CRACK TIP EFFECTIVE STRAIN RATES IN RATE SENSITIVE MATERIALS

C. S. Lee Battelle-Columbus Laboratories, Columbus, Ohio 43201 USA tel: 614/424-6424 R. M. Caddell and A. G. Atkins Department of Mechanical Engineering, University of Michigan Ann Arbor, Michigan 48104 USA tel: 313/764/1817

Simultaneous pairs of values of K (stress intensity factor) and R (fracture toughness, or equivalently strain energy release rate G) for quasi-statically running cracks in compact tension testpieces of polymethylmethacrylate have been determined at various crack velocities (\dot{a}) and temperatures (T) [1,2]. K were arrived at from the Gross-Srawley expression (see, for example [3]) using current crack lengths and loads, and R was measured by Gurney's segmental area methods [4].

Presuming that they are related by $K^2 = ER$ (forgetting the Poisson's ratio term), where E (Young's modulus) and R are both rate and temperature sensitive, it follows that E (\dot{a} ,T) can be determined from K^2/R and references to E found in independent simple tensile tests carried out at various T and $\dot{\epsilon}$ (tensile strain rates). In this way, experimental effective $\dot{\epsilon}(\dot{a}$,T) at the crack tip can be established. Crossplotting (E = K^2/R)_T, \dot{a} and ($E_{tensile}$ tensile test) $\dot{\epsilon}$,T in the range $10^{-4} < \dot{a} < 10^{-2}m/s$, gave

έ ≈ 0.13a

where the relationship was independent of temperature within our accuracy. For $a > 10^{-2}$ m/s, ϵ rose less steeply, again independent of T and flattened off at about $\epsilon = 10^{-2}$ as seen in Figure 1.

Williams [5] has given for the strain rate at a moving crack tip, $\dot{\epsilon} \approx \pi \epsilon_v^{3} (E/K)^2 \dot{a}$ (2)

(1.)

where ε_y is the yield strain, and where E and K are rate dependent. A similar expression may be arrived at from Irwin's crack tip stress rate equation [6]. In [1,2] independent relationships were derived for R(å,T) and the tangent modulus E($\dot{\varepsilon}$,T), which are a toughnessbiased Ree-Eyring expression

$$\dot{a} = A_1 \exp[-(U - \lambda R)/kT]$$
(3)

and

$$E = 12.12\dot{\epsilon}^{0.0087} - 0.0268T$$
 (GN/m²) (4)

where A_1 is a constant, k is Boltzmann's constant, λ is the activation area, and U is the activation energy. Amending (2) to

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$$\dot{\epsilon} \approx \pi \epsilon_{\rm y}^{3} ({\rm E/R}) \dot{a}$$
(5)

and substituting for E and R, we obtain a relation between ε and a, viz:

$$\dot{a} \approx (U/\lambda) \dot{\epsilon} / \{ \pi \epsilon^3 10^9 (12.12 \dot{\epsilon}^{0.0087} - 0.0268T)$$
 (6)

where $T(k/\lambda) \ln a/A_1$ is omitted since it is small in comparison with U/λ , which is some 1.62 kJ/m² [2]. Notice that this ε vs a relation is dependent upon temperature, whereas our experimental relation is too coarse to pick that up. On Fig. 1 are superimposed the predictions of (6) using $\varepsilon_y \approx 0.003$, the value of which was found by trial and error to bracket the experimental results. The trends are acceptable, but two comments must be made.

The value of ε_y seems very low in comparison with the critical strain level for room temperature craze initiation of 0.013 quoted by Kambour [7], or typical PMMA yield strains of $\varepsilon_y = 0.02$. In fact $\varepsilon_y = 0.003$ corresponds with the *offset* which produces $\varepsilon_y = 0.02$ in PMMA. Again, (6) could be made to agree with the experimental ε vs a relation, independent of temperature, by using different ε_y at every temperature. For example, at $\varepsilon = 10^{-5}\text{s}^{-1}$, $\varepsilon_y \rightarrow 0.0027$ for T = 283 deg K, but $\varepsilon_y \rightarrow 0.0035$ for T = 353 deg K. Such changes are very small, and demonstrate the sensitivity of (2) to cubing ε_y . A discussion of the definition of yield stress and strain in polymers, and how secant moduli (used by Williams [5]) may be affected, is presented in [8].

Secondly, (6) predicts a continuously increasing ε vs a relation whereas in fact there are limiting velocities for each temperature beyond which crack tip adiabatic heating produces instabilities (see e.g., [2,9]. Thus, there are cut-off points in the R(a) relation (3); for example, at T = 283 deg K, a \neq 3×10^{-2} m/s; at T = 353 deg K, a \neq 5 m/s. It is interesting that these limiting velocities coincide with the region of the experimental ε vs a data where the ε level off, to some $\varepsilon = 10^{-2}$ s⁻¹. The fact that such lower ε occur than are predicted by (6) fits in with the occurrence of adiabatic heating, because in general terms the equation predicts that lower ε are produced at higher T.

REFERENCES

- [1] C. S. Lee, Ph.D. Dissertation, University of Michigan, 1974.
- [2] A. G. Atkins, R. M. Caddell, and C. S. Lee, Journal of Materials Science 10 (1975) 1381, 1394.
- [3] W. F. Brown, Jr., Editor, *Review of Developments in Plane Strain Fracture Toughness Testing*, STP 463, American Society for Testing and Materials, Philadelphia (1970) 264.
- [4] C. Gurney and J. Hunt, Proceedings of the Royal Society A299 (1967) 508-524.

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- [5] J. G. Williams, International Journal of Fracture Mechanics 8. (1972), 393-401.
- [6] G. R. Irwin, Applied Materials Research 3 (1964) 65-81.
- [7] R. P. Kambour, A Review of Crazing and Fracture in Thermoplastics, GE Report No. 72CRD285, Schenectady, New York (1972).
- [8] C. S. Lee, R. M. Caddell, and A. G. Atkins, Materials Science and Engineering 18 (1975) 213-220.
- [9] J. G. Williams and G. P. Marshall, Polymer Letters 15 (1974) 251-252.
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Figure 1. Experimental and Theoretical Relations Between ε and a. 0-experimental Points