### ENGINEERING RESEARCH INSTITUTE THE UNIVERSITY OF MICHIGAN ANN ARBOR, MICH.

### REPORT

ON

### RUPTURE AND TOTAL-DEFORMATION CHARACTERISTICS OF M252 (VM), UDIMET 500, INCONEL 700, INCONEL 713 AND STELLITE 31 ALLOY

(PHASE VII)

by

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### RUPTURE AND TOTAL-DEFORMATION CHARACTERISTICS OF M252 (VM), UDIMET 500, INCONEL 700, INCONEL 713 AND STELLITE 31 ALLOYS

(PHASE VII)

The objective of the investigation was to establish rupture and total deformation strengths for representative samples from production heats of five alloys considered for turbine blade applications by the Wright Aeronautical Division. The properties were evaluated at temperatures considered to be of main interest for each particular alloy for the application. These temperatures were 1350° and 1550°F for the M252 (vacuum melted), 1350° and 1640°F for Udimet 500 and Inconel 700 alloys, 1550° and 1700°F for Inconel 713 alloy, and 1650°F for Stellite 31 alloy. Evaluation of properties were based on the rupture strengths and stresses for total deformations of 1.0, 0.5, 0.2 and 0.1 percent in 30, 100 and 300 hours.

### **SUMMARY**

The rupture and total deformation characteristics were within expected ranges in properties for the individual alloys, except for Inconel 713 alloy. Only limited data were developed for Inconel 713 because further testing was cancelled when the initial tests indicated that the strength of the specimens submitted was far below that considered characteristic of the alloy.

The ductility of the Udimet 500 and Inconel 700 specimens were very low at 1350°F.

The strengths for a total deformation of 0, l percent are sparse and limited to short-time periods. The same limitation applied to a lesser extent to the values for 0, 2 percent total deformation. The incompleteness of these small deformation strengths resulted from a number of causes as the investigation developed.

### EXPERIMENTAL MATERIALS

The reported chemical compositions available for the particular heats of material investigated are given in Table I. Errors were found for the reported analyses in one case and there is, therefore, doubt concerning the analyses given in Table I.

The specimens used were 0.250-inch in diameter by 1-inch long at the gage section. These were supplied as machined specimens except for the Inconel 713 and Stellite 31. Specimens of the Inconel 713 alloy were machined at the University from the cast bars submitted. Investment cast specimens of Stellite 31 were furnished. Additional descriptive information supplied was as follows:

### M252 Alloy

The specimens had been machined from 21/32-inch bar stock from vacuum melted Heat KA223. Only trace amounts of silicon and manganese were present in the heat. The stock had been heated 4 hours at 1950°F, air cooled and aged 15 hours at 1400°F. The stock was fully heat treated before machining. Eighteen specimens coded Al through Al8 were furnished.

### Udimet 500 Alloy

The stock used was from Utica Drop Forge and Tool Corporation Heat 4131, a vacuum melted heat which contained normal amounts of silicon and manganese. The heat treatment consisted of solution for 4 hours at 1975°F, air cooled, and double aged at 1550°F for 24 hours plus 1400°F for 15 hours. Specimens were machined after full heat treatment. Eighteen specimens coded El through E18 were supplied.

### Inconel 700 Alloy

The stock was solution treated at 2160°F for 2 hours, air cooled and aged at 1600°F for 4 hours to WAD 7827. The specimens, coded Cl through C9, were apparently from Heat Y7952.

### Inconel 713 Alloy

Specimens were machined from investment cast bars. The castings were made by the Wright Aeronautical Division in three heat lots, 78, 79 and 80, from a single lot of master alloy furnished by The International Nickel Company. The casting conditions were reported to be as follows:

Casting Temp. (*F)	Mold Temp. (*F)
2650	1900
2700	1900
2740	1900
	2650 2700

The castings were very coarse grained. As is shown later in the report, the stock was much weaker than is considered characteristic of the alloy. The reasons for this were not established, although it was indicated orally by WAD representative that the actual analysis after remelting for casting was off.

### Stellite 31 Alloy

As-cast investment cast specimens made by Haynes Stellite were used for the tests. Chemical analysis for the stock was not available. The material was indicated to be in accordance with AMS 5382 and the specimens furnished were coded Sl through S7.

### General Comment

It should be noted that the chemical analyses and heat numbers for the materials were submitted at a date considerably later than the specimens. Actual errors were found in this information. For instance, the composition reported for Stellite 31 was obviously wrong. There is, therefore, doubt that the others are entirely correct. The only definitely correct identification available is the coding marks.

### PROCEDURE

The temperatures at which the properties of the alloys investigated were to be established had been determined during preliminary discussion with representatives of the Wright Aeronautical Division. At the same time it had been decided that the required data would be curves of stress versus time over the range of 30 to 300 hours for rupture and total deformations of 1.0, 0.5, 0.2 and 0.1 percent. Total deformation was defined as all deformation including the elastic and plastic deformation during loading.

The stress-rupture time curves were first established. Creep data were taken during the rupture tests and supplemented with sufficient additional creep tests to establish the total deformation curves.

The tests were conducted in single specimen units with the load applied through a simple beam. Temperatures variations along the gage length and during the tests were automatically maintained at  $\pm 3^{\circ}F$  of the nominal indicated temperature. In conducting the tests, about 16 hours were used for temperature adjustment before the load was applied. Creep was measured by a modified Martens type optical extensometer with the extension rods attached to collars threaded on the specimens ahead of the pull rods. The measured deformation was corrected for the fillets and shoulder sections of the specimens on the basis of well established correction factors for the test conditions. Elongations after rupture were based on changes in the overall dimensions of the gage length and not on a punch marked gage length.

### RESULTS

The results of the investigation provide the following information:

- 1. A table for each alloy which gives the following:
  - (a) Stress-rupture time data and the elongation and reduction of area values for the rupture tests.
  - (b) Time at which creep tests were discontinued.
  - (c) Deformation during loading of each test.
  - (d) Time to reach deformations of 1.0, 0.5, 0.2 or 0.1 percent during the rupture and creep tests. The values were read from time-elongation curves plotted from the original creep data.
- 2. Curves of stress versus time for rupture and time for the total deformations plotted from the data described in "l". The values of elongation after rupture are shown on the figures near the rupture time points.
- 3. Values of stress for rupture and total deformations of 1.0, 0.5, 0.2 and 0.1 percent in 30, 100 and 300 hours were interpolated from the figures and included in the tables.

The stress versus time for total deformation curves for the smaller deformations are extrapolated back to short-time periods asymptotic to the stress which gave the deformation during loading. The tests generally did not attain the larger deformation during loading. The values for total deformations of 0.1 percent are generally few and limited to short-time periods. The curves were not completely established for the smaller deformations because the specimens supplied were too few, results as they were obtained did not justify the additional testing, or because review of the data with WAD representatives indicated further testing was not necessary.

The following comments apply to the data for the individual alloys investigated.

### M252 Alloy (vacuum melted)

The results obtained indicated a high level of rupture strength and ductility for the alloy at both 1350° and 1550°F (Table II and Fig. 1).

At 1350°F, the stress versus time curves for total deformations of 1.0 and 0.5 percent were close to the rupture curve, while the curves for 0.2 and 0.1 percent were at considerably lower stress levels. The larger percentage of deformation during loading involved in the total deformations was responsible for this distribution. At 1550°F when creep was a more predominant factor in total deformation, there was not as wide a separation in the curves and it was not necessary to stress specimens so near to the rupture strength to obtain total deformations of 1.0 and 0.5 percent.

### Udimet 500 Alloy

The Udimet 500 material investigated had a high level of rupture strength and ductility at 1640°F (Table III and Fig. 2). Ductility in the rupture tests was, however, very low at 1350°F. Because the deformation during loading was the predominant factor in the total deformation values at 1350°F the stress for a given deformation was only slightly lowered by increasing time. Data for total deformations of 0.2 and 0.1 percent were not established because creep was a relatively very small factor for these deformations at 1350°F.

### Inconel 700 Alloy

The properties of the material tested was reasonable for the alloy at 1640°F (Table IV and Fig. 3). The specimens were, however, quite brittle at 1350°F.

Due to this brittleness and high resistance to creep at 1350°F only a few tests were conducted.

### Cast Inconel 713 Alloy

The rupture strengths of the specimens provided (Table V and Fig. 4) proved to be much lower than is considered characteristic of the alloy at 1550° and 1700°F. When limited tests established this fact and it was brought to the attention of representatives of WAD further testing was cancelled.

### Cast Stellite 31 Alloy

Evaluation of properties were limited to 1650°F (Table VI and Fig. 5). The results appeared to be reasonable for the alloy at this temperature.

### DISCUSSION

The choice of materials used for the investigation was intended to give representative samples of the alloys as they would be supplied for production purposes. It was quite evident that this was not achieved in the case of Cast Inconel 713 alloy. The results in the other cases appeared to be reasonable for the alloys.

There are two main limitations to the results. The number of specimens tested were too few in several cases for complete establishment of the curves. It is far more important to recognize, however, that there can be variations in properties between heats of a given alloy. Thus the values established do not define the ranges in properties which would be expected.

It should be recognized that the M252 material was vacuum melted with only traces of silicon and manganese. This alloy is often made with normal amounts of these two elements.

# CHEMICAL COMPOSITION OF FIVE ALLOYS INVESTIGATED

Alloy	M252 vacuum melted	Udimet 500 vacuum melted	Inconel 700	Inconel 713	Stellite 31
Heat No.	KA223	4131	Y7952	78,79,80	
Chemical Composition (percent)					
C Cr Co Ni Mo Ti Al Mg Cb + Ta Si	0.13 19.13 9.90 56.55 10.0 0.88 Trace 2.68 1.05  Trace	0.08 18.7 13.8  4.15 1.02 0.1 2.9 2.9 0.01 0.01	0. 12 15. 7 28. 69 46. 25 3. 08 0. 07 2. 02 3. 13 	0.1 14.5  Bal. 4.95 3.14 0.15 0.75 5.5 2.78	9ldslisvs ton
ď	0,01		. !	0,01	

TABLE II

## CREEP-RUPTURE DATA FOR M252 ALLOY

### Original Data

h Specified ion in % rs)	0,5		(	2 °	),4  3,4  3,4  3,4  4,6	108	374.	<u>.</u> _		т - -	ָרָ . י	6 48,	57, 5 92, 0	,0 142,	633,	0.		(1) Extrapolated by creen rates	(2) Contraction occurred dur	•	a Deformation exceeded during loading	b Deformation less than indi-	cated value when rupture occurred or the test was	discontinued,	() Extrapplated.	
Time to Reach Specified Total Deformation in % (hours)	0, 2		,	<b>т</b> Б	<b>ರ ಗು</b>	ા ત	ಡ	10,0 840,0		0	e	າ້ ເ	17, 5	73,0	75,	275,0		ations		1 0	74	32	64	25	55	2.0
T	0, 1		(	<b>.</b>	ರ ರ	ď	ಹ	a 8, 0		ď	<b>ร</b> ิ	ಡ	ಡ	ಹ	<b>5</b> , 0	e	Data	Total Deformations		0, 5	. 29	59	58	23	51	18
Deformation on Loading	(%)	1350 F.	5	388	0,344	0,284	0,250	0, 176 0, 088	1550°F	0 197	-	<b>-</b>	0, 133		0,093	080 0	Interpolated I	Specified		0, 2	32	22	2.7	<b>∞</b>	22	(15)
Reduction of	Area (%)		α-	12	23	15	22	discontinued discontinued		34		) C	, ,	00 .	discontinued	dıscontınued		for Rupture or	in Specified Time (in	0° I	0 0	(14)	<b>1</b> 0	0	0	0
Elongation	(%)		α-	20	2.4	12		disco		37	3.2	) \ 1 \	40		disco	disco		Strength	$\sin Spe$	Rupture	83	(36)	1 2	31	61	24
Rupture Time	(hours)			31,5		222,3	934,9	963, 4 987, 3(2)		39, 0	113 3	0,000	77°, 7	C 200	0.706	1 */06			Test	Temp (F)	$\mathcal{C}$	1550	1350		1350	1550
Stress	(ps1)		90,000	82,000	73,000	000,09	53,000	37,500 19,000		37,000	30,000	25,000	22,000	11,000	1 , 500	13,000			$\operatorname{Time}$	(hr)	30		100		300	

## CREEP-RUPTURE DATA FOR UDIMET 500 ALLOY

### Original Data

ch Specified nation in %	7.0		6	5 23.	94.	.5 2.76.	483, U 1088, U b b			.6 6.	.0 28,	55,	0.	13% U 243, U 500 0 b	) •						_	a Deformation exceeded during	b Deformation less than indi-	occurred or the test was	() Extrapolated,
me tal	7 0		ď	ಡ	ಡ	ଟ (	പ് പ	م و		e	ທັ້ນ			56,5 97,5			ıtions	ļ	0,1	7 <u>]</u> 24	- 29	19	56	1 - 1	
ļ	0		ช	ಡ	ಡ	ದ (	ಡ ಡ	r q		ત્ય	ಡ	ಡ		ა 4 ი դ	ŧ	Data	Total Deformations	k	0,5	64 21	82	17	52	ļ	
Deformation on Loading	9	1350°F	0,385	0,362	0,309	0,275	0, 242 0, 193	60	1640°F	_	0, 138		10	0, 083 0, 055	) •	Interpolated D	ified	0	7 0	15	8	10	(2)	•	
	Area (%)		4	4	4,	3	aiscontinuea discontinued	discontinued		10	16	71	17	6. discontinued		71	0	ecified	0,1	(7)	. (	) 0 ) 0	0 0 8 0		
Elongation	(%)		5	<b>,</b>			discol discol	disco		11	18	01	19				Strength for Rupture	ı	Kupture	75 28	65	22	58 18		
Rupture Time	(nours)		25, 3	33, 0	132, 9		1422, 2 $1180, 0(2)$	940, 0(2)		18,7	61,0			598, 0					Temp (*F)	1350 1640	1350	1640	1350 1640		
	(ps1)		80,000	75,000	64,000	57,000	50°,000 40,000	19,000		30,000	25,000	21,000	18,000	10,000	•			Time	(hr)	30	100		300		

## CREEP-RUPTURE DATA FOR INCO 700 ALLOY

## Original Data

fied n %	0 1	1, 4	: :	i 1 ,	Ω	10, 2	67, 8 115, 5	206,0
Time to Reach Specified Total Deformation in % (hours)	0,0	0,5	0.67	ئے ہ	Ω		47.2 75.0	
Time to Total D	7,0	ત	ದ ದ	ಡ.	d	0.7	20.0 5.0	282,0
¢	0	ರ	ದ ದ	ಡ (	ಸ	ત	a <0,05	190,0
Deformation on Loading	(%) 1350*F	0,325	0, 295 0, 250	0, 220	0.205 1640°F	0.143	0.114 0.096	0,072
Reduction of	Area (%)	1	;	\$ B	discontinued	14	12 11	œ
Elongation	(%)	2	¦ ~	1 1	discoi	12	(9) <sub>8</sub>	œ
Rupture Time	(hours)	11, 1	> 50, 2(5) 216, 2(4)	> 93.0(5)(2)	845, 27=7	22,5	109, 3 203, 7,3,	704,0(4)
Stress	(psi)	72,000	65,000 55,000	48,000	45,000	30,000	24,000 20,000	15,000

	(2) Contraction occurred during	early stages of test	(3) Fractured in threads at	50, 2 hours	(4) Test interrupted at 145, 3	hours by thread failure	(5) Fractured in threads at	93,0 hours	(6) Fracture damaged in re-	moval from furnace	a Deformation exceeded dur-	ing loading	b Deformation less than indi-	cated value when rupture
		nations		1,0		0	26		0	21.5		8	17	
ed Data		otal Deform	0 psi)	0,5		(64)	24	l	j t	20	•	0	16	
Interpolated Data		· Specified T	Time (in 100	2.0		0	22		8	18	)	0	(15)	•
		Strength for Rupture or Specified Total Deformations	in Specified Time (in 1000 psi)	1 0		9	16		8	15	) 	0	(14)	
		Strength f	)	Rupture		29	29		61	24		(52)	18	
			Test	$Temp(^{\bullet}F)$		1350	1640		1350	1640		1350	1640	
			Time	(hr)		30			100			300		

occurred or the test was discontinued Extrapolated

TABLE V

## CREEP-RUPTURE DATA FOR INCO 713 ALLOY

Original Data

ied 1 % 1.0		0 0	8		S B	ŧ			
Time to Reach Specified Total Deformation in % (hours)		2,3	403,0		4,9	42,0			
Time to Total I		а О 3	10,0		ત્ય	0,3		nations	1°0
0.		<i>თ</i> თ	ઇ ત		ಡ	ಡ	ata	al Deforn psi)	0,5
Deformation on Loading (%)	1550°F	0,310	0, 140	1700°E	0,240	0, 150	Interpolated Data	or Rupture or Specified Total Lin Specified Time (in 1000 psi)	0,2
Elongation Reduction of (%) Area (%)			·		1	1	•	Strength for Rupture or Specified Total Deformations in Specified Time (in 1000 psi)	0,1
Elongation (%)		11	) [		4	4		Strength	Rupture
Rupture Time (hours)		7, 6	491,9		13, 6	49.2		Test	Temp (•F)
Stress (psi)		50,000	25,000		27,000	20,000		$\operatorname{Time}$	(hr)

a Deformation exceeded during loading.

8 1

0 I

| | | |

32

1550 1700

100

8 8

26

1 I

0 I

27

1550 1700

300

1 1

36 21

(23)

8 I I I

39 23

1550 1700

30

## CREEP-RUPTURE DATA FOR STELLITE 31 ALLOY

### Original Data

	I°0		8 0	2,3	3,4	11,5	q	q	Ф
Time to Reach Specified Total Deformation in % (hours)	0,2 0,5		0.03 0.2	0, 1 0, 6	0, 1 0, 6	0,3 2,1	13, 0 b	21,5 2100,0	64,0 b
Tin T	0,1		ಡ	ď	0.05	0, 05	3,0	7, 0	0.6
Deformation on Loading	(%)	1650°F	0, 122	0, 108	0,089	0.071	0,042	0,034	0.028
Reduction of	Area (%)		19	21	17	9	ntinued	discontinued	discontinued
Elongation Reduc	(%)		10	10	13	2	disco	disco	disco
Stress Rupture Time	(hours)		20.8	70.2	124,4	386, 2	1445,0	2448,0	601,5
Stress	(psi)		20,000	18,000	16,000	13,500	8,000	6,500	5,500

## Interpolated Data

		Strength f	Strength for Rupture or Specified Total Deformations	Specified Tot	tal Deform	ations
Time	Test	)	in Specified Time (in 1000 psi)	me (in 1000	psi)	
(hr)	Temp (*F)	Rupture	0.1	0° 2	0.5	1.0
30	1650	20	(2°0)	9	11	(13)
100	1650	16	8 0	5,3	9.2	(11)
300	1650	14	8	(4,4)	8	ŧ ŧ

- a Deformation exceeded during loading
- Deformation less than indicated value when rupture occurred or the test was discontinued م







