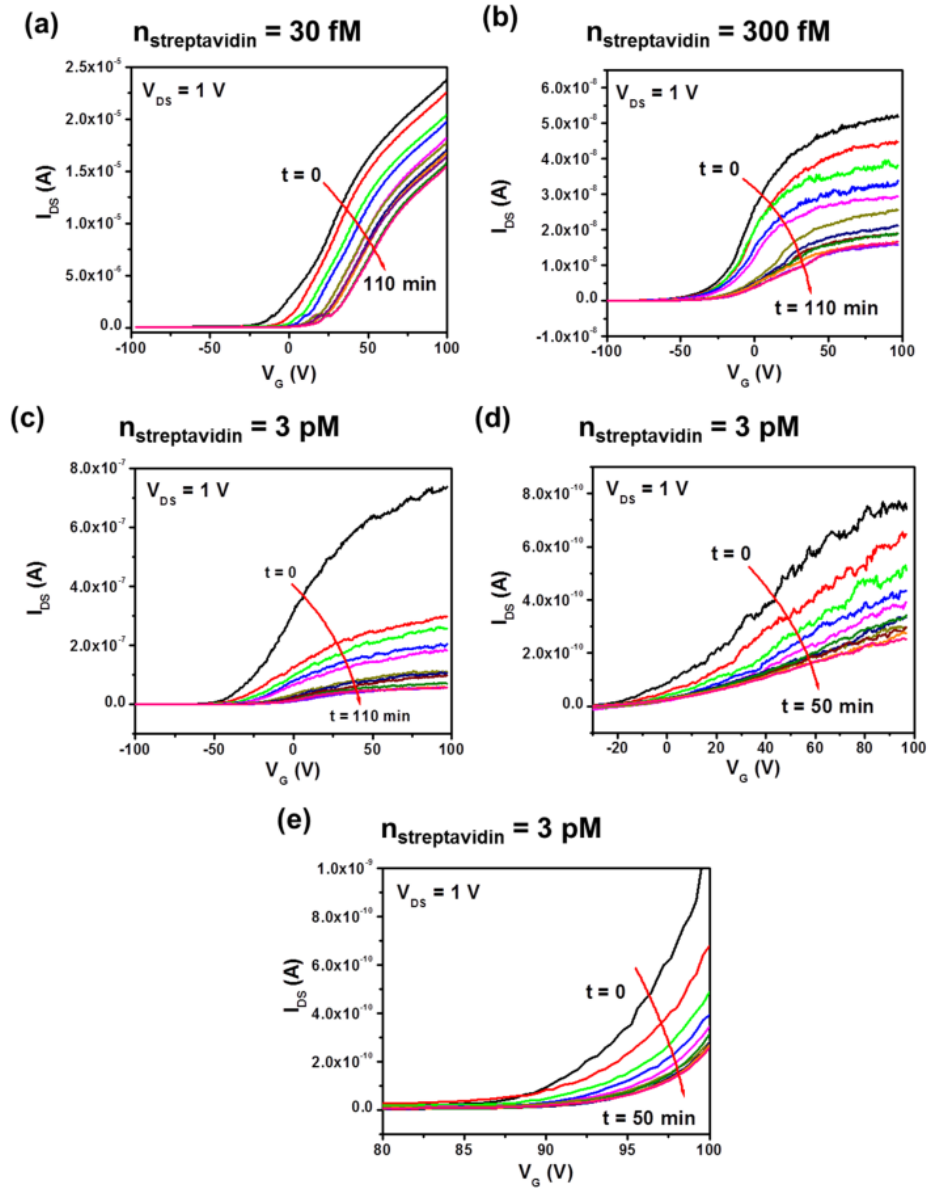


Figure 5.8 t -dependent I_{DS} - V_G characteristic curves of a set of MoS₂ FET sensors, which were measured at different streptavidin concentrations ($n_{streptavidin} = 30 \text{ fM}$, 300 fM , and 3 pM).

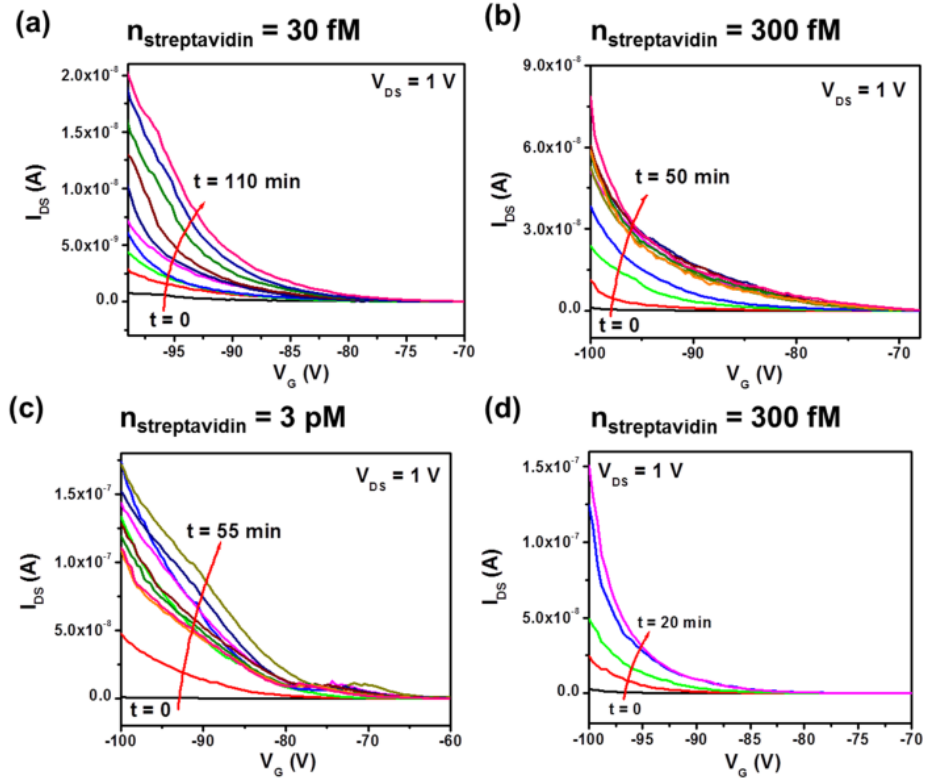


Figure 5.9 p-type branches of the t -dependent I_{DS} - V_G characteristic curves of a set of WSe_2 FET sensors, which were measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM).

used to generate an additional S - t response curve at $n_{streptavidin} \sim 300$ fM, which is also consistent with the one already plotted in Figure 5.10 (b).

In addition, we also used our MoS_2 FET sensors and the IFDM cycle-wise method for testing the quantification of other clinical biomarkers, which usually exhibit a much weaker affinity to their antibody receptors in comparison with streptavidin. Here, we demonstrated Interleukin 1beta ($IL1\beta$) quantification. $IL1\beta$ is a cytokine protein and serves as an important mediator of the inflammatory response. It is related to many cellular activities, including cell proliferation, differentiation, and apoptosis [165]. $IL1\beta$

has a relatively weak affinity to its antibody in comparison with streptavidin (*i.e.*, K_D , *Streptavidin* $\sim 10^{-14}$ M (or 10 fM), and K_D , *IL1 β* $\sim 10^{-9}$ M (or 1 nM)). Therefore, the state-of-the-art biodetection methods usually result in LODs in the range of 10s pM to 1 nM for IL1 β quantification, and fM-level IL1 β quantification is still a challenge.

Figure 5.11 displays the t -dependent I_{DS} - V_G characteristic curves measured from a set of six MoS₂ FET sensors, which were subjected to a set of pure IL1 β solutions with different IL1 β concentrations (*i.e.*, $n_{IL1\beta} = 0$ fM, 1 fM, 4 fM, 20 fM, 100 fM, and 500 fM), respectively. From these I_{DS} - V_G curves, the corresponding S - t response curves were extracted at $V_G = 100$ V and plotted in Fig. 5.12 (a). It should be noted that for $n_{IL1\beta} = 0$, the S - t response curve is not completely flat. This indicates that the pure buffer solution results in moderate nonspecific absorption of ions or molecules on the MoS₂ channel. Currently, we have not fully understood the mechanism of such nonspecific absorption, but temporarily attributed it to the slow reaction between the crystal defects on the MoS₂ channel and the buffered solution. Such a zero-concentration S - t response curve serves as the baseline for IL1 β quantification. Figure 5.12 (a) shows that the S - t response curve measured at $n_{IL1\beta} = 1$ fM is significantly distinct from that measured at $n_{IL1\beta} = 0$. This indicates that our MoS₂ FET biosensors in combination with the IFDM cycle-wise method can result in a LOD of ~ 1 fM for IL1 β quantification. As discussed for streptavidin quantification, the initial slopes of the S - t response curves shown in Fig. 5.12 (a) can be also used as response signals for the rapid IL1 β quantification. Figure 5.12 (b) shows the extracted dS/dt data plotted as a function of the IL1 β concentrations ($n_{IL1\beta}$). In our work, to unambiguously determine the dS/dt values for IL1 β quantification, the minimum incubation time is about 20 min, and the corresponding total assay time is

about 23 min. Therefore, this work has demonstrated that the TMDC FET biosensors operated using the IFDM cycle-wise method can enable rapid fM-level quantification of IL1 β and other biomolecules with relatively weak affinity (*i.e.*, nM-level K_D values) to their antibody receptors.

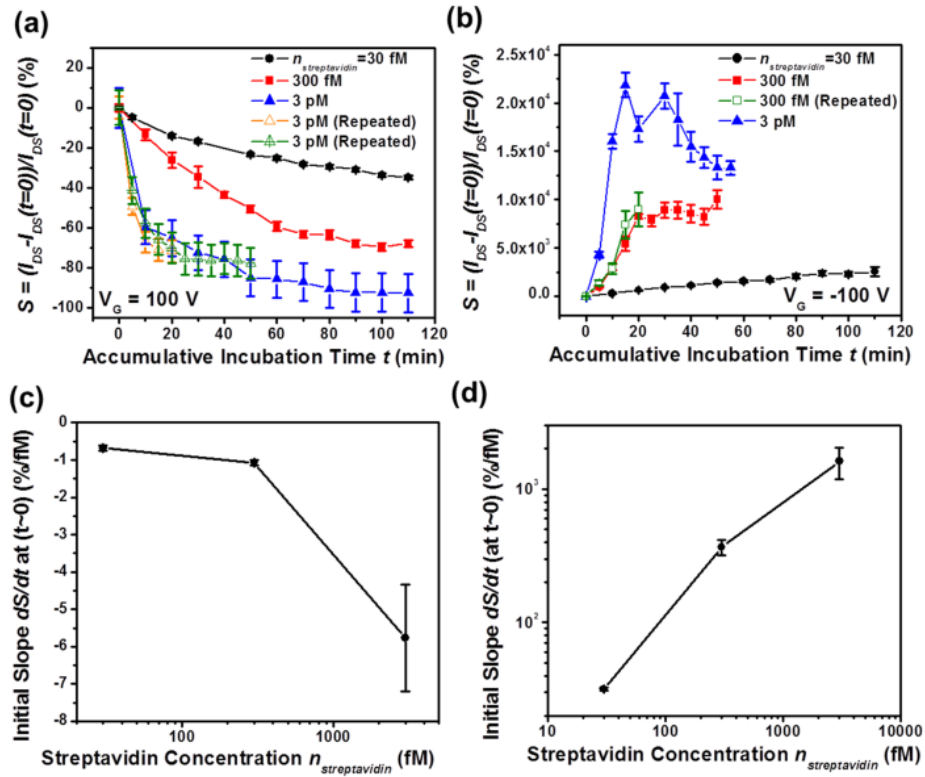


Figure 5.10 S - t response curves measured at different streptavidin concentrations ($n_{streptavidin} = 30$ fM, 300 fM, and 3 pM), which are captured from (a) the n-type branches of the t -dependent I_{DS} - V_G curves of a set of MoS₂ sensors and (b) the p-type branches of the I_{DS} - V_G curves of a set of WSe₂ sensors; (c) and (d) plots the dS/dt - $n_{streptavidin}$ curves extracted from the S - t response curves measured from MoS₂ and WSe₂ sensors, respectively.

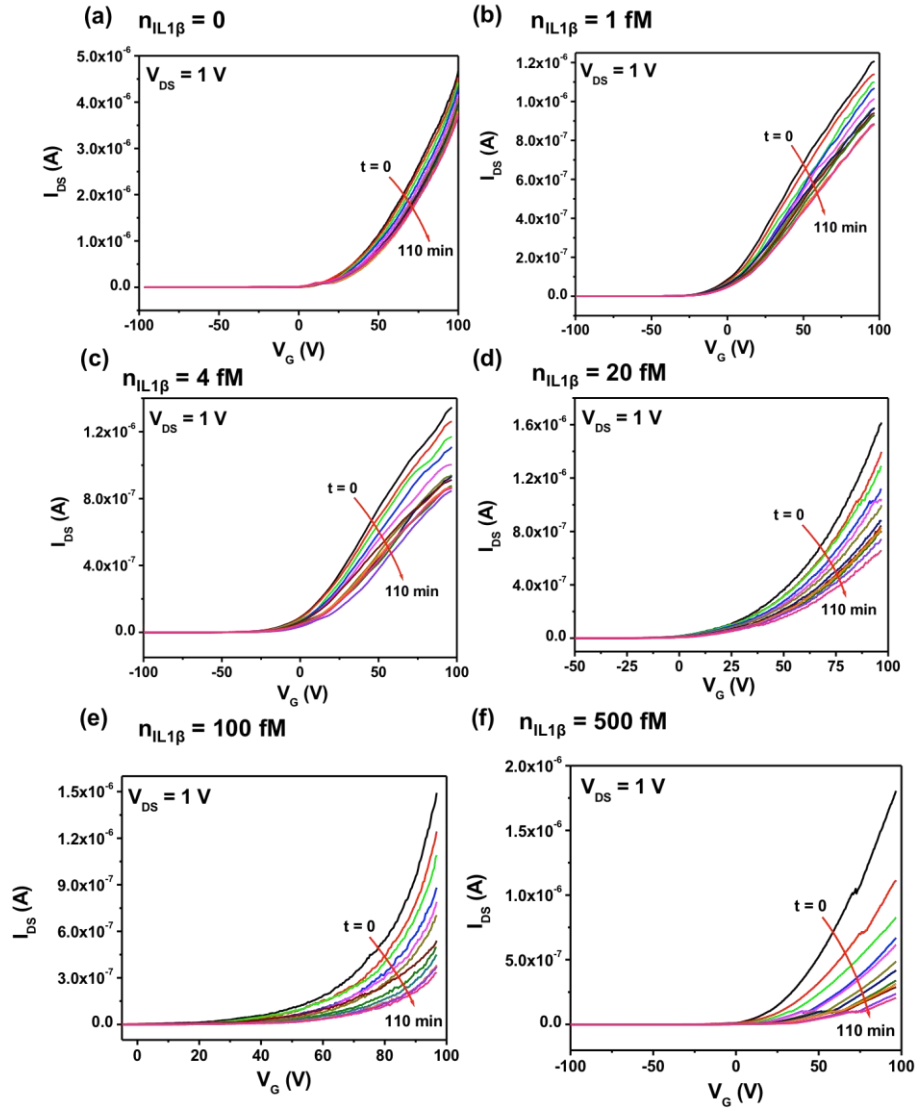


Figure 5.11 t -dependent I_{DS} - V_G characteristic curves of a set of MoS₂ FET sensors, which were measured at different IL1 β concentrations ($n_{IL1\beta} = 0$ fM, 1 fM, 4 fM, 20 fM, 100 fM, and 500 fM).

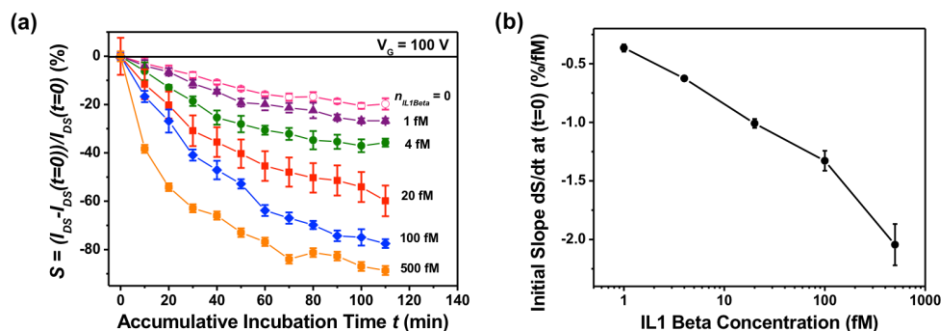


Figure 5.12 (a) S - t response curves measured at different IL1 β concentrations ($n_{IL1\beta} = 0$ fM, 1 fM, 4 fM, 20 fM, 100 fM, and 500 fM), which are captured from the t -dependent I_{DS} - V_G curves of a set of MoS₂ sensors at $V_G = 100$ V, $V_{DS} = 1$ V; (b) plots the dS/dt - $n_{IL1\beta}$ curve extracted from the S - t response curves shown in (a).

5.4 Summary

In summary, we established a cycle-wise approach for operating MoS₂ and WSe₂ FET biosensors. This approach enabled rapid, low-noise, highly specific biomolecule quantification and analyte-receptor binding kinetics analysis. The cycle-wise detection process utilized a microfluidic structure integrated with a MoS₂ or WSe₂ FET sensor that delivers a time-sequenced series of reagent fluids to the sensing FET channel. Periodically delivering the reagent fluids into the device resulted in assay cycles, each with four stages: incubation, flushing, drying, and measurement (*i.e.*, IFDM cycles). This process can measure time-dependent sensor responses to analyte-receptor binding reaction. The IFDM cycle-wise approach can physically separate incubation, flushing, and electrical measurement steps from each other in the time sequence. Thus, it can prevent liquid-solution-induced electrochemical damage, electronic noise, signal screening, and nonspecific adsorption, therefore improving the sensors' durability,

detection limit, sensitivity, and specificity. Combined with the high sensitivity and low detection limit of MoS₂/WSe₂-based FET biosensors, this cycle-wise detection method can realize rapid biomolecule quantification at femtomolar levels. In this work, we have demonstrated the quantification of femtomolar-level streptavidin and IL1 β suspended in pure and complex solutions (*e.g.*, serum) with a detection limit of \sim 1 fM and a total incubation time less than 20 min (or total assay time less than 23 min). This work leverages the superior electronic properties of emerging layered semiconductors for label-free biosensing applications and also advances the sensor operation technology toward realizing fast real-time biomolecule quantification at femtomolar levels. In addition, the presented cycle-wise method could be generally applied to other nanoelectronics-based biological and chemical sensors while overcoming the issues resulting from poor compatibility between reagents and electrical detection processes in a fluidic environment.

Chapter 6

Conclusions and Future Work

6.1 Summary of Thesis

This thesis work successfully advanced nanofabrication technologies and device physics for biosensing applications based on atomically layered materials. In our first study, we developed the plasma-assisted nanoprinting approach for producing ordered and pristine few-layer-MoS₂ flake pixels over large areas. This work lays an important foundation for future scale-up nanoelectronic applications of few-layer-MoS₂ micro- and nanostructures. In our second study, we fabricated MoS₂ transistor sensors with insulating layers, which can be used for quantifying femtomolar-level biomolecules and molecule interactions. This work advanced the critical device physics for leveraging the excellent electronic/structural properties of TMDCs in biosensing applications as well as the research capability in analyzing the biomolecule interactions with fM-level sensitivities. In our third study, we compared between insulating-layer-coated and

insulating-layer-free MoS₂ transistor biosensors in their sensor response behaviors. This work advances the device physics for simplifying the transistor biosensor structures targeting for femtomolar-level quantification of biomolecules. In our fourth study, we developed cycle-wise operation of MoS₂ and WSe₂ transistor biosensors for enabling rapid biomolecule quantification at femtomolar levels. This work leverages the superior electronic properties of emerging layered semiconductors for biosensing applications and advances the biosensing techniques toward realizing fast real-time immunoassay for low-abundance biomolecule detection.

6.2 Future Research

Our TMDC FET sensors, especially those fabricated in different fabrication batches, still exhibit observable device-to-device variation in sensor response signals. It is anticipated that such variation could be further minimized by using calibrated response quantities that are completely independent of the FET performance parameters. We currently lack such a calibration approach, however, because we have not established a complete device physics model that can be used for quantitatively analyzing the responses of TMDC FET biosensors to specific analyte-receptor binding reaction events. In our previous work, we qualitatively identified that the electrical responses measured from a few-layer-TMDC FET biosensor are attributed to the net effect of (i) analyte-molecule-introduced doping to the TMDC channel and (ii) surface scattering due to binding-reaction-induced disordered potentials. Future work will seek to quantitatively calculate the contributions of these two mechanisms to the final sensor response signals

and establish a calibration method based on such calculation to further improve the sensor-to-sensor consistency in calibrated sensor readings. Furthermore, such a quantitative device physics model is also needed for extracting critical molecular interaction properties, such as the affinity constant (K_D), and association/dissociation rates (k_{on}/k_{off}) of analyte-receptor pairs, from electrical sensor responses. Another approach to explore device-to-device variation in sensor response signals is to study crystal structures of TMDC materials. Unlike silicon-based FET, TMDC FETs have polycrystalline structures. Therefore, the aspects of crystal domain boundaries on TMDC channel surfaces are different in each sensor, which induce the device-to-device variation in sensor response signals. Since we expect the receptors are functionalized on the domain boundaries, another future study will pursue the fabrication of our TMDC FET sensors with different channel lengths, which may affect the total number of binding sites on the sensing channels, resulting in a change of sensor response signals. Based on the aforementioned research, we can contribute to the development of a multiplexed cytokine detection platform with high throughput field-effect transistor biosensors.

APPENDICES

Appendix A

Finite Element Analysis for Simulating Electric Field around the Edge ribbon of a MoS₂ Pixel and Calculating Effective Gate Capacitance of Edge-Based FETs (C_g)

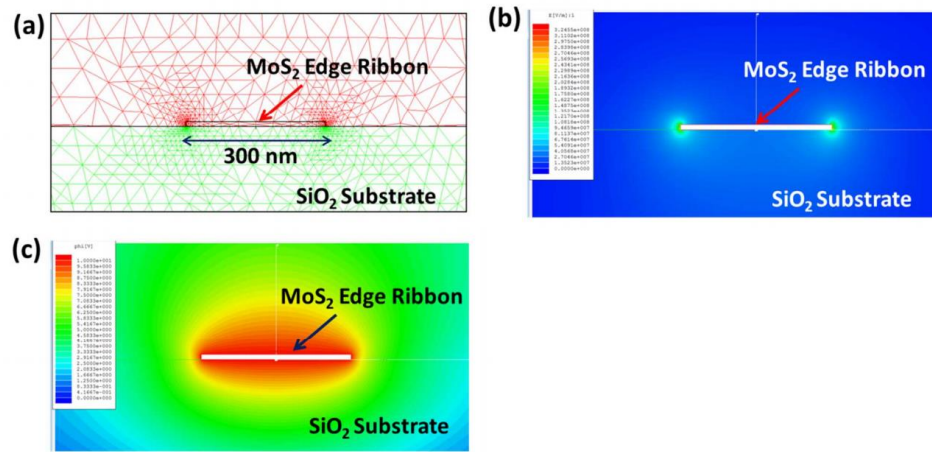


Figure A1 (a) Simulation model based on finite element analysis (FEA) for calculating the effective gate capacitance of the edge ribbon of a MoS₂ pixel, which takes into account the fringe effect at the feature edges. (b) Cross-sectional viewgraph of simulated electric field around the MoS₂ edge ribbon. (c) Cross-sectional viewgraph of simulated equipotential lines of 6 electric field around the MoS₂ ribbon.

Figure A1 shows the simulation of electric field around the edge ribbon of a MoS₂ pixel (cross-sectional view). The simulation was performed by using a finite element analysis (FEA) software tool (MAXWELL SV from Ansoft, Inc.). The simulation can

give out the effective gate capacitance associated with a single MoS₂ ribbon per unit channel length (C_g) [unit: F/m].

In the simulation, the dielectric constant of SiO₂ is $\epsilon_r \sim 3.9$; the thickness of the gate dielectric is $d \sim 330$ nm; the width and thickness of the MoS₂ edge ribbon are 300 nm and 10 nm, respectively. The average gate capacitance associated with a 300 nm wide, 10 nm thick MoS₂ ribbon per unit channel length is calculated to be 187.9 pF/m.

Figure B1 shows the dual-gate thin-film transistor biosensor model. The binding of TNF- α molecules with the antibody receptors functionalized on the HfO₂ effective layer can cause a potential change ($\Delta\Phi$) on this effective layer. $\Delta\Phi$ can be evaluated using $\Delta\Phi = qN_{TNF}/C_{HfO_2}$, where q is the effective charge brought to the HfO₂ effective layer through a single antibody-(TNF- α) binding event (here, q is the effective charge sensed by the transistor, and the screening effect due to the electrical double layer in solvent has been involved into q); N_{TNF} is the total number of TNF- α molecules bound to the HfO₂ effective layer; C_{HfO_2} is the total capacitance of the HfO₂ effective layer. This $\Delta\Phi$ induces a change in the conductive charge ($\Delta Q = C_{HfO_2}\Delta\Phi$) in the MoS₂ channel. This $\Delta\Phi$ can cause a shift of the threshold voltage (ΔV_T) measured from the back gate (note: not measured from the top gate), and ΔV_T can be evaluated by $\Delta V_T = \Delta Q/C_{SiO_2} = (C_{HfO_2}/C_{SiO_2})\Delta\Phi = qN_{TNF}/C_{SiO_2}$, where C_{SiO_2} is the capacitance of the back-gate dielectric layer. Furthermore, N_{TNF} can be calculated using $N_{TNF} = \sigma_{TNF}A$, where σ_{TNF} is the areal density of bound TNF- α molecules on the effective layer and A is the total sensor area. C_{SiO_2} can be calculated using $C_{SiO_2} = k_{SiO_2}\epsilon_0A/d_{SiO_2}$ and k_{SiO_2} are the thickness and dielectric constant of the SiO₂ back-gate dielectric layer, respectively; ϵ_0 is the vacuum permittivity. Therefore, $\Delta V_T = qN_{TNF}/C_{SiO_2} = qd_{SiO_2}\sigma_{TNF}/k_{SiO_2}\epsilon_0$.

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