

## Electrical conductivity and thermopower of Cu–SiO<sub>2</sub> nanogranular films

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We have measured the thermopower  $S$  and electrical conductivity  $\sigma$  in a series of Cu<sub>*x*</sub>(SiO<sub>2</sub>)<sub>1–*x*</sub> nanogranular films between 2 and 300 K with Cu volume fraction  $x$  varying from 0.43 up to 1.0. At low temperatures, disorder-enhanced electron–electron interaction effects dictate the behavior of  $\sigma$ . A crossover of the temperature dependence from  $\sigma \propto \sqrt{T}$  to  $\sigma \propto T^{1/3}$  is observed as  $x$  is lowered and the metal–insulator transition is approached.  $S$  is small, shows linear temperature dependence, and is rather insensitive to the change of  $x$ . Effects of annealing are also discussed. © 2002 American Institute of Physics. [DOI: 10.1063/1.1493668]

Metal–insulator composites have demonstrated interesting physics when their constituent grain size is reduced to only a few nanometers. Recently, the giant Hall effect has been discovered in nonmagnetic Cu<sub>*x*</sub>(SiO<sub>2</sub>)<sub>1–*x*</sub> nanogranular systems.<sup>1</sup> Near 3 orders of magnitude enhancement in the Hall coefficient was observed when the Cu volume fraction  $x$  was reduced down to 0.51. This effect disappeared after annealing the samples that significantly enlarges the grain size. The giant Hall effect has been previously observed<sup>2–6</sup> also in magnetic (NiFe)–SiO<sub>2</sub> and Fe–SiO<sub>2</sub> nanogranular films. However, its discovery in the nonmagnetic Cu–SiO<sub>2</sub> films shows that quantum-interference effects associated with the small grain size could be responsible for the apparent decrease in the effective charge carrier concentration. In one study<sup>7</sup> of the electrical conductivity of Cu–SiO<sub>2</sub> composites with  $0.17 \leq x \leq 0.33$ , a variable range hopping conduction of the form  $\sigma \propto \exp\{-(T_0/T)^{1/2}\}$  was observed and explained through the Coulomb interaction and the presence of a large random potential. Investigations of thermopower provide complimentary information to what one obtains through the study of electrical conductivity. However, there exist a limited number of experimental studies of thermopower in metal–insulator nanocomposites, especially at liquid-helium temperatures where the signal is small and measurement is very difficult. Hurvits *et al.*<sup>8</sup> measured the room-temperature thermopower of Al–Ge films and found it consistent with the theoretical predictions given by Bergman and Levy.<sup>9</sup> Jing and Yan<sup>10</sup> observed a small and temperature insensitive thermopower for magnetic (NiFe)–SiO<sub>2</sub> and Fe–SiO<sub>2</sub> composites near the percolation threshold between 70 and 300 K.

In this letter, we present detailed studies of electrical conductivity  $\sigma$  and thermopower  $S$  of the nonmagnetic Cu–SiO<sub>2</sub> nanogranular films from 2 up to 300 K and with Cu volume fraction  $x$  from 1.0 down to 0.434, which is just above the classical percolation threshold<sup>1</sup>  $x_c \approx 0.43$ . The effect of annealing is also discussed.

The films were prepared by co-sputtering the source materials onto glass substrates held at 50 °C. The base pressure of the sputtering chamber was below  $2 \times 10^{-7}$  Torr. The Cu volume fraction was determined from energy-dispersive x-ray spectroscopy analysis. All investigated films were about 1  $\mu\text{m}$  thick. The annealing condition was 450 °C for 1 h. Resistivity was measured from 2 to 300 K using the standard four-probe ac technique with the aid of a 16 Hz excitation of a Linear Research bridge. Thermopower measurements were performed using a longitudinal steady-state technique. At low temperatures, thermopower signals were very small. Therefore, fine NbTi superconducting wires were used as our voltage leads from 2 to 8 K to avoid the contribution to the thermopower from the wires. Furthermore, we employed Ge thermometers, which have a resolution of 1 mK or better, to accurately determine the temperature gradient across the sample. From 8 to 300 K, we used copper–constantan thermocouples with the copper legs serving also as voltage leads. The thermopower was corrected for the contribution of the copper. A miniature strain gauge served as a heater in both cases.

The room-temperature value of electrical conductivity  $\sigma$  decreases by 4 orders of magnitude, from  $3 \times 10^4$  down to about 4 S/cm, when the Cu volume fraction  $x$  is lowered from 0.804 down to 0.434. Figure 1 shows the temperature dependence of the normalized conductivities for four representative samples with  $x = 0.434, 0.510, 0.726,$  and  $0.804$ .

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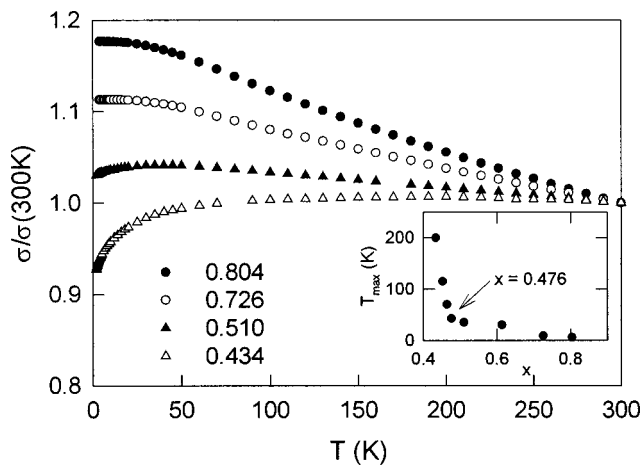


FIG. 1. Temperature dependence of normalized conductivity. The inset shows the temperatures  $T_{\max}$ , where  $\sigma$  reaches a maximum for a given sample, plotted against  $x$ .

One general observation is that for a given sample,  $\sigma$  has a maximum value at a certain temperature  $T_{\max}$ , which can be taken as a rough measure of the strength of disorder in the system.<sup>11</sup>  $T_{\max}$  as a function of  $x$  is depicted in the inset of Fig. 1 where a rather abrupt change can be seen at the Cu volume fraction of about 0.47. Shown in Fig. 2 is the normalized electrical conductivity  $\sigma$  as a function of the square root of the temperature  $\sqrt{T}$  from 2 to 10 K for  $0.476 \leq x \leq 0.613$ . A temperature dependence,  $\sigma \propto \sqrt{T}$ , is unambiguously demonstrated in the plot and can be well understood as the result of electron–electron interactions in a three-dimensional weakly disordered system.<sup>12</sup> However, as  $x$  is reduced below 0.47, the temperature dependence of  $\sigma$  deviates from  $\sqrt{T}$  and it is found that  $T^{1/3}$  is in fact a better description, as illustrated in the inset of Fig. 2. When the system is close to the metal–insulator transition, the diffusion coefficient  $D$  can no longer be treated as a constant and should be renormalized. One may take this into account by using the Einstein equation  $\sigma = N(E_F)e^2D$ , which leads to the  $T^{1/3}$  dependence.<sup>13</sup> Here  $N(E_F)$  is the electronic density of states at the Fermi level. The diffusion constant  $D$  enters the problem because the motions of carriers are diffusive in

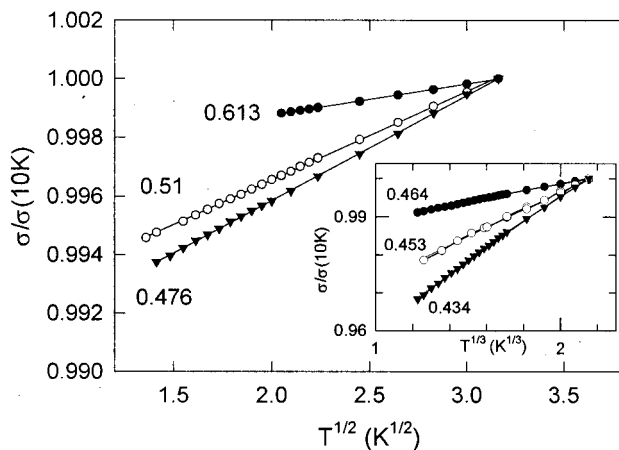


FIG. 2. Normalized conductivity as a function of  $\sqrt{T}$  from 2 to 10 K for  $x > 0.47$ . The inset represents normalized conductivity as a function of  $T^{1/3}$  for  $x < 0.47$  over the same temperature range. All straight lines through the data points are guides for the eye.

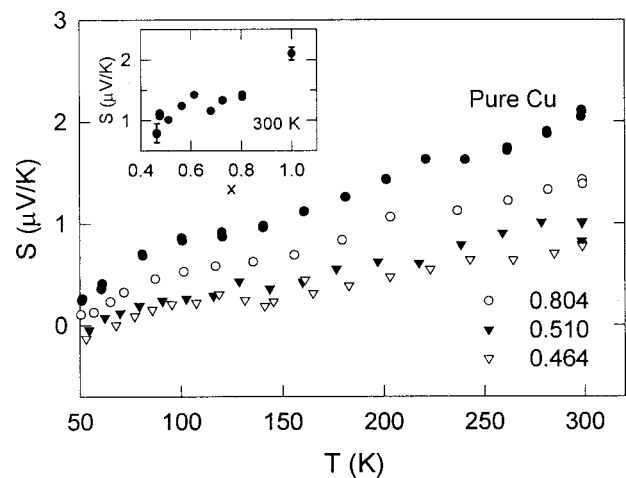


FIG. 3. Temperature dependence of thermopower  $S$  from 50 to 300 K. The thermopower of a pure Cu film is also plotted (filled circles) for comparison. The inset displays the room-temperature value of  $S$  as a function of  $x$ .

the presence of disorder, which results in multiple elastic scattering. Note that the change of temperature dependence coincides with the sudden rise seen on  $T_{\max}$  versus  $x$  (see the inset of Fig. 1), indicating that stronger disorder indeed gives rise to the transition to the  $T^{1/3}$  dependence.

To study the effect of grain size, we annealed the sample with  $x = 0.510$ , which is at the quantum percolation threshold.<sup>1</sup> After annealing, the sample shows a ten times higher conductivity, still following the  $\sqrt{T}$  dependence but with a much smaller slope, and  $T_{\max}$  shifts from about 35 down to 20 K. These results are consistent with the transmission electron microscopy analysis which shows that the average grain size grows from 3 to about 10 nm upon annealing. The chance for a charge carrier to be scattered by grain boundaries is hence greatly reduced at a given temperature.

Figure 3 shows the thermopower  $S$  as a function of temperature for three representative samples with  $x = 0.804$ , 0.510, and 0.464 from 50 to 300 K. A pure Cu film is fabricated and measured in the same way, and the result is also plotted for comparison in Fig. 3. All samples have small thermopower values.  $S$  displays essentially linear temperature dependence and is rather insensitive to the amount of Cu in the system, in sharp contrast to the very rapid decrease of  $\sigma$  with decreasing  $x$ . The linear temperature dependence can be understood as the behavior of the diffusive thermopower of the Cu matrix, where charge transport takes place. The mean-free path of phonons is greatly limited by the small grain size. Therefore, the phonon drag effect, otherwise notable on pure Cu with a large crystalline size at around 50 K, does not seem to be present in these nanograin-size films. The inset of Fig. 3 plots the room-temperature values of  $S$  for samples with different copper content. The insensitivity of  $S$  to  $x$  is due to the fact that it is the thermopower of the Cu matrix that we are essentially measuring. Unlike  $\sigma$ , the behavior of  $S$  in metal–insulator composites has not been much explored. The slightly decreasing trend of  $S$  with respect to decreasing  $x$  might be due to the disorder-induced modification to the density of states around the Fermi energy. A slow variation of  $S$  across the percolation threshold is also in line with the prediction of Bergman and Levy.<sup>9</sup>

Shown in Fig. 4 is the thermopower as a function of

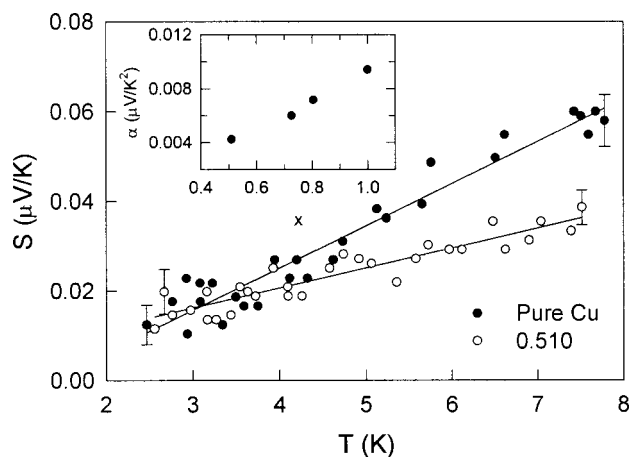


FIG. 4. Temperature dependence of thermopower  $S$  from 2 to 8 K. The straight lines are linear fits and error bars are included. The inset shows the slope  $\alpha$  of the lines plotted against  $x$ .

temperature for a representative sample with  $x=0.510$  and for a pure Cu film from 2 to 8 K. Error bars are indicated in Fig. 4. One can see that  $S$  has very small values and varies linearly with temperature. It is again believed to be the behavior of the diffusive thermopower of the Cu matrix. The slope  $\alpha = dS/dT$  is extracted and plotted against  $x$ , as shown in the inset of Fig. 4. It is clear that  $S$  becomes less sensitive to temperature as  $x$  is lowered, consistent with our observation at higher temperatures. Annealing introduces little change to the magnitude and temperature dependence of  $S$ , in sharp contrast to what it does to the electrical conductivity. Considering that  $S$  is shown to be rather insensitive to the strength of disorder in the system, this pronounced difference in the behavior of  $S$  and  $\sigma$  is perhaps not surprising.

In conclusion, the electrical conductivity  $\sigma$  and thermopower  $S$  are studied for  $\text{Cu}_x(\text{SiO}_2)_{1-x}$  nanogranular films from 2 to 300 K with the Cu content between 0.43 and 1.0.

At low temperatures, disorder-enhanced electron–electron interaction effect plays an important role in the charge transport. A crossover from  $\sqrt{T}$  to  $T^{1/3}$  dependence of  $\sigma$  is observed as the system approaches the metal–insulator transition.  $S$  is small and varies linearly with temperature at both low and high temperatures. Annealing has considerable influence on the behavior of  $\sigma$ .

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