

# A Processable Mullite Precursor Prepared by Reacting Silica and Aluminum Hydroxide with Triethanolamine in Ethylene Glycol: Structural Evolution on Pyrolysis

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**A simple, processable precursor to mullite can be synthesized in quantities of 100 g in a few hours by direct reaction of silica, aluminum hydroxide, and triethanolamine in ethylene glycol. To delineate a processing window whereby precursor shapes can be transformed into mullite, the chemical and phase microstructural evolution of this precursor on pyrolysis to selected temperatures in air is followed by thermal gravimetric analysis, differential thermal analysis, diffuse reflectance infrared Fourier transform spectroscopy, solid-state  $^{27}\text{Al}$  and  $^{29}\text{Si}$  nuclear magnetic resonance, X-ray diffractometry, and Brunauer–Emmett–Teller analytical methods. The precursor behaves as a single-phase, atomically mixed material that initially transforms to a porous, amorphous aluminosilicate when heated to temperatures as high as 950°C. Above 950°C, the precursor first transforms to tetragonal mullite, based on comparison with the literature, and, on continued heating above 1200°C, to orthorhombic mullite with coincident loss of porosity.**

## 1. Introduction

MULLITE ( $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$ ) is an important and widely studied ceramic material. It is used in a diverse number of applications, including microelectronic packaging, high-temperature protective coatings, microwave dielectrics, and infrared-transmitting materials.<sup>1–10</sup> Mullite substrates are expected to be used increasingly instead of alumina ( $\text{Al}_2\text{O}_3$ ) substrates, which are currently used in high-performance midrange computers, because of a lower thermal-expansion mismatch in the mullite–alumina–silicon system compared to alumina.<sup>3,11</sup> In addition, mullite is considered to be a candidate material for high-temperature structural applications, because of its low coefficient of thermal expansion ( $5 \times 10^{-6}/^\circ\text{C}$ ), good thermal-shock fracture resistance, low true density, high creep resistance, good chemical and thermal stability, high melting point ( $1828^\circ \pm 10^\circ\text{C}$ ), and excellent toughness and strength.<sup>9–18</sup> Thus, mullite is an excellent target ceramic material to dem-

onstrate the utility of our recently discovered synthetic route to aluminosilicate precursors directly from silica ( $\text{SiO}_2$ ) and aluminum hydroxide ( $\text{Al}(\text{OH})_3$ ) (*vide infra*).<sup>19–22</sup>

Mullite can be synthesized by either traditional powder or chemical processing methods. In the traditional approach, mullite is formed from naturally occurring aluminosilicate minerals, e.g., sillimanite.<sup>9</sup> In minerals processing, relatively coarse (micrometer-range) powders are used; thus, high temperatures ( $>1700^\circ\text{C}$ ) are required to complete the solid-state reactions. Despite the high temperatures, segregated  $\text{Al}_2\text{O}_3$ - and  $\text{SiO}_2$ -rich regions often remain. Mineral sources often contain high impurity levels (iron oxide, titanium oxide, etc.) that are unacceptable in advanced ceramic applications. Contamination problems can be avoided by sintering mixtures of high-purity oxides or hydroxides of aluminum and silicon. Unfortunately, high sintering temperatures ( $>1650^\circ\text{C}$ ) are still required and mixing is still at the micrometer level.<sup>1,9</sup> Thus, homogeneous microstructures remain difficult to obtain.

In contrast, in chemical processing, the intimacy of mixing dictates the mullitization temperature and phase purity in the final product.<sup>23–28</sup> If the homogeneity is at the atomic level (“single phase”), as in chemical vapor deposition (CVD), spray pyrolysis, and polymer precursor methods, direct mullitization can be observed, as evidenced by an exothermic reaction at  $\sim 980^\circ\text{C}$ . In contrast, if the homogeneity is in the nanometer-to-micrometer range (“colloidal” or “diphasic”), mullitization occurs through a transient  $\text{Al}_2\text{O}_3$  phase and is often evidenced by a second exotherm at a temperature  $>1200^\circ\text{C}$ .<sup>24,28</sup> For example, powders or gels that are formed by the fast hydrolysis of various alkoxides or salts are generally diphasic.<sup>4,10,23,24</sup> Despite the lower processing temperatures (compared to traditional approaches), segregation during microstructural development is often observed.

The exact mechanism(s) of microstructural evolution (nucleation and growth) in the formation of mullite remain(s) a subject of much debate. In the traditional approach, mullite forms via an interfacial reaction between  $\text{Al}_2\text{O}_3$  and  $\text{SiO}_2$ .<sup>1,9</sup> Growth seems to occur by diffusion of aluminum and silicon through the mullite layer. In nanoscale diphasic systems, mullite nucleation is suggested to occur within the siliceous amorphous matrix rather than at the interface.<sup>8,9</sup> Alternately, mullite nucleation is reported to occur via an aluminum–silicon spinel phase, which is responsible for the exotherm that is observed at temperatures  $>1200^\circ\text{C}$ . Mullite formation by direct solid-state reaction between  $\text{Al}_2\text{O}_3$  and amorphous  $\text{SiO}_2$  also is proposed to occur.<sup>9</sup>

Despite the controversy about the mechanism(s) of nucleation and microstructural development, there continues to be a need for low-cost, high-purity processable precursors for forming mullite shapes. As briefly noted above, we recently described a low-cost route to easily processed aluminosilicate

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precursors by reaction of  $\text{SiO}_2$ ,  $\text{Al}(\text{OH})_3$ , and Group I/II metal hydroxides/oxides in ethylene glycol (EG) and triethanolamine (TEA) using the “oxide one-pot synthesis” (OOPS) process.<sup>9–22</sup> The purpose of the current work is to provide details about the chemical and microstructural evolution that occurs during the pyrolytic transformation of the OOPS mullite precursor to amorphous and then ceramic (mullite) materials and to delineate an appropriate processing window. The processing protocols that are developed in the following study will help to identify pyrolysis processing parameters for ceramic shapes such as coatings, fibers, and thin films. We will report on the use of this precursor to process mullite/carbon coatings on graphite fibers in the near future.<sup>21</sup>

## II. Experimental Procedure

### (I) General Procedures

All chemicals were purchased from standard vendors and used as-received, except for EG, which was distilled under nitrogen gas before use. EG that was recovered from the reaction also was redistilled and recycled. Care was taken not to expose the reaction contents to the atmosphere, because of their slight moisture sensitivity.

(A) *Synthesis of Mullite ( $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$ ) Precursor:* The mullite precursor was synthesized directly from aluminum hydroxide hydrate ( $\text{Al}(\text{OH})_3 \cdot x\text{H}_2\text{O}$ ) (53.5 wt%  $\text{Al}_2\text{O}_3$ , Aldrich Chemical Co., Milwaukee, WI),  $\text{SiO}_2$  (99.99 wt%, Cabot Corp., Tuscola, IL), analytical-grade EG (Mallinckrodt, Paris, KY), and TEA (Aldrich).  $\text{Al}(\text{OH})_3 \cdot x\text{H}_2\text{O}$  (55.0 wt%  $\text{Al}_2\text{O}_3$ ) (338 mmol, 31.4 g), TEA (338 mmol, 50.4 g), and excess EG (300–350 mL) were placed in a schlenk flask (capacity of 500 mL). The mixture was heated to the EG distillation temperature (200°C), under nitrogen gas, with constant mechanical stirring. The  $\text{Al}(\text{OH})_3 \cdot x\text{H}_2\text{O}$  dissolved readily with continuous, slow distillative removal of both EG and byproduct  $\text{H}_2\text{O}$ . The reaction became clear in 1–2 h, after 250–300 mL of EG and water distilled off, and a clear yellow solution was obtained. The reaction was allowed to cool to room temperature.

$\text{SiO}_2$  (113 mmol, 6.8 g), TEA (113 mmol, 16.8 g), and excess EG (200 mL) were added to the above-mentioned reaction. The mixture was heated again under nitrogen gas with constant mechanical stirring. As distillation continued,  $\text{SiO}_2$  dissolved and, in 1–2 h, the reaction became clear. The reaction was stopped after 200–250 mL of EG and byproduct water had distilled off, which gave a clear, yellow viscous solution. The solvent was removed by vacuum evaporation, and the resulting residue was vacuum dried ( $10^{-2}$  torr ( $\sim 1.3$  Pa)) at  $\sim 210^\circ\text{C}$  for 2–3 h to remove residual EG and TEA, if any. On cooling, a hard, brittle polymer was obtained that was freeze fractured (alternate cooling in liquid nitrogen and warming in hot water), transferred to a container, and stored in the inert atmosphere of a glove box (Model MO40-2-Dri-Lab, Vacuum Atmospheres, Co., Hawthorne, CA). The yield of the reaction was  $\sim 100$  g (ceramic yield was  $\sim 23$  wt%).

(B) *Pyrolyses:* All pyrolyses were performed in a single-zone tube furnace (Model 58114, Lindberg, Watertown, WI) (heating zone was  $\sim 7$ – $8$  cm long) that was equipped with a programmable temperature controller (Model 818P, Eurotherm, Northring, U.K.). Samples (1–2 g) were placed in an  $\text{Al}_2\text{O}_3$  boat (3 cm  $\times$  6 cm) and then inside a quartz tube. The quartz tube was sealed with a ported cap, inserted into the furnace, and connected to a supply of synthetic air. The samples were heated at  $10^\circ\text{C}/\text{min}$  in air to selected temperatures, followed by a hold of 1 h, and then furnace cooled. The samples were taken out of the quartz tube and stored in air for further characterization.

### (2) Materials Characterization

The phase transformations and chemical changes that occur during pyrolysis of the mullite precursor were characterized. The characterization methods included thermal gravimetric

analysis (TGA) and differential thermal analysis (DTA), powder X-ray diffractometry (XRD), solid-state magic-angle spinning  $^{29}\text{Si}$  and  $^{27}\text{Al}$  nuclear magnetic resonance ( $^{29}\text{Si}$  and  $^{27}\text{Al}$  MAS NMR), diffuse reflectance infrared Fourier transform (DRIFT) spectroscopy, scanning electron microscopy (SEM), and surface area and porosimetry, as described below.

(A) *Thermal Analyses:* Thermal analyses were performed (Model Thermal Analyst 2200 with a Hi-Res TGA 2950 Thermogravimetric Analyzer and a DSC 2910 Differential Scanning Calorimeter equipped with a 1600 DTA Differential Thermal Analyzer module; TA Instruments, New Castle, DE). TGA samples that averaged 15–20 mg were loaded in a platinum pan and heated in flowing (60  $\text{cm}^3/\text{min}$ ) synthetic air to  $1000^\circ\text{C}$ , in “Hi-Res 4.0 mode,” at  $50^\circ\text{C}/\text{min}$ . Hi-Res mode decreases the heating rate as the rate of mass loss increases, which provides sharply defined thermal events (decomposition, oxidation, etc.). DTA measurements (sample sizes of  $\sim 30$  mg) were performed using platinum crucibles, under a continuous flow (50  $\text{cm}^3/\text{min}$ ) of synthetic air, at  $10^\circ\text{C}/\text{min}$  to  $1400^\circ\text{C}$ . Calcined  $\text{Al}_2\text{O}_3$  (Alcoa, Pittsburgh, PA) was used as a reference.

(B) *XRD:* XRD powder patterns were determined using a rotating anode goniometer (Rigaku Denki Co., Tokyo, Japan). Powder samples (100–200 mg) were ground with an  $\text{Al}_2\text{O}_3$  mortar and pestle, packed in a glass specimen holder, and placed in the goniometer. Scans were continuous from  $5^\circ$ – $80^\circ 2\theta$  at a scan rate of  $10^\circ 2\theta/\text{min}$  using  $0.05^\circ 2\theta$  increments and  $\text{CuK}\alpha$  ( $\lambda = 1.542 \text{ \AA}$ ) radiation. Scans that were used for comparing the integrated peak intensities for the (120) and (210) peaks were taken at  $1^\circ 2\theta/\text{min}$ , using  $0.05^\circ 2\theta$  increments. The peak positions and the relative intensities of the powder patterns were identified by comparison with ICDD<sup>†</sup> File Card No. 15-776 to identify the crystalline phases.

(C) *Solid-State MAS NMR:*  $^{29}\text{Si}$  and  $^{27}\text{Al}$  solid-state MAS NMR spectra were obtained for pyrolyzed samples to study changes in chemical structure during pyrolytic transformation.<sup>‡</sup> The  $^{29}\text{Si}$  MAS NMR spectra were recorded on either of two spectrometers (Models MSL 300 (59.6 MHz) or MSL400 (79.5 MHz), Bruker) using 7 mm MAS probes (Bruker, Wissembourg, France). The spinning rate was 5 kHz. The repetition time was 60 s with a pulse width of 2  $\mu\text{s}$  ( $30^\circ$  pulse angle). The  $^{27}\text{Al}$  MAS NMR spectra were recorded on the MSL300 spectrometer (78.2 MHz) with a 4 mm MAS probe (Bruker). The spinning rate was 15 kHz. A pulse width of 1  $\mu\text{s}$  was used, with repetition times of 1 s. The  $^{29}\text{Si}$  MAS NMR spectra were simulated using the WIN-FIT software program.

(D) *DRIFT Spectroscopy:* DRIFT spectra were recorded on a Galaxy Series FTIR 3000 spectrometer (Mattson Instruments, Madison, WI). Optical-grade random cuttings of potassium bromide (KBr) (International Crystal Laboratories, Garfield, NJ) were ground using an  $\text{Al}_2\text{O}_3$  mortar and pestle. DRIFT spectroscopy samples were prepared by mixing 0.3–1.0 wt% of the sample to be analyzed with KBr.

Sample concentrations were maintained at  $<1$  wt% to ensure adherence to Beer’s law. The dilute samples that were prepared in KBr were subsequently packed firmly in the sample holder, leveled off at the upper edge to provide a smooth surface, and transferred to the Fourier transform infrared spectroscopy (FTIR) sample chamber, which was constantly flushed with nitrogen gas. A minimum of 64 scans were collected for each sample, at a resolution of  $\pm 4 \text{ cm}^{-1}$ . Peak positions were identified using a standard peak-searching program.

(E) *Surface Area and Porosimetry:* Surface area and micropore analyses were conducted at a temperature of 77 K using porosimetry equipment (Model ASAP2000, Micromeritics Instruments Group, Norcross, GA) with nitrogen as the

<sup>†</sup>International Centre for Diffraction Data, Newtowne Square, PA (formerly Joint Committee on Powder Diffraction Standards (JCPDS), Swarthmore, PA).

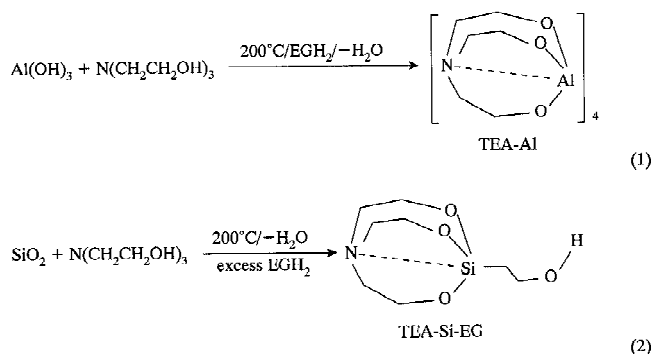
<sup>‡</sup>The solid-state NMR experiments were performed by Florence Babonneau and Jocelyne Maquet, Laboratoire Chimie de la Matière Condensée, Université de Pierre et Marie Curie, Paris, France.

adsorbate gas. Prior to analysis, samples were degassed at  $400^\circ\text{C}$  under high vacuum for 12 h. Surface areas were calculated using the five-point method. Micropore analysis was performed using differential function theory (DFT) with slitlike pores in carbon as the model; thus, pore size is reported as a pore width.

(F) SEM: The bulk material microstructure was examined by using an SEM microscope (Model S800, Hitachi, Tokyo, Japan) that was operating at 5 keV. SEM samples were prepared by mounting pyrolyzed powder samples on an aluminum stub using a double-stick tape. Powder samples were sputter coated with a layer of Au-Pd for ~45 s at 10 mV, to enhance their conductivity.

### III. Results and Discussion

The mullite precursor can be synthesized directly by reacting stoichiometrically correct mixtures of  $\text{SiO}_2$  and  $\text{Al}(\text{OH})_3$  in EG with TEA.<sup>19,21</sup> Alternately, the precursor can be prepared by reacting the individual components (TEA-Al and TEA-Si-EG, which are synthesized via reactions (1) and (2).



TEA-Al and TEA-Si-EG both have been characterized in detail.<sup>19,22</sup> Mixtures of TEA-Al and TEA-Si-EG, in a 3:1 ratio, that are heated briefly in EG, followed by vacuum drying, provide the mullite precursor. One-hundred-gram quantities can be prepared in 2–3 h using either method. The theoretical ceramic yield is 28.2%, based on this 3:1 mixture.

Following synthesis, precursor samples were first analyzed by TGA and DTA to delineate the decomposition patterns, which, in turn, were used to define an approximate processing window for complete pyrolytic transformation of the precursor to ceramic material. Samples that were heated to selected temperatures were characterized by XRD, DRIFT spectroscopy,  $^{29}\text{Si}$  and  $^{27}\text{Al}$  MAS NMR, and SEM. The XRD data permit identification of crystalline phases and microstructural development that occur at selected temperatures. MAS NMR provides complementary information on the evolution of the local environments of silicon and aluminum sites in the pyrolyzed samples. DRIFT spectroscopy can be used to derive information about the phase or atomic transformations, crystallinity, impurities, and surface characteristics such as the presence of adsorbed  $\text{H}_2\text{O}$ , etc. Complete characterization of the precursor as it is pyrolyzed in air to selected temperatures is presented below.

#### (1) Thermal Analyses

The TGA (Fig. 1) of the mullite precursor shows a  $1000^\circ\text{C}$  ceramic yield of 23.0%, which is ~5% below theory (28.2%). The lower experimental ceramic yield is likely caused by some retained, unreacted TEA and free EG in the sample. Even though the sample has been vacuum dried ( $210^\circ\text{C}$  for 2–3 h), the highly viscous precursor resists solvent removal as it hardens to a hard plastic on drying. Some support for solvent retention is found in the TGA, which shows three major losses.

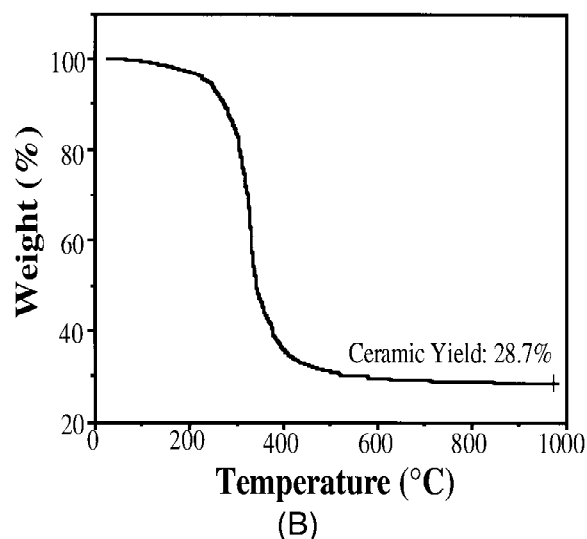
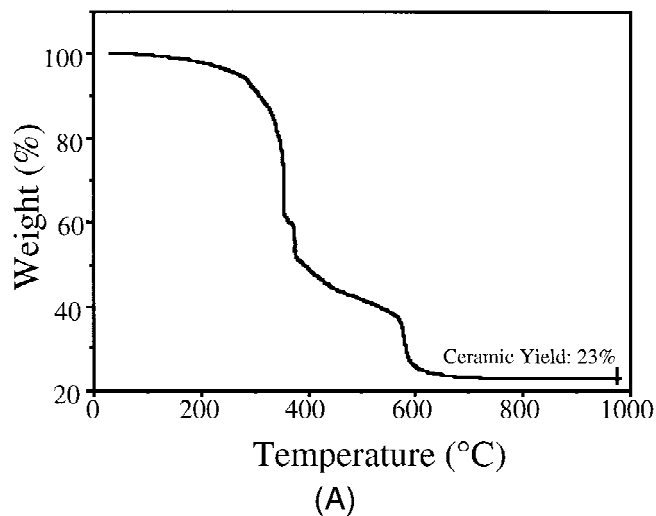


Fig. 1. TGA of mullite precursor heated in “Hi-Res 4.0 mode” at  $50^\circ\text{C}/\text{min}$  (A) in air and (B) in nitrogen to  $1000^\circ\text{C}$ .

The first mass loss (~6%) in the temperature range of  $80^\circ\text{C}$ – $260^\circ\text{C}$  likely results from volatilization of unreacted TEA and free EG. The second mass loss (~42%) occurs in the temperature range of  $270^\circ\text{C}$ – $370^\circ\text{C}$  and is due to the decomposition of organic ligands (TEA and EG). If samples of the precursor are heated in nitrogen (Fig. 1(B)), the second mass loss is greater and the third mass loss (~30%) is not observed.<sup>21</sup> Thus, we attribute the mass loss at  $375^\circ\text{C}$ – $610^\circ\text{C}$  to oxidation of char that is formed by oxidative crosslinking of the organic ligands, which cannot occur in nitrogen.

As the organic ligands decompose, a number of reactions probably occur simultaneously. The specific decomposition reactions have not, as yet, been delineated.<sup>20</sup> The precursor consists primarily of ethyleneoxy ligands ( $\text{Si}-\text{OCH}_2\text{CH}_2\text{O}-$  and  $\text{M}-\text{OCH}_2\text{CH}_2-\text{N}=\text{}$ ) that can thermally fragment to form volatile compounds ( $\text{CH}_2=\text{CH}_2$ ,  $\text{CH}_2\text{O}$ ,  $\text{CH}\equiv\text{CH}$ ,  $\text{CH}_4$ , etc.) and hydroxylated species (e.g.,  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$ ).<sup>20</sup>

When samples are heated in air, the decomposition process seems to be complicated by oxidative crosslinking.<sup>20</sup> In oxidative crosslinking, carbon radicals are generated and subsequently couple to form extended polymeric materials (char). On continued heating, the char can undergo oxidative chain fragmentation, which leads to more volatiles or more crosslinked char. The retained C–H bonds in the char generate  $\text{H}_2\text{O}$  (and CO or  $\text{CO}_2$ ) on oxidation, which generates fresh



M–OH groups. Because the char becomes more highly crosslinked with time and at higher temperature, it is hard to oxidize; hence, an extended temperature range (375°–610°C) is required for its removal.

The DTA (Fig. 2) shows a sharp exotherm at 380°C at the end of the major mass loss event (200°–360°C) that is observed in the TGA. The TGA shows that, by 370°C, ~50% of the mass is lost; thus, it is unlikely that this exotherm results from organic ligand decomposition. One possible explanation is that as the organic ligands decompose to volatiles, inorganic species—e.g.,  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$ —form concurrently (see “DRIFT Spectroscopy,” Section III (5)). These species must react at some point to form the Si–O–Al network (by loss of  $\text{H}_2\text{O}$ ). We suggest that network formation is most likely responsible for the observed exotherm at 380°C.<sup>20</sup> The  $^{29}\text{Si}$  NMR analysis (*vide infra*) indicates the presence of ~80% Si–O–Al sites at the lowest temperature (400°C) that is examined. Thus, an Si–O–Al network does form at these temperatures, which provides some support for this interpretation of the exotherm at 380°C. The exotherm at 610°C coincides with the third mass loss and likely corresponds to char oxidation. This exotherm is not observed when the precursor is pyrolyzed in nitrogen.<sup>21</sup>

The sharp exotherm that is observed at 990°C is typical of crystallization to mullite and/or a  $\gamma\text{-Al}_2\text{O}_3$ -type spinel phase.<sup>1,4,5,9,10,14,15,17,23–28</sup> No other exotherm is observed at higher temperatures (>1200°C), as expected for single-phase precursors that are reported to exhibit only one exotherm at temperatures of 980°–990°C, because of spinel or mullite (or tetragonal mullite) phase formation.<sup>9,23–25</sup> A more detailed discussion of the meaning of this exotherm is presented below.

## (2) XRD

The powder patterns for mullite samples that have been pyrolyzed in air to selected temperatures are shown in Fig. 3. The samples at temperatures <1000°C are X-ray amorphous, as indicated by the broad maxima. The 1000°C XRD pattern indicates crystallization, as evidenced by sharp peaks at  $2\theta$  values of  $\sim 16^\circ$ ,  $26^\circ$ ,  $35^\circ$ ,  $41^\circ$ , etc. These peaks correspond to ICDD File Card No. 15-776 for stoichiometric mullite,  $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$ .<sup>4</sup> These data appear to be in contrast with the 1000°C  $^{29}\text{Si}$  NMR spectrum (*vide infra*), which suggests phase separation, although it correlates with the DTA exotherm at 990°C, commonly thought to result because of mullitization. Also, there is no evidence of the spinel phase that was observed by Gerardin *et al.*<sup>24</sup> and Chakravorty<sup>28</sup> in single-phase precursors synthesized by the sol–gel method. This suggests that the OOPS-precursor-derived material has a higher degree of homogeneity because of better atomic mixing. At 1200° and

1300°C, the peaks sharpen (peak width decreases and intensity increases) because of grain growth. Thomson *et al.*<sup>25</sup> used the splitting of the peak at  $26.15^\circ 2\theta$  in tetragonal mullite (on heating at temperatures >1000°C) as a means of estimating the extent of conversion of the tetragonal phase to the orthorhombic phase. Thus, the ratio of the integrated peak intensities for the (120) and (210) peaks (at  $25.97^\circ$  and  $26.26^\circ 2\theta$ , respectively) for ICDD File Card No. 15-776 (mullite crystallized at 1700°C) is used as the reference. On this basis, the samples that are produced at selected temperatures are as follows: only tetragonal at 1000°C (no splitting), 80% orthorhombic at 1200°C, and 87% orthorhombic at 1400°C. These results are qualitatively in agreement with the  $^{29}\text{Si}$  NMR results that are presented below.

## (3) $^{29}\text{Si}$ MAS NMR

Figure 4 displays  $^{29}\text{Si}$  MAS NMR spectra for samples that have been pyrolyzed to selected temperatures. The 400°C sample spectrum is characterized by a primary, broad signal with a maximum at  $-85$  ppm (~80% of the total silicon sites by integration), and a smaller component at  $-106$  ppm (~20%). The  $-85$  ppm chemical shift is typical for  $\text{Si}(\text{OAl})_4$  units<sup>24,27,29–32</sup> and shows that these units dominate early evolutionary stages. However, the asymmetry of the peak strongly suggests that mixed units ( $\text{Si}(\text{OAl})_{4-x}(\text{OSi})_x$  ( $x \neq 0$ )) also are present, and, indeed, the  $-106$  ppm signal results from  $\text{Si}(\text{OSi})_4$  sites. The spectra do not change much up to 950°C, although a shift in the main peak position (Fig. 5) toward  $-90$  ppm suggests an enrichment of the aluminosilicate phase in  $\text{SiO}_2$ , which is coincident with the broadening of the signals at 950°C.

The main change occurs at 1000°C: the spectrum clearly shows two broad signals, centered at  $-88$  ppm (~53%) and  $-108$  ppm (~47%), which strongly suggests the presence of an  $\text{Al}_2\text{O}_3$ -rich, aluminosilicate phase and a  $\text{SiO}_2$ -rich region. The amount of  $\text{SiO}_2$ -rich sites increases drastically in the temperature range of 950°–1000°C, from 20% to 50% (Fig. 6), in the temperature range that corresponds to the onset of crystallization (by DTA and XRD). However, despite the crystallization of mullitelike peaks in the XRD pattern, the  $^{29}\text{Si}$  NMR signals are broad, even the  $-88$  ppm peak, which is typical of silicon sites in mullite. This suggests that a large distribution of sites (some disorder) exists in the material. Note that XRD detects only the long-range order in a sample, whereas NMR detects the local environment around the silicon atom. Similar results were reported by Gerardin *et al.*,<sup>24</sup> as discussed below.

In contrast to the 1000°C samples, the 1300°C sample shows a  $^{29}\text{Si}$  MAS NMR spectrum that is typical of crystalline mullite, with three sharp peaks at  $-87.2$ ,  $-90.5$  and  $-94.7$  ppm and

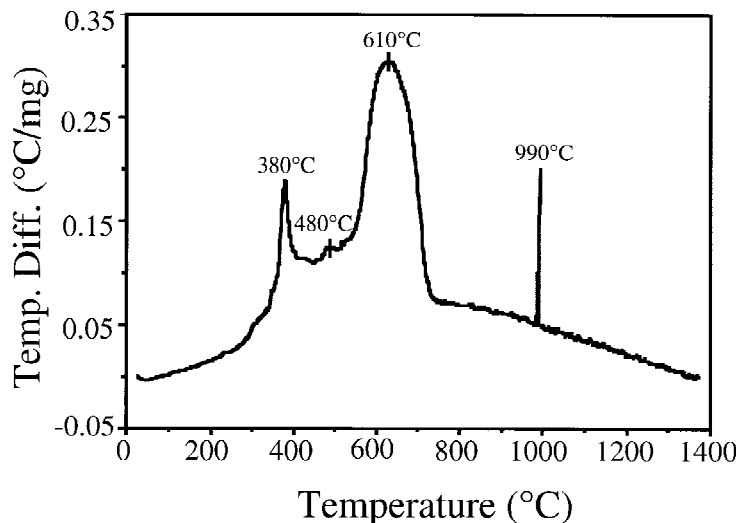


Fig. 2. DTA of mullite precursor heated at 10°C/min in air to 1400°C.

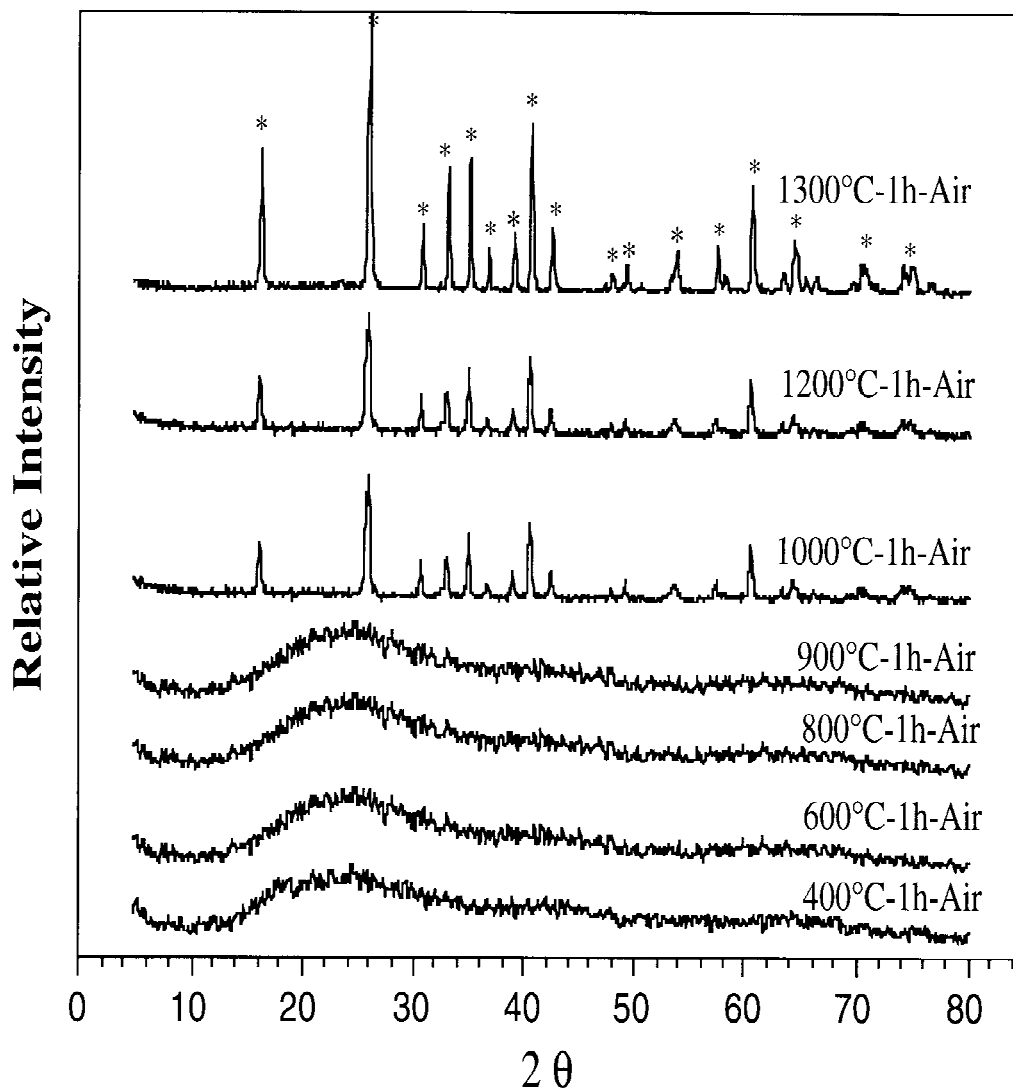


Fig. 3. XRD patterns of mullite precursor pyrolyzed to selected temperatures ((\*))  $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$ , ICDD File Card No. 15-776).

a shoulder at  $-81$  ppm, in good agreement with previous NMR studies of mullite.<sup>24,27,29–32</sup> A signal at approximately  $-108$  ppm is still present, albeit with a much-lower intensity ( $\sim 18\%$ , Fig. 6), which shows that the  $\text{SiO}_2$ -rich and  $\text{Al}_2\text{O}_3$ -rich sites that are formed at  $1000^\circ\text{C}$  recombine to approach a stoichiometric mullite. As the precursor is heated to  $1400^\circ\text{C}$ , the peaks become more defined, because of further crystallization, and no  $\text{SiO}_2$  sites are detected.

Gerardin *et al.*<sup>24</sup> and Jaymes *et al.*<sup>27</sup> reported the formation of  $\text{Al}_2\text{O}_3$ - and  $\text{SiO}_2$ -rich regions (by  $^{29}\text{Si}$  NMR) for pyrolyzed single-phase precursors that exhibit single exotherms at  $\sim 980^\circ\text{C}$  and mullite diffraction patterns in the XRD (at  $1000^\circ\text{C}$ ). Samples in the work of Gerardin *et al.*<sup>24</sup> were first heated at a rate of  $1^\circ\text{C}/\text{min}$  to  $800^\circ\text{C}$  and then heated at a rate of  $3^\circ\text{C}/\text{min}$  to selected temperatures ( $950^\circ$  and  $1000^\circ\text{C}$ ), followed by holds of 6 h, and further annealed for 24 h. The  $950^\circ$  and  $1000^\circ\text{C}$  spectra show two main features: a broad peak, centered at  $-110$  ppm, which is attributed to  $\text{SiO}_2$ -rich sites, and a narrow peak at  $-86$  ppm, with shoulders at  $-90$  and  $-93$  ppm, which corresponds to mullite. Integration indicates that 35% of the  $\text{SiO}_2$  sites are  $\text{Si}(\text{OSi})_4$ , as opposed to 47% in the OOPS-derived mullite sample. Although the Jaymes *et al.*<sup>27</sup> work also shows the same segregation at  $1000^\circ\text{C}$ , no integration was reported.

Gerardin *et al.*<sup>24</sup> observed shoulders ( $-90$  and  $-93$  ppm) for mullite at  $1000^\circ\text{C}$  that were not observed in our samples or by Jaymes *et al.*<sup>27</sup> This difference in the  $\text{SiO}_2$ -rich sites (35% vs.

47%) and the absence of shoulders (due to mullite) in the OOPS-derived mullite precursor sample that was heated to  $1000^\circ\text{C}$  may be due to different heating schedules. OOPS-derived mullite samples were heated at  $10^\circ\text{C}/\text{min}$  to selected temperatures (e.g.,  $1000^\circ\text{C}$ ), followed by a hold of only 1 h.<sup>20</sup> The slower heating rates and longer holds (6 h) for the Gerardin *et al.*<sup>24</sup> samples allow diffusion and may result in a more-homogeneous and better-crystallized material, in comparison to the OOPS-derived  $1000^\circ\text{C}$  mullite sample.

#### (4) $^{27}\text{Al}$ MAS NMR

The  $^{27}\text{Al}$  MAS NMR spectra (Fig. 7) were recorded on pyrolyzed samples, with special attention to the  $900^\circ$ – $1000^\circ\text{C}$  temperature range, where crystallization occurs. The spectrum of the  $400^\circ\text{C}$  sample shows three main components with maxima at 1, 30, and 55 ppm, which are assigned to hexacoordinated, pentacoordinated, and tetraordinated aluminum sites, respectively.<sup>24,29–32</sup> As samples are heated progressively to  $950^\circ\text{C}$ , the spectra show a relative increase in the peak intensity due to pentacoordinated aluminum sites. At  $1000^\circ\text{C}$ , the intensity of the pentacoordinated aluminum peak decreases suddenly. At  $1300^\circ$  and  $1400^\circ\text{C}$ , the spectra are characteristic of a crystalline mullite phase with one hexacoordinated and two tetraordinated sites.

Gerardin *et al.*<sup>24</sup> discussed the importance of pentacoordinated aluminum sites in the amorphous phase in obtaining low-temperature mullitization. Their homogeneously mixed

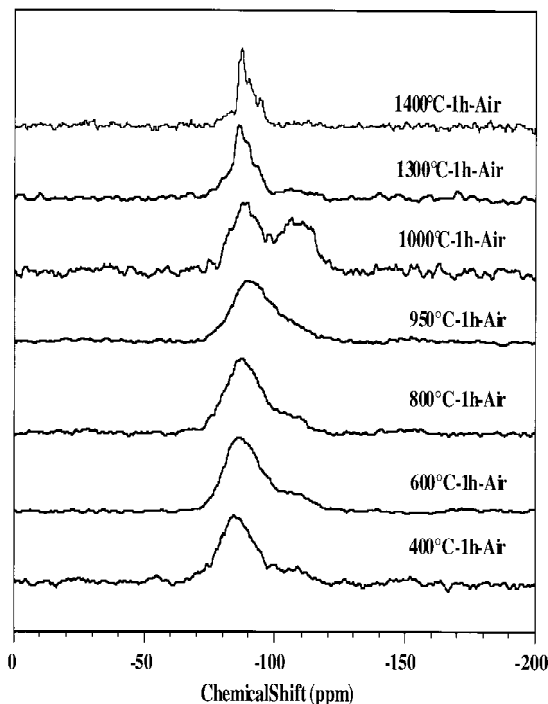


Fig. 4.  $^{29}\text{Si}$  MAS NMR spectra of mullite precursor pyrolyzed to selected temperatures.

precursor also contained considerable amounts of pentacoordinated sites in the sample that was pyrolyzed to 900°C, which disappeared at 1000°C, as is observed in our samples. However, phase-pure mullite was not obtained at 1000°C (by XRD) and a small amount of a spinel phase was obtained. The pentacoordinated sites were absent in a less-homogeneously mixed precursor that showed an exotherm at 1295°C and a crystalline pattern for mullite only at 1300°C. Pentacoordinated sites seem to be critical to the direct mullitization of amorphous mullite composition materials (*vide infra*).

#### (5) DRIFT Spectroscopy

The DRIFT patterns for the mullite precursor that has been pyrolyzed in air to selected temperatures are displayed in Fig. 8. The various vibrational modes that are present in the pyrolyzed materials at selected temperatures are discussed below.

The broad envelope of  $\nu(\text{Si-O})$  and  $\nu(\text{Al-O})$  bands at  $<1100\text{ cm}^{-1}$ , which are observed at temperatures  $\leq 1000^\circ\text{C}$ , indicate a diverse number of atomic environments, typical of amorphous materials. The peaks grow sharper at temperatures  $\geq 1200^\circ\text{C}$  as the samples become more crystalline. The long-range order (periodicity) that is found in crystalline materials limits the number of vibrational modes that are available to each bonding pair in a particular atomic environment, which results in sharp infrared (IR) bands.

The spectra at 400°, 600°, and 800°C show  $\nu(\text{O-H})$  bands at  $3200\text{ cm}^{-1}$ , with an overtone at  $1600\text{ cm}^{-1}$ . These bands probably do not result from adsorbed  $\text{H}_2\text{O}$  species from the atmosphere; otherwise, they also would appear in the higher-temperature pyrolyzed samples. Their presence indicates the formation of  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$  species during pyrolysis. These bands weaken at 800°C and disappear above this temperature. This observation supports the idea that oxidative removal of char generates  $\text{H}_2\text{O}$ , which, in turn, generates new MOH groups, as suggested above. Further support for *in situ* generation of MOH groups comes from a comparison of the pyrolysis behavior of an OOPS  $\text{K}_2\text{O}\cdot\text{Al}_2\text{O}_3\cdot 2\text{SiO}_2$  precursor with a hydrolytically produced  $\text{K}_2\text{O}\cdot\text{Al}_2\text{O}_3\cdot 2\text{SiO}_2$  gel. Pyrolysis of gel-derived samples does not produce new M-OH groups.<sup>20</sup>

As noted above, oxidative crosslinking leads to char, which is stable to  $\sim 600^\circ\text{C}$ . Thus, at temperatures  $<800^\circ\text{C}$ , oxidation of retained C-H bonds generates  $\text{H}_2\text{O}$ , which, in turn, generates new M-OH groups. Continual generation of  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$  species, as evidenced by the presence of  $\nu(\text{-OH})$  bands, means that Si-O-Al network formation is not complete. Only at higher temperatures ( $\sim 800^\circ\text{C}$ ), where all the char has oxidized, will the remaining  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$  species combine (by condensation reactions) to form a complete and extended Si-O-Al network, as evidenced by the disappearance of  $\nu(\text{-OH})$  bands.

The literature reports standard mullite IR spectra as consisting of the following: tetrahedral ( $\text{SiO}_4$ )  $\nu(\text{Si-O})$  vibrations at  $1130$  and  $1170\text{ cm}^{-1}$ , tetrahedral ( $\text{AlO}_4$ )  $\nu(\text{Al-O})$  vibrations at

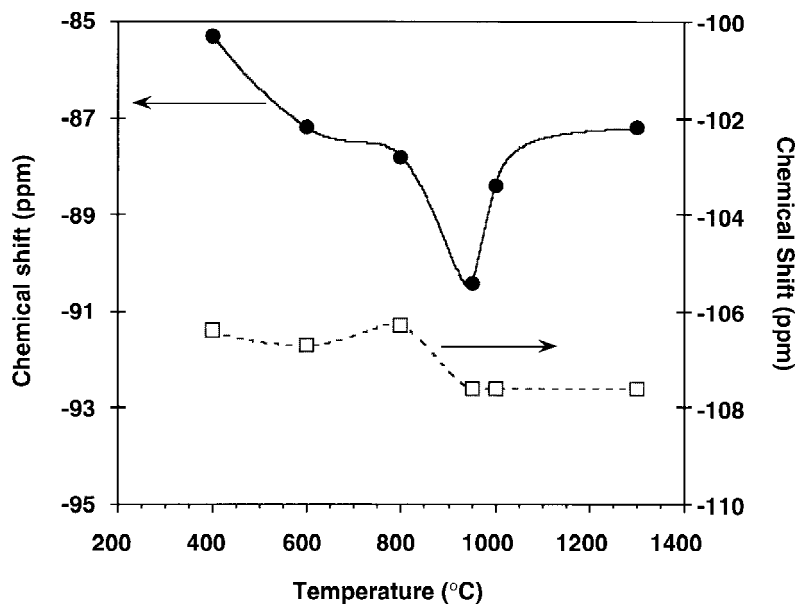
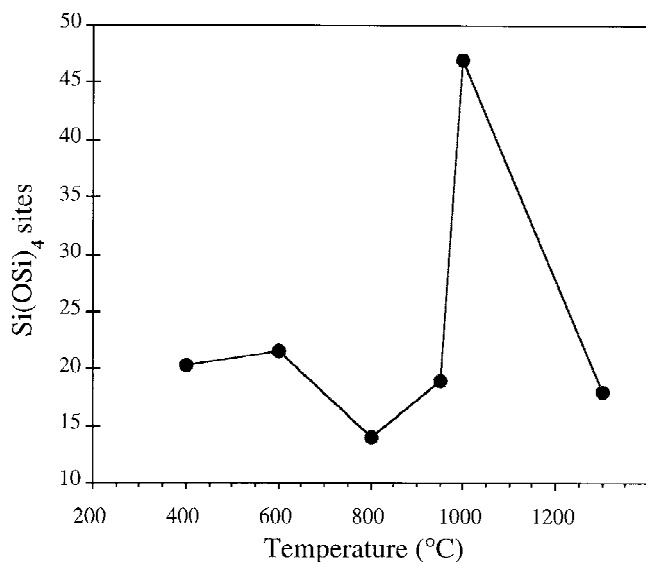


Fig. 5. Plot of chemical shifts versus selected temperatures from  $^{29}\text{Si}$  MAS NMR spectra of pyrolyzed mullite precursor ((●) Si-O-Al peak and (□) Si-O-Si peak).

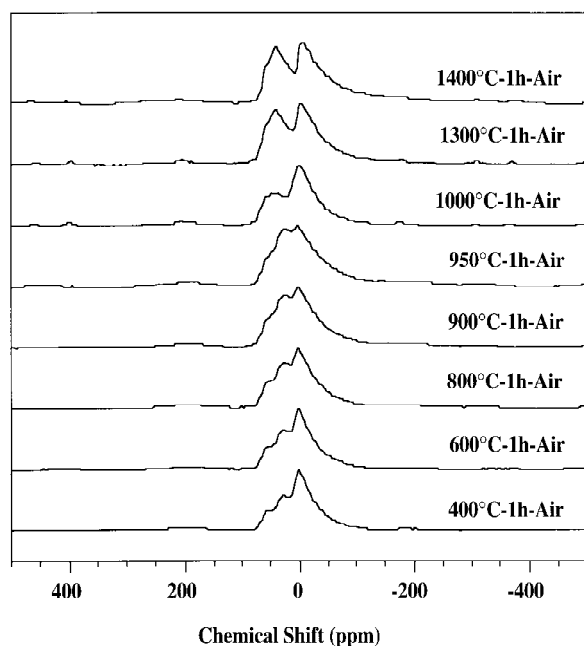


**Fig. 6.** Plot of percentage of  $\text{Si}(\text{OSi})_4$  sites versus selected temperatures from  $^{29}\text{Si}$  MAS NMR spectra of pyrolyzed mullite precursor.

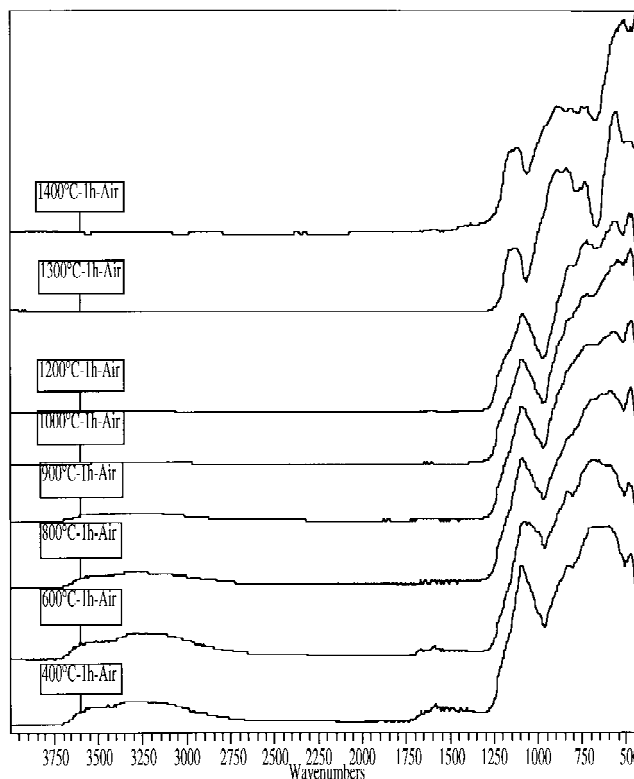
740 and  $830\text{ cm}^{-1}$ , and octahedral ( $\text{AlO}_6$ )  $\nu(\text{Al}-\text{O})$  vibrations in the range of  $530\text{--}680\text{ cm}^{-1}$ .<sup>33–35</sup> The two bands for  $\text{AlO}_4$  and  $\text{AlO}_6$  probably each result from  $\text{Si}-\text{O}-\text{Al}$  and  $\text{Al}-\text{O}-\text{Al}$  bonds. In the following, each type of vibration and the changes that occur at different pyrolysis temperatures are discussed.

(A)  $\text{SiO}_4$   $\nu(\text{Si}-\text{O})$  Vibrations: At temperatures in the range of  $400\text{--}1200^\circ\text{C}$ , the spectra show a band at  $\sim 1100\text{ cm}^{-1}$ , which is attributed to tetrahedral  $\nu(\text{Si}-\text{O})$  vibrations, as found in  $\text{SiO}_2$  (*vide infra*). At  $1300^\circ\text{C}$ , this band shifts and splits to  $1170$  and  $1120\text{ cm}^{-1}$ . These values correspond to the literature values for tetrahedral  $\nu(\text{Si}-\text{O})$  vibrations in mullite.<sup>33–35</sup>

(B)  $\text{AlO}_4$   $\nu(\text{Al}-\text{O})$  Vibrations: No well-defined bands due to  $\text{AlO}_4$   $\nu(\text{Al}-\text{O})$  vibrations are observed at  $400^\circ$ ,  $600^\circ$ , or  $800^\circ\text{C}$ . At these temperatures, a broad envelope, centered at  $700\text{ cm}^{-1}$ , most likely results from mixed bands due to  $\text{AlO}_4$  and  $\text{AlO}_6$   $\nu(\text{Al}-\text{O})$  vibrations. At  $900^\circ\text{C}$ , a weak band at  $720$



**Fig. 7.**  $^{27}\text{Al}$  MAS NMR spectra of mullite precursor pyrolyzed to selected temperatures.



**Fig. 8.** DRIFT spectra of mullite precursor pyrolyzed to selected temperatures.

$\text{cm}^{-1}$  can be attributed to  $\text{AlO}_4$   $\nu(\text{Al}-\text{O})$  vibrations. At  $1000^\circ\text{C}$ , the band shifts to  $740\text{ cm}^{-1}$  (literature value), because a more-ordered structure forms (coincides with the crystallization that is indicated by XRD), and continues to sharpen at  $1200^\circ$  and  $1300^\circ\text{C}$ . Another band at  $830\text{ cm}^{-1}$  (absent at lower temperatures), also typical of  $\text{AlO}_4$   $\nu(\text{Al}-\text{O})$  vibrations that are associated with mullite, is first observed at  $1200^\circ\text{C}$  and sharpens at  $1300^\circ\text{C}$ .

(C)  $\text{AlO}_6$   $\nu(\text{Al}-\text{O})$  Vibrations: In the temperature range of  $400\text{--}800^\circ\text{C}$ , no bands due to  $\text{AlO}_6$   $\nu(\text{Al}-\text{O})$  vibrations can be observed, although, as explained above, they are probably part of the broad envelope centered at  $700\text{ cm}^{-1}$ . At  $900^\circ$  and  $1000^\circ\text{C}$ , a broad (ill-defined) band at  $580\text{ cm}^{-1}$  is attributed to  $\text{AlO}_6$   $\nu(\text{Al}-\text{O})$  vibrations. At  $1200^\circ$  and  $1300^\circ\text{C}$ , the band sharpens with crystallization.

#### (6) Precursor Evolution on Pyrolysis

The above-mentioned results, considered *in toto*, can be used to provide a detailed picture of the processes that occur during pyrolytic conversion of the OOPS precursor to crystalline, orthorhombic mullite. In particular, we are interested in how the amorphous  $\text{Si}-\text{O}-\text{Al}$  network that is formed at  $\sim 400^\circ\text{C}$  transforms. All the analytical studies indicate that the most-dramatic changes occur as the heat-treatment temperature reaches  $\sim 1000^\circ\text{C}$ . Before  $1000^\circ\text{C}$ , only DRIFT spectroscopy and NMR provide any indications of change.

The primary changes in the DRIFT spectra (Fig. 8) are the slow disappearance of  $\nu(\text{O}-\text{H})$  vibrations, as  $\text{M}-\text{OH}$  groups slowly condense, eliminate  $\text{H}_2\text{O}$ , and complete formation of the three-dimensional  $\text{Si}-\text{O}-\text{Al}$  network at  $\sim 900^\circ\text{C}$ . Much of this likely occurs on the surface of the material<sup>20</sup> as the surface area decreases from  $300\text{ m}^2/\text{g}$  to  $<150\text{ m}^2/\text{g}$  (*vide infra*). The  $^{29}\text{Si}$  NMR provides a more-detailed picture of network formation and behavior than DRIFT spectroscopy.

The  $^{29}\text{Si}$  MAS NMR, at lower temperatures, indicates that the  $400^\circ\text{C}$  material contains  $\sim 20\%$   $\text{Si}(\text{OSi})_4$  sites; however, the primary sites are due to the  $\text{Si}-\text{O}-\text{Al}$  network (80%, Figs. 5 and 6). If we start with a homogeneously mixed precursor with



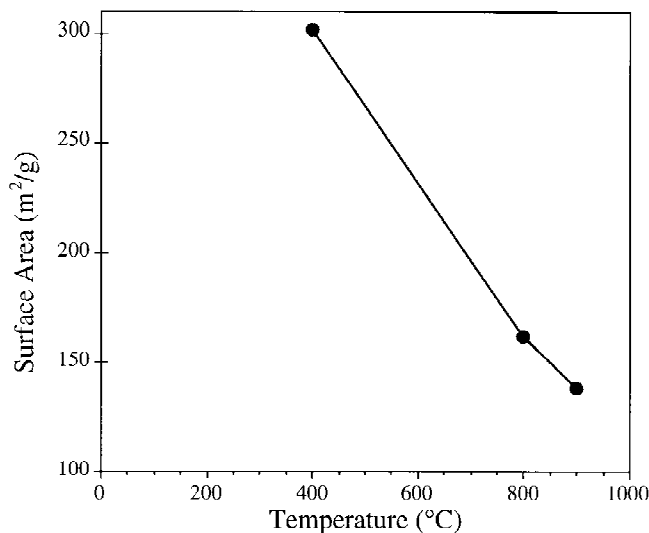


Fig. 9. Plot of surface area as a function of temperature for pyrolyzed mullite precursor.

the stoichiometric composition of mullite (3:1 Al:Si ratio), then, on pyrolysis, we should obtain a homogeneous, but amorphous, phase.  $^{29}\text{Si}$  NMR of this material will give an envelope of environments for silicon wherein the most common has an average of three aluminum atoms that are bound, through oxygen, to one silicon atom. However, tetrahedral silicon has a fourth position that can be occupied by another O–Si, another O–Al, or OH. If we ignore OH for the moment, a statistical argument would have the fourth position being occupied most frequently by aluminum; this will result in an envelope of  $^{29}\text{Si}$  environments with a maximum at a 3:1 Al:Si ratio and skewed toward a small component that consists of  $\text{Si}(\text{OSi})_4$  sites, as shown in Fig. 4. The smoothest envelopes occur at 800°–900°C. Why is a smooth envelope that reflects a statistical average of environments not obtained at lower temperatures?

As shown by DRIFT spectroscopy, some OH groups are present at moderate temperatures. Thus, species such as  $\text{Si}(\text{O})_x(\text{OH})_y$  and  $\text{Al}(\text{O})_m(\text{OH})_n$  are present. On heating, these groups react (by condensation) to form Si–O–Si, Al–O–Si, or Al–O–Al bonds. It is quite possible that Si–O–Si and Al–O–Al bonds form in preference to Al–O–Si bonds; this would account, at least in part, for the lower-temperature segregation. On heating to higher temperatures, diffusion rates finally become sufficiently high to give (by 800°C) a truly statistical distribution. Based on  $^{29}\text{Si}$  NMR and statistical arguments, the material seems to be truly homogeneous.

In the  $^{27}\text{Al}$  NMR studies, the only definitive change that occurs as the samples are heated to 950°C is the growth of the pentacoordinated-aluminum peak at 30 ppm. This change occurs coincident with the homogenization of the  $^{29}\text{Si}$  NMR signals. Thus, a high percentage of pentacoordinated-aluminum sites may be an indicator for the formation of a homogeneous, amorphous material of mullite composition.<sup>24,27,29–32</sup> At this point, additional heating causes tremendous changes.

As the samples are heated from 950° to 1000°C, (i) an exotherm is observed in the DTA, (ii) a mullite powder XRD pattern appears, (iii)  $^{29}\text{Si}$  NMR shows extensive segregation, and (iv)  $^{27}\text{Al}$  NMR shows the complete disappearance of the pentacoordinated aluminum species. For example, when going from 950°C to 1000°C, the number of silicon atoms in  $\text{Si}(\text{OSi})_4$  environments increases from 20% to 47%. In keeping with the segregation that is shown by  $^{29}\text{Si}$  NMR, the 1000°C sample provides a DRIFT spectrum with a broad  $\nu(\text{Si–O})$  band at 1100  $\text{cm}^{-1}$ , which is typical of  $\text{SiO}_4$  sites in silicates, rather than bands at 1130 and 1170  $\text{cm}^{-1}$ , which are typical of aluminosilicates (e.g., the 1300°C sample). In addition, the envelope of  $\nu(\text{Al–O})$  vibrations (centered at 700  $\text{cm}^{-1}$ ) suggests a wide

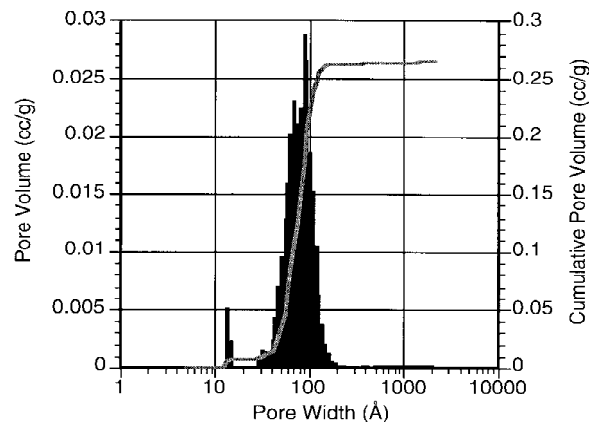


Fig. 10. Plot of pore-size distribution for mullite precursor pyrolyzed to 900°C for 1 h in air.

variety of vibrational modes that are typical of amorphous rather than crystalline materials (see the 1300°C sample). Hence, the 1000°C sample seems to contain significant amounts of amorphous material.

The processes that occur at 1000°C remain the subject of much controversy. Some researchers<sup>9,10,28</sup> have suggested that the exotherm at 990°C results from the reaction of an  $\text{Al}_2\text{O}_3$ -rich spinel phase with the segregated amorphous  $\text{SiO}_2$  to form mullite. Thomson *et al.*,<sup>24</sup> Jaymes *et al.*,<sup>27</sup> and Huling and Messing<sup>26</sup> suggested that it results from the formation of  $\text{Al}_2\text{O}_3$ -rich tetragonal mullite coincident with the demixing of  $\text{SiO}_2$ . Tetragonal mullite seems to have an Al:Si composition of ~4:1, although it could be more aluminum-rich.<sup>25,26</sup>

The complete disappearance of pentacoordinated aluminum coincides with the exotherm at 980°C. Thus, another explanation for this exotherm is that it results from the transformation of pentacoordinated aluminum atoms to more-stable tetrahedral and octahedral coordination.<sup>24–27</sup> This argument is not inconsistent with a crystallization event; it simply explains what occurs at the atomic level.

XRD studies correlate the exotherm at 990°C with the crystallization of mullite. However, NMR and DRIFT spectroscopy indicate that whatever crystallization event does occur, sudden demixing of  $\text{SiO}_2$  (~47% vs. 20%) occurs coincidentally and considerable amounts of amorphous material remain. If tetragonal mullite crystallizes with an Al:Si composition closer to 4:1 than 3:1, then  $\text{SiO}_2$  demixing should be expected. For a 4:1 ratio, one can calculate that ~25 mol% excess  $\text{SiO}_2$  should be present as a second phase. The fact that we observe 47 mol% suggests that tetragonal mullite, if it forms, is more aluminum-rich. If the ratio were 5:1, then demixing should produce ~40 mol% free  $\text{SiO}_2$ , which is closer to 47 mol%. Thus, if tetragonal mullite forms initially, its Al:Si composition is closer to 5:1.<sup>25,26</sup> Thus, the crystallization of  $\text{Al}_2\text{O}_3$ -rich tetragonal mullite can account for the exotherm at 980°C, the mullite powder XRD pattern at 1000°C, and the demixing of  $\text{SiO}_2$ . Its formation also explains the DRIFT spectra that are obtained at temperatures of  $\geq 1000^\circ\text{C}$ .

The formation of considerable amounts of free  $\text{SiO}_2$  explains the 1100  $\text{cm}^{-1}$   $\nu(\text{Si–O})$  band that is typical of pure  $\text{SiO}_2$  in the 1000°C spectrum. Because the material seems to be homogeneous at 950°C, demixing at 1000°C should occur homogeneously, e.g., as in spinodal decomposition. Therefore, tetragonal mullite grains should form in an amorphous  $\text{SiO}_2$  matrix. The matrix/mullite interfacial interactions will influence the Al–O bonding environments and could cause the broad envelope of  $\nu(\text{Al–O})$  vibrations that is observed at 1000°C.

As the samples are heated to 1300° and then to 1400°C, the  $^{29}\text{Si}$  NMR indicates that the demixed  $\text{SiO}_2$  readsorbs (18% at 1300°C) as the literature peaks for phase-pure orthorhombic mullite appear.<sup>24–28</sup> In the DRIFT spectra, the 1100  $\text{cm}^{-1}$



$\nu(\text{Si}-\text{O})$  band splits into the 1130 and 1170 cm<sup>-1</sup> bands, as expected for mullite, and another tetrahedral  $\nu(\text{Si}-\text{O})$  band that is typical of mullite appears at 900 cm<sup>-1</sup>.<sup>33-35</sup> Similarly, the powder XRD pattern peaks sharpen and increase in intensity.

Based on the above-mentioned analysis and the work of Gerardin *et al.*,<sup>24</sup> Jaymes *et al.*,<sup>27</sup> and Huling and Messing,<sup>26</sup> we conclude that the OOPS precursor, when heated in air to temperatures of 400°–600°C, first decomposes to a homogeneous, amorphous material with a mullite composition. When heated further to 1000°C, this material crystallizes to an Al<sub>2</sub>O<sub>3</sub>-rich tetragonal mullite with coincident demixing of considerable amounts of amorphous SiO<sub>2</sub>. Finally, at temperatures of  $\geq 1300^\circ\text{C}$ , the demixed SiO<sub>2</sub> readsorbs as stoichiometric orthorhombic mullite finally crystallizes.

### (7) Specific Surface Area and Porosimetry

Figure 9 shows a plot of the surface area as a function of precursor pyrolysis temperature. This characterization technique gives Brunauer–Emmett–Teller method (BET) surface area, pore size, and pore-size distribution. As the pyrolysis temperature increases, densification occurs via diffusion, which results in the reduction of surface area.<sup>20</sup> The exceptionally high surface area for the 400°C sample is probably due to some porous char (20% from TGA) that remains in the material, as shown in related efforts.<sup>36</sup> Figure 10 gives the pore-size distribution for the mullite sample pyrolyzed to 900°C for 1 h in air. The mean pore size is  $\sim 80 \text{ \AA}$  (8 nm).

SEM micrographs (not shown) of as-fired surfaces were taken of mullite precursor that was pyrolyzed to 1000° and 1400°C. At 1000°C, no visible pores were observed and only hints of microstructural development were observed. The SEM micrographs at 1400°C revealed some texture that was associated with the appearance of small ( $\sim 0.2\text{--}0.3 \mu\text{m}$ ) grains and some pores that probably resulted from grain growth without densification. These results are consistent with all the other analytical data, which indicates that fully crystalline mullite forms only at temperatures of  $\geq 1300^\circ\text{C}$ , not at 1000°C.

## IV. Conclusions

The mullite precursor used in the studies reported here was synthesized directly from SiO<sub>2</sub>, Al(OH)<sub>3</sub>, and triethanolamine in ethylene glycol by a direct, low-temperature route. Based on the above-mentioned analyses and the work of Gerardin *et al.*,<sup>24</sup> Jaymes *et al.*,<sup>27</sup> and Huling and Messing,<sup>26</sup> we conclude that the OOPS precursor, when heated in air to temperatures of 400°–600°C, first decomposes to a homogeneous, amorphous material with a mullite composition. When heated further to 1000°C, this material crystallizes to an Al<sub>2</sub>O<sub>3</sub>-rich tetragonal mullite with coincident demixing of considerable amounts of amorphous SiO<sub>2</sub>. Finally, at temperatures of  $\geq 1300^\circ\text{C}$ , the demixed SiO<sub>2</sub> readsorbs as stoichiometric orthorhombic mullite finally crystallizes.

Thus, the OOPS-derived precursor behaves as a “single-phase” (as defined in the literature), atomically mixed precursor. The results that have been presented indicate that this precursor behaves identically to the mullite precursors that were synthesized by various methods (sol–gel, slowly hydrolyzed gels, etc.) that have been reported in the literature.<sup>1,9,24–27</sup> Perhaps most importantly, this precursor can be made in large quantities in a few hours; in contrast, sol–gel processing of atomically mixed mullite materials can require weeks.<sup>25–27</sup>

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