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# BIMONTHLY PROGRESS REPORT NO. X THERMAL-SHOCK INVESTIGATION

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#### THERMAL-SHOCK INVESTIGATION

#### OBJECT

The object of this research is to evaluate optimum design of test specimens and criteria which will permit correlation of thermal-shock data with performance of the material in the form of turbine buckets.

#### SUMMARY

Thermal-shock tests were run on wrought S-816 alloy that had been solution-treated and aged. Nominal test temperatures were 1600, 1700, and 1800°F. The resistance to cracking of these heat-treated specimens was the same as that of specimens tested in the as-received condition.

Thermal-shock resistance of S-816 alloy decreased monotonically and smoothly with increasing temperature.

On a comparative basis, S-816 alloy is inferior in resistance to cracking by thermal shock to the best N-155 alloy at all temperatures and to practically all Inconel specimens at 1700°F.

The scatter of results was smaller than in most previous tests.

#### INTRODUCTION

Previous tests on wrought S-816 alloy in the thermal-shock rig had been conducted on the alloy in the as-received condition. Inasmuch

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as commercial uses of the alloy frequently dictate a heat-treatment, it was felt that tests should be run on specimens that had been solution-treated and aged. Moreover, it was believed that the test temperature of 1500°F previously employed was too low for tests of reasonable duration.

Tests were conducted on heat-treated specimens at nominal temperatures of 1600, 1700, and 1800°F during the period December 11, 1953, to February 11, 1953.

#### APPARATUS AND PROCEDURE

The test rig employed in these tests was identical with that described in the latest progress reports. One test unit consisting of air nozzle, specimen holder, and radiation pyrometer was used for all tests in order to eliminate differences in results that might be introduced by variations in test units.

The test specimens were machined from 3/4-inch diameter rod. After machining, the specimens were solution-treated simultaneously for 1 hour at 2150°F, water-quenched, and aged for 16 hours at 1800°F. Four specimens were tested at each of three maximum cycle temperatures, 1600, 1700, and 1800°F. Of the three possible edges available for use as the cooled edge on each specimen, the most nearly perfect edge was selected as the cooled edge.

Specimens were viewed by a telescope, as previously reported, to detect cracks.

The criterion of cracking was the presence of a crack completely across the cooled edge of a specimen.

#### RESULTS

The results of tests on heat-treated specimens are given in Fig. 1 together with previous results on specimens tested in the as-received condition. The specimens reported in the present report are Nos. 9 through 20; the specimens from previous tests used for comparison are Nos. 3 through 8.

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Two specimens which were to have been run at 1600°F were actually run as follows: No. 16 1615°F

No. 20 1660°F

The resistance of these specimens to cracking is in line with the results from other specimens if the higher test temperature is taken into consideration. The tabular results are as follows.

Number of 1600°F	Cycles to First Crack	at 1800°F
1000 1	1100	1000 F
5130	2426	<b>10</b> 69
4600	1903 <sup>+</sup>	956 <b>-</b>
3620 (1615°F)	1956-	1146 <del>-</del>
3100 <sup>-</sup> (1660°F)	2 <b>300</b> -	784

The specimen run at 1700°F that shows 1903 cycles to failure was actually found not to be cracked completely through the cooled edge; rather, a crack 0.031 inch long was present in a 0.034-inch edge. The test was stopped because it was felt that oxide had filled the lower 0.003 inch of the crack, but microscopic observation of the cleaned edge showed the incompleteness of the crack. Based on past experience with crack propagation, it is felt that the actual number of cycles to failure would have been closer to 2000 cycles than to 1900 cycles.

The three specimens tested at 1800°F that show 1146, 1049, and 956 cycles to failure all had cracked somewhat beyond the edge thickness. Oxide obscured the cracks to some extent, so that crack detection was difficult. The actual numbers of cycles should thus be less than those shown, probably below 956 cycles.

The two specimens tested at 1700°F showing 2300 and 1956 cycles to first cracking and the specimen tested at 1660°F all had passed the number of cycles to first cracking by the time the test was stopped. Oxide obscured these cracks so that actual determination of crack length was difficult.

It is estimated that scatter of results is about 15% at 1600 and  $1800^\circ F$ , and about 20% at  $1700^\circ F$ . The data are relatively consistent within this range of scatter.

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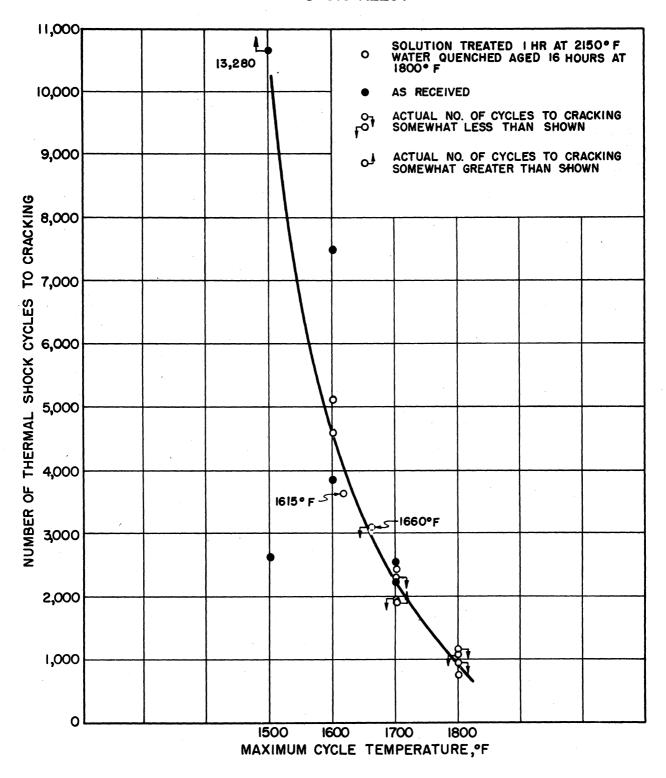
Specimens previously reported were tested in the as-received condition at 1500, 1600, and 1700°F. Results from these tests fall along the line of results from the present tests. It appears that the heat-treatment used did not appreciably alter the thermal-shock resistance of the wrought S-816 alloy, but the treatment may have increased the reproducibility of results.

The wrought S-816 alloy specimens were poorer in resistance to thermal cracking than the N-155 alloy, lot I, previously reported, and about the same as the N-155 alloy, lot II, previously reported. Incomel specimens were tested extensively only at 1700°F; practically all of the Incomel specimens were superior to the S-816 alloy at that temperature.

#### CONCLUSION

- 1. The heat-treated wrought S-816 alloy specimens manifested thermal-shock resistance that decreased smoothly with increase in test temperature from 1600 to 1800°F.
- 2. The results from wrought S-816 alloy in the heat-treated and as-received conditions are about the same at 1600 and 1700°F, the only two test temperatures available for comparison. Scatter of results from the heat-treated specimens was smaller than from the as-received specimens.
- 3. Heat-treated wrought S-816 alloy is inferior to N-155 alloy, lot I, but is about the same as N-155 alloy, lot II, in resistance to thermal cracking at temperatures from 1600 to 1800°F. At 1700°F, Incomel in most cases outperforms S-816 alloy in resistance to thermal cracking.

FIG. I S-816 ALLOY



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#### KEY TO LOG

Column (1)	(1) Relative position on bar stock 1 Specimen number
W .045 P.F.	Arrow indicates direction and location of cooling jet; cooling medium is air unless otherwise stated Cooling medium is water Width of cooled edge, inches Previously subjected to rotating beam fatigue as shown in column (6) Failed during pre-fatigue
Column (3) M 1500/5 P1800 +10/100 40.5K to 1800	Thermal shock cycle manually controlled Automatic cycle control; maximum temperature, °F, and length of cooling period, seconds  Dead load, 1800 lbs  Starting with stated maximum temperature, maximum temperature was increased 10°F after each 100 cycles  Reversed-bending (rotating-beam) fatigue tests; maximum stress, 40,500 psi  Maximum temperature held constant after 1800°F was reached
Column (4) A W no symbol	Air cooling for stated number of cycles Water cooling for stated number of cycles Air cooling for stated number of cycles
Column (5) O F C G FC PC	No failure visible Fracture Cracks Grooves Face crack Possible crack
Column (6)  B A 0.14 T300/1600 G1500 OH  BT	Specimen warped due to thermal strains Area of cross section, square inch Heat treated before testing 300 hr at 1600°F Grooves first appeared at 1500 cycles Stated maximum temperature was exceeded due to malfunction of control unit Broke through to thermocouple hole

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Previously subjected to 82000 cycles at 40,500 psi

R Reproducibility test

N Specimen formed a neck due to tensile strain.

+100/5108 Maximum temperature was increased 100°F at 5108 cycles.

Check II Second test to determine the effect of alteration of testing procedure.

P Study of crack propagation

PT1 Previously subjected to tensile strain of 1% at room temperature IRSI Long-time test at reduced severity, Test No. I

T{}I Heat treated as shown in braces { }. Lot No. I

C20/1700 Heat treated for 20 hours by heating to 1700°F and allowing to cool for 5 seconds by natural convection.

TEST LOG

Specimen Number (1)	Cross Section (2)	Cycle (3)	Number of Cycles (4)	Type of Failure (5)	Remarks
Type 304 St	ainless Steel				
1	045	M		0	В
2	V <sub>w</sub>	1600/10	4400 A 300 <b>W</b>	С	В
3	<b>√</b> ○/	1600/4	1783	C ·	
4а. 4ъ	Fatigue Specimens	40.5К 40.5К	3300 2600	F F	
5	<b>→</b>	1700/4 1800/4	1100 675	O C	
6		1600/4 1900/4	6240 1240	O C	G6500
7	ightharpoonup	1500/4 P600	4130	F	A 0.16

TEST LOG (cont)

Specimen Number (1)	Cross Section (2)	Cycle	Number of Cycles (4)	Type of Failure (5)	Remarks
8	$\Diamond$	1600/5 1800/4	3082 517	0 C	Т300/1600
9	$\rightarrow$	1500/3	5753	0	
10		1600/4 1700/4 1800/4	1000 1000 80	0 0 C	
11	$\Diamond$	1500/5 P1800	1000	F	A 0.132
12	<b>†</b>	1500/5 P600 P900 P1800	5000 1200 203	0 0 <b>F</b>	A 0.133
13	$\rightarrow \diamondsuit$	1600/4	1284	C	G 115
14	$\rightarrow$	1500/4	1000	F	ОН
15		1600/5	1900	С	т300/1600
16	→�	1600/5	409	С	

TEST LOG (cont)

Specimen Number	Cross Section	Cycle	Number of Cycles	Type of Failure	Remarks
(1)	(2)	(3)	(4)	(5)	(6)
17	$\bigcirc$	1500/5 <b>P1</b> 800	300	F	A 0.140
18		1800/4	1950	С	G 1500
19	V w 1	1700/3	5 <b>30</b> W	c	
20		1500/3	1000	0	BT
ype 347 Sta	inless Steel				
1	. 045	1600/4 +10/100	866	С	
2	.020	1 <i>6</i> 00/4 +10/100	1147	С	
3	→ <b>?</b>	1500/4 +10/100	575	С	BT
4a 4b	Fatigue Specimens	54K 54K	5200 10400	F F	40.5к 82000

TEST LOG (cont)

Specimen Number (1)	Cross Section (2)	Cycle (3)	Number of Cycles (4)	Type of Failure (5)	Remarks
5		1500/4 +10/100	1326	C	
6	<u> </u>	1500/4 +10/100	1990	С	
7	$\rightarrow$ $\bigcirc$	1600/3.5 +10/100 to 1800	2700	G	
8	(Defective)				
9	.035	1600/4	2863	С	R
10	.020	1600/4	3787	С	Check II
11	.050	1600/4	2580	С	
12	.020	1600/4	3162	С	G 736
13	.020	1600/4	2204	С	G 2072
14	020	1600/4	2707	С	G 2604

Specimen Number (1)	Cross Section (2)	Cycle	Number of Cycles (4)	Type of Failure (5)	Remarks
15	.035	1600/4	3003	C	G2820 R
16	020	1600/4	2518	C	R
17	.023	1600/4	4850	0	Check I
18		Fatigue 64K	7200	F	54 <b>k</b> 103300
19	035	1600/4	1825	С	R
20		Fatigue 64K	4300	F	37K/217100 42K/11000 48K/35600 54K/10000 59K/10400
21		1600/4	4430	С	
22	(Defective)				
23		1600/5	2962	C	
24	V.010	Fatigue 59K	52900	F	

Specimen Number (1)	Cross Section (2)	Cycle (3)	Number of Cycles (4)	Type of Failure (5)	Remarks
25	.010	1600/5 P.F.	1562	С	54K/50000
26	V.010	1600/5	1960	С	53K/52000 59K/12000 64K/1000 70K/1000 75K/500
27	010	Х Р. <b>F</b> .	-	F	53K/52000 59K/11300
28	, 010	1600/5 P.F.	1594	С	53K/52000 59K/12000 64K/1000 70K/1000 75K/500
29	. 010	X P.F.		С	53K/52000 59K/12000 64K/1000 70K/1000 75K/300
30	.010	1600/5	1973	С	
31	.010	1600/5	2764	С	
32	V.010	1600/5	1500	C	

TEST LOG (cont)

Specimen Number (1)	Cross Section (2)	Cycle (3)	Number of Cycles (4)	Type of Failure (5)	Remarks (6)
33 (4)	V. 040	X P.F.	-	F	59 <b>K/3</b> 2600
3 <sup>4</sup> (3)	→V•036	P.F.	1811	C	60K/39000
35 (2)	(Used for o	calibration	of Heat-Eye)		
36 (1)	→ V.oto	1600/5 P.F.	1859	c	58K/30000
37 (5)		1600/5	4635	С	
<i>.</i> <b>3</b> 8	.025	1600/5	2114	С	T2/2000
39 (7)		1600/5	5 <del>/1</del> /10	G	G 2440 Rigid Support Nozzle No. 3
40 (8)	<b>→</b> ♡	1600/5	3143	G	Nożzle No. 4
41	$\nabla \leftarrow$	1600/5	2710	C	G 2000 Rigid Support Nozzle No. 3

TEST LOG (cont)

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
42		(	Used for calib	ration)	
43 (11)	.025	1600/5	10708	. C	P Rigid Support Nozzle No. 4
44	.035	1600/5	2046	С	т2/2000
45	.025	1600/5	1956	C	T2/2000
I.S. 21 (vi	tallium) Cast				
1	<b>→</b>	1500/3.5	, <b>i000</b>	C	BT
Inconel					
1	→ 0.015	1500/3	1450	C	
2	→ 0,030	1500/3 +10/100	2730	d	· · · · · · · · · · · · · · · · · · ·
3	→ 0.035	1500/3 +10/100	428	Ċ	ÞŤ
4	.035 🗸	L700/5	3167	C	T2/500 T1/3/1400
5	.035	1700/5	1819	<b>c</b>	T2/500 T1/3/1400

TEST LOG (cont)

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Tailure	(6)
6	→ 035	1600/4	7449	С	
7	.035 4	1700/5	4706	C	T2/500 T1/3/1400
8	.025	1700/5	2090	c	T1/3/1400 PTI
9		1700/5	6465	С	T2/800
10	.035	1700/5	3680	d	T1/3/1400 PT10
11	.028	1700/5	2860	C	T1/3/1400 PT5
12	.030	1700/5	1884	С	T1/3/1400 C20/1700
13	.025	1700/5	2500	<b>C</b> .	T1/3/1400 PT1
14	.030	1700/5	2527	C	T1/3/1400 PT5
15		1700/5	2804	C	T1/3/1400 PT10
16	.025 ←	1700/5	3590	С	T1/3/1400 PTO
17		1700/5	2270	С	T1/3/1400 PTI

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
18	V.	1700/5	2576 3015	PC C	T1/3/1400 PT5
19		1700/5	1850	C	T1/3/1400 PT10
20	.030	1700/5	2898	C	T1/3/1400 PTO
21		1700/5	7498 11265	F.C.(?)	T1/3/1400 LRSI
22	.035 🗲	1700/5	4339 6866	FC? C	T1/3/1400 flex. pipe to nozzle
23	.035	1700/5	2250	C	T1/3/1400
24		1700/5	8145	FC	T1/3/1400 LRSII
25	.035	1700/5	3538 4229	FC C	T1/3/1400
S-816 Alloy	(wrought)			and the second s	
1		1500/4 12700 No load	1788 18391	O C	A 0.08 N +100/5108 +100/10000
2		1500/4 P1100 to P700	2657	F	80.0 A
3	<b>→</b>	1700/4	2256	C	

TEST LOG (cont)

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
4	<b>→</b>	1700/4	2250	С	
5	→ (V)	1600/4	3870	C	
6	<b>↓</b>	1500/4	2630	С	
7	<b>→</b>	1500/4	13280	C	
8	<b>→</b>	1600/4	7497	Ç	
9	.0371	1800/5	1069	С	T 1/2150 W 16/1800
10	.037	1700/5	2426	C	T 1/2150 W 16/1800
11	.036	1600/5	5130	С	τ{1/2150 w 16/1800
12	.0388	1800/5	956 <b>-</b>	С	T 1/2150 W 16/1800
13	.034	1700/5	1903 <sup>+</sup>	C .003 short	т 1/2150 w 16/1800

TEST LOG (cont)

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
14	.0350	1800/5	1146	С	T 1/2150 W 16/1800
15	.036	1600/4	4600	C	T 1/2150 W T 16/1800
16	.0335	1600/4	3620	C	T 1/2150 W I 16/1800 Average test temp. was 1615° F.
17	.0362	1700/5	1956	C	T 1/2150 W I 16/1800 }
18	.0384	1800/5	784	С	T [1/2150 W]I 16/1800 }
19	.0345	1700/5	2300	С	T [1/2150 W] I 16/1800]
20	.0331	1600/5	3100	C	T 1/2150 W I 16/1800 } Average test temp. was 1660° F.

(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
N-155 Alloy	(Wrought)				
ĩ	.038	1700/5	3764 3878 4949	FC C 2C	T 1/3/2200 W]I
2	.040	1700/5	3211	С	T \[ 1/3/22001 W] I \\ 50/1400 \]
3	.038	1700/5	3248	C	T 1/3/2200 W I 50/1400
4	•034	1800/5	1508	C	T 1/3/2200 WIT 50/1400
5	.036	1600/5	3886	0	T 1/3/2200 W I 50/1400 Removed for check; No crack
6	.01:0	1700/5	3105	С	T 1/3/2200 W T 50/1400
7	.o.i.2	1800/5	1818	С	T{1/3,2200 W}I 50/1400
8	.039	1700/5	3195	C	T 1/3/2200 W I 50/1400
9	.037	1700/5	2888	С	T 1/3/2200 W T 50/1400
10	.041	1600/5	10124	0	T{1/3/2200 W}T 50/1400}
11	.045	1800/5	2052	C	T{1/3/2200 W}I 50/1400

-			*		
(1) Specimen Number	(2) Cross Section	(3) Cycle	(4) Number of Cycles	(5) Type of Failure	(6) Remarks
12	.038	1800/5	1228	С	T 1/3/2200 W II 50/1400
13	.048	1800/5	1095	С	T 1/3/2200 W II 50/1400
14	.035	1800/5	1042	С	T 1/3/2200 W II 50/1400
15	.0385	1800/5	990	С	T[1/3/2200 W]II 50/1400
16	.0415	1800/5	1130	C	T {1/3/2200 W ] II 50/1400 }
17	.040	1700/5	2229	С	T{1/3/2200 W}II 50/1400
18	.0365	1700/5	1995	С	T{1/3/2200 W}II 50/1400
19	.0395	1600/5	5153	С	T[1/3/2200 W]II 50/1400
20	.0465	1700/5	2320	С	T{1/3/2200 W}II 50/1400
21	.0433	1600/5	3530	С	T\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\

