WEAR OF AGED DENTAL COMPOSITES

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Summary

Surface morphology and wear characteristics of five dental composites subjected to accelerated aging were investigated. Aging conditions caused degradation of the surface as evidenced by the formation of microcracks and the exposure and loss of filler particles. Wear parameters such as track width, transition load and mode of surface failure caused by single-pass sliding were influenced by aging and by the composition of the composite.

1. Introduction

Wear of dental composite restorative materials has been characterized in vitro by single-pass sliding [1, 2]. Wear patterns and surface morphology of dental composite restorations in vivo have been reported by Kusy and Leinfelder [3] and by O'Brien and Yee [4]. Clinical wear characteristics may result from changes in the surface properties of the composites on aging in the oral environment. Dental composites subjected to accelerated aging *in* vitro showed changes in surface profile and exposure of filler particles [5]. The wear characteristics of the aged composites differed from the unaged composites in single-pass sliding [6].

The purpose of this investigation was to evaluate the morphology and wear characteristics of dental composites after different amounts of accelerated aging.

2. Materials and methods

Two conventional dental composites C and PR and three microfilled composites FN, I and SI were used (Table 1). Each composite was mixed according to the manufacturer's instructions and packed into a stainless steel mold 20 mm in diameter and 1.5 mm thick. The surfaces were covered with Mylar and the composite was allowed to polymerize for 7 min.

Accelerated aging of the sample disks was performed in a weathering chamber (Weather-Ometer 25-WR, Atlas Electric Devices Co., Chicago,

TABLE 1

Code	Product	Batch no.	Manufacturer		
Conver	ntional comp	osites			
С	Çoncise	Paste A, 9K3 Paste B, 9M3	3M Co., St. Paul, MN 55101		
PR	Profile	Universal, 47812 Catalyst, 37901	S.S. White Dental Products International, Philadelphia, PA 19102		
Microf	illed compos	ites			
FN	Finesse	Base, 071279 Catalyst, 071279	L. D. Caulk Co., Milford, DE 19963		
I	Isopast	Base, 661178 Catalyst, 660179	Vivadent (U.S.A.), Inc., Buffalo, NY 14217		
SI	Silar	Paste A, 9EE1 Paste B, 9E3	3M Co., St. Paul, MN 55101		

Code, batch number and manufacturer of products tested

IL 60613) at 43 °C and 90% relative humidity with UV radiation (Xenon Burner 12-2881, Atlas Electric Devices Co., Chicago, IL 60613) of 11.2 mW cm⁻² and intermittent water spray for 18 min every 120 min. Samples were tested after 300, 600 and 900 h of aging. Samples used as controls were not subjected to accelerated aging but were tested after storage for 7 days in 37 °C distilled water.

Single-pass sliding experiments on the samples were performed using an apparatus described elsewhere [1]. Four replications were used for each material and time of aging. The surfaces of the dental composites were subjected to the sliding of a diamond hemisphere ($360 \mu m$ in diameter) at a sliding speed of 0.25 mm s⁻¹. The normal loads ranged from 1 to 10 N in increments of 1 N. The wear track widths were measured using a calibrated eyepiece in a metallurgical microscope. Optical and scanning electron microscopy were used to examine the modes of surface failure. Classification of the failure modes were as follows: ductile, class 1; tensile cracking, class 3; extensive chevron formation, class 5. Classes 2 and 4 were intermediate classifications. Classes 4 and 5 involved a distinct loss of material.

The transition loads for different surface failure classifications were analyzed by the Spearman-Karber statistic [7]. Multiple comparisons of the transition loads for different aging periods within each material were performed using the Student-Newman-Keuls test [8] at the 95% level of confidence.

3. Results

Figures 1 - 4 show the average wear track widths at various normal loads for aged and unaged specimens of PR, FN, I and SI respectively.



Fig. 1. Wear track widths of PR vs. normal load: ★, 0 h; ○, 300 h; ●, 600 h; □, 900 h.
Fig. 2. Wear track widths of FN vs. normal load: ★, 0 h; ○, 300 h; ●, 600 h; □, 900 h.



Fig. 3. Wear track widths of I *vs.* normal load: ★, 0 h; ○, 300 h; ●, 600 h; □, 900 h.
Fig. 4. Wear track widths of SI *vs.* normal load: ★, 0 h; ○, 300 h; ●, 600 h; □, 900 h.



Fig. 5. Scanning electron micrographs of wear tracks of C, PR, FN, I and SI for a normal load of 8 N at (a) 0, (b) 300, (c) 600 and (d) 900 h. The arrow indicates the direction of sliding.

Similar wear track data were obtained for C and PR. The transition loads from class 1 to class 4 modes of surface failure are listed in Table 2. Scanning electron micrographs of wear tracks at a normal load of 8 N are shown in Fig. 5.

The wear track widths of the two conventional composites C and PR decreased after 300 and 600 h of aging. After 900 h of aging the wear tracks were wider at lower normal loads but narrower at higher normal loads than those of the unaged samples.

The microfilled composites showed an initial decrease in wear track width on aging and then the track widths approached those of the unaged samples on further aging. There were differences in the wear track widths between each aged and unaged microfilled composite and between the microfilled composites.

The two conventional composites had similar transition loads from class 1 to class 4 modes of surface failure. Both had lower transition loads

TABLE 2

Transition load from class 1 to class 4 modes of surface failure for aged and unaged dental composites

Condition	Material transition load (N)					
	C	PR	FN	Ι	SI	
Unaged	2.8 (0.5)	2.3 (0.5)	8.0 (0.6)	4.3 (1.5)	7.0 (1.7)	
300 h	1.8(0.5)	1.5(0.0)	3.3 (0.5)	4.3(0.5)	6.0 (0.6)	
600 h	1.5(0.0)	1.5(0.0)	2.8(0.5)	5.0 (1.3)	5.0 (0.6)	
900 h	1.5 (0.0)	1.5 (0.0)	3.0 (0.6)	4.3 (0.5)	2.0 (0.6)	

The mean values of the transition load are given with the standard deviations in parentheses.

The values connected by the same vertical line are not significantly different at the 95% level of confidence.

after accelerated aging. There was no significant difference between the transition loads for the composites aged for 300, 600 and 900 h, but they were significantly different from those of the unaged composites.

The effect of aging on the transition loads of the three microfilled composites was different for each composite. The transition load for FN decreased from 8 to 3 N on aging for 300 h and then there was no further change. There were no significant differences in the transition loads for aged and unaged I. The transition load for SI showed gradual changes. There was no significant difference between unaged samples and 300 h samples, nor between 300 and 600 h samples. The transition load for SI at 900 h was significantly lower than that of any other SI sample.

The surface morphology of the composite materials was altered by accelerated aging conditions. Changes in morphology became more severe with increased aging as shown in Fig. 6 for the conventional composites and Fig. 7 for the microfilled composites. Degradation of the conventional composites was characterized by the formation of microcracks and exposure and loss of filler particles over the entire surface. Degradation of the microfilled composites was characterized initially by the formation of isolated microcracks that became more extensive with continued aging.

4. Discussion

Accelerated aging of dental composites causes chemical degradation of the surface characterized by microcracks and exposure and loss of filler particles. The degradation causes changes in the mechanical properties of the surface layer characterized by lower track width, lower transition load and greater loss of material during single-pass sliding.

The composition of the composites appears to influence their chemical degradation. The conventional composites C and PR contain 76.2% and 78.4%



(a)



(b)



(c)



(d)

Fig. 6. Scanning electron micrographs of surface morphology of PR and C at (a) 0, (b) 300, (c) 600 and (d) 900 h.

by weight of filler particles respectively that are several microns in size [9]. The microfilled composites FN, I and SI contain 33.2%, 37.2% and 49.7% by weight of silica particles respectively that are less than 1 μ m in size. The polymer matrix of the composites is a dimethacrylate containing Bisphenol A, with the exception of I which has a urethane dimethacrylate polymer matrix. The conventional composites appeared to be most susceptible to degradation, whereas the microfilled composite with the urethane dimethacrylate matrix appeared to be the least susceptible.



(d)



Composite restorations in the oral environment are subject to aging, although the conditions are not identical with the *in vitro* conditions used in this investigation. The rate of surface degradation *in vivo* is slow; thus, excess wear may not be noticeable until months after the placement of the restoration. However, a degraded surface layer would be removed more easily by abrasion, erosion or other mechanisms of wear. The subsequent exposure of a freshly worn surface to the degradation process would continue the cycle.

5. Conclusions

Surface morphology and wear characteristics of dental composite restorative materials were studied by accelerated aging *in vitro*. The observed

changes in wear track widths, transition loads and modes of surface failure caused by single-pass sliding were attributed to surface degradation of the dental composites as a result of aging.

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